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**Authors**

Zhang, Liming

Liu, Chong

Wong, Andrew Barnabas

et al.

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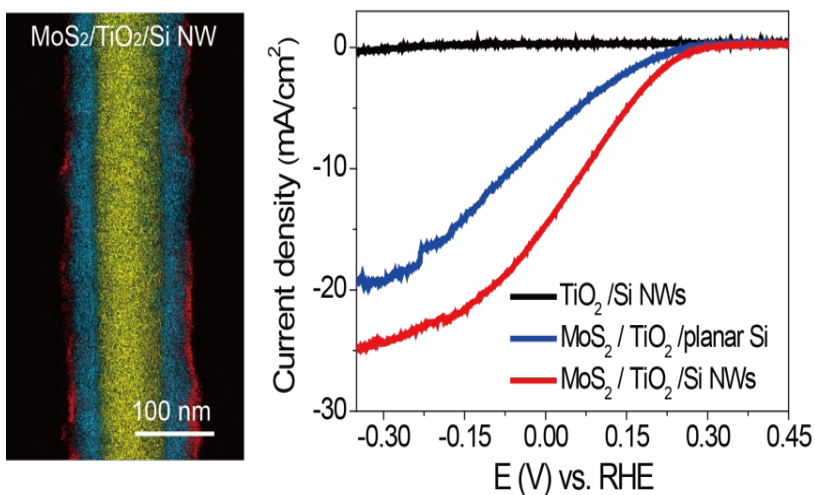
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### **MoS<sub>2</sub>-Wrapped Silicon Nanowires for Photoelectrochemical Water Reduction**



Silicon (Si) NW arrays were employed as a model photocathode system for MoS<sub>2</sub> wrapping, and their solar-driven HER activity was evaluated. The photocathode is comprised of a well-defined MoS<sub>2</sub>/TiO<sub>2</sub>/Si coaxial NW heterostructure, which yielded photocurrent density up to 15 mA/cm<sup>2</sup> at 0 V vs. the reversible hydrogen electrode (RHE) with good stability under operating conditions.

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# MoS<sub>2</sub>-Wrapped Silicon Nanowires for Photoelectro-chemical Water Reduction

*Liming Zhang,<sup>†,‡,Δ</sup> Chong Liu,<sup>†,‡</sup> Andrew Barnabas Wong,<sup>†,§</sup> Joaquin Resasco,<sup>‡</sup> and Peidong Yang<sup>\*,†,‡,§</sup>*

<sup>†</sup>Department of Chemistry, <sup>‡</sup>Department of Chemical Engineering and <sup>Δ</sup>Department of Materials Science and Engineering, University of California, Berkeley, CA 94720, United States

<sup>§</sup>Materials Science Division, Lawrence Berkeley National Laboratory, Berkeley, CA 94720, United States

<sup>Δ</sup>These authors contributed equally to this work.

E-mail: p\_yang@berkeley.edu.

**ABSTRACT:** Integration of molybdenum disulfide ( $\text{MoS}_2$ ) onto high surface area photocathodes is highly desired to minimize the overpotential for solar-powered hydrogen evolution reaction (HER). Semiconductor nanowires (NWs) are beneficial for use in photoelectrochemistry because of a large electrochemically available surface area and their inherent ability to decouple light absorption and the transport of minority carriers. Here, silicon (Si) NW arrays were employed as a model photocathode system for  $\text{MoS}_2$  wrapping, and their solar-driven HER activity was evaluated. The photocathode is comprised of a well-defined  $\text{MoS}_2/\text{TiO}_2/\text{Si}$  coaxial NW heterostructure, which yielded photocurrent density up to  $15 \text{ mA/cm}^2$  at 0 V vs. the reversible hydrogen electrode (RHE) with good stability under operating conditions. This work reveals that earth-abundant electrocatalysts coupled with high-surface area NW electrodes can provide performance comparable to noble-metal catalysts for photocathodic hydrogen evolution.

**KEYWORDS:**  *$\text{MoS}_2$ , Si nanowire (NW) array, coaxial heterostructure, Photoelectrochemistry, Hydrogen evolution reaction (HER)*

## 1 Introduction

The vision of utilization of hydrogen as a future energy source requires a sustainable, highly efficient, and cost-effective production method [1]. To use renewable energy to produce hydrogen fuel, solar-driven water splitting is one of the most promising approaches [2, 3]. Effective photoelectrochemical (PEC) water splitting devices must be able to absorb a large fraction of incident sunlight, generate high photocurrent with sufficient photovoltage, and exhibit long-term stability. One challenge in designing such a device is to reduce the reaction overpotential, by increasing the reaction kinetics on the surface. While existing noble metal catalysts, such as Pt, are highly active at catalyzing the hydrogen evolution

reaction (HER) [4, 5], the high cost of these materials limits the economic viability of their use for hydrogen production. The intensive search for earth-abundant, inexpensive, and nontoxic catalysts with comparable performance for the HER has led to significant progress in the development of new catalysts, including metal alloys [6, 7], nitrides [8], borides [9], carbides [9, 10], chalcogenides [11-16], and phosphides [17]. In particular, metal chalcogenides, such as molybdenum sulfide ( $\text{MoS}_x$ ), are an exciting group of HER catalysts that exhibit promising activity in both crystalline and amorphous forms, as well as in the form of grafted molecular clusters [11, 12, 18-20]. Despite the considerable efforts dedicated to investigating and optimizing the catalytic activity of various  $\text{MoS}_x$  materials, the effective integration of  $\text{MoS}_x$  onto photocathodes is still a challenge. To achieve high geometric area-normalized HER activity, it is necessary to carefully engineer  $\text{MoS}_x$  functional nanostructures to maximize the density of active sites, which have been identified both theoretically [21, 22] and experimentally [11, 12] to be the uncoordinated sulfur edge sites.

High-surface area photoelectrodes, such as semiconductor nanowire (NW) arrays, are attractive because of their large semiconductor/electrolyte interfacial area, enhanced light scattering effect, and efficient transport of charge carriers [23-25]. Silicon (Si), with bandgap of 1.12 eV, is a promising candidate as a light-absorber due to its relatively low cost, excellent carrier transport properties, and suitable band edge with respect to HER potential [25]. As atomically thin  $\text{MoS}_2$  has limited light absorption, the successful integration of  $\text{MoS}_2$  with semiconductor NWs constitutes an ideal system for PEC hydrogen production.

In this work, using a *p*-Si NW array as scaffold, a well-defined  $\text{MoS}_2/\text{TiO}_2/\text{Si}$  NW integrated coaxial heterostructure was constructed as a model photocathode system for PEC hydrogen generation. Multilayer  $\text{MoS}_2$  with a small domain size of  $\sim 10$  nm was uniformly wrapped on the surface of the NWs, exposing a high density of active uncoordinated edge sites. An onset of photocurrent at  $\sim 0.30$  V

vs. the reversible hydrogen electrode (RHE), together with a short-circuit current density of 15 mA/cm<sup>2</sup> under simulated 1 sun illumination was achieved. Additionally, this functional device exhibits good stability under operating conditions. This system demonstrates the principle that earth-abundant electrocatalysts coupled with high-surface area NW electrodes can provide performance comparable to noble-metal catalysts for photocathodic hydrogen evolution. This principle could serve as a guide for future studies in the field of solar-powered chemical fuel production, by offering an additional dimension for device engineering through the benefits of the NW morphology.

## 2 Results and Discussion

Metal oxide overlayers can assist to preserve the photo-response of Si [26], consequently for our study, Si/TiO<sub>2</sub> core-shell structures were prepared via atomic layer deposition (ALD) of 30 nm of crystalline TiO<sub>2</sub> on the NW surface at 573 K. The MoS<sub>2</sub> layer was synthesized by thermolysis of ammonium tetrathiomolybdate [(NH<sub>4</sub>)<sub>2</sub>MoS<sub>4</sub>] at low temperature [27, 28]. (NH<sub>4</sub>)<sub>2</sub>MoS<sub>4</sub> was dissolved in an organic solvent and drop-casted onto the NW surface [additional details are provided in the Electronic Supplementary Materials (ESM)]. After drop-casting, the coated NWs were annealed at 573~673 K in a N<sub>2</sub>/H<sub>2</sub> (80/20 sccm) atmosphere at ambient pressure to form MoS<sub>2</sub>.

A single TiO<sub>2</sub>/Si NW wrapped with MoS<sub>2</sub> was characterized by high-angle annular dark field (HAADF) imaging (Figure 1a). The MoS<sub>2</sub> and TiO<sub>2</sub> shells appear brighter as Ti and Mo atoms located in the shell have a higher atomic number than the Si atoms in the core. High-resolution transmission electron microscopy (HRTEM) revealed multiple layers of MoS<sub>2</sub> on the NW surface, as evidenced by the lattice fringes with a spacing of 0.62 nm, in agreement with the expected MoS<sub>2</sub> interlayer distance of 0.614 nm (Figure 1b) [29]. The crystallinity of MoS<sub>2</sub> highly depends on the annealing temperature, with

increasing temperature resulting in higher crystallinity (Figure S-1 in the ESM). Although the surface is mostly covered with MoS<sub>2</sub> basal planes, the domain size is smaller than 10 nm and crystal edges of MoS<sub>2</sub> are evidently abundant (indicated by the red arrows in Figure 1b). Relative to MoS<sub>2</sub> coated on planar electrode surfaces [18, 30], this core-shell structure exposes more MoS<sub>2</sub> edges because of the large surface area afforded by the NW morphology. Further analysis of the elemental distribution was obtained from energy dispersive X-ray spectroscopy (EDS) mapping. The elemental map shows a homogenous distribution of Si in the core (Figure 1c), Ti in the inner shell (Figure 1d), and a thin layer of Mo in the outer shell (Figure 1e). The combined elemental mapping in Figure 1f demonstrates a well-defined coaxial heterostructure of MoS<sub>2</sub>/TiO<sub>2</sub>/Si along a single NW. The uniform coating of MoS<sub>2</sub> distributes the solar-generated electron flux evenly along the semiconductor light-absorber. The effective utilization of the high surface area of the NW decreases the surface flux of electrons, leading to a reduced HER kinetic overpotential.

Additional characterization was performed to understand the nature of the synthesized MoS<sub>2</sub>, from which deeper insight into the performance may be gained. Thermolysis of (NH<sub>4</sub>)<sub>2</sub>MoS<sub>4</sub> at low temperature in the presence of H<sub>2</sub> will result in the conversion initially to MoS<sub>3</sub> and finally to MoS<sub>2</sub> [27]. X-ray photoelectron spectroscopy (XPS) (Figure 2) and Raman spectroscopy (Figure S-2 in the ESM) were used to investigate the chemical states of Mo and S before and after annealing. The binding energy of the Mo 3d<sub>5/2</sub> photoelectron peak remained constant (229.2 eV) before and after annealing, indicating a 4+ oxidation state for Mo [13]. The S 2p region of MoS<sub>x</sub> before annealing shows two doublets, consistent with that seen for MoS<sub>3</sub> amorphous catalysts [13]. The doublet at lower binding energy (162.2 and 163.4 eV) is attributed to S<sup>2-</sup> at the basal plane, characteristic of MoS<sub>2</sub>. The doublet at higher binding energy (163.8 and 165.0 eV) is attributed to S<sub>2</sub><sup>2-</sup> at the bridge sites. After annealing, only the doublet at lower binding energy corresponding to MoS<sub>2</sub> is observed. These results demonstrate that this

low temperature annealing process is a simple method for the synthesis of stoichiometric MoS<sub>2</sub> catalysts with small domain sizes and uniform coverage, even on highly corrugated surfaces.

The electrocatalytic performance of MoS<sub>2</sub> layers was first evaluated using planar  $p^+$ -Si substrate as an electrode. Different volumes of precursor solution, corresponding to different loadings of the MoS<sub>2</sub> cocatalyst, were drop-casted onto the electrode surface, followed by annealing at 673 K in a N<sub>2</sub>/H<sub>2</sub> atmosphere at ambient pressure. The TiO<sub>2</sub> interlayer was required to achieve stable current; otherwise, the underlying Si will be gradually oxidized, leading to a decrease in activity (Figure S-3 in the ESM). The catalytic performance was measured using iR-compensated linear scan voltammetry in 0.5 M H<sub>2</sub>SO<sub>4</sub> electrolyte in a three-electrode configuration. Onset of cathodic current was observed at approximately -0.20 V vs. the reversible hydrogen electrode (RHE) (Figure 3a). The cathodic current increased with increasing loading of MoS<sub>2</sub> and exhibited a Tafel slope of 75~84 mV per decade (Figure 3b). The apparent exchange current densities  $j_0$  were obtained from the Tafel plots and are summarized in the Supporting Information (Table S-1 in the ESM). The Tafel slope observed here compares favorably with MoS<sub>2</sub> synthesized by chemical vapor deposition (CVD) on glassy carbon electrodes [18, 31]. After evaluating the dependence of HER activity on the MoS<sub>2</sub> loading density, the optimal condition was applied to the NW photoelectrodes for subsequent PEC investigation.

To evaluate the impact of MoS<sub>2</sub> cocatalyst deposition on the solar-to-hydrogen activity of the Si NW photocathodes, PEC measurements were performed under the same conditions as the electrochemical measurements described above. A well-defined high-surface area Si NW arrays electrode was fabricated via deep reactive ion etching (DRIE) (experimental details are provided in the ESM). The scanning electron micrographs are shown in Figure 3c-d. The etched NW arrays are approximately 30  $\mu$ m in length and 800 nm in diameter, and are arranged on a square lattice with periodicity of 2  $\mu$ m. In order to achieve a positive photovoltage, a heavily doped  $n^+$  emitter layer was diffused into the surface region of



Si NWs via rapid temperature annealing at 1173 K for 3 min, using arsenic (As) as an n-type dopant. Following the procedure outlined above, a  $\text{MoS}_2/\text{TiO}_2/n^+p\text{-Si}$  NW coaxial heterostructure was created from the Si NW array. For comparison,  $\text{Pt}/\text{TiO}_2/n^+p\text{-Si}$  NW was also fabricated by sputtering Pt nanoparticles onto the NW surface (Figure S-4 in the ESM). The PEC performance of these photoelectrodes was tested under identical conditions, using  $100 \text{ mW/cm}^2$  simulated AM 1.5G irradiation. Figure 3e shows the activity comparison of Si photocathodes before and after deposition of  $\text{MoS}_2$ . Si NW arrays exhibit nearly no photo-activity in the absence of a cocatalyst. When  $\text{MoS}_2$  was deposited onto the NW surface, the onset of photocurrent shifted substantially to a more positive potential ( $\sim 0.30\text{V}$  vs. RHE), only 100 mV cathodic of the Pt nanoparticle decorated electrode. For the affordable application of overall water splitting by coupling of dual light absorbers, a current density on the order of  $10 \text{ mA/cm}^2$  is necessary [20]. The  $\text{MoS}_2$  layers wrapped Si NW array photocathode provides a current density of  $\sim 15 \text{ mA/cm}^2$  at 0 V vs. RHE, indicating that this photocathode can be applicable to an affordable integrated water-splitting system. To study the impact of NW surface area on the required catalytic activity, an equal loading of  $\text{MoS}_2$  was deposited on the surface of a planar  $\text{TiO}_2/n^+p\text{-Si}$  electrode. As shown in Figure 3e, the photo-activity was substantially improved in the Si NW array, indicating that increasing the surface area will decrease the surface flux of electrons, leading to a lower requirement for the catalytic activity on the surface. Figure 3f shows the typical spectral response for  $\text{MoS}_2/\text{TiO}_2/n^+p\text{-Si}$  NWs (experimental details are provided in the ESM). The external quantum efficiency reaches a maximum at 500 nm, consistent with the external quantum efficiency observed for Si wires coated with a thin layer of Pt nanoparticles [4]. The maximum external quantum efficiency is larger than 0.75,  $\sim 25\%$  higher than previously reported for Ni-Mo coated Si wire arrays [7].

Another important criterion for PEC energy conversion is the long-term stability of the photoelectrode under operating conditions. The short-term stability of the coaxial  $\text{MoS}_2/\text{TiO}_2/\text{Si}$  NW structure and its

faradaic efficiency for hydrogen reduction were evaluated by chronoamperometry. The potential of the photocathode was held at -0.33 V vs. RHE under 100 mW/cm<sup>2</sup> simulated AM 1.5 G irradiation, and both the photocurrent and hydrogen gas concentration in the He carrier gas were monitored. Bubbles were continuously evolved from photocathode, and the accumulation and release of these bubbles led to the observed variations in photocurrent over the duration of the measurement. The photocurrent of the coaxial MoS<sub>2</sub>/TiO<sub>2</sub>/Si NW structure remained at approximately 21 mA/cm<sup>2</sup> for 75 min without degradation (Figure 4), and the faradaic efficiency for hydrogen evolution to H<sub>2</sub> was found to be ~100% (more details are provided in the ESM and Figure S-5). This result shows that the coaxial MoS<sub>2</sub>/TiO<sub>2</sub>/Si NWs are stable and are indeed reducing water into H<sub>2</sub>.

### **3 Conclusions**

In brief, this work shows the feasibility of functionalizing a Si NW array photocathode with a MoS<sub>2</sub> cocatalyst to drive solar hydrogen production. The exploration of the earth-abundant catalyst MoS<sub>2</sub>, and its application to high surface area Si NW array electrodes establishes a general strategy to replace precious-metal cocatalysts for efficient and affordable solar-to-fuel application at a large scale.

### **Acknowledgments**

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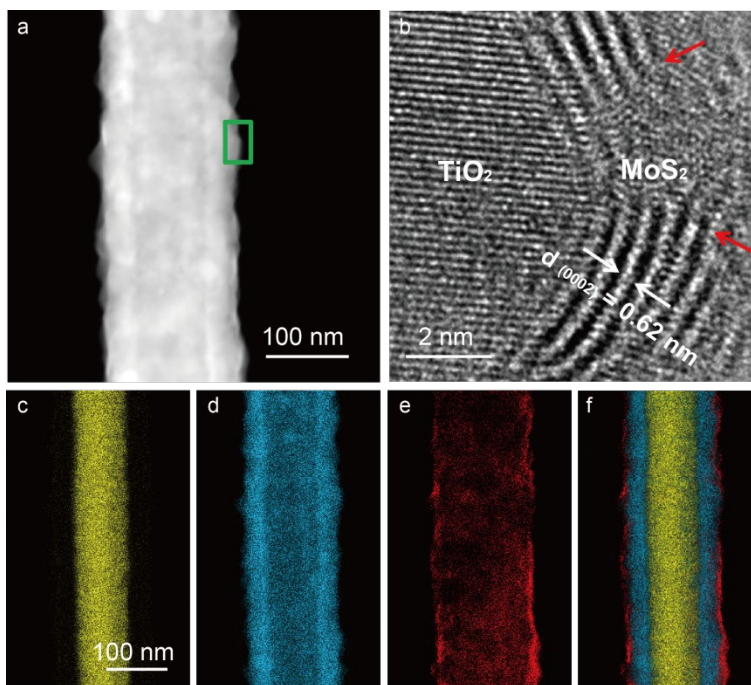
<sup>Δ</sup> **Present address:** Department of Materials Science and Engineering, Stanford University, Stanford, CA 94305, USA

**Notes:** The authors declare no competing financial interests.

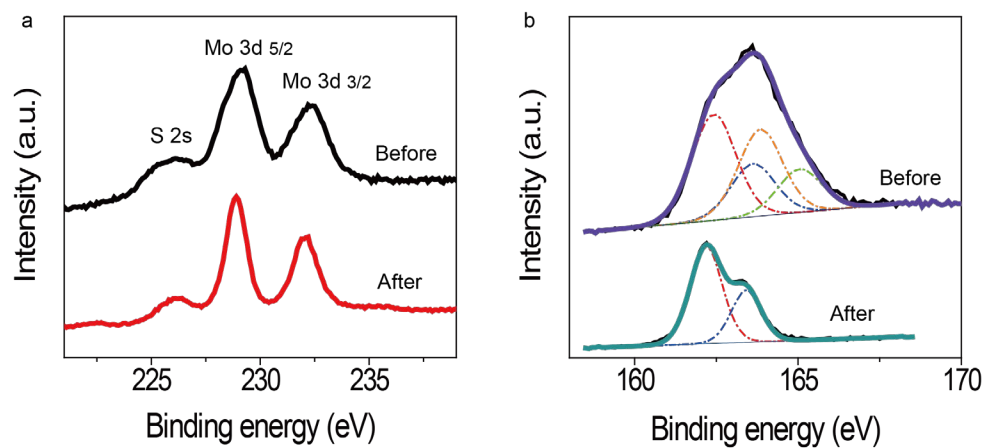
**Electronic Supplementary Materials:** Experimental details, additional TEM, Raman data, stability measurement, and calculations of electrochemical performance. This material is available in the online version of this article at [http://dx.doi/10.1007/\\*\\*\\*\\*\\*](http://dx.doi/10.1007/*****)

## Figures

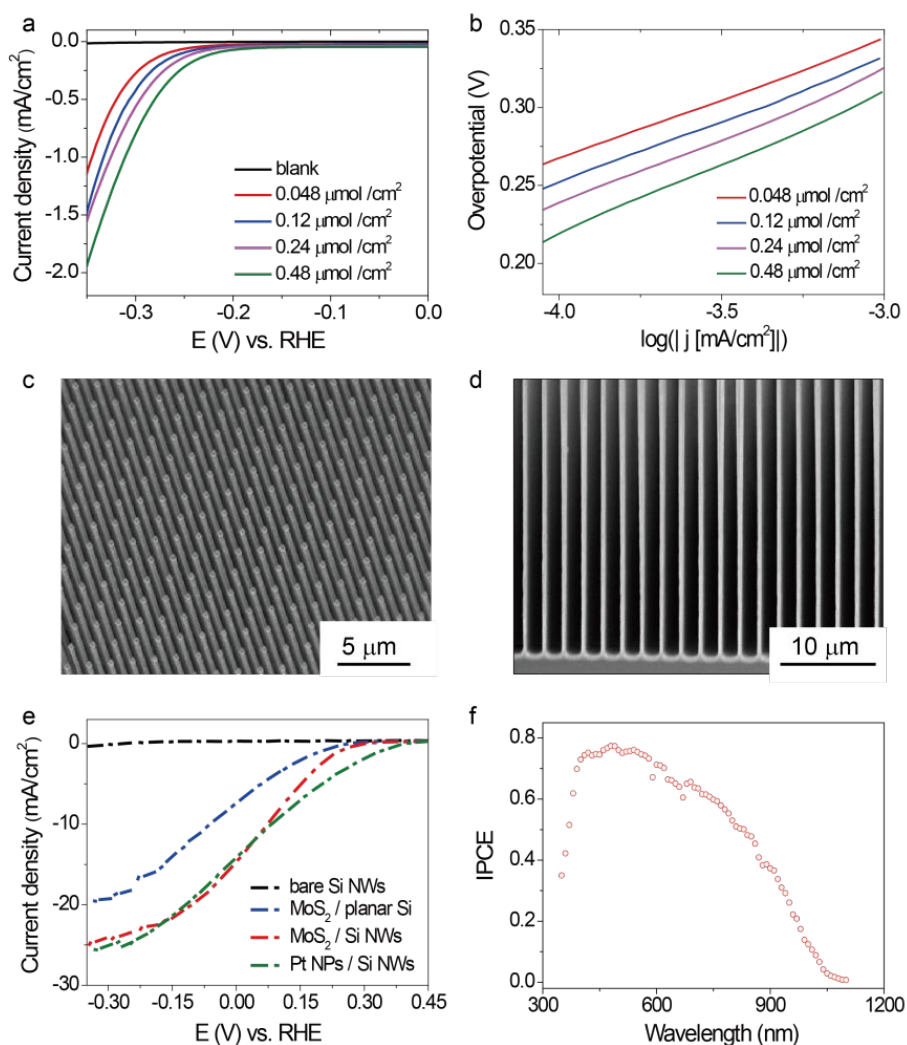
**Figure 1** (a) HAADF image of the MoS<sub>2</sub>/TiO<sub>2</sub>/Si coaxial structure on a single NW. (b) HRTEM image taken at the MoS<sub>2</sub>/TiO<sub>2</sub> interface as shown in the green box of (a). The lattice fringes show a spacing of 0.62 nm between adjacent layers, which is in agreement with the expected MoS<sub>2</sub> interlayer distance. The red arrows indicate the edge of a MoS<sub>2</sub> cluster where HER-active MoS<sub>2</sub> edge sites are likely exposed on the NW surface. (c-f) EDS elemental mapping along a single NW, which shows the elemental distribution of Si (c), Ti (d), Mo (e), and the combined signal of Mo/Ti/Si (f), respectively, demonstrating the well-defined integrated coaxial heterostructure of MoS<sub>2</sub>/TiO<sub>2</sub>/Si NW.



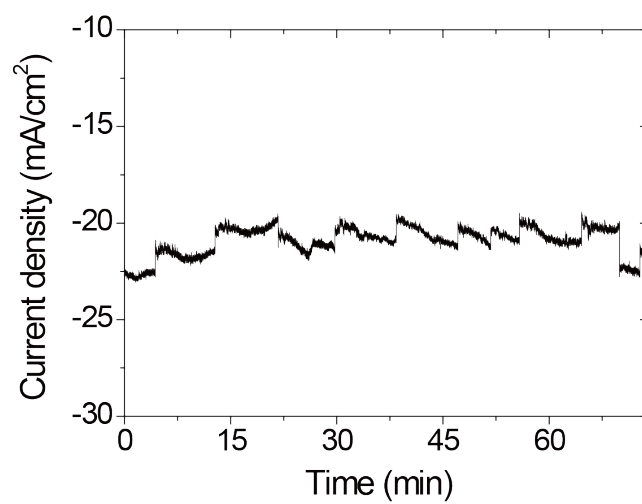
**Figure 2** XPS characterization of Mo (a) and S (b) signal in the  $\text{MoS}_x$  cocatalyst before and after annealing. The Mo 3d region shows the Mo oxidation state is 4+ before and after annealing while the S 2p region shows a transition from  $\text{MoS}_3$  to  $\text{MoS}_2$ .



**Figure 3** (a-b) Electrochemical performance of MoS<sub>2</sub> catalysts in 0.5 M H<sub>2</sub>SO<sub>4</sub>. (a) Linear sweep voltammograms and (b) Tafel plots for TiO<sub>2</sub>-coated *p*<sup>+</sup>-Si planar surface with different MoS<sub>2</sub> loadings. (c) 45° tilt and (d) cross-sectional SEM images of as-grown Si NW arrays. (e) Comparison of HER performance (under 100 mW/cm<sup>2</sup> simulated AM 1.5G irradiation) of Si NW array photocathodes coated with MoS<sub>2</sub> to an MoS<sub>2</sub> coated planar Si substrate, and a Si NW array loaded with Pt nanoparticles. (f) Spectral response [incident photon to current conversion efficiency (IPCE)] of Si NW arrays after coating with MoS<sub>2</sub> layers.



**Figure 4** Stability test of MoS<sub>2</sub> wrapped Si NW array. Current vs. time during the controlled potential photoelectrolysis under 100 mW/cm<sup>2</sup> simulated AM 1.5G irradiation at -0.33 V vs. RHE.



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