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### Authors

Moussa, S. G

Stern, A. C

Raff, J. D

et al.

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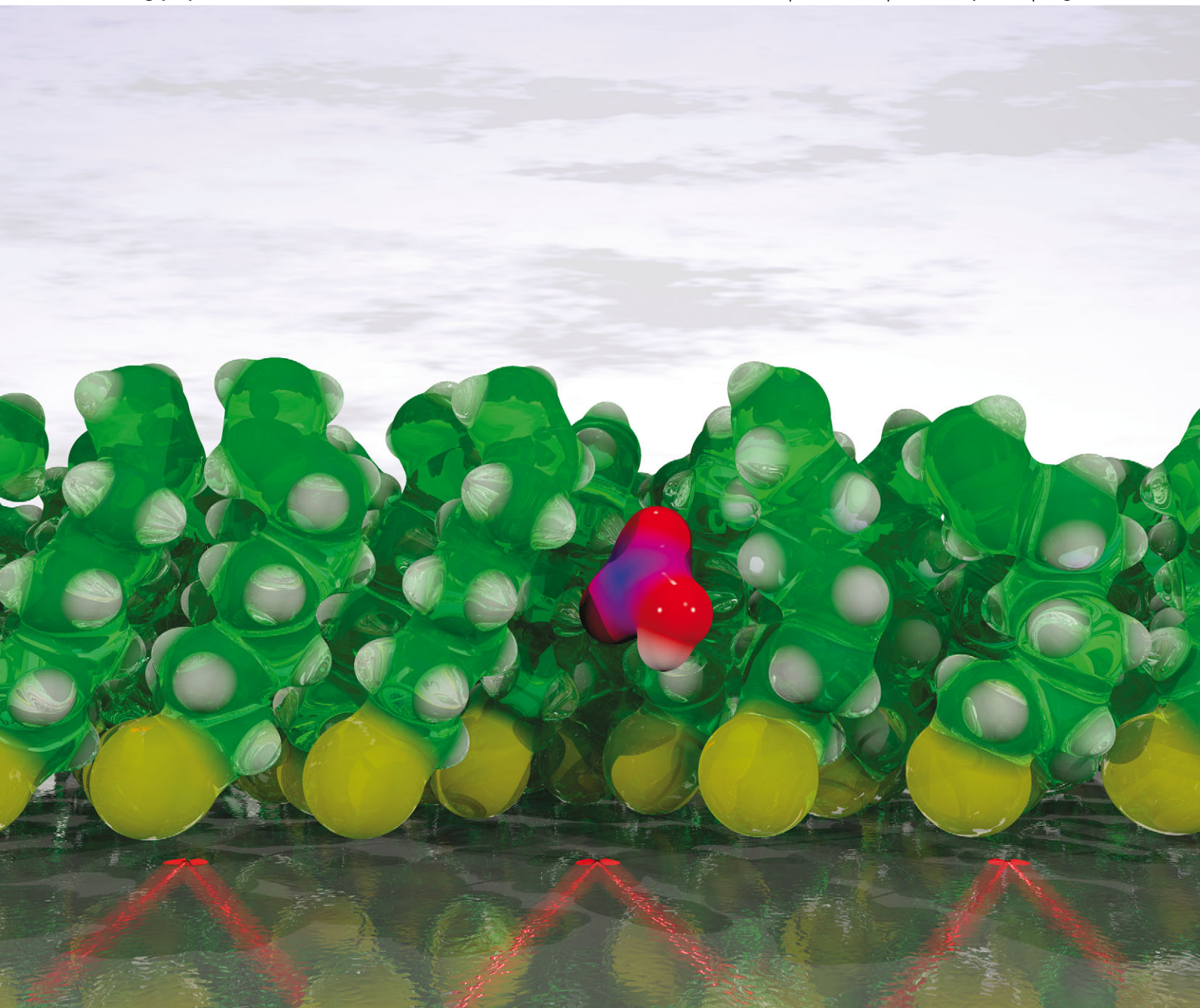
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**PAPER**

D. J. Tobias, B. J. Finlayson-Pitts *et al.*

Experimental and theoretical studies of the interaction of gas phase nitric acid and water with a self-assembled monolayer



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# Experimental and theoretical studies of the interaction of gas phase nitric acid and water with a self-assembled monolayer

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S. G. Moussa, A. C. Stern, J. D. Raff,<sup>†</sup> C. W. Dilbeck, D. J. Tobias\* and B. J. Finlayson-Pitts\*

Nitric acid in air is formed by atmospheric reactions of oxides of nitrogen and is removed primarily through deposition to surfaces, either as the gas or after conversion to particulate nitrate. Many of the surfaces and particles have organic coatings, but relatively little is known about the interaction of nitric acid with organic films. We report here studies of the interaction of gaseous HNO<sub>3</sub> with a self-assembled monolayer (SAM) formed by reacting 7-octenyltrichlorosilane [H<sub>2</sub>C=CH(CH<sub>2</sub>)<sub>6</sub>SiCl<sub>3</sub>] with the surface of a germanium infrared-transmitting attenuated total reflectance (ATR) crystal that was coated with a thin layer of silicon oxide (SiO<sub>x</sub>). The SAM was exposed at 298 ± 2 K to dry HNO<sub>3</sub> in a flow of N<sub>2</sub>, followed by HNO<sub>3</sub> in humid N<sub>2</sub> at a controlled relative humidity (RH) between 20–90%. For comparison, similar studies were carried out using a similar crystal without the SAM coating. Changes in the surface were followed using Fourier transform infrared spectroscopy (FTIR). In the case of the SAM-coated crystal, molecular HNO<sub>3</sub> and smaller amounts of NO<sub>3</sub><sup>-</sup> ions were observed on the surface upon exposure to dry HNO<sub>3</sub>. Addition of water vapor led to less molecular HNO<sub>3</sub> and more H<sub>3</sub>O<sup>+</sup> and NO<sub>3</sub><sup>-</sup> complexed to water, but surprisingly, molecular HNO<sub>3</sub> was still evident in the spectra up to 70% RH. This suggests that part of the HNO<sub>3</sub> observed was initially trapped in pockets within the SAM and shielded from water vapor. After increasing the RH to 90% and then exposing the film to a flow of dry N<sub>2</sub>, molecular nitric acid was regenerated, as expected from recombination of protons and nitrate ions as water evaporated. The nitric acid ultimately evaporated from the film. On the other hand, exposure of the SAM to HNO<sub>3</sub> and H<sub>2</sub>O simultaneously gave only hydronium and nitrate ions. Molecular dynamics simulations of defective SAMs in the presence of HNO<sub>3</sub> and water predict that nitric acid intercalates in defects as a complex with a single water molecule that is protected by alkyl chains from interacting with additional water molecules. These studies are consistent with the recently proposed hydrophobic nature of HNO<sub>3</sub>. Under atmospheric conditions, if HNO<sub>3</sub> is formed in organic layers on surfaces in the boundary layer, e.g. through NO<sub>3</sub> or N<sub>2</sub>O<sub>5</sub> reactions, it may exist to a significant extent in its molecular form rather than fully dissociated to nitrate ions.

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## Introduction

Nitric acid is considered to be the end-product of oxidation of oxides of nitrogen in the atmosphere.<sup>1</sup> This acid reacts with ammonia and amines to form solid or aqueous phase nitrate particles, and can also be taken up into other types of

atmospheric particles. Removal of gaseous HNO<sub>3</sub> and particulate nitrates takes place primarily through wet and dry deposition, in what were once considered to be terminal removal processes. However, there are now a number of laboratory and field studies that suggest that nitric acid can be converted back into gaseous oxides of nitrogen such as NO, NO<sub>2</sub> and HONO.<sup>2–8</sup>

Nitric acid is a strong acid, but it requires a cluster of at least four water molecules to dissociate in the gas phase.<sup>9–11</sup> In bulk aqueous solutions and on ice, nitric acid can form hydrates with 1, 2 or 3 water molecules, depending on the concentration.<sup>12–24</sup> Such hydrates, as well as the HNO<sub>3</sub> dimer, have also been seen on solid surfaces during the hydrolysis of NO<sub>2</sub> at

Department of Chemistry, University of California, Irvine, CA 92697-2025, USA.  
E-mail: [bjfinlay@uci.edu](mailto:bjfinlay@uci.edu) (for experiments), [dtobias@uci.edu](mailto:dtobias@uci.edu) (for theory);  
Fax: +1 949 824-2420; Tel: +1 949 824-7670, +1 949 824-4295

<sup>†</sup> Current address: School of Public & Environmental Affairs and Department of Chemistry, Indiana University, Bloomington, IN 47405-2204.

50% relative humidity.<sup>25</sup> Both theory and experiment suggest that nitric acid is not fully dissociated at the interface of aqueous solutions.<sup>26–34</sup> Indeed, recent results suggest that undissociated HNO<sub>3</sub> interacts very weakly with water.<sup>33,34</sup>

Because of the presence of significant concentrations of water vapor in the lower atmosphere, there is an abundance of water available on surfaces such as roads, buildings, *etc.*, as well as in airborne particles.<sup>35–37</sup> Surfaces in the tropospheric boundary layer also hold a variety of adsorbed organic compounds.<sup>38–42</sup> However, there are relatively few studies of the interactions of HNO<sub>3</sub> with such surface-bound organics. Shultz and coworkers<sup>43</sup> showed that HNO<sub>3</sub> in CCl<sub>4</sub> was undissociated, while Donaldson and coworkers<sup>4</sup> have shown that HNO<sub>3</sub> adsorbed into an organic film could photodissociate to generate gas phase NO<sub>2</sub> and/or HONO. In this case, the organic acted as a photosensitizer.

The goal of this study is to understand the interaction of HNO<sub>3</sub> with a relatively well-characterized organic film, and the effect of water on these interactions using attenuated total reflectance Fourier transform infrared spectroscopy (ATR-FTIR).<sup>44</sup> Self-assembled monolayers (SAMs) are used as a model for organic compounds that are adsorbed on atmospheric surfaces.<sup>45–53</sup> We show, *via* a combination of experimental spectroscopic data and molecular dynamics simulations, that some of the gas phase HNO<sub>3</sub> intercalates between the hydrocarbon chains and is not fully available to interact with water that is added subsequently. However, simultaneous exposure of the SAM to HNO<sub>3</sub> and H<sub>2</sub>O only generates protons and nitrate ions. These studies suggest that if nitric acid is formed in an organic matrix, for example on boundary layer surfaces, it may exist in an undissociated molecular form whose chemistry and photochemistry is different than that of the nitrate ion formed upon dissociation.

## Experimental

### Preparation of ATR crystals

A germanium attenuated total reflectance (ATR) crystal (Pike Technologies, 80 mm × 10 mm × 4 mm) allowing for 10 reflections along the length of the crystal was used both as a substrate and as an optical element with which to detect adsorbed H<sub>2</sub>O, HNO<sub>3</sub>, NO<sub>3</sub><sup>−</sup>, and the attached SAMs. While well-ordered SAMs are readily synthesized on silica,<sup>54–58</sup> SiO<sub>x</sub> does not transmit below 1500 cm<sup>−1</sup>, a region where both nitric acid and nitrate ions have strong absorptions that are readily followed experimentally. As a result, a germanium ATR crystal coated with a thin layer of SiO<sub>x</sub> was used for these studies, which extended the IR window down to 860 cm<sup>−1</sup>. The Ge crystal had a thin layer of SiO<sub>x</sub> (average of 20 nm based on ellipsometry measurements) on its surface that was deposited using plasma-enhanced chemical vapor deposition (PECVD). The SiO<sub>2</sub> was generated by the oxidation of SiH<sub>4</sub> (2.75 mL min<sup>−1</sup>) with N<sub>2</sub>O (50 mL min<sup>−1</sup>) in the presence of nitrogen. Prior to use and derivatization, ATR crystals were rinsed with double distilled water (Milli-Q Plus, 18.2 MΩ cm), dried with nitrogen (Oxygen Service Co., UHP, 99.999%) and placed in an

argon plasma discharge cleaner (Harrick Scientific Plasma Cleaner/Sterilizer PDC-32G, medium power) for ~10 min to remove organic contaminants. The SAM, designated hereafter as C8=, was formed by reacting the SiO<sub>x</sub>-coated ATR crystals with 7-octenyltrichlorosilane. [H<sub>2</sub>C=CH(CH<sub>2</sub>)<sub>6</sub>SiCl<sub>3</sub>] (Sigma-Aldrich, mixture of isomers, 96%). A detailed description of the procedure for depositing the SAMs onto the ATR crystal has been described previously.<sup>45,46,59,60</sup>

### Exposure of surfaces to HNO<sub>3</sub> and H<sub>2</sub>O

The ATR crystal was placed in a horizontal flow-through cell (Pike Technologies) with a head space of ~500 μL,<sup>60</sup> where it was purged overnight with a dry stream of N<sub>2</sub> inside the sample compartment of a Mattson Galaxy 5020 FTIR spectrometer. The stainless steel parts of the ATR cell were coated with halocarbon wax, and Teflon tubing and connections were used to minimize loss of HNO<sub>3</sub> on the sampling lines and walls of the cell. For the 4000–850 cm<sup>−1</sup> region, the depth of penetration of the evanescent wave from the Ge ATR crystal into air was calculated to be 0.13–0.61 μm, much greater than the thickness of the combined SAM (~1.3 nm) and SiO<sub>x</sub> layer (~20 nm). The infrared beam thus interrogates the entire organic monolayer and provides spectra similar to a transmission spectrum. All ATR measurements were taken at a total pressure of 1 atm. Single beam spectra were collected over the spectral range 4000–600 cm<sup>−1</sup>. For each experiment, a total of 128 scans were collected at 4 cm<sup>−1</sup> resolution.

Dry gaseous HNO<sub>3</sub> was generated by flowing N<sub>2</sub> (20 mL min<sup>−1</sup>) over the surface of a solution of HNO<sub>3</sub>/H<sub>2</sub>SO<sub>4</sub> (1 : 3 v/v) in a glass trap (69.7 wt% HNO<sub>3</sub>, Fisher; 96.0 wt% H<sub>2</sub>SO<sub>4</sub>, Fisher). The flow of HNO<sub>3</sub> was further combined and diluted with a flow of dry or humid nitrogen with known relative humidity, making the total flow ~220 mL min<sup>−1</sup>. A portion, (70–74 mL min<sup>−1</sup>) of this total flow, was diverted to the ATR cell. The concentration of HNO<sub>3</sub> over the ATR crystal after dilution was 1.7 × 10<sup>16</sup> molecules cm<sup>−3</sup> (~700 ppm). Desired water vapor concentrations (relative humidity) were achieved by mixing measured flows of dry and humid N<sub>2</sub> gas. Humid N<sub>2</sub> flows were obtained by bubbling dry nitrogen through water in two borosilicate fritted glass bubblers in a thermostated water bath. The temperatures of the spectrometer, cell, and water bath and the trap containing the HNO<sub>3</sub> were kept constant at 298 ± 2 K as measured by a thermocouple (OMEGA HH506).<sup>37,59</sup> For some experiments, the SAM was exposed simultaneously to HNO<sub>3</sub> and H<sub>2</sub>O at 50% RH by redirecting the flow of N<sub>2</sub> to pass over the headspace of a 5 M aqueous solution of HNO<sub>3</sub>. The HNO<sub>3</sub> concentration in the gas phase in these experiments was calculated to be 1.7 × 10<sup>13</sup> molecule cm<sup>−3</sup> (~700 ppb) based on known H<sub>2</sub>O–HNO<sub>3</sub> equilibria.<sup>61,62</sup>

### Scanning electron microscopy (SEM)

SEM images were collected on a Zeiss Ultra 55 field emission scanning electron microscope using an in-column energy selective backscattered electron detector. An accelerating voltage of 2000 V was used, and a bias of −1500 V was applied to the collector grid to eliminate secondary electrons and low-energy



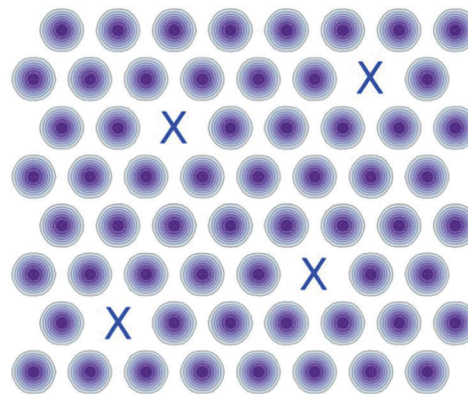
backscattered electrons. The images therefore are produced only by backscattered electrons. A working distance of 3.9 mm was used.

### Molecular dynamics (MD) simulations

To shed light on the experimental findings, simulations were performed using a self-assembled monolayer composed of 64 eight-carbon, vinyl-terminated alkylthiolates (1-octenethiolate) chemically adsorbed on a gold (111) lattice coexisting with humid  $\text{HNO}_3$  vapor. As in our previous work,<sup>45,59</sup> the choice of alkylthiolates on gold in computer simulations rather than alkylsilanes on germanium silicate glass substrate as in the experiment was driven by methodological considerations. While potentials for accurately describing the interactions between alkanethiolates and the gold(111) surface are well established,<sup>63</sup> the relatively complex bonding between alkylsiloxanes and silicon oxide surfaces introduces uncertainties in the development of atomistic models for alkylsiloxane SAMs.<sup>64</sup> Moreover, it has been shown that the structural differences between SAMs composed of long alkane chains on silica and on gold are minor.<sup>65–67</sup>

Chemisorption of the alkylthiolates to the gold surface was modeled using a two-dimensional corrugation potential in conjunction with a Lennard-Jones 9–6 potential acting along the vector normal to the SAM surface. The alkylthiolate, gold, and corrugation force field parameters used were those employed successfully in several previous studies.<sup>63,68</sup> The van der Waals interactions of  $\text{HNO}_3$  were modeled using the Lennard-Jones parameters of the Amber generalized force field,<sup>69</sup> while the electrostatic interactions were accounted for with atom-centered point charges fit to the electrostatic potential of  $\text{HNO}_3$ .<sup>69</sup> All intramolecular degrees of freedom within an  $\text{HNO}_3$  molecule were held rigid using high force constants for motions associated with planarity, bending, and stretching, with the sole exception of the torsional potential involving the proton, which was modeled as a cosine with a barrier height of  $2.86 \text{ kcal mol}^{-1}$  and a periodicity of two. Water molecules were modeled using the rigid TIP3P parameters.<sup>70</sup>

The silica surface that results from the PECVD process is amorphous and rough on the molecular scale. By extension, it can be safely assumed that the SAM coating will be far more disordered than an atomically smooth alkylthiolate SAM. Although it was impractical to consider all possible arrangements of SAM defects, the goal of this study was more modest: to simulate one example of a defective monolayer to investigate the possibility that  $\text{HNO}_3$  may be able to intercalate between the hydrocarbon chains of the monolayer and remain buried and protected from water vapor. To introduce defects in the monolayer surface, four SAM molecules were removed. Fig. 1 shows the sites of the defects denoted by X's that were used in the MD simulations. After equilibrating the monolayer, a single  $\text{HNO}_3$  was added to the simulation cell and simulated until the  $\text{HNO}_3$  settled into a defect, at which point the simulation was then continued with between 0 and 8 added water molecules. Canonical ensemble simulations (constant particle number, volume, and temperature) were performed at a temperature



**Fig. 1** Two dimensional contour plots viewed from the top of the corrugation potential that acts on the sulfur atoms of the alkylthiols. The lattice sites corresponding to the defects are marked by an "X".

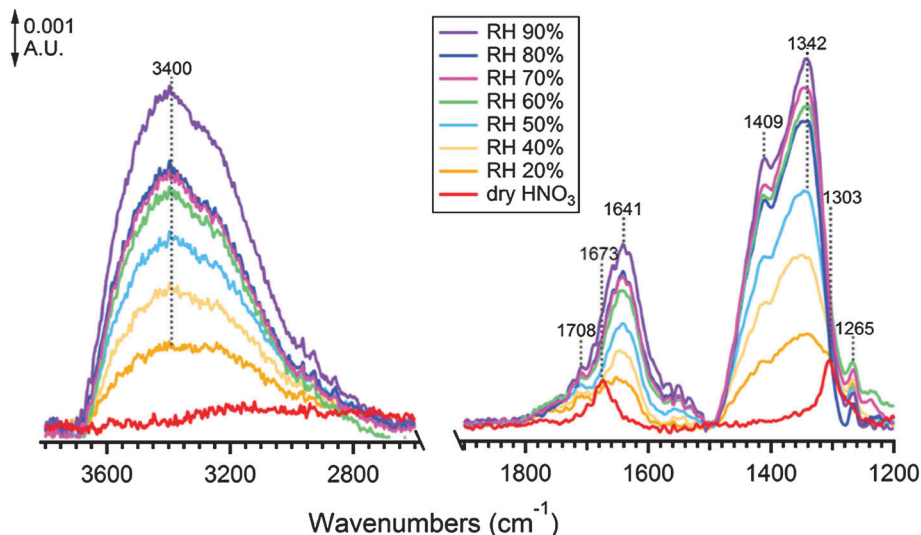
of 300 K maintained by running Langevin dynamics with a damping coefficient of  $1 \text{ ps}^{-1}$ . A reversible, multiple timestep algorithm<sup>71</sup> was used to integrate the equations of motion with time steps of 2 fs for the electrostatic forces and short-ranged nonbonded forces, and 1 fs for the bonded forces. The smooth particle mesh Ewald method<sup>72</sup> was used to calculate the electrostatic interactions. The short-ranged, real space interactions were cut off at 11 Å using a switching function. Simulations were conducted using the NAMD software package.<sup>73</sup>

## Results and discussion

### A. Interaction of gaseous $\text{HNO}_3$ and water vapor with a bare $\text{SiO}_x/\text{Ge}$ surface

Fig. 2 shows the absorbance spectrum (red trace) obtained when dry gaseous nitric acid flows over the  $\text{SiO}_x/\text{Ge}$  ATR crystal. Absorbance is defined here as  $\log(S_0/S_1)$  where  $S_0$  is the single beam spectrum of the crystal prior to addition of  $\text{HNO}_3$  and  $S_1$  is the single beam spectrum after exposure to  $\text{HNO}_3$ . Thus, positive peaks signify new species on the crystal due to uptake and reactions of  $\text{HNO}_3$ . Three distinct peaks appear. Those at  $1673 \text{ cm}^{-1}$  and  $1303 \text{ cm}^{-1}$  are assigned to well known  $\nu_2$  ( $\text{NO}_2$  antisymmetric stretch) and  $\nu_4$  ( $\text{NO}_2$  symmetric stretch) vibrational modes of  $\text{HNO}_3$ , which have been reported for  $\text{HNO}_3$  in aqueous solutions,<sup>26,74,75</sup> in the gas phase<sup>76,77</sup> and on ice.<sup>23,24,78,79</sup> The peak at  $1265 \text{ cm}^{-1}$  is assigned to the nitrite ion,<sup>80</sup> and indicates that the uptake of  $\text{HNO}_3$  onto the crystal is accompanied by a surface reaction that forms  $\text{NO}_2^-$ . Underwood *et al.* also observed the presence of nitrite ion on mineral oxide surfaces at  $\sim 1250 \text{ cm}^{-1}$ .<sup>81</sup>

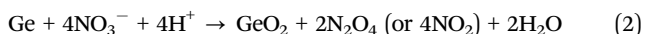
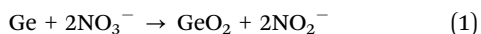
The bare  $\text{SiO}_x/\text{Ge}$  surface continued to take up nitric acid for approximately two hours, at which time peaks due to molecular  $\text{HNO}_3$  no longer changed in intensity. Once equilibrium was established, a flow of  $\text{N}_2$  containing various amounts of water vapor that increased in stepwise fashion from 20 to 90% RH was then added to the  $\text{HNO}_3$  flow. The concentration of  $\text{HNO}_3$  above the crystal was held constant to minimize evaporation of nitric acid from the surface. New peaks appear around  $3400 \text{ cm}^{-1}$  and  $1641 \text{ cm}^{-1}$  with a shoulder at  $1708 \text{ cm}^{-1}$ . A broad feature in



**Fig. 2** Infrared spectra of a  $\text{SiO}_x/\text{Ge}$  ATR crystal after exposure to dry  $\text{HNO}_3$  ( $1.7 \times 10^{16}$  molecule  $\text{cm}^{-3}$ ) followed by the addition of water vapor to the flow to give 20–90% RH at 298 K. The  $y$ -axis is  $\log(S_0/S_1)$  where  $S_0$  is the single beam spectrum of a  $\text{SiO}_x/\text{Ge}$  ATR crystal prior to exposure to  $\text{HNO}_3$  or  $\text{H}_2\text{O}$  and  $S_1$  is the single beam spectrum of the same  $\text{SiO}_x/\text{Ge}$  ATR crystal after exposure to  $\text{HNO}_3$  before and then after addition of  $\text{H}_2\text{O}$ .

the 1300–1500  $\text{cm}^{-1}$  region also appears. The peaks due to molecular  $\text{HNO}_3$  at 1673 and 1303  $\text{cm}^{-1}$  are strongly overlapped by the new product peaks, and become essentially undetectable when water vapor is added. The peak at 1708  $\text{cm}^{-1}$  is assigned to  $\text{H}_3\text{O}^+$ .<sup>23,82</sup> Mashburn *et al.*<sup>83</sup> and Baltrusaitis *et al.*<sup>84</sup> similarly assigned peaks in this region to  $\text{H}_3\text{O}^+$  on clays and aluminum oxide surfaces. Those centered at 3400 and 1641  $\text{cm}^{-1}$  are assigned to the stretching and bending modes of  $\text{H}_2\text{O}$ , respectively, where water may be complexed in part to nitrate ions.<sup>24,85–87</sup> The peaks in the 1300–1500  $\text{cm}^{-1}$  region are due to nitrate ions complexed to water. As discussed in detail elsewhere,<sup>24,87</sup> the antisymmetric stretching vibration ( $\nu_3$ ) of  $\text{NO}_3^-$  is degenerate for the planar structure, but the degeneracy is removed and two bands appear if the symmetry is changed by interaction with other species such as solvent molecules or a surface.<sup>88</sup> At 90% relative humidity, the splitting in the  $\nu_3$  mode is 67  $\text{cm}^{-1}$ , which is typical of  $\text{NO}_3^-$  bound to approximately five water molecules.<sup>24,87</sup>

It is well known that  $\text{HNO}_3$  readily adsorbs to silica.<sup>89,90</sup> However, SEM micrographs of the crystal surface (not shown) revealed that the  $\text{SiO}_x$  coating was not uniform, leaving regions of exposed germanium. Nitric acid is known to oxidize Ge.<sup>91–93</sup> This forms the metal oxide and reduced nitrogen oxides such as  $\text{NO}_2/\text{N}_2\text{O}_4$  and  $\text{NO}_2^-$  via thermodynamically favored net reactions such as:



Thus, the spectra in Fig. 2 are consistent with a reaction of  $\text{HNO}_3$  with the uncoated germanium to form  $\text{NO}_2^-$ . This remains on the surface, as does  $\text{HNO}_3$  which, upon the addition of water, dissociates to  $\text{H}_3\text{O}^+$  and  $\text{NO}_3^-$ . The spectra in the nitrate ion region are very similar to those reported for nitrate ions on metal oxide surfaces and clays.<sup>83,84,90</sup> Gaseous products

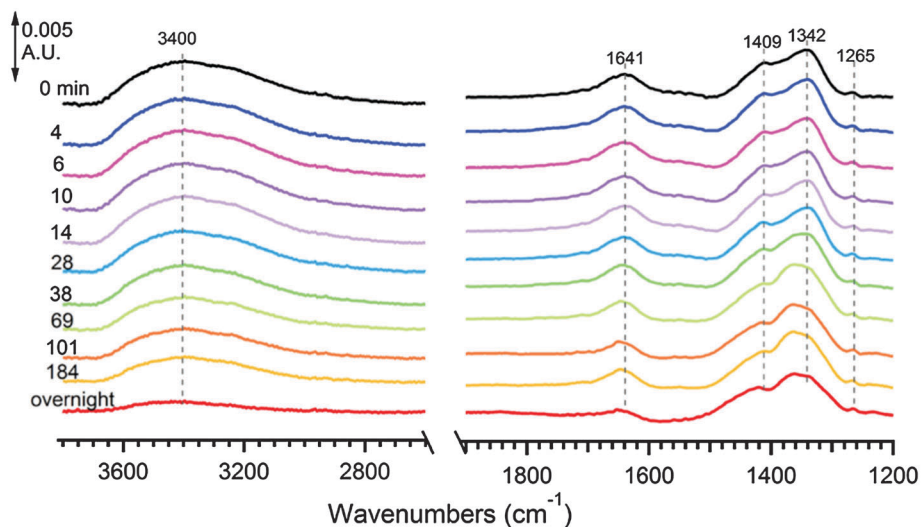
such as  $\text{NO}_2/\text{N}_2\text{O}_4$  would be removed in the gas stream, and likely also undergo some hydrolysis in the water film to form  $\text{HONO}/\text{NO}_2^-$  and  $\text{HNO}_3/\text{NO}_3^-$ .<sup>94</sup>

Fig. 3 shows the sequence of spectra when the flow of the gaseous  $\text{HNO}_3$ - $\text{H}_2\text{O}$  mixture is replaced by a flow of dry  $\text{N}_2$ . The peaks due to water at 1641  $\text{cm}^{-1}$  and  $\sim 3400$   $\text{cm}^{-1}$  decrease as the surface dries out, but there are no other significant changes in the spectrum. This indicates that water evaporates during the drying process, leaving nitrate and nitrite ions behind on the surface, along with residual water.

## B. Interaction of gaseous $\text{HNO}_3$ and water vapor with a self-assembled monolayer on $\text{SiO}_x/\text{Ge}$

Fig. 4 shows spectra similar to those in Fig. 2 but for a C8= SAM coated  $\text{SiO}_x/\text{Ge}$  crystal exposed first to dry  $\text{HNO}_3$  then to humidified  $\text{N}_2$  at various RH. Organic trichlorosilanes form SAMs on both silica surfaces and on germanium,<sup>54,60</sup> although those on Ge are less stable.<sup>54,60,95–101</sup> Nevertheless, SAMs are expected to cover the entire surface. Given the presence of the double bond in the SAM, the irregular  $\text{SiO}_x$  coating of the Ge, and the instability of the SAM on Ge, some disorder is expected in the film. This will be reflected in the peak positions and widths.<sup>54,55,102,103</sup> The antisymmetric  $-\text{CH}_2-$  stretch in a well-ordered SAM has a peak position at 2917  $\text{cm}^{-1}$  and a full width at half-maximum (fwhm) of 14–16  $\text{cm}^{-1}$ .<sup>55</sup> For the C8= SAM on borosilicate glass,<sup>59</sup> the peak is at 2919 with a fwhm of 21  $\text{cm}^{-1}$ . For the C8= SAM on a germanium ATR crystal,<sup>60</sup> the peak is at 2924 with a fwhm of 30  $\text{cm}^{-1}$ . In the present case of  $\text{SiO}_x/\text{Ge}$ , the peak is at 2920  $\text{cm}^{-1}$ , with a fwhm of 23–25  $\text{cm}^{-1}$ .

There are a number of significant differences between experiments conducted in the presence of a SAM coating compared to those on bare Ge- $\text{SiO}_x$  surfaces. For example, the time required for  $\text{HNO}_3$  to reach equilibrium with the surface was much shorter than for the bare crystal ( $\sim 10$ –15 min compared to several hours),



**Fig. 3** Infrared spectra of a  $\text{SiO}_x/\text{Ge}$  ATR crystal exposed to  $\text{HNO}_3$  and  $\text{H}_2\text{O}$ , and then to dry  $\text{N}_2$  for the times shown. The y-axis is  $\log(S_0/S_1)$  where  $S_0$  is the single beam spectrum of the  $\text{SiO}_x/\text{Ge}$  ATR crystal prior to exposure to  $\text{HNO}_3$  and  $\text{H}_2\text{O}$  and  $S_1$  is the single beam spectrum after exposure to  $\text{HNO}_3$  and  $\text{H}_2\text{O}$ , and then to dry  $\text{N}_2$  for the times shown.

and a peak due to  $\text{NO}_2^-$  was not observed. The longer time needed for the adsorbed nitric acid to stabilize on the bare crystal reflects the time required to first convert the germanium to its oxide, and then to saturate the surface with  $\text{HNO}_3$ . For the SAM-coated crystal,  $\text{HNO}_3$  is not reacting directly with Ge on the underlying crystal surface, suggesting that the SAM coverage is relatively complete and blocks these sites. It should be noted, however, that exposure of the SAM-coated crystal to water vapor alone resulted in a negative peak due to the surface free O–H stretch at  $3695\text{ cm}^{-1}$ . This indicates that the smaller water molecule can still access some of the crystal surface (data not shown).

Exposure to dry  $\text{HNO}_3$  produces a broad shoulder in the  $1300\text{--}1500\text{ cm}^{-1}$  region where the  $\nu_3$  antisymmetric stretching mode of  $\text{NO}_3^-$  occurs. There are two possible sources of  $\text{NO}_3^-$  in the dry system: dissociation of  $\text{HNO}_3$  or its self-reaction/autoionization to form  $\text{NO}_3^- + \text{NO}_2^+ + \text{H}_2\text{O}$ . Traces of water will be present in this system which can promote some dissociation of  $\text{HNO}_3$ . Autoionization generates  $\text{NO}_2^+$  which has an infrared absorption band<sup>94</sup> at  $\sim 2300\text{ cm}^{-1}$ , but it is sufficiently weak that it may not be detectable under the present conditions. The  $\text{NO}_2^+$  ion reacts with two molecules of water to regenerate  $\text{HNO}_3 + \text{H}_3\text{O}^+$ , so the net effect of these two possibilities is similar when water vapor is added.

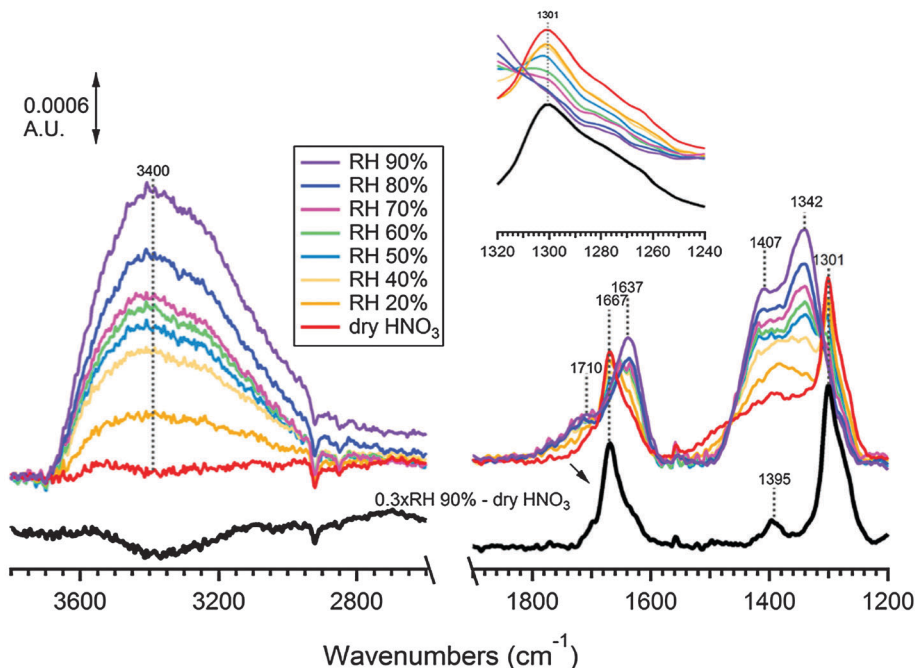
The shoulder attributed to  $\text{NO}_3^-$  is structureless and centered at  $\sim 1395\text{ cm}^{-1}$  for RH below 50%. This is in contrast to that in the absence of the SAM (Fig. 2) where two bands due to splitting of the degeneracy are detectable even at 20% RH and the maximum peak absorbance is at  $1342\text{ cm}^{-1}$ . This suggests there is an additional underlying peak in this region for the SAM-coated crystal. In order to probe this, a fraction (0.30) of the spectrum at 90% RH was subtracted from that at the 0% RH spectrum, in order to compensate for the contribution due to the nitrate ion. As seen by the black line shown below the original data in Fig. 4, a small peak at  $1395\text{ cm}^{-1}$  emerges from this subtraction. This is attributed to the HON

bend of  $\text{HNO}_3$  which is also observed in the liquid<sup>74</sup> and at anhydrous interfaces.<sup>26</sup> This peak is not obvious in the absence of the SAM, suggesting that the interaction of the  $\text{HNO}_3$  with the  $\text{SiO}_x/\text{Ge}$  surface is different. A strong interaction with the surface in the absence of the SAM *via* the hydrogen and nitrogen may shift and broaden the HON absorption to the point that it is not evident in the spectrum.

The peak at  $1301\text{ cm}^{-1}$  due to molecular  $\text{HNO}_3$  is detectable up to 70% RH in the presence of the SAM (Fig. 4 inset), whereas it was not detectable on the bare crystal after exposure to water vapor, even at 20% RH. Based on previous studies<sup>25</sup> where  $\text{HNO}_3$  was formed on surfaces in the presence of water vapor *via* the hydrolysis of  $\text{NO}_2$ , it seems likely that some of the nitric acid is present in the form of hydrates. Such hydrates may contribute to the peak at  $1637\text{ cm}^{-1}$  (Fig. 4), since complexation of  $\text{HNO}_3$  with water is expected to red-shift the antisymmetric  $\text{NO}_2$  stretch in the acid by  $\sim 10\text{ cm}^{-1}$ .<sup>24,26,86,104</sup> The O–H stretching absorptions due to nitric acid–water complexes ( $2975\text{ cm}^{-1}$  and  $2812\text{ cm}^{-1}$ ) are also expected to shift relative to anhydrous nitric acid,<sup>24,25</sup> but they are overwhelmed by the large contribution due to water in this region. However, regardless of a potential contribution of nitric acid hydrates to the spectra, it is clear that the SAM coating inhibits reaction of  $\text{HNO}_3$  with the Ge substrate as well as the dissociation of  $\text{HNO}_3$  on the surface in the presence of water vapor compared to the bare  $\text{SiO}_x/\text{Ge}$  surface.

The intensities of both the nitrate and water peaks on the SAM-coated crystal are significantly smaller, by about a factor of two, than those for the bare crystal. This is not surprising since the uptake of water would be expected to be greater on a bare oxide crystal than on a surface that is largely covered with a hydrophobic organic layer. The presence of large amounts of water would promote more dissociation of  $\text{HNO}_3$ .

It is clear from the spectra in Fig. 4 that the amount of molecular  $\text{HNO}_3$  relative to nitrate ions decreases as the relative



**Fig. 4** Infrared spectra of a C8= SAM-coated  $\text{SiO}_x/\text{Ge}$  crystal after exposure to dry  $\text{HNO}_3$  followed by water vapor at different RH at 298 K. The y-axis is  $\log(S_0/S_1)$  where  $S_0$  is the single beam spectrum of the C8= SAM on a  $\text{SiO}_x/\text{Ge}$  ATR crystal prior to exposure to  $\text{HNO}_3$  or  $\text{H}_2\text{O}$  and  $S_1$  is the single beam spectrum of the C8= SAM after exposure to  $\text{HNO}_3$  or  $\text{H}_2\text{O}$ . The dark black trace is the spectrum resulting from the subtraction of  $0.3 \times$  the 90% RH spectrum from that for dry  $\text{HNO}_3$ . The inset expands the region around  $1301 \text{ cm}^{-1}$ .

humidity increases, as expected for water-promoted dissociation of the acid. If the transition intensities of the surface species were the same as for the gas phase, a rough estimate of the relative amounts of  $\text{HNO}_3$  and  $\text{NO}_3^-$  can be made as follows. The theoretical infrared intensity of the  $\text{NO}_2/\text{NO}_3$  symmetric stretch of  $\text{HNO}_3$  (at  $1301 \text{ cm}^{-1}$  in the present studies) using MP2 TZ2P and assuming harmonicity was reported to be  $206 \text{ km mol}^{-1}$ .<sup>105</sup> The theoretical intensities (anharmonic) for the nitrate ion are  $845 \text{ km mol}^{-1}$  for nitrate coordinated to one water molecule and an average of  $592 \text{ km mol}^{-1}$  for nitrate coordinated to 4–5 water molecules.<sup>25</sup> The spectra in the  $1200\text{--}1500 \text{ cm}^{-1}$  region in Fig. 4 were separated into two overlapping contributions, one from  $\text{HNO}_3$  centered at  $1301 \text{ cm}^{-1}$  and a broader contribution from  $\text{NO}_3^-$ . Using the peak areas of the two,  $\text{HNO}_3$  is 1.6 times greater than  $\text{NO}_3^-$  in the “dry” case. At 50% RH when it was assumed that  $\text{NO}_3^-$  is coordinated to 4–5 water molecules but  $\text{HNO}_3$  is not complexed to water, the  $\text{HNO}_3$  would be present at 30% of the nitrate ion concentration. However, calculations by Staikova and Donaldson<sup>104</sup> suggest that the theoretical intensity of the molecule  $\text{HNO}_3$  band could increase by a factor of five when complexed to one water. If this were the case for  $\text{HNO}_3$  at 50% RH, then the  $\text{HNO}_3$  would be present at only 6% of the nitrate ions. These estimates should be taken as order-of-magnitude, given the use of theoretical intensities for gas phase species, the uncertainty in the number of water molecules complexed to  $\text{HNO}_3$  and  $\text{NO}_3^-$ , and the use of a mixture of harmonic and anharmonic calculations.

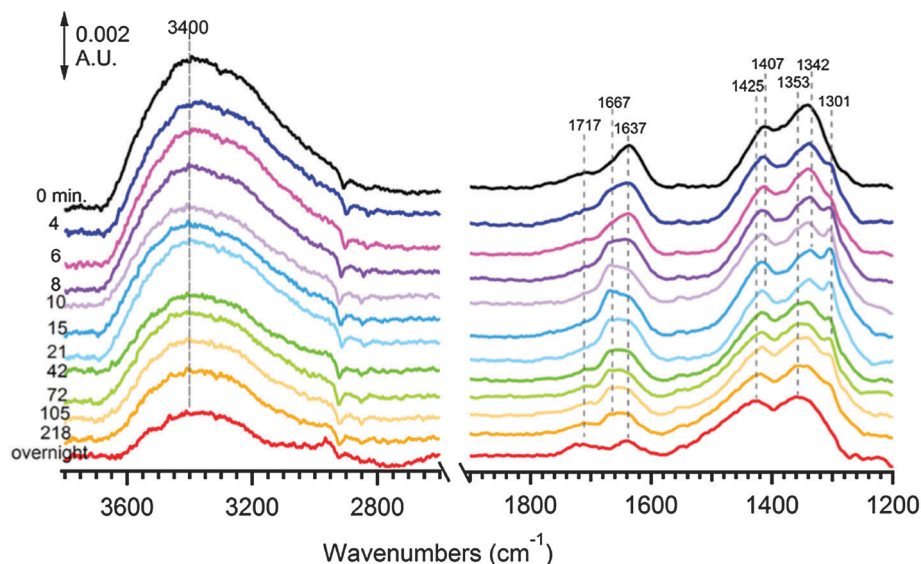
Fig. 5 shows the sequence of spectra obtained when the flow of the gaseous  $\text{HNO}_3\text{--H}_2\text{O}$  mixture is replaced by a flow of dry

$\text{N}_2$  on the SAM-coated crystal. In contrast to the case of the bare crystal, peaks at  $1667$  and  $1301 \text{ cm}^{-1}$  assigned to molecular  $\text{HNO}_3$  appear shortly after drying is initiated. Their relative intensities increase and then decrease over the course of three hours. They completely disappear after drying overnight. This indicates that nitrate ions and protons must be in sufficiently close proximity that they can recombine while water is evaporating and the aqueous thin film becomes more concentrated. Recombination leads to molecular  $\text{HNO}_3$ , which desorbs into the gas phase and is removed in the gas stream.

The following model is consistent with these data. On the bare crystal, dry nitric acid oxidizes exposed Ge in the portions of the crystal that were not coated with  $\text{SiO}_x$ . This converts Ge to its oxide and simultaneously generates reduced forms of oxides of nitrogen such as  $\text{NO}_2^-$ , which remains on the surface ( $1265 \text{ cm}^{-1}$  peak, Fig. 2), and  $\text{N}_2\text{O}_4/\text{NO}_2$ , which will desorb into the gas stream. Once the surface has been completely oxidized, molecular  $\text{HNO}_3$  physisorbs ( $1673$  and  $1303 \text{ cm}^{-1}$ , Fig. 2). The subsequent addition of water leads to dissociation of surface-adsorbed  $\text{HNO}_3$  so that the degenerate bands due to  $\text{NO}_3^-$  complexed to water then appear on the surface in the  $1300\text{--}1500 \text{ cm}^{-1}$  range (Fig. 2). Interaction of the proton and  $\text{NO}_3^-$  with the ionic oxide surface is sufficiently strong so that replacement of the  $\text{HNO}_3/\text{H}_2\text{O}$  with dry  $\text{N}_2$  simply leads to removal of much of the water. This leaves nitrate ions bound to  $\text{Ge}^{4+}$  on the surface (Fig. 3).

When most of the surface is initially covered with the hydrophobic SAM, the  $\text{SiO}_x/\text{Ge}$  crystal is largely protected from exposure to nitric acid. As a result, the equilibrium between





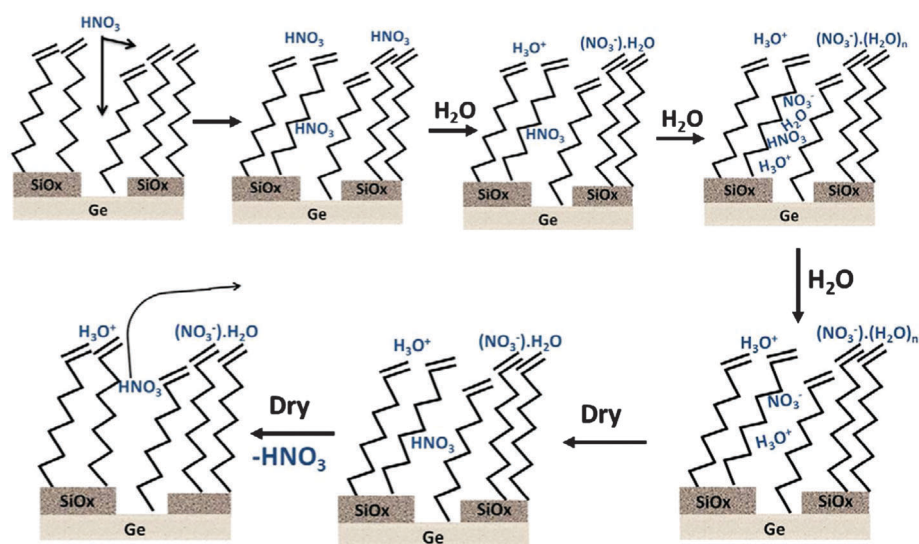
**Fig. 5** Infrared spectra of C8= SAM on SiO<sub>x</sub>/Ge ATR crystal exposed to HNO<sub>3</sub> and H<sub>2</sub>O then to dry N<sub>2</sub> for the times shown. The y-axis is log( $S_0/S_1$ ) where  $S_0$  is the single beam spectrum of a dry C8= SAM on a SiO<sub>x</sub>/Ge ATR crystal prior to exposure to HNO<sub>3</sub> and H<sub>2</sub>O and  $S_1$  is the single beam spectrum of C8= exposed to HNO<sub>3</sub> and H<sub>2</sub>O, and then to dry N<sub>2</sub> for the times shown.

physisorbed HNO<sub>3</sub> and the gas phase is reached much faster and the formation of nitrite ions that accompanies oxidation of Ge on the crystal is not observed. Interestingly, the amount of molecular HNO<sub>3</sub> on the dry SAM-coated surface based on the 1301 cm<sup>-1</sup> peak is about double that on the bare crystal (Fig. 2 vs. Fig. 4) and NO<sub>3</sub><sup>-</sup> is observed immediately (peak at 1395 cm<sup>-1</sup>), even in the absence of water.

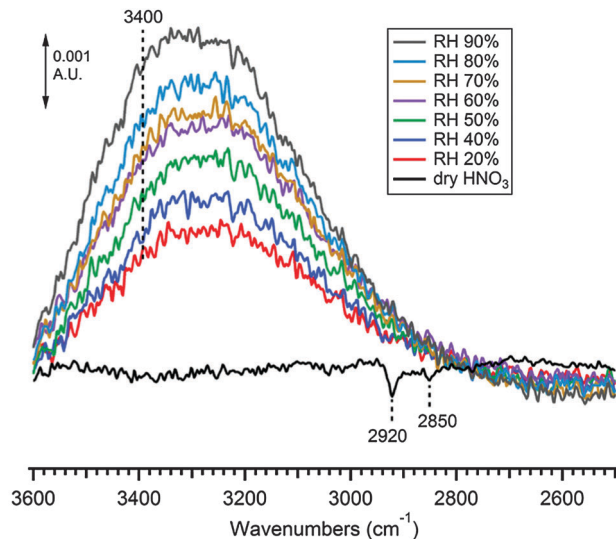
We propose that HNO<sub>3</sub> intercalates between the chains of the SAM, perhaps in pores created due to the disorder in the monolayer. The formation of nitrate ions in the absence of water could reflect a high local concentration of HNO<sub>3</sub> in the pores, leading to autoionization to NO<sub>2</sub><sup>+</sup>NO<sub>3</sub><sup>-</sup> and the generation of nitrate ions as discussed earlier. This “trapped HNO<sub>3</sub>” is

not readily accessible to added water, leading to less dissociation as the RH is increased. Consistent with this model is the persistence of the bands due to molecular HNO<sub>3</sub> up to at least 70% RH (Fig. 4). When dissociation does occur within these pores or between chains at high RH, the proton and NO<sub>3</sub><sup>-</sup> will be in relatively close proximity, so that subsequent drying allows them to readily recombine to form molecular HNO<sub>3</sub> which eventually evaporates as shown in (Fig. 5). This mechanism is summarized schematically in Fig. 6.

Support for this hypothesis is found in the impact of HNO<sub>3</sub> on the order/disorder of the monolayer. Decreases in the intensity of C–H stretching vibrations for the SAM in the 2800–3000 cm<sup>-1</sup> region can indicate either reaction with



**Fig. 6** Schematic of interaction of HNO<sub>3</sub> and H<sub>2</sub>O with a C8= SAM.



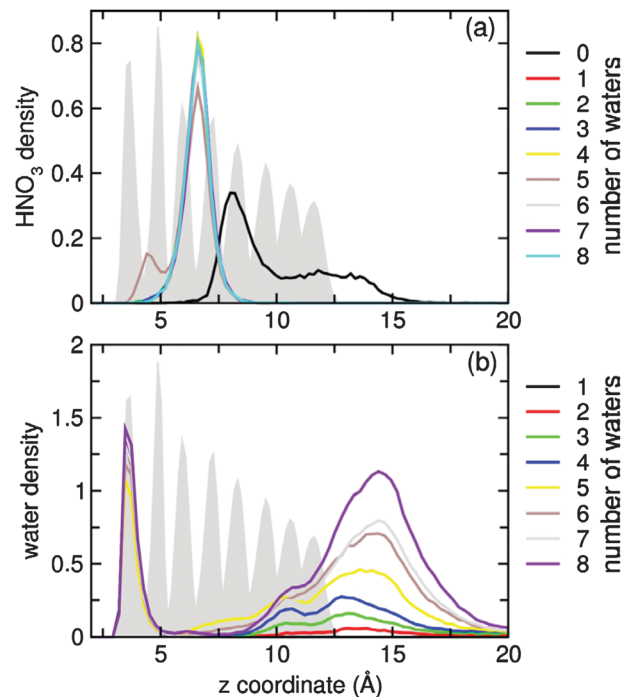
**Fig. 7** Infrared spectra of a  $\text{SiO}_x/\text{Ge}$  ATR crystal coated with a C8 SAM in equilibrium with water vapor at 20–90% RH in the absence of  $\text{HNO}_3$ . For comparison, the black trace is the spectrum on exposure to dry  $\text{HNO}_3$ .

removal of some of the SAM, or increased disorder<sup>54,55,102,103,106</sup> in the organic monolayer. Although it is possible that  $\text{HNO}_3$  reacts with the unsaturated portion of the SAM, peaks due to expected reaction products (*e.g.*, organic nitrates or carbonyl-containing compounds) were never observed in the IR spectra. However, decreases in the  $\nu_{\text{as}}$  and  $\nu_{\text{s}}$  stretches<sup>45,54,55,59</sup> of  $-\text{CH}_2-$  at 2920 and 2850  $\text{cm}^{-1}$  were observed when the SAM-coated crystal was exposed to  $\text{HNO}_3$  either dry or after the addition of water (Fig. 4); as seen in Fig. 7, such decreases did not occur upon exposure to water in the absence of  $\text{HNO}_3$ . In addition, upon drying overnight where the molecular  $\text{HNO}_3$  has been removed (Fig. 5), the spectra in the  $-\text{CH}_2-$  stretching region revert to that of the unperturbed SAM prior to exposure to  $\text{HNO}_3$ . This is consistent with  $\text{HNO}_3$  penetrating into the SAM, causing a disordering of the monolayer. The fact that  $\text{HNO}_3$  intercalates in this manner is consistent with the reported interactions of  $\text{HNO}_3$  with  $\text{CCl}_4$ <sup>43</sup> and with condensed phase organics,<sup>4</sup> and with the recent proposal that undissociated  $\text{HNO}_3$  interacts weakly with water.<sup>33,34</sup>

### C. Molecular dynamics simulations of the interaction of gaseous $\text{HNO}_3$ and water vapor with a self-assembled monolayer

Fig. 8 shows density profiles of  $\text{HNO}_3$  and water (if present) as a function of distance ( $z$ ) away from the substrate. Relative density profiles for carbon are also shown, from which the point of separation between the gas phase and the SAM can be seen to occur at 12.5 Å.

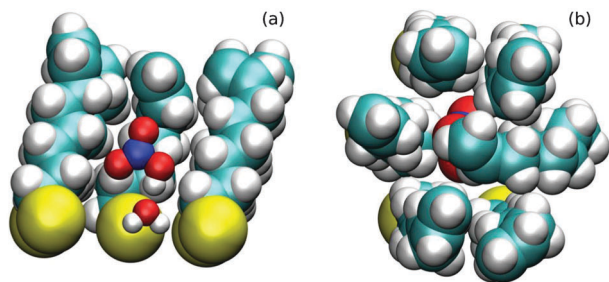
Upon exposure of the defective self-assembled monolayer to  $\text{HNO}_3$  with no water present, nitric acid preferentially resides in the pockets of the monolayer formed by the defects. However, it is prevented from interacting directly with the substrate by SAM molecules which adopt arrangements which protect the defect (Fig. 8a, black line). However, after water vapor is introduced



**Fig. 8** (a)  $\text{HNO}_3$  and (b) water density profiles along the vector normal to the surface (distance  $z$  from the substrate) for the four defect simulation with the indicated number of water molecules added. The density profiles are normalized to the number of molecules present in the simulation. The (un-normalized) density profile of the carbon atoms is shown in grey for reference. Note that in (a), the density profiles corresponding to  $n = 1, 2, 3, 4, 6, 7, 8$  are nearly indistinguishable.

into the simulation cell, the formation of a water– $\text{HNO}_3$  dimer is observed that resides deep in the monolayer. The water– $\text{HNO}_3$  dimer, formed *via* a hydrogen bond involving the acidic proton, is consistently oriented such that the coordinating water is buried deepest, followed by  $\text{HNO}_3$  which has its proton pointed toward the oxygen of the water. The arrangement of the water– $\text{HNO}_3$  dimer can be clearly seen in the density profiles shown in Fig. 8 where the sharp peaks in Fig. 8b at  $\sim 4$  Å correspond to a single water and the associated  $\text{HNO}_3$  peaks are observed at  $\sim 7$  Å in Fig. 8a. Calculation of the radial distribution functions (not shown) between the proton and oxygen of water verifies that the hydrogen bond of the dimer is persistent over the course of the simulation, an indication that the dimer is highly stable. Shown in Fig. 9 is a representative snapshot taken from simulation of the water– $\text{HNO}_3$  dimer buried in the monolayer. From the snapshot in side-view Fig. 9a,  $\text{HNO}_3$  can be seen to reside in a protective pocket of the monolayer, with its proton pointing away from the gas phase. The snapshot in the top-view in Fig. 9b shows the hydrocarbon chain in an arrangement which encapsulates the  $\text{HNO}_3$ .

Although the model of nitric acid does not permit dissociation, we hypothesize that molecular  $\text{HNO}_3$  could be stabilized in the defect. Our simulations suggest the dimer is oriented such that the acidic proton of  $\text{HNO}_3$  is pointing away from the gas phase and therefore unable to interact with the minimum number of water molecules needed to stabilize the contact ion pair.<sup>9–11</sup>



**Fig. 9** Representative snapshots taken from the simulations that represent the most probable configuration when any water is present. A side view is shown in (a) and a top view in (b). The atoms are colored as follows: sulfur (yellow), carbon (cyan), oxygen (red), nitrogen (blue), and hydrogen (white). For clarity, only the nearest 5 (a) or 6 (b) alkythiols are shown.

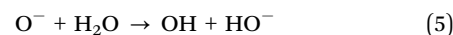
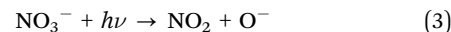
The simulations indicate that  $\text{HNO}_3$  interacts directly with only one isolated water molecule and only sparingly with the remaining water molecules present in the gas phase. It is important to note that although we speculate here that dissociation is unlikely to occur in the defect configuration modeled, this does not preclude the possibility that chemistry may occur with the siloxanes, or that variations in defect size and shape may stabilize contact ion pairs. Indeed, our simulations of other defect sizes and shapes exhibit some variations in the density profiles of both sequestered  $\text{HNO}_3$  and water.

There is additional support for the role of the SAM in trapping  $\text{HNO}_3$  under dry conditions and preventing it from interacting with water vapor. In an experiment in which the SAM was exposed directly to a mixture of  $\text{HNO}_3$  and  $\text{H}_2\text{O}$  corresponding to 50% RH, only the formation of nitrate ions on the surface was observed; there was no evidence of molecular  $\text{HNO}_3$  on the surface. Under these conditions, water is readily available when  $\text{HNO}_3$  is taken up on the surface and dissociation to  $\text{NO}_3^-$  and  $\text{H}_3\text{O}^+$  occurs immediately. No change in the  $-\text{CH}_2-$  stretch absorptions was observed in this case, indicating the SAM was not further perturbed as was the case when dry  $\text{HNO}_3$  was introduced. This suggests that the combination of nitric acid and water leads to rapid dissociation on the surface of the SAM, before  $\text{HNO}_3$  can penetrate into the film.

### Atmospheric implications

In the troposphere, organic thin films are common on surfaces in the boundary layer<sup>38–42</sup> and on airborne particles.<sup>1</sup> These coexist with gas phase nitric acid formed *via*  $\text{NO}_x$  oxidation.<sup>1</sup> Water vapor is also present in significant concentrations, even in relatively dry environments (*e.g.* deserts). Thus, it is likely that the simultaneous deposition of nitric acid and water on organic-coated surfaces or on particles will lead to rapid dissociation of the acid on the surface, rather than migration into the film where it could be trapped as molecular  $\text{HNO}_3$ . In this case, the chemistry and photochemistry of deposited  $\text{HNO}_3$  will be dominated by that of the nitrate ion, rather than molecular  $\text{HNO}_3$ . (Note, however, that hydrates of molecular  $\text{HNO}_3$  were observed from the hydrolysis of  $\text{NO}_2$  in earlier studies on silica surfaces<sup>25</sup> which suggests they could play a

role in the absence of organic coatings.) There is evidence that the photochemistry of the nitrate ions formed on surfaces may differ from that in bulk aqueous solutions. Thus, the quantum yields for photolysis are larger when the ion is at the interface of aqueous solutions where there is an incomplete solvent cage.<sup>107–116</sup> The production of OH radicals *via* the well known photochemistry in bulk aqueous solutions<sup>117,118</sup>



could therefore lead to more efficient oxidation of the surface. On the other hand, one might expect efficient recombination of the photo-fragments of nitrate ions trapped within the SAM, leading to smaller effective quantum yields and less oxidation initiated along the chains. The composition of urban surface films is a complex mixture of organic compounds ranging from short chain aliphatics to long chain fatty acids,<sup>42</sup> and as a result, will be more disordered than a SAM. A more disordered film will have more pores and pockets that allow gases to more readily penetrate the film and be trapped there, increasing the efficiency of recombination of the photo-fragments of nitrate ions and hence trapping of nitric acid molecules.

There may be cases where  $\text{HNO}_3$  is formed inside an organic film and hence contribute to the chemistry and photochemistry, along with nitrate ions. For example, ozone,<sup>119</sup> and likely  $\text{NO}_2$ , can be trapped in organic media. Reaction between  $\text{NO}_2$  and  $\text{O}_3$  forms the nitrate radical which readily abstracts hydrogen atoms from organics to form  $\text{HNO}_3$ .<sup>1</sup> Similarly, uptake and hydrolysis of  $\text{NO}_2$  and  $\text{N}_2\text{O}_5$  form  $\text{HNO}_3$ .<sup>1,94</sup> There may be circumstances where there is sufficient water associated with the organic film for the hydrolysis to occur, but not enough to dissociate the  $\text{HNO}_3$  product. While the photolysis of nitrate ions is reasonably well understood,<sup>117,118</sup> that of adsorbed nitric acid is less certain. Zhu and coworkers,<sup>120–123</sup> for example, report that the absorption cross sections and quantum yields for nitric acid photolysis adsorbed on inorganic surfaces such as silica are much larger than in the gas phase. Whether the same is true for nitric acid on or in an organic film remains to be investigated. Efficient recombination of the initially formed fragments trapped inside a defect/pocket of the organic layer could provide a counter-balancing effect that leads to less, rather than more, efficient photochemistry compared to the gas phase.

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