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Hierarchical Design Enables Sufficient Activated CO2 for Efficient Electrolysis of Bicarbonate to CO

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SUMMARY

Bicarbonate electrolyzers (BCEs) offer a promising approach to reducing the energy cost of $CO₂$ reduction by integrating upstream carbon capture and downstream electrochemical utilization. However, the faradaic efficiency of $CO₂$ electrolysis in BCEs has been limited by insufficient activated CO2 on the catalyst surface. We report a hierarchical design strategy combining molecular and systemlevel innovations to ensure sufficient activated $CO₂$ on the catalyst in BCEs. At the molecular scale, we introduce a single-atom catalyst $CoPc@CNT$ with strong $CO₂$ adsorption to prevent $CO₂$ desorption from the catalyst surface. Systemically, a cathodic electrolyte cross-flow strategy further enhances CO2 mass transfer. This approach achieves a faradaic efficiency exceeding 96.2% for CO at 50-300 mA cm⁻², with a 36.0% and 35.3% reduction in overall energy cost compared to conventional BCEs and $CO₂$ gas-fed electrolyzers, respectively. This innovative strategy represents a significant advancement of low-energy consumption exhaust conversion technologies for carbon neutrality.

INTRODUCTION

Electrochemical CO₂ reduction reaction $(CO_2RR)^1$ typically requires high-purity and high-pressure gaseous CO_2 as the raw material whose attainment includes energy-intensive steps such as CO_2 release and compression (**Fig. S1**).²⁻⁴ A promising alternative is directly using bicarbonate electrolyzers (BCEs) to integrate CO_2 capture and electrochemical reduction and to avoid the energy-intensive steps.^{2,3,5-7}

According to the techno-economic analysis, the commercialization of the $CO₂$ electrolysis system needs to ensure the yield and economics of products, requiring a current density greater than 200 mA $cm⁻²$ and a FE of approximately 90% at operating voltages lower than 3.0 V.⁸⁻¹⁰ In pursuit of these targets for BCEs, significant efforts^{11,12} have been devoted to improving the faradaic efficiency of CO (FE $_{\text{CO}}$) at industrially relevant current densities. Various innovative approaches to promote CO_2 mass transfer have been applied such as screening CO_2 -capture medium, $13-18$ optimizing electrode hydrophobicity,^{3,19,20}, and changing the fluid flow pattern.³ Despite these advancements, which have elevated FE $_{\text{CO}}$ from 20% to 60% at 200 mA cm⁻², BCEs have yet to fulfill the criteria for industrial applicability.8-10

Achieving a significant enhancement in faradaic efficiency necessitates a detailed mechanistic insight into the primary limitations. In the context of BCEs for $CO₂RR$, the supply of $CO₂$ is markedly lower than that in traditional gas-fed electrolyzers (**Fig. S2**). This difference arises because, in BCEs, $CO₂$ generation hinges on the reaction between bicarbonate ions (HCO₃⁻) and protons (H⁺) within the bulk electrolyte (Eq. S1 and Eq. S2).²¹ Subsequently, the insufficient CO₂ is transferred to the catalyst surface, where its adsorption and activation occur, facilitating the following conversion process (**Fig. S3**).²²⁻²⁴ Up until now, efforts to boost the faradaic efficiency in BCEs have primarily focused on improving CO2 mass transfer rates. Yet, there has been rare attention paid to investigating how catalystrelated mechanisms can influence this process. This gap highlights the critical need for further exploration into how enhancing catalyst efficiency and selectivity could significantly impact the overall performance of BCEs.

One of the greatest challenges impeding the faradaic efficiency of $CO₂RR$ in BCEs is to achieve an efficient CO_2 activation process, especially at low CO_2 concentrations. Enhancing the binding ability of the catalyst site towards CO_2 could avoid CO_2 desorption from the catalyst surface and inhibit the hydrogen evolution reaction (HER), which is of great significance for improving faradaic efficiency.²⁵

Currently, although the bulk Ag catalyst is widely used in $CO₂$ gas-fed electrolyzers with FE_{CO} exceeding 90%,^{8,26} it exhibits significantly diminished efficiency in BCEs due to their inadequate $CO₂$ adsorption strength, particularly evident at low $CO₂$ concentrations where FE $_{CO}$ drops below 60% at 200 mA cm-2 under ambient conditions (**Fig. 1a**). 3,21 This stark contrast underscores the need for catalyst optimization in BCEs to address the unique challenges posed by reduced $CO₂$ availability.

Therefore, there is a need for a strategy that tailors the adsorption energy of the catalyst for $CO₂$ to achieve highly efficient conversion in direct capture medium electrolysis. Additionally, the limited $CO₂$ mass transfer leads to less collision between $CO₂$ and the catalyst surface, also resulting in insufficient activated $CO₂$ on the catalyst surface.²⁷ In conventional BCEs, OH⁻ is generated at the cathode along with the CO2RR process (**Eq. S1-S3**). Driven by electric field forces, the OH- aggregates near the surface of the cathodic catalyst act as a CO₂-consuming layer that barriers CO₂ release and transport to the catalyst surface (**Eq. S4 and Eq. S5**) (**Fig. 1a**). Therefore, to improve the faradaic efficiency of $CO₂$ conversion in BCEs, strategies should be adopted to enhance the catalyst's binding ability to CO_2 and promote CO_2 mass transfer, ultimately increasing the amount of activated CO_2 on the catalyst surface.

Herein, we report a hierarchical strategy integrating molecular-level and system-level designs to ensure sufficient activated CO₂ on the catalyst for the BCEs system (Fig. 1b). On the molecular level, the single-atom catalyst (CoPc@CNT) with strong $CO₂$ binding ability promotes $CO₂$ activation and conversion. On the system level, our BCEs device adopts a cathodic electrolyte cross-flow strategy. This design channels HCO₃ directly to the membrane surface through two distinct flow paths to generate CO₂, significantly enhancing mass transfer compared to conventional BCEs reliant on HCO₃ diffusion. Meanwhile, the cathodic electrolyte vertically passes across the cathodic electrode which not only eliminates the CO_2 -consuming layer on the catalyst surface but further promotes CO_2 mass transfer as well. This innovative BCEs system achieves an impressive FE_{CO} of over 96.2% from 50 to 300 mA cm⁻². Remarkably, the significantly improved FE_{CO} makes BCEs highly competitive in terms of CO2 conversion overall energy consumption, 36.0% and 35.3% lower compared to conventional BCEs and CO_2 gas-fed electrolyzers,^{2,14,18} respectively.

RESULTS

Figure 1: Hierarchical design enables sufficient activated CO2 for efficient electrolysis of CO2 to CO in BCEs.

a) In conventional BCEs, conventional Ag catalysts with weak adsorption capability to CO₂ cause the insufficient activation of $CO₂$. At the same time the OH- covers the catalyst surface and forms a $CO₂$ consuming layer, which impedes the CO2 mass transfer to the catalyst. **b**) In hierarchically designed BCEs, a single-atom catalyst $CoPc@CNT$ with strong $CO₂$ adsorption capability is introduced to enhance the catalyst binding ability to $CO₂$ and to facilitate $CO₂$ activation and conversion. Meanwhile, a strategy of cathodic electrolyte cross-flow is implemented to promote HCO₃ mass transfer to eliminate the CO2-consuming layer that developed on the cathodic catalyst surface to further promote $CO₂$ mass transfer.

Figure 2: Enhanced CO2 adsorption by a single-atom-catalyst strategy (CoPc@CNT) enables excellent catalytic performance in low CO₂ concentrations environment.

a) Comparative analysis of CO_2 adsorption energies between $CoPe@CNT$, Ag, and Cu, as a function of potential under different CO2 partial pressures. **b**) Investigation of the adsorption energy of CO2 on CoPc@CNT, considering CO2 partial pressure and electrode potential. **c**) Co K-edge EXAFS spectra of CoPc@CNT and Co Foil. **d**) Wavelet transform analysis of EXAFS spectra of CoPc@CNT **e**) Evaluation of the performance of CoPc@CNT catalysts without CoPc clusters and CoPc+CNT (two

powders physically mixed in a 1 to 1 ratio by mass without a single-atom dispersion process) catalysts with CoPc clusters at low CO₂ concentrations in a CO₂ gas-fed flow cell. **f**) CO₂RR performance comparison between CoPc@CNT, Ag, and Au at various $CO₂$ concentrations in a $CO₂$ gas-fed flow cell, maintaining a constant current density of 100 mA cm^{-2} .

Molecular Level Strategy for Enhancing CO2 Adsorption

Catalysts with stronger interactions with $CO₂$ facilitate $CO₂$ activation and thereby improve the electrocatalytic CO₂RR activity even at low CO₂ concentrations.²⁸ At present, Ag-based catalysts are usually used as cathode catalysts in BCEs.^{3,19,21} However, the low CO_2 affinity on Ag catalysts limits the CO₂RR activity. Transition-metal single-atom catalysts (SACs) generally show stronger $CO₂$ adsorption compared to bulk catalysts.^{27,29} As the transition metal atoms become atomically isolated, there is a concomitant upward shift of the d-band center. This shift results in a decreased density of electrons in the antibonding state, which stabilizes the intermediates. ³⁰ Here, carbon nanotube-loaded cobalt phthalocyanine (CoPc@CNT) (**Fig. S4**) is chosen as a representative single-atom catalyst, which may possess stronger CO₂ adsorption capability. Although these single-atom catalysts have been demonstrated with excellent performance in CO_2 gas-fed electrolyzers, $31-35$ their research in BCEs and the relationship between $CO₂$ adsorption capability and the BCEs performance are still blank.

We first performed grand canonical density functional theory (GCDFT) calculations to compare the potential dependent CO_2 adsorption free energy on the $CoPe@CNT$ catalyst and commonly used Ag and Cu catalyst in $CO₂RR$. Under a $CO₂$ pressure of 1 atm, it is clear that the adsorption energy of *CO₂ on CoPc@CNT is considerably lower, than that on Ag or Cu, and the difference is more than 0.5 eV in a wide range of potential. It indicates stronger adsorption of $CO₂$ by $CoPc@CNT$ when $CO₂$ is under high concentrations (**Fig. 2a**). The adsorption energy of all three catalysts, CoPc@CNT, Ag, and Cu, increases as the $CO₂$ concentration decreases (from 1 atm to 0.1 atm, further to 0.01 atm). Notably, under low CO_2 partial pressures (0.01 atm, 0.001 atm), the adsorption of *CO_2 on Ag proves to be exceptionally difficult, with reaction energy exceeding 0.75 eV on Ag (111) even at substantially negative potentials. This suggests that $CO₂RR$ through bicarbonate reduction will be particularly challenging to achieve. While the *CO_2 adsorption on CoPc@CNT at low CO₂ pressure is still considerably stronger than that on Ag at 1 atm, demonstrating the superior $CO₂$ adsorption capability

of CoPc@CNT. In addition, although $CO₂$ can be significantly stabilized at the CoPc site, the stabilization effect for CO is not as pronounced. This difference is attributed to the unique ability of the single atom site to stabilize intermediates with large dipole moments as bent $CO₂$. In contrast, *COOH and *CO exhibit much smaller dipole moments, making them less responsive to single-atom site stabilization, a phenomenon detailed in Chan's study.³⁶ Consequently, the adsorption energy of CO on CoPc is relatively weak, facilitating its easy desorption, as shown in **Fig. S5**.

As illustrated in Fig. 2b, The *CO_2 adsorption free energy on $CoPe@CNT$ is highly dependent on the electrode potential and CO_2 pressure. With the applied voltage being more negative and the CO_2 partial pressure increasing, there is a general enhancement in adsorption capability. This phenomenon can be partially attributed to the substantial dipole moment of bent *CO_2 , which renders it highly responsive to changes in the interfacial electric field, thereby elucidating the strengthened adsorption observed under negative potential conditions. Furthermore, it is noteworthy that pressure exerts a profound influence on the translational entropy of gaseous $CO₂$. The adsorption process imposes constraints on both the translational and rotational components of the gas's entropy, contributing to the observed enhancement in adsorption energy as pressure undergoes variations. In addition, our previous study25 showed that the Co single-atom site exhibits a much more energy-localized 3*d* state compared to the dispersed *d* band states for Ag, which is believed to strengthen the interaction between the adsorbate and the metal according to the Newns–Anderson–Grimley model. These DFT findings indicate that CoPc has a strong adsorption capability at low $CO₂$ concentrations, which makes it an excellent candidate for bicarbonate conversion in BCEs.

The CoPc@CNT was synthesized and characterized to guarantee uniform distribution of Co singleatomic sites on the CNT. A facile method was utilized by mixing the CNT into nearly saturated CoPc solutions with the assistance of magnetic stirring (**Fig. S6**). The X-ray absorption near-edge structure (XANES) spectra (**Fig. S7**) and extended X-ray absorption fine structure (EXAFS) spectra (**Fig. 2c**) show that the oxidation state of the Co atoms in hybrids was approximately $+2$ and no Co-Co coordination is found, suggesting that CoPc was uniformly anchored onto the CNTs surface at the single molecular level. Comparing wavelet transform analysis of EXAFS spectra of CoPc@CNT with that of CoO or Co foil, only Co-N and Co-N-C bonds are present in CoPc@CNT, which indicates a mono-dispersion of CoPc at CNTs (**Fig. 2d and Fig. S8**). Scanning electron microscopy (SEM) images also indicate that no CoPc clusters are present in CoPc@CNT, whereas rod-like CoPc clusters are observed in CoPc+CNT (**Fig. S9**). TEM images and corresponding EDS elemental maps of CoPc@CNT indicate N and Co elements' uniform dispersion at CNTs (**Fig. S10**). XRD patterns and Raman spectra are further used to characterize the CoPc@CNT and CoPc+CNT (**Fig. S11**), and it shows similar signature vibrational peaks and crystal structure of CoPc+CNT and CoPc. It is noted that some of the CoPc vibrational features are not observed with CoPc@CNT, suggesting strong CoPc-CNT electronic interactions that prohibit some of the vibrational modes of the CoPc molecules on CNT. After 3600 s of reaction in cross-flow BCEs at a flow rate of 240 ml min⁻¹ and a current density of 100 mA cm⁻², the XRD pattern as well as Raman spectra of $CoPe@CNT$ show that the reaction process did not change the distribution state of CoPc on CNT.

To verify the high CO_2 adsorption capability of $CoPe@CNT$ at low CO_2 concentration, and to demonstrate the necessity of single-atom dispersion, we tested the $CO₂RR$ electrochemical performance of $CoPc@CNT$ in a $CO₂$ gas-fed flow cell under varying $CO₂$ partial pressure mixing CO2 with Ar in a different ratio and compared it with CoPc+CNT and bulk Ag catalyst. At a 100% $CO₂$ concentration, both the CoPc@CNT and CoPc+CNT show a similar FE_{CO} over 90% from 10 to 50 mA cm⁻² (**Fig. 2e**). However, there is a significant difference in their performance when the $CO₂$ concentration decreases to 20%. The CoPc@CNT maintains a high FE $_{CO}$ of more than 96% until 50 mA cm[−]² at the low CO2 concentration, under which the CoPc+CNT only achieves roughly 20%. The difference trend becomes evident when the $CO₂$ concentration is turned to 10%, at which $CO₂$ mass transport plays a more critical role in CO2RR. Meanwhile, for the single-atom catalyst CoPc@CNT, the partial current densities of CO remain constant despite a 40% to 10% decrease in CO2 concentration (**Fig. S10**). The CoPc@CNT was subjected to a constant current density test of 100 $mA \text{ cm}^{-2}$ at different CO_2 concentrations and was compared with bulk Ag and Au catalysts which exhibit excellent CO_2RR performance in CO_2 gas-fed electrolyzers. It turns out that the FE $_{CO}$ of the $CoPe@CNT$ catalyst was maintained at close to 100% when the $CO₂$ concentration was higher than 10%. In contrast, the bulk Ag catalyst exhibits a significant decline in FE_{CO} as $CO₂$ concentration drops below 50%, while the bulk Au catalyst' $CO₂RR$ undergoes continuous decay with decreasing $CO₂$ concentration **(Fig. 2f)**. Meanwhile, the bulk Ag catalyst exhibits a significant decline in CO partial

current densities as the CO_2 concentration decreases, reaching a 50% reduction at -1.7 V with 10% CO2 **(Fig. S12)**. Therefore, it is confirmed that different from the CoPc+CNT with aggregated CoPc clusters and Ag catalyst with the active sites densely packed on the surface of nanoparticles, $CoPe@CNT$ has uniformly dispersed active sites across the whole carbon matrix, whose stronger $CO₂$ binding ability enables a higher proportion of $CO₂$ to exist in the activated form, especially when the CO2 mass transfer is limited.

Figure 3: Performance of CoPc@CNT catalyst in conventional BCEs device.

a) Structure of conventional BCEs devices. The cathode catalyst tightly adheres to the proton exchange membrane, with OH- localized on the catalyst surface, impeding the mass transfer of CO2. **b**) Partial current density and FE_{CO} of CoPc@CNT at various cell voltages in conventional BCEs devices. **c**) Comparative performance of the FE_{CO} between $CoPe(\omega CNT)$, $CoPe+CNT$, Ag, and Au catalysts in conventional BCEs devices at different current densities. The right coordinates indicate the Ecell of CoPc@CNT at different current densities in conventional BCEs. **d**) Comparison of the FE_{CO} for

CoPc@CNT in conventional BCEs at different current densities with other Ag-based catalysts reported in the literature under room temperature and standard atmospheric pressure.

Molecular Level Strategy for Application in BCEs

To further evaluate the performance of the single-atom catalyst CoPc@CNT in BCEs and confirm the importance of its high $CO₂$ adsorption capability, we incorporated it into a conventional BCEs device. In this setup, the cathode catalyst layer directly interfaces with the ion exchange membrane, and the bicarbonate solution flows over the back of the porous electrode, where it combines with protons through diffusion to the catalyst surface of the catalyst, resulting in the production of CO2 (**Fig. 3a**). The CoPc@CNT, prepared with a catalyst loading capacity of 0.6 mg cm⁻², was coated onto the porous carbon cloth substrate (**Table S4**). The porous carbon cloth exhibits significant porosity and hydrophilicity (**Fig. S13**), promoting efficient electrolyte flow through the catalyst layer during the reaction. The partial current density of CO experiences a notable increase as the voltage rises from - 2.7 V to -3.0 V, reaching a peak of 153 mA cm⁻² (**Fig. 3b**). Concurrently, the FE_{CO} significantly decreases with more negative voltages, attributed to limited $CO₂$ mass transfer at higher current densities. Comparison between the performance of CoPc@CNT with CoPc+CNT catalysts containing CoPc clusters, bulk Ag, and Au catalysts reveals interesting insights. At a low current density of 50 mA cm⁻², both CoPc@CNT and CoPc+CNT exhibit similar FE_{CO} values exceeding 85% (**Fig. 3c**). However, a substantial disparity emerges as the current density surpasses 100 mA cm⁻². At 200 mA cm⁻², CoPc@CNT exhibits a remarkable FE_{CO} of 69%, whereas CoPc+CNT only registers 13%. Further amplifying the current density to 300 mA cm⁻², CoPc@CNT maintains a FE_{CO} of 51%, while CoPc+CNT plummets to 3%, dominated by the HER. This trend aligns with the results observed in CO2 gas-fed flow cells with varying CO2 concentrations (**Fig. 2e**), where CoPc@CNT consistently demonstrates high FE $_{CO}$ across different $CO₂$ concentrations, while $CoPc+CNT$ exhibits low FE $_{CO}$ at lower $CO₂$ concentrations. In contrast, for bulk Ag and Au catalysts, the FE_{CO} remains lower even at low current densities due to the weak adsorption capability of CO₂ and decreases as the current densities increase, with the FE_{CO} being below 5% at 300 mA cm⁻² as well. These findings indicate that the single-atom dispersed catalyst $CoPc@CNT$ effectively enhances $CO₂$ adsorption and activation in BCEs, owing to its superior $CO₂$ adsorption capability. Additionally, a performance comparison

between CoPc@CNT and bulk Ag catalysts reported in the literature reveals CoPc@CNT superior FE_{CO} across the range of 50 to 300 mA cm⁻² (**Fig. 3d**). While the molecular-level strategy to enhance $CO₂$ adsorption capability proves effective, achieving satisfactory FE_{CO} under high current density requires further system optimization.

Understanding the limitations in $CO₂$ mass transfer within conventional BCEs devices is crucial for addressing the issue of low faradaic efficiency. At elevated current densities, increased H^+ production from the anode migrates to the cathode. While a surplus of H^+ has the potential to release more CO_2 by reacting with the KHCO₃ solution, the inadequate supply of KHCO₃ solution from the rear of the catalyst layer results in an accumulation of excess H^+ near the cathode catalyst. This accumulation may inadvertently promote the hydrogen evolution reaction (HER), leading to a reduction in FE_{CO} . Furthermore, under high current density conditions, the increased production of OH- accumulates on the catalyst surface, forming a CO_2 -consuming layer. This layer impedes the mass transfer of CO_2 to the catalyst surface, further diminishing FE_{CO} . Therefore, achieving high FE_{CO} at high current density requires further optimization of the device structure to promote efficient $CO₂$ mass transfer. This optimization should enhance CO_2 release by fully utilizing H^+ while simultaneously mitigating the presence of OH^- to prevent its consumption of $CO₂$.

Figure 4: The cross-flow strategy for eliminating the CO2-consuming layer on the catalyst surface and CO2RR performance of CoPc@CNT catalyst in the hierarchical design BCEs.

a) The diagram of cathodic electrolytes cross-flow strategy. **b**) COMSOL simulation illustrating the localized pH distribution around the catalyst for different flow rates. Shaded regions indicate the location of the catalyst layer. The value of 0 on the X-axis represents the geometric center spatial position of the catalyst layer. A positive value represents proximity to the anode layer, while a negative value represents proximity to the cathode. c), **d**) Partial current density of CO and H₂, and the FE_{CO} as a function of applied electrode potential. **c**) showcase a 90 ml min-1 flow rate, while **d**) depict a 240 ml min-1 flow rate. **e**) Comparison of performance between various flow plate geometries in conventional BCEs and our hierarchical design BCEs. **f**) Comparison of FE_{CO} achieved in this work with reported literature at varied current densities under room temperature and standard atmospheric pressure. **g**) Stability test of CoPc@CNT catalysts in our hierarchical design BCEs: the catalysts

demonstrate stability for more than 13 hours at a current of 100 mA cm⁻² and a flow rate of 150 ml min^{-1} .

System Level Strategy to Further Promote CO2 Mass Transfer

To eliminate the CO_2 -consuming layer formed by OH $\overline{}$ on the catalyst surface and enhance CO_2 mass transfer dynamics, the configuration of the cell is crucial. In this regard, a cell configuration has been designed that incorporates a cathodic electrolyte cross-flow between the catalyst layer and the cation exchange membrane (**Fig. 4a**) (**Fig. S14 and Fig. S15**) This cross-flow strategy allows the sufficient supply of HCO₃ through two special channels directly to the membrane surface, where it combines with $H⁺$ to enhance $CO₂$ mass transfer. The electrolyte in the cathode chamber then flows out after passing across the cathode electrodes vertically. This cross-flow strategy not only enhances the supply of HCO₃⁻ and CO₂ but also prevents neutralization by the OH⁻ generated from the cathodic catalyst (**Eq. S4**), allowing for the optimal utilization of H^+ to release ample CO_2 in situ (**Eq. S2**). Furthermore, the electrolyte flow in the direction from the membrane to the cathode serves the dual purpose of removing OH⁻ formed on the catalyst surface and transporting CO₂ from the electrolyte bulk to the catalyst surface.

To investigate the impact of the electrolyte cross-flow on CO₂ mass transfer, we constructed a device model and performed COMSOL simulations to calculate the pH around the catalyst layer. As depicted in **Fig. 4b**, the pH value gradually increases from the anode side of the catalyst layer to the back side, confirming that the electrolyte cross flow with the direction from anode to cathode plays an important role in driving OH- transfer away from the catalyst layer. In the absence of the electrolyte cross flow, the pH on the catalyst surface is as high as 10.7. As the flow rate increases to 240 ml min⁻¹, the pH on the catalyst surface decreases to 8.42, which represents an order of magnitude decrease in the concentration of OH⁻. Accordingly, based on the dissolution equilibrium of CO₂ in bicarbonate (Eq. S6 and Eq. S7) (Fig. S16), theoretical results demonstrated that the CO₂ partial pressure can be significantly increased from 0.11 to 1.14 atm by enhancing the electrolyte cross-flow rate, ensuring sufficient CO₂ supply on the catalyst surface (**Fig. S17**).

To validate the proposed electrolyte cross-flow mechanism, we conducted measurements of the pH

in the single-pass cathodic tailings under various flow rates (**Fig. S18**). To mitigate the buffering effect of bicarbonate that could potentially affect pH accuracy, an alternative K_2SO_4 solution was employed as the electrolyte. Simultaneously, CoPc@CNT served as the cathode catalyst to generate OH⁻ on the catalyst surface through the hydrogen evolution reaction (HER). The total amount of OH- remained consistent, regulated by adjusting the current density and reaction time. As depicted in **Fig. S18**, based on the calculated total amount of OH- and the measured results in the tail bulk solution, the residual OH on the catalyst surface gradually decreased by approximately 50% (from 2.3 µmol to 1.2 µmol) as the intensity of the flow rate increased from 60 to 180 ml min⁻¹. This observation demonstrates that the cross-flow strategy effectively removes OH- from the catalyst surface. It further indicates that the cross-flow strategy eliminates the CO_2 -consuming layer and enhances CO_2 mass transfer to the catalyst surface in the BCEs system, aligning with the COMSOL simulation results (**Fig. 4b**).

Building upon the hierarchical strategy outlined, which includes the design of a molecular-level strong CO2 absorption catalyst and a system-level cross-flow cell, we meticulously evaluated the $CO₂RR$ performance in the new BCEs device. At a low flow rate of 90 ml min⁻¹, the partial current density of CO markedly increases as the voltage rises from -2.7 V to -3.0 V, reaching a maximum of 150 mA cm⁻² and then slightly decreases. Simultaneously, the FE_{CO} significantly decreases when the voltage is higher than exceeded -3.0 V due to the limited $CO₂$ mass transfer at high current densities (**Fig. 4c**), which is consistent with the test results in conventional BCEs devices (**Fig. 3b**). Upon increasing the electrolyte flow rate to 240 ml min^{-1} , the partial current densities of CO show a consistent rise with increasing voltage. Especially, the FE_{CO} consistently maintains higher than 96.2% at various voltages (Fig. 4d). With increasing flow rates, both the partial current densities and FE_{CO} exhibit improvements, as depicted in **Fig. S19 and Fig. S20**. This trend suggests that the cross-flow of electrolytes effectively eliminates the $CO₂$ -consuming layer on the catalyst surface, facilitating enhanced CO₂ supply.

However, this increased flow rate and subsequent performance enhancement are only effective in our hierarchical-design BCEs, where the mutual coupling of CoPc@CNT and cross-flow strategies is essential. It is found that when $CoPc@CNT$ was used in conventional BCEs, increasing the flow rate did not improve performance, due to the insufficient CO2 supply and OH- aggregation on the catalyst surface at high current densities (**Fig. S21**). Comparing the performance of bulk Ag and Au catalysts, or other single-atom catalysts such as $NiPc@CNT^{37}$ and $FePc@CNT^{37,38}$ in conventional BCEs and cross-flow BCEs, respectively, we found that at the same flow rate, cross-flow strategy did not improve FE_{co} (Fig. S22), which indicates that it is effective only with a catalyst with relatively strong CO₂ adsorption capability, but not excessive, like CoPc@CNT. These comparisons further highlight the innovation of our hierarchical design.

To confirm the effectiveness of our cross-flow BCEs device design, we compared its catalytic performance with those of conventional BCEs device structures featuring serpentine flow plates and optimized interdigitated flow plates (**Fig. 4e**). As expected, our cross-flow BCEs device delivers a 96.2% FE_{co} under 300 mA cm⁻², which is significantly higher than the other two, demonstrating the advantage of our cross-flow BCEs device. Our device achieves the highest FE_{CO} over a wide current range compared with the state-of-the-art BCEs demonstration under room temperature and standard atmospheric pressure (**Fig. 4f**). The hierarchical design BCEs was further tested for stability (**Fig. 4g**), and the FE_{CO} was stable for more than 13 hours at a current density of 100 mA cm⁻² and a flow rate of 150 ml min⁻¹. XRD and Raman spectra of the catalyst after the reaction showed that the CO₂RR in our hierarchical BCEs did not change the distribution of the catalyst molecules themselves (**Fig. S11**). The decreased stability of the system stemmed from the detachment of the catalyst from the carbon cloth caused by the flow. An effective approach to further improve device stability in future work involves growing the catalyst on the substrate, which prevents the catalyst from being washed away by the highrate electrolyte flow.

Figure 5: Practical application demonstration of our hierarchical design BCEs.

a) Schematic illustration of CO_2 capture and conversion cycle in BCEs system. **b**) CO_2 capture rate using K2CO3 solution for different simulated exhaust gas compositions. **c**) Photograph and XRD analysis of CO_2 capture production by K_2CO_3 solution. **d**) Performance of KHCO₃ formed after the capture of CO2 from different simulated exhaust gas in our hierarchical design BCEs. **e**) Energy consumption comparison between our hierarchical design BCEs and other reported BCEs systems. Energy calculation includes $CO₂$ capture, compression, and reduction processes. $CO₂$ conversion efficiency is 50% in $CO₂$ gas-fed cells and 100% in BCEs.

Practical Application Demonstration

To achieve carbon neutrality in authentic industrial production, it is crucial to design a comprehensive system that integrates $CO₂$ capture and electrochemical conversion. We have developed a novel, sustainable carbon capture and conversion system based on our hierarchical design BCEs which is suitable to use in industrial exhaust gas with approximately 14% CO₂, as illustrated in **Fig. 5a**. Due to issues such as the low solubility of CO_2 in K_2CO_3 solution and slow reaction rates, the absorption rate of $CO₂$ by $K₂CO₃$ solution is relatively slow at room temperature, especially for low concentrations of $CO₂$ in exhaust gases. The addition of DEA as a promoter can effectively increase

the rate of CO_2 absorption (Eq. $S8-S12$).³⁹⁻⁴¹ According to proportion based on literature reports and industrial production, $3.1M K₂CO₃$ solution containing the 2.5 wt% DEA promoter was employed to capture and convert CO_2 from simulated industrial exhaust gas.³⁹⁻⁴¹ After electrolysis, KHCO₃ was converted to K_2CO_3 which can be used for subsequent CO_2 capture.^{13,39,42}

To verify the feasibility of our hierarchical design BCEs linked carbon capture and conversion, we conducted experiments using pure $CO₂$ and other two simulated exhaust gases which contain a mixture of oxygen and nitrogen and different concentrations of CO2. **Fig. 5b** illustrates that the CO2 absorption rate K_2CO_3 solution has a positive correlation to the CO_2 concentration. With DEA introduced, the rate of $CO₂$ absorption has been accelerated.⁴³ However, the DEA in BCEs devices poses a challenge to $CO₂RR$ due to its rapid binding to $CO₂$, resulting in a decline in FE_{CO} (**Fig. S23**). Consequently, we chose to separate $KHCO₃$ from the DEA-containing $CO₂$ capture solution before it entered the BCEs device. After introducing the simulated exhaust gas into the K_2CO_3 solution for 90 minutes, the initially clear absorption solution gradually became cloudy, accompanied by the formation of white precipitates as depicted in **Fig. 5c**. These white solids were subsequently filtered and characterized via XRD analysis, confirming the generation of KHCO₃. Elemental analysis and titration results confirm that the KHCO₃ solid produced by capturing CO_2 with K_2CO_3 is of high purity, exceeding 99.9%, and can be directly used for the subsequent conversion in BCEs (**Table S7** and **Table S8**). The KHCO₃ solids produced from the carbon capture of different $CO₂$ concentrations nearly have consistent $CO₂RR$ performance in our hierarchical design BCEs (**Fig. 5d**). This demonstrates that our BCEs system has encouraging applicability of different actual exhaust gases with different $CO₂$ concentrations.

Developing a system that minimizes energy requirements for a fully integrated CO₂ capture and conversion process is challenging. By optimizing the device structure and reducing the thickness of the HCO₃ flow plate and proton exchange membrane (Fig. S24), we achieve energy efficiencies of 35% at 100 mA cm-2 and 52% at 10 mA cm-2 in BCEs (**Fig. S25**). We counted the total energy cost in our hierarchical design BCEs and compared it to conventional BCEs and conventional gas-fed cells. Due to the significant improvement in FE_{CO} and energy efficiency of our BCEs, energy consumption has been drastically reduced by 36.0% compared to conventional BCEs. Furthermore, Our BCEs system averts the energy-intensive steps such as $CO₂$ release and compression, resulting in a 35.3%

energy consumption reduction compared to CO₂ gas-fed electrolyzers (**Fig. 5e, Table. S2 and Table. S3**). Therefore, our hierarchically designed BCEs present attractive prospects for practical industrial applications.

DISCUSSION

In summary, our study demonstrates the efficient electrolysis of bicarbonate into CO with faradaic efficiencies reaching 96.2% across the range of 50 to 300 mA cm⁻². This success is attributed to a hierarchical design incorporating a single-atom catalyst and a cross-flow cell. Theoretical calculations and electrochemical studies reveal that the single-atom dispersed CoPc@CNT catalyst and electrolyte cross-flow synergistically enhance binding energy interactions with $CO₂$ and promote $CO₂$ mass transfer, ensuring an sufficient supply of activated $CO₂$ for electrochemical reduction. Leveraging the unprecedented FE_{CO}, our hierarchical design BCEs offer significant advantages, including reduced energy consumption, within the integrated $CO₂$ capture and conversion system. Consequently, this marks a tangible advancement in the practical application of BCEs for efficiently capturing and converting exhaust gas into valuable products.

EXPERIMENTAL PROCEDURES

Resource availability

Lead contact

Further information and requests for resources should be directed to and will be fulfilled by the lead contact,

Jia Zhu (jiazhu@nju.edu.cn).

Materials availability

This study did not generate new unique reagents.

Data and code *availability*

The datasets generated in this study are available from the lead contact on reasonable request.

Materials

All the Chemicals were purchased from commercial sources and used without further purification unless otherwise noted. Ethanol (AR), Nafion solution (5 wt%), Nafion 115 membrane, KHCO₃ (AR), and H₂SO₄ (AR) were purchased from commercial suppliers. Ir@Ti mesh electrode (loading: 0.5 mg cm⁻², Kunshan Shengmolong Electrolytic Equipment) was purchased from Kunshan Shengmolong Electrolytic Equipment. Milli-O water of 18.2 M Ω cm⁻¹ was used in all experiments.

Synthesis of CoPc@CNT Single-atom Catalysts

50 mg of CoPc molecules were dispersed in 100 ml of N-N dimethylformamide (DMF) with the assistance of stirring. After removing the insoluble particles from the solution by filtering, a uniformly saturated CoPc solution was obtained. Then, 200 mg of CNTs was added to the above solution and stirred for 7 days to obtain a well-mixed suspension. Subsequently, the mixture was centrifuged and the precipitate was washed with DMF three times and ethanol twice. Finally, the precipitate was lyophilized to yield the final powder of CoPc@CNT.

Cathode Preparation and Electrochemical Measurements

The catalyst ink was prepared as follows. 2 mg of CoPc@CNT powder and 50 µl of 5 wt% Nafion solution were dispersed in 950 µl anhydrous ethanol assisted with ultrasonication for 60 min. Then the catalyst ink was dropped onto carbon cloth (loading 0.6 mg cm⁻²) as the working electrode. The electrodes in the flow cell were based on hydrophobic carbon paper (29BC) and other factors remained unchanged.

Performance Testing at Different CO2 Concentrations.

All the performance tests at different $CO₂$ concentrations of the catalyst were carried out with a three-electrode system in a flow cell. The saturated Ag/AgCl electrode was used as the reference electrode, while an Ir@Ti electrode was used as the counter electrode. The anodic and cathodic electrolytes were 0.5 M KHCO₃. All potentials were converted to the reversible hydrogen electrode (RHE) scale using:

$$
E(\text{potential}, \text{versus RHE}) = E(\text{versus Ag}/AgCl) \times 0.197V + 0.0591 \times pH
$$

Material Characterizations.

The microscopic structures are characterized by SEM (Tescan, MIRAS3 FE-SEM). Energy dispersive X-ray spectroscopy was performed on FEI Titan G2 60-300. Raman spectra of materials were taken with a homeassembled Raman spectrometer. X-ray diffraction patterns were recorded on D8 Discover (Bruker).

Electrochemical Measurement

Electrochemistry measurements were measured with a two-electrode liquid cross-flow cell system with commercial Ir@Ti as the anode and CoPc@CNT on carbon cloth as the cathode. The geometric areas of the anode and cathode were 4 cm2 and 1 cm2 respectively. Our hierarchical design BCEs were built in-house and consist of housing, gaskets, anode and cathode, an anode proton exchange membrane N115 (3 cm \times 3 cm), and a bicarbonate feed plate. The electrolyzer shells were constructed from acrylic sheets. 0.005 M H_2SO_4 (pH = 2) solution was employed as the anolyte for the oxygen evolution reaction and was delivered by a peristaltic pump at 50 mL min⁻¹. The catholyte solution (3.0 M K HCO_3) was circulated by a peristaltic pump at a constant flow rate ranging from 90 to 240 ml min⁻¹ into the flow-cell electrolyzer, which was then vented back into the flask. The fresh electrolyte was used for each set of experiments. The headspace of the cathode electrolyte reservoir was purged with Ar at 50 sccm throughout each experiment. The gaseous outlet of the catholyte bottle was vented into a gas chromatograph (GC). As references, a conventional cell and an interdigitated cell were also assembled. All the other procedures were the same as those for the cross-flow cell. All the measurements were performed at room temperature (about 25 °C) and ambient pressure, and three independent experiments were conducted.

Experiments on the Action of Cross-flow Strategy

Electrochemistry measurements were made with a two-electrode liquid cross-flow cell system with commercial Ir@Ti as the anode and CoPc@CNT on carbon cloth as the cathode. The geometric areas of the anode and cathode were 4 cm² and 1 cm² respectively. The hierarchical design BCEs were built in-house and consisted of housing, gaskets, anode and cathode, an anion exchange membrane FAB-130 (3 cm \times 3 cm), and a K₂SO₄ feed plate. The electrolyzer shells were constructed from acrylic sheets. Pure water was employed as the anolyte for the oxygen evolution reaction and was delivered by a peristaltic pump at 50 mL min⁻¹. The catholyte solution $(0.1$ M K₂SO₄) was circulated by a peristaltic pump at a constant flow rate ranging from 60 to 180 ml min⁻¹ into the hierarchical design BCEs, then into the tail-end bulk solution. Measurement of pH using a pH meter. The fresh electrolyte was used for each set of experiments. All the measurements were performed at room temperature (about 25 °C) and ambient pressure, and three independent experiments were conducted.

CO2 Capture Experiments by K2CO3 Solution.

3.1 M K₂CO₃ solution served as the medium for capturing CO_2 from the simulated exhaust gas (8%, 14%, and pure CO2), and diethanolamine (DEA) was introduced as a catalyst in the solution. Accompanied by carbon capture progress, $KHCO₃$ precipitates were formed and then separated from the DEA-containing solution through filtration. These precipitates were subsequently dissolved in water yielding a fresh 3.0 M KHCO_3 solution, which served as the feedstock for the hierarchical design BCEs. After undergoing electrochemical conversion in the hierarchical design BCEs, the regenerative K_2CO_3 solution was 1.5 M and would be concentrated by a photothermal evaporation process to obtain a 3.1 M K_2CO_3 solution, thereby completing the circulation.

DFT Calculation

DFT calculations were performed using the Vienna Ab-Initio Simulation Package (VASP)⁴⁴ with the RPBE functional. The core electrons were described with the projector augmented wave (PAW) method.⁴⁵ The convergence criteria for electronic and force minimization were set to 10^{-6} eV and 0.02 eV/Å for the structure optimization. The cutoff energy for the kinetic energy of the plane waves was 450 eV. The entropy corrections determined from frequency calculations were performed by using the harmonic oscillator approximation. ($3 \times$ 3)-4 layers (111) surfaces were chosen for modeling Ag, where the Brillouin zone was sampled using the $(4 \times$ 4×1) Gamma-centered k-point grid. For CoPc@CNT, we established a graphene-supported CoPc molecular catalyst model. Only the Γ k-point is sampled.

Grand Canonical DFT calculations (GCDFT) were performed to obtain the potential-dependent energetics, where the number of electrons is allowed to change to adapt to the change of work function along the reaction pathway. The potential-dependent grand canonical free energy can be expressed by a surface charging model:

$$
\Omega(U) = \Omega(U) - q(U) \cdot FU = \Omega(U_0) - \frac{1}{2}C(U - U_0)^2
$$

where $\Omega(U)$ is the electronic energy of the surface at the U potential in SHE scales, q(U) is the charge difference against the neutral condition, and F is the Faradaic constant. C stands for the effective capacitance and U_0 represents the potential of zero charge (pzc). For more details on GCDFT, see our previous work.⁴⁶⁻⁵⁰ The selfconsistent implicit solvation model VASPsol34 is used to represent the polarizable electrolyte region. The dielectric constant 78.3, and the Debye screening length 3 Å, are used, as they correspond to a bulk ion concentration of 1 M.

SUPPLEMENTAL INFORMATION

Supplemental information can be found online at

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AUTHOR CONTRIBUTIONS

J.Z. and X.W. conceived the project and designed the experiments. M.S., L.J., Q.X., S.F., S.C., Z.W., and J.W. carried out the synthesis, material characterizations, and electrocatalytic measurements. D.C. and P.S. carried out and analyzed the DFT calculation. Y. L. carried out the COMSOL simulation. M.S. and L.J. analyzed the data and wrote the manuscript. All authors discussed the results and commented on the manuscript.

DECLARATION INTERESTS

The authors declare no competing interests.

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