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Introducing Solubility Control for Improved Organic P-Type Dopants

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7 ABSTRACT: To overcome the poor solubility of the widely 8 used p-type dopant 2,3,5,6-tetrafluoro-7,7,8,8-tetracyanoquinodimethane (F4TCNQ), we have synthesized a series of 9 10 structure-modified, organic p-type dopants to include alkyl ester groups designed to enable solubility and miscibility 11 control. UV-vis-NIR and cyclic voltammetry measurements 12 show increased solubility of mono- and diester substituted 13 dopants with only modest changes to acceptor strength. Using 14 UV-vis-NIR, photoluminescence, and in-plane conductivity 15 measurements, we demonstrate that the new dopants can 16 successfully p-type dope poly(3-hexylthiophene-2,5-diyl) 17 (P3HT). Monoester substituted dopants are characterized by 18 only slightly reduced electron affinity relative to F4TCNQ, but 19 2.0



greater doping effectiveness due to increased miscibility with P3HT. Diester substituted dopants undergo a dimerization reaction

before assuming their doped states, which may help anchor dopants into position post deposition, thus decreasing the negative 21

effect of dopant drift and diffusion. We conclude that increased dopant solubility/miscibility increases the overall effectiveness of 2.2

doping in solution-cast polymer films and that ester modification is a practical approach to achieving solubility/miscibility control 23

24 in TCNQ-type dopants.

INTRODUCTION 25

26 Organic semiconductors have recently drawn much interest 27 because of a number of advantages they have over conventional 28 inorganic materials, which may include low cost, light weight, 29 compatibility with flexible substrates, biocompatibility, low 30 environmental impact, and chemical tailorability.¹⁻⁴ Organic 31 electronic devices, such as organic light-emitting diodes (OLEDs),^{5,6} organic photovoltaics (OPV)^{7,8} and organic 32 33 field-effect transistors (OFETs),^{9,10} show significant improve-34 ments in performance with the addition of dopants. Intrinsic 35 organic semiconductors in general have low free-charge 36 densities and therefore low conductivity compared to inorganic 37 semiconductors. To increase conductivity, conjugated polymers 38 or small molecule semiconductors can be doped via the 39 addition (n-type) or removal (p-type) of electrons. One 40 method of doping a polymer is to form an organic salt using 41 acidic or basic doping groups for p- and n-type doping, 42 respectively.¹¹ Polyethylenedioxythiophene polystyrenesulfo-43 nate (PEDOT:PSS) is a well-known example of an acidically 44 doped p-type organic conductive material.^{12,13} An alternative 45 method to dope organic semiconductors is to add a neutral 46 molecule with an electron affinity (EA) higher than the 47 ionization energy (IE) of the organic semiconductor.¹⁴ In this 48 case, an electron from the semiconductor is spontaneously 49 donated to the high EA molecule to create a hole state, a classic 50 example being the use of iodine to dope polythiophenes.^{15,16}

High EA organic molecular dopants have been studied since 51 the 1960s. Recent studies on such dopants have focused on the 52 role that ground state and excited state charge transfer (CT) 53 states play in quenching excited state fluorescence and in 54 recombination processes in OPV devices.^{17–20} Assuming no 55 CT-state formation, charge transfer occurs when the lowest 56 unoccupied molecular orbital (LUMO) of the dopant is 57 accessible to the highest occupied molecular orbital 58 (HOMO) of the organic semiconductor matrix. The dopant 59 molecule must also have a stable structure that can be 60 reoxidized without reaction with nearby molecules. The 61 quinone structure has these characteristics and early studies 62 of molecular organic dopants focused on tetrachlorobenzoqui- 63 none (chloranil; LUMO: -2.76 eV)²¹⁻²³ and tetracyanoqui- 64 nodimethane (TCNQ; LUMO: -2.8 eV).^{24,25} More recently, 65 dopants with higher EA were prepared by incorporating 66 electron-withdrawing groups into the quinone ring. Dopants 67 in this category include 2,3-dichloro-5,6-dicyano-1,4-benzoqui- 68 none (DDQ; LUMO: -4.6 eV)²⁶⁻²⁸ and the more widely used 69 2,3,5,6-tetrafluoro-7,7,8,8-tetracyanoquinodimethane 70 (F4TCNQ; LUMO: -5.24 eV).^{14,29,30} Following the same 71 concept of molecular design, even higher EA dopants, such as 72 1,3,4,5,7,8-hexafluorotetracyanonaphthoquinodimethane (F6-73

Supporting Information 6

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⁷⁴ TNAP; LUMO: -5.37 eV)^{31,32} and 3,6-difluoro-2,5,7,7,8,8-⁷⁵ hexacyanoquinodimethane (F2-HCNQ; LUMO: -5.59 eV),³³ ⁷⁶ have been recently synthesized. Other recently reported p-type ⁷⁷ dopants with high EA include C₆₀ fullerene (LUMO: -3.6⁷⁸ eV),³⁴ its fluorinated derivative C₆₀F₃₆ (LUMO: -5.38 eV),^{35,36} ⁷⁹ and hexaazatriphenylene hexacarbonitrile (HAT-CN6; LUMO: ⁸⁰ 5.7 eV).^{37–39} However, most of these dopants have low ⁸¹ solubility in common solvents, and are incorporated into ⁸² devices structures using either evaporation processes^{14,40} or ⁸³ premixing in an inert matrix⁴¹ because they cannot be ⁸⁴ effectively solution processed. This limitation significantly ⁸⁵ restricts further development of solution-processed and mass-⁸⁶ produced organoelectronics applications.⁴²

In this paper, we first demonstrate a straightforward synthetic route to soluble versions of F4TCNQ-type dopants by substituting the cyano groups with either methyl or n-octyl 90 esters (Figure 1). A comprehensive study of the electro-



Figure 1. Molecular structures of F4TCNQ and its ester analogues.

91 chemical properties of these F4TCNQ analogues is performed 92 using cyclic voltammetry. With a combination of optical 93 absorption spectroscopy, photoluminescence spectroscopy, and 94 conductivity measurements, we not only demonstrate the p-95 type doping of P3HT using these new dopants, but also show 96 that comparable doping efficiency can be achieved even with 97 slightly reduced electron affinity. These novel molecular 98 dopants establish that the introduction of solubility control is 99 a successful strategy to tailor the properties of organic p-type 100 dopants.

101 EXPERIMENTAL SECTION

102 Materials. 1,4-Bis(chloromethyl)-2,3,5,6-tetrafluorobenzene was purchased from Oakwood Products. Sodium hydride (60% dispersion 103 104 in mineral oil) was purchased from Alfa Aesar. Bromine, 105 perfluorobenzene, potassium carbonate, dimethyl carbonate, malono-106 nitrile, sodium acetate, sodium hydroxide, and phosphorus tribromide 107 were purchased from Sigma-Aldrich. Anhydrous dimethyl sulfoxide 108 (DMSO) and dimethylformamide (DMF) were purchased from Acros 109 Organics. Hydrochloric acid (37% aqueous), p-toluenesulfonic acid 110 monohydrate, ethanol, tetrahydrofuran (THF), dichloromethane (DCM), ethyl acetate, toluene, and hexane were purchased from 111 112 Fisher Scientific. Acetic acid and sodium cyanide were purchased from 113 Fluka. Octanol was purchased from EM Science. Trifluoroacetic acid 114 (TFA) and acetic anhydride were purchased from EMD. P3HT 115 (Regioregular >98%, $M_n = 54-75$ kDa, HOMO 5 eV and LUMO 3 116 eV) was purchased from Sigma-Aldrich. F4TCNQ (>98%) was purchased from TCI. All chemicals were used as received unless 117 118 otherwise indicated. All solvents were dried over molecular sieves (3 119 Å) before use.

¹²⁰ **Characterization.** ¹H NMR (300 MHz), ¹³C NMR (75 MHz) and ¹²¹ ¹⁹F-NMR (282 MHz) spectra were recorded on a Varian Mercury 300

NMR spectrometer. Spectral data were processed using MestReNova 122 (version 6.2.0) desktop NMR data processing software. 123

Cyclic voltammograms (CVs) were recorded on a BASi Epsilon 124 MF-9092 Electrochemical Workstation. The redox potentials of all of 125 substrates were measured in anhydrous acetonitrile (MeCN) solution 126 containing tetramethylammonium tetrafluoroborate (Me₄NBF₄, 127 0.05M) using a platinum disk ($\Phi = 1.6$ mm) as the working electrode 128 and Ag/AgCl as the reference electrode. The concentration of 129 substrates in the working solution was 0.5 mM, and the electro- 130 chemical potential sweep rate was fixed at 100 mV/s. 131

High resolution mass spectra (HRMS) were obtained on a Thermo 132 Fisher Hybrid LTQ-Orbitrap XL mass spectrometer equipped with 133 electrospray. Liquid chromatography-mass spectra (LC-MS) were 134 recorded on Qtrap LC/MS instruments. 135

UV-vis-NIR and photoluminescence spectra were recorded on a 136 PerkinElmer Lambda 750 spectrophotometer and a Varian Eclipse 137 photoluminescence spectrophotometer, respectively. For the solubility 138 measurements, 5 mg of F4TCNQ and 100 mg of the four new dopants 139 were introduced into 1.0 mL of chloroform. The mixtures were stirred 140 on a hot plate at 60 °C for 24 h, and then rested at room temperature 141 for another 24 h. Saturated solutions were diluted accordingly in order 142 to be measured in a useful absorbance range. Calibration curves were 143 also measured for each dopant. For characterization of the blended 144 films, glass substrates were cleaned in ultrasonic baths of acetone, 145 Mucasol detergent (5%), and deionized water, followed by drying with 146 nitrogen. The substrates were then exposed to UV/ozone for 30 min 147 before use. Solutions of 5 mg/mL P3HT and p-type dopants (0.5 mg/ 148 mL for F4TCNQ and 5 mg/mL for the ester derivatives) in 149 chloroform were mixed in appropriate ratios to achieve the desired 150 mole fraction of dopants to P3HT. Except neat F4TCNQ, all other 151 films were spin-coated from blend solutions at 60 °C inside a nitrogen 152 glovebox. Neat F4TCNQ films were deposited using an MBruan 153 thermal evaporator at deposit rate ~0.2 Å/s. Film thicknesses were 154 measured with a Veeco Dektak 150 Surface profilometer. All UV-vis- 155 NIR and photoluminescence spectra were measured under ambient 156 conditions in air. 157

Conductivity measurements were performed with a four-point 158 probe setup using a Keithely 2420 source sourcemeter unit. Four 159 electrodes (5 nm Cr/95 nm Au, 1×5 mm², 1 mm spacing) were 160 deposited through a shadow mask by thermal evaporation. The same 161 procedure was used for substrate cleaning and blended film deposition 162 described above. All conductivity measurements were performed in the 163 dark under a nitrogen atmosphere in a glovebox.

RESULTS AND DISCUSSION

Synthesis of P-Type Dopants. Synthesis of the sym- 166 metrical di(alkoxycarbonyl)dicyanotetrafluoroquinodimethanes 167 F4DMCDCNQ and F4DOCDCNQ starts from commercial 168 1,4-bis(chloromethyl)-2,3,5,6-tetrafluorobenzene 1 (Scheme 169 s1 1). Conversion to the required bis(cyanomethyl) derivative 3 170 s1 does not take place by direct substitution with cyanide, so the 171 intermediate bis(bromomethyl)tetra-fluorobenzene 2 was 172 prepared. Remarkably, direct substitution of 1 with bromide 173 also proves challenging, but 2 could be made via the 174 corresponding (bis)benzyl alcohol. Substitution of 2 with 175 cyanide is facile, and methoxycarbonylation of 3 by 176 deprotonation and reaction with dimethyl carbonate gives 177 diester 4. Oxidation of 4 to the dimethyl ester F4TCNQ 178 analogue F4DMCDCNQ with bromine proceeds in high yield. 179 Alternatively, transesterification of 4 with octanol gave the 180 longer-chain diester 5, which could likewise be readily oxidized 181 to F4DOCDCNQ. 182

The synthesis of the mono(alkoxycarbonyl)- 183 tetrafluorotricyanoquinodimethanes F4MCTCNQ and 184 F4OCTCNQ starts from commercial hexafluorobenzene 185 (Scheme 2). Reaction with methyl cyanoacetate in the presence 186 s2 of a base gives the intermediate methyl 2-cyano-2- 187

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Scheme 1. Synthesis of F4DMCDCNQ and F4DOCDCNQ^a



^aReagents and conditions: (a) NaOAc, Ac₂O, AcOH, 105 °C, 36 h; (b) NaOH, EtOH/THF/H₂O, 75 °C, 24 h; (c) PBr₃, DCM, 10 h, 76% over 3 steps; (d) NaCN, CF₃COOH, DMSO, 5 h, 51%; (e) (MeO) ₂CO, NaH, THF, 75 °C, 18 h, 83%; (f) Br₂, H₂O, 2 h, 94%; (g) octanol, TsOH·H₂O, toluene, 110 °C, 6 h, 72%; (h) Br₂, H₂O, 3 h, 93%.

188 (pentafluorophenyl)acetate, which is decarboxylated in situ to 189 give 2-(pentafluorophenyl)ethanenitrile **6**. S_NAr substitution 190 with the sodium salt of malononitrile provides 7, which is 191 methoxycarbonylated by reaction with dimethyl carbonate and 192 base to give **8**. Oxidation of **8** with bromine gives 193 **F4MCTCNQ**, the monomethyl ester analogue of F4TCNQ. 194 Transesterification of **8** with octanol followed again by 195 oxidation gives the mono-octyl ester dopant **F4OCTCNQ**. 196 Detailed synthetic procedures are provided in the Supporting 197 Information.

Characterization of P-Type Dopants. Our dopant design 198 199 goal was to improve the solubility of the dopant molecules 200 without compromising doping efficiency. To demonstrate the 201 improved solubility of the new F4TCNQ analogues, saturated 202 solutions were prepared in chloroform (Figure 2a) and their absorbance spectra were recorded by UV-vis-NIR at room 2.03 temperature. The solubility limits (Figure 2b) were then 204 determined based on calibration curves of each compound. The 205 details of the calibration curves and measurements can be found 206 207 in the Figure S47. As can be seen from Figure 2b, the solubility of the ester-substituted dopants is significantly greater than 208 209 F4TCNQ itself, by a factor of at least 55 for F4MCTCNQ and 210 100 for F4OCTCNQ. Interestingly, although the solubility of 211 the diesters are still \sim 30 times higher than that of F4TCNQ, 212 they are less soluble than the monoesters, which is attributed to 213 the higher symmetry of the diesters.⁴³

²¹⁴Because the EA of the dopant is an important predictor of ²¹⁵doping efficiency, the next thing we needed to show is what



^aReagents and conditions: (a) MeOCOCH₂CN, K₂CO₃, DMF, 115 °C, 6 h, 72%; (b) 50% aq HOAc, conc. H₂SO₄, reflux, 15 h, 75%; (c) CH₂(CN)₂, NaH, THF, 70 °C, 18 h, 66%; (d) (MeO)₂CO, NaH, THF, 80 °C, 48 h, 89%; (e) Br₂, H₂O, 1 h, 82%; (f) octanol, TsOH-H₂O, toluene, 85 °C, 9 h, 71%; (g) Br₂, H₂O, 1.5 h, 81%.

effect substitution of an ester group for a cyano group on the 216 F4TCNQ structure would have on the reduction potential. The 217 electrochemical states of the dopants were investigated using 218 cyclic voltammetry (CV). Solutions of dopants in acetonitrile 219 were measured under reducing potentials and the CV data are 220 shown in Figure 3a. As expected, F4TCNQ itself undergoes 221 f3 two reversible one-electron reduction steps corresponding to 222 F4TCNQ^{•-} and F4TCNQ²⁻. The monoester-substituted 223 dopants F4MCTCNQ and F4OCTCNQ also exhibit two 224 reduction peaks, but interestingly, they are located between the 225 first and second reductions of F4TCNQ, and the first reduction 226 peak $E_{\rm red1}$ of the monoester-substituted dopants is greater than 227 the second reduction peak E_{red2} . Unlike F4TCNQ, only one 228 reduction peak is seen in the CV of diester-substituted dopants. 229 Closer inspection of the CV data reveals that both the mono 230 and diesters undergo redox processes that are either irreversible 231 or quasi-reversible. Therefore, instead of using the mean values 232 of anodic and corresponding cathodic peak potentials $(E_{1/2})$, 233 reduction potentials were used for comparison. The reduction 234 potentials for all dopants vs Ag/AgCl are summarized in Table 235 t1 236 t1

As can be seen from Table 1, E_{red1} shows a decreasing trend 237 from 0.51 to ~0.4 and ~0.2 V as cyano groups are substituted 238 by one or two ester groups, respectively, showing that 239 F4TCNQ is a stronger p-type dopant than its ester analogues. 240 This is expected since the cyano group is more electron 241 withdrawing than the ester group, which results in a more 242 electron deficient quinoid ring in the F4TCNQ than in the 243

f2



Figure 2. (a) Photos of saturated solutions (from left to right: F4TCNQ, F4MCTCNQ, F4OCTCNQ, F4DMCDCNQ, and F4DOCDCNQ) in chloroform. (b) Measured solubility limits of the dopants in chloroform at room temperature.

Table 1. Summary of Reduction Potentials from CV and Determined LUMO Levels of F4TCNQ and Its Ester Derivatives

compd	$E_{\rm red1}(V)$	$E_{\rm LUMO1}(\rm eV)$	$E_{\rm red2}(V)$	$E_{\rm LUMO2}~({\rm eV})$
F4TCNQ	0.51	-5.23	0	-4.72
F4MCTCNQ	0.42	-5.14	0.12	-4.84
F4OCTCNQ	0.40	-5.12	0.11	-4.83
F4DMCDCNQ	0.25	-4.97		
F4DOCDCNQ	0.21	-4.93		

ester-substituted derivatives. Comparing the octyl and methyl 244 ester substituted dopants, subtle differences in $E_{\rm red1}$ are 245 observed. Longer alkyl chains are more electron donating so 246 the $E_{\rm red1}$ values of the methyl ester-substituted dopants are 247 slightly greater than those of the octyl ester derivatives, 248 increasing from 0.40 to 0.42 V and from 0.21 to 0.25 V, for the 249 monoester and diester derivatives, respectively. 250

On the basis of the CV data, the LUMO levels for each 251 dopant can be determined from their reduction potentials 252 according to following relationship: $E_{\rm LUMO} = -e(E_{\rm red} + E_{\rm ref})$, 253 where $E_{\rm ref}$ is 4.72 V for the Ag/AgCl reference electrode used in 254 our measurements.^{33,44,45} The measured LUMO energies are 255 also listed in Table 1. The measured LUMO level of F4TCNQ 256 (-5.23 eV) is in good agreement with the published values 257 from CV (5.33 eV)³³ as well as ultraviolet photoelectron 258 spectroscopy (UPS) (5.24 eV),⁴⁶ which verifies the accuracy of 259 our assignments. 260

To enable a visual comparison, the LUMO levels of all 261 investigated p-type dopant molecules are compiled into one 262 energy diagram (Figure 3b). The HOMO and LUMO levels of 263 P3HT and P3HT positive polaron (P3HT⁺) reported from 264 literature^{15,47–49} are also shown to scale. Further discussion of 265



Figure 3. (a) Cyclic voltammograms of F4TCNQ and its monoester derivatives (top) and diester derivatives (bottom). Reduction potentials of F4TCNQ (F1, F2), monoester derivatives (M1, M2), and diester derivatives (D1) are marked. (b) Molecular orbital energy level diagram of P3HT (left: solid line for neutral P3HT and dashed line for P3HT polaron) and p-type dopants (right).

266 the P3HT matrix/dopants system is deferred to the next 267 section, as we focus here on the comparison of LUMO levels of 268 the dopants. Similar to F4TCNQ, E_{red1} and E_{red2} for the ester 269 analogues can be assigned to the formation of anion radicals 270 from the neutral molecules and the formation of dianions from 271 the anion radicals, respectively. Therefore, we assign singly and doubly reduced dopants as compound^{•-} and compound²⁻, 272 respectively. Among the compounds considered, F4TCNQ 273 shows the deepest LUMO level, F4MCTCNQ^{•-} and 274 275 F4OCTCNQ^{•-} show the next deepest levels, and 276 F4DMCDCNQ^{•-} and F4DOCDCNQ^{•-} show relatively shallow levels. In addition, doubly reduced dopants show 277 weaker reduction strength (smaller EA) and thereby shallower 278 LUMO levels than singly reduced dopants, as expected. 279 280 Another highlight of the CV measurements is that there is only one reduction peak for diester-substituted dopants. Thus, 281 the LUMO levels of F4DMCDCNQ²⁻ and F4DOCDCNQ²⁻ 282 are not shown because these species do not form based on the 283 CV data. This suggests that the diester-substituted dopants 2.84 285 undergo a different chemical process during reduction than the 286 F4TCNQ and monoester-substituted dopants, the details of 287 which will be discussed below. Finally, only subtle differences in 288 LUMO levels are found between the methyl ester and octyl 289 ester analogues, merely 0.01-0.04 eV, indicating that the R 290 group on the ester does not strongly influence the electronic nature of the TCNQ dopants. 291

A mechanism is proposed to explain the charge transfer processes of F4TCNQ and ester-substituted dopants, as illustrated schematically in Figure 4. Assuming only one selectron is involved at each step, radical anions are formed



Figure 4. Proposed mechanism for charge transfer for F4TCNQ and its derivatives. The -R group can be a methyl or octyl group. Resonance structures for the radical anions are also displayed.

from neutral molecules in the first reduction. Resonance 296 structures are also shown. Due to the molecular symmetry of 297 the F4TCNQ and the diester molecules, the two resonance 298 structures of their anion radicals are identical, whereas for the 299 monoester molecules, the spin and negative charge density may 300 be asymmetrically distributed. By accepting a second electron, 301 dianions are formed subsequently for F4TCNQ and the 302 monoesters. However, this is not the case for the diester 303 molecules. Instead, radical coupling takes place to give a 304 dimeric dianion. Only one reduction potential is observed, 305 suggesting that the radical coupling is rapid and irreversible. 306

To support the above hypothesis, we measured the masses of 307 the fully reduced dopant molecules using mass spectrometry 308 (MS). F4TCNQ and all four ester-substituted analogues were 309 mixed with potassium iodide (KI) to ensure full reduction, and 310 the resulting mass spectra are shown in the Figures S37–S46. 311 From these data, it is clear that diester dopants form dimers 312 after reductionk while monoester dopants and F4TCNQ do 313 not. The structure of the dimer shown in Figure 4 is a proposal 314 based on the most likely site for spin coupling, involving no 315 interference with the aromaticity of the benzene rings and with 316 maximum possible separation of the charges. However, for the 317 purposes of this work, the exact structure of the dimer is of no 318 real consequence. 319

Characterization of Doped Films. It has been shown in 320 the above sections that F4TCNQ can be synthetically altered to 321 incorporate esters in place of nitrile groups, where the ester can 322 support different alkyl chains that can be used to affect the 323 solubility of the molecule. In this section we demonstrate 324 successful p-type doping of the well-characterized conjugated 325 polymer P3HT by the above-described ester-substituted 326 dopants. In particular, we investigate the effect of altering the 327 solubility of the dopant on the charge density and conductivity 328 of the polymer. 329

Measuring Charge Density. Absorption spectroscopy (UV– 330 vis–NIR) is commonly used to observe changes in the 331 crystalline to amorphous content of P3HT samples.^{50,51} The 332 doping level can also be monitored using UV–vis–NIR.³⁰ 333 Grazing incidence X-ray spectra and concentration-dependent 334 conductivity were recently measured for the P3HT/F4TCNQ 335 system.⁵² For doping levels below 4 mol %, F4TCNQ occupies 336 locations within the amorphous part of the P3HT⁵² but dopes 337 exclusively in the crystalline domains.²⁰ Between 4 mol % and 338 17 mol % doping ratios, F4TCNQ intercalates between the 339 P3HT chains within crystalline domains, increasing the (010) 340 crystal spacing (along the π -stacking direction) from 3.83 341 to7.18 Å.⁵² The 17 mol % blend ratio is the point when this 342 blend system reaches the upper limit on both conductivity and 343 low energy absorptivity (1–1.7 eV) according to literature.⁵² 344

The normalized UV–vis–NIR absorption spectra for P3HT/ 345 dopant at selected doping ratios are shown in Figure 5. The full 346 fs doping concentration series of UV–vis–-NIR spectra can be 347 found in the Supporting Information (Figure S48). These data 348 exactly reproduce the literature results for P3HT/F4TCNQ. 52 349 Undoped P3HT has a band gap of ~1.9 eV. Upon oxidation by 350 F4TCNQ, two broad sub-bandgap absorbance peaks centered 351 around 0.4 and 1.5 eV are observed, which can be assigned to 352 the polarons of P3HT. 53,54 In addition, two sharper peaks at 353 1.43 and 1.62 eV can be detected that correspond to 354 absorbance by F4TCNQ^{•–}. As shown in Figure 5, subgap 355 subsorptions similar to those of F4TCNQ^{•–} are seen in 356 F4MCTCNQ^{•–} and F4OCTCNQ^{•–}. The double peaks at 357 1.43 and 1.62 eV are replaced by a single absorbance at the 358



Figure 5. Normalized UV–vis–NIR absorbance spectra of neat P3HT (black), dopants (magenta dashed line), and P3HT/dopant blend films at 3 mol % (red) and 17 mol % (blue) doping ratios. Arrows show the dedope peak for P3HT/dopant system.

359 same energy that increases with increased dopant concen-360 tration. In fact, the NIR absorbance is stronger at both 0.4 and 361 1.5 eV in films doped with the monoester dopants than with 362 F4TCNQ at the same mol % doping concentration. For 363 instance, at 17 mol % the ratios between polaron absorbance at 364 1.5 eV to ground-state absorbance at 2.3 eV are 1:1, 1.04:1, and 365 1.12:1 for F4TCNQ, F4MCTCNQ and F4OCTCNQ, 366 respectively. The increased polaron/ground-state ratio is an indication that higher P3HT polaron density is achieved for 367 monoester derivatives. Conversely, only a slight increase in 368 P3HT polaron peaks is observed in films containing 369 370 F4DMCDCNQ and F4DOCDCNQ, even as the doping concentration reaches 17 mol %. This result indicates that 371 372 there is less charge transfer between P3HT and the diester dopants compared with the F4TCNQ and monoester dopants 373 at the same doping concentration. 374

Changes in the short-range ordered regions (aggregates)^{55,56} 375 of P3HT can be probed by monitoring the absorbance ratio of 376 the 0-0 to 0-1 vibronic peaks at 2.03 eV (610 nm) and 2.25 377 eV (550 nm) because these peak are associated with interchain-378 delocalized excitation^{47,57} and the degree of P3HT order-379 ing.^{55,58,59} The concentration-dependent spectra in the Figure 380 S48 show that the relative intensity of the peak at 610 nm 381 382 compared to the other P3HT vibronic peaks increases with 383 increasing doping concentration up to 10 mol %. This result 384 implies one of two things. It could indicate that crystalline 385 domains of P3HT are becoming more ordered (more planar) 386 upon doping. Alternatively, the relative increase in the peak at 387 2.03 eV could indicate an increased probability of doping in less 388 crystalline domains. The ground-state signal is bleached upon

doping and only the most crystalline sections of the sample are 389 isolated from contact with the dopants. A relative increase in 390 the peak at 610 nm occurs in all P3HT/acceptor systems, 391 although the diester dopants induce less pronounced shifts in 392 the vibronic spectra. 393

Besides the characteristics of the doping effects mentioned 394 above, UV-vis-NIR spectra for all neat dopants are also 395 presented. The absorbance spectra of these five dopants do not 396 show many differences other than the broader and red-shifted 397 absorption peak for F4TCNQ. The blueshift of the absorption 398 peak from F4TCNQ (~2.8 eV) to the monoester derivatives 399 $(\sim 3.3 \text{ eV})$ and diester derivatives $(\sim 3.4 \text{ eV})$ is expected due to 400 less delocalization of charge resulting from replacing cyano with 401 ester groups. The appearance of an additional peak in the high 402 energy range of blend systems is also noted, as indicated by 403 arrows (Figure 5). These peaks are assigned to the dopant 404 anion, although in monoester systems an overlapping of the 405 monoester anion absorbance and neutral dopant molecule 406 absorbance is observed. A recent study from our group showed 407 that this optical transition can be used to dedope P3HT/ 408 F4TCNQ films with complete recovery of optical character- 409 istics and mobilities.⁶⁰ The related transitions for ester- 410 substituted dopants are blue-shifted with respect to F4TCNO 411 by ~0.5 eV. The UV-vis-NIR spectra of the dopant anions 412 can be found in the Figure S49. 413

The differential solubility of the mono- and diester dopants 414 with methyl and octyl chains enables us to evaluate the effect of 415 dopant solubility on doping efficiency, with the long chain 416 esters showing higher solubility. Comparison of the $P3HT^+/$ 417 dopant⁻ CT-state absorbance features to the ground-state 418 absorbance of bulk P3HT gives a direct measure of the charge 419 density. The results in this section clearly indicate that a dopant 420 that is more miscible in the polymer has a higher probability of 421 doping the polymer.

Measuring Dopant Miscibility. Photoluminescence spec- 423 troscopy (PL) is another technique that has been extensively 424 used to evaluate doping interactions in organic materials 425 because photoluminescence is quenched by the presence of 426 dopants. 53,61 Dopants increase the free charge density in the 427 polymer and free charges react with and quench excitons 428 through a dark process with $\sim 1/r^3$ distance dependence.⁶² PL 429 experiments were performed using 2.4 eV excitation to optically 430 excite P3HT and the PL emission for undoped P3HT (1.4-2.0 431 eV) was observed. Figure 6 shows the PL intensity as a function 432 f6 of mol fraction for all dopants. The inset shows typical PL 433 spectra for neat and doped P3HT at 1 mol % doping ratio. As 434 can be seen, F4DMCDCNQ and F4DOCDCNQ show only 435 partial PL quenching of ~20% and ~40%, respectively. 436 F4TCNQ is a stronger exciton quencher (~80%), whereas 437 monoester dopants F4MCTCNQ and F4OCTCNQ are able to 438 fully quench PL. This result clearly indicates that 20% of the 439 P3HT volume remains undoped by F4TCNQ at 1 mol % 440 doping ratio, while all P3HT domains are infiltrated by 441 monoester-substituted dopants at the same concentration. The 442 full concentration series of PL spectra can be found in the 443 Figure S48. Unlike the monoester dopants where complete 444 quenching occurs at lower than 1 mol % doping, complete 445 quenching for diester dopants is observed at around 17 mol %. 446 This is consistent with dimerization of the dopants, which 447 causes charging of P3HT at higher dopant concentrations. 448

To summarize, on the basis of the reduction potential of the 449 dopants, F4TCNQ should be a more effective dopant than the 450 mono- and diester dopants by 0.2 and 0.4 eV, respectively. 451



Figure 6. Normalized photoluminescence intensity as a function of doping concentration for all P3HT/dopant systems. Inset: PL quenching at 1 mol % doping ratio. The full doping concentration series of spectra can be found in the Supporting Information.

452 However, UV-vis-NIR results indicate that monoester-453 substituted TCNQs generate more charge density in P3HT 454 at the same doping ratio. PL quenching demonstrates that 455 monoester dopants infiltrate all domains of P3HT at a lower 456 concentration than F4TCNQ. These results are in consistent 457 with the increased solubility of dopants in CHCl₃.

458 **Conductivity.** In this section, we describe concentration-459 dependent conductivity measurements that were performed to 460 determine whether the increase in dopant miscibility translates 461 to enhanced electronic properties. Figure 7a shows a log–log 462 plot of conductivity versus mole fraction of dopant for P3HT 463 doped with F4TCNQ and each of the new dopants. Blend films at different doping ratios were spin-cast on prepatterned 464 substrates and then measured by the four-point probe method, 465 the details of which have been described in earlier 466 publications.^{63,64} It is clear that the in-plane conductivity of 467 P3HT can be tuned by at least 3 orders of magnitude by all 468 dopants. All dopants also show a similar trend of increased 469 conductivity with increasing dopant concentration. After 470 reaching a peak, the addition of more dopant leads to an 471 increase in neutral dopant domains and conductivity decreases 472 with increased concentration. The upper limit of conductivity 473 in P3HT when doped with F4TCNQ or monoester dopants is 474 ~10° S/cm while for diester dopants the maximum 475 conductivity is ~1 × 10⁻¹ S/cm. 476

Over most of the doping range, the conductivity of P3HT 477 doped with the monoester dopants is higher at the same mole 478 fraction than P3HT doped with F4TCNQ, which is consistent 479 with UV-vis--NIR and PL measurement results. For 480 comparison to the earlier work of Duong et al.,⁵² the 481 conductivity data are also plotted on a log linear scale (Figure 482 7b). Just as was seen for P3HT doped with F4TCNQ, there is a 483 clear change in the slope of conductivity vs concentration for all 484 dopants at around 4 mol % dopant loading. We assume here 485 that the assignments made by Duong et al. are correct in that 486 the initial high slope comes from dopants mixing into the 487 amorphous P3HT domains.^{20,52} At higher doping ratios, the 488 dopants intercalate into the crystalline P3HT domains, 489 increasing the crystal spacing between P3HT chains. The 490 initial slopes for loading of dopant up to 4 mol % are identical 491 for the monoester dopants and F4TCNQ. However, when the 492 doping ratio is increased beyond 4 mol %, F4OCTCNQ in 493 particular shows a larger increase in conductivity with increased 494



Figure 7. (a) In-plane conductivity measurements of P3HT/dopant blend films as a function of doping concentration. The neat P3HT film conductivity is also shown (orange). (b) Fits to the conductivity data in the weak (dashed lines) and strong (solid lines) doping regimes for F4TCNQ and the monoester derivatives (top) and diester derivatives (bottom).

495 doping than F4TCNQ, confirming that the modified dopant is 496 more efficient at intercalating between P3HT chains. This 497 result demonstrates that increased miscibility of the dopant has 498 a clear advantage for doping efficiency. For F4TCNQ, the peak 499 in conductivity occurs at 17 mol % doping followed by a rapid 500 drop in conductivity with increased dopant concentration, 501 presumably because increased dopant content is neutral and 502 pure F4TCNQ crystals are formed in the polymer.⁵² By 503 comparison, both monoester dopants show a much broader 504 peak in conductivity with increased dopant concentration, 505 which also argues for better miscibility between the dopant and 506 P3HT.

Another interesting observation is that the conductivity of 507 508 P3HT samples doped with diester dopants increases exponen-509 tially up to 40 mol % doping ratio, whereas the F4TCNQ and 510 monoester dopants show an exponential increase in con-511 ductivity until 10 mol % for F4OCTCNO and 17 mol % for 512 F4MCTCNQ and F4TCNQ, respectively. This is consistent 513 with the UV-vis-NIR results for the diesters (Figure S48) 514 which show that P3HT polaron density continues to increase 515 with up to 64 mol % dopant loading. This trend cannot be 516 explained solely by the relatively shallow LUMO level of 517 diesters. Instead, the increase in conductivity with such high 518 doping levels implies that the dimerization reaction stabilizes 519 the doped states. The conductivity does not increase until there 520 is high enough dopant concentration that they are able to 521 dimerize and form stable dianions. This dimerization reaction 522 shifts equilibrium toward the dianion, which is trapped by the 523 irreversibility of the process and then cannot give the electron 524 back into the polymer. Although it may be considered disadvantageous that large doping ratios are necessary to 525 526 achieve higher conductivity with the diesters, there is a possible processing advantage to the dimerization reaction. This 527 528 advantage is that the large dianion is unlikely to diffuse with 529 thermal stress or drift with electric field stress. The smaller 530 F4TCNQ is known to diffuse against a concentration gradient 531 and thereby to dope regions of the sample outside of the 532 originally intended area, thus reducing device lifetimes.³⁵ In 533 addition, because the diester-substituted dopants do not 534 dimerize until after deposition, they can be efficiently mixed 535 into the polymer and then doping can occur post film 536 formation through the dimerization reaction. The dopants, 537 once dimerized, will provide a much higher barrier to diffusion.

538 CONCLUSIONS

539 In conclusion, we have developed a method to introduce 540 solubility control into p-type organic dopants based on the 541 F4TCNQ structure. Solubility control is achieved by replacing 542 cyano groups on the F4TCNQ framework with ester groups. 543 The results show that the solubility of the new ester dopants is 544 improved by a factor of at least 30 and up to 100 compared 545 with F4TCNQ at room temperature. In addition, these new 546 dopants possess a range of interesting properties. First, 547 monosubstituted ester dopants show only slightly lower $(\sim 0.2 \text{ eV})$ electron affinity than F4TCNQ. However, in spite 548 549 of the lower EA, monoester dopants lead to increased polaron 550 density and more efficient doping in P3HT than F4TCNQ. 551 F4OCTCNQ, for instance, has the same conductivity at 10 mol 552 % doping as F4TCNQ at 17 mol % doping. The increased 553 doping efficiency is proposed to be a result of the ester-554 substituted dopants being both more soluble in good solvents 555 for P3HT and more miscible in P3HT itself. This study not 556 only shows that molecular tailoring of organic dopants for

increased miscibility with the desired polymer is a sound 557 approach to improving doping efficiency, but also that single 558 ester replacements are a viable method to modify common 559 dopant structures with cyano attachments. Additionally, we also 560 introduce diester-substituted dopants, which have distinctly 561 different electrochemical properties than F4TCNQ and the 562 monoester dopants due to a dimerization reaction that occurs 563 between their radical anions. Despite the shallow LUMO level 564 of the diesters, this irreversible dimerization also results in 565 effective doping. However, much higher doping levels are 566 needed for these diester dopants to maximize conductivity in 567 P3HT, as shown by increased polaron density in optical 568 spectra. Viewed broadly, these results provide a pathway for 569 future dopant molecule design with controlled solubility/ 570 miscibility as a synthetic design parameter. Future studies will 571 include a variety of ester -R groups that could tailor the 572 placement of molecular dopants with respect to particular 573 conjugated molecules/polymers. 574

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S Supporting Information 576

The Supporting Information is available free of charge on the 577 ACS Publications website at DOI: 10.1021/acs.chemma- 578 ter.5b02340. 579

Detailed synthetic procedures and NMR, mass, UV–vis, 580 and photoluminescence spectra (PDF) 581

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