

Communication

¹ Oxidation State and Surface Reconstruction of Cu under CO₂ ² Reduction Conditions from *In Situ* X-ray Characterization

³ Soo Hong Lee,^{||} John C. Lin,^{||} Maryam Farmand, Alan T. Landers, Jeremy T. Feaster,
⁴ Jaime E. Avilés Acosta, Jeffrey W. Beeman, Yifan Ye, Junko Yano, Apurva Mehta,* Ryan C. Davis,*
⁵ Thomas F. Jaramillo,* Christopher Hahn,* and Walter S. Drisdell*



6 ABSTRACT: The electrochemical CO_2 reduction reaction (CO_2RR) using Cu-based catalysts holds great potential for producing 7 valuable multi-carbon products from renewable energy. However, the chemical and structural state of Cu catalyst surfaces during the 8 CO_2RR remains a matter of debate. Here, we show the structural evolution of the near-surface region of polycrystalline Cu 9 electrodes under *in situ* conditions through a combination of grazing incidence X-ray absorption spectroscopy (GIXAS) and X-ray 10 diffraction (GIXRD). The *in situ* GIXAS reveals that the surface oxide layer is fully reduced to metallic Cu before the onset potential 11 for CO_2RR , and the catalyst maintains the metallic state across the potentials relevant to the CO_2RR . We also find a preferential 22 surface reconstruction of the polycrystalline Cu surface toward (100) facets in the presence of CO_2 . Quantitative analysis of the 13 reconstruction profiles reveals that the degree of reconstruction increases with increasingly negative applied potentials, and it persists 14 when the applied potential returns to more positive values. These findings show that the surface of Cu electrocatalysts is dynamic 15 during the CO_2RR , and emphasize the importance of *in situ* characterization to understand the surface structure and its role in 16 electrocatalysis.

opper is known as the only monometallic heterogeneous 17 ✓ electrocatalyst that can convert CO₂ into more valuable 18 19 multi-carbon products.¹ To produce C₂₊ products at 20 economically feasible reaction rates requires a significant $_{21}$ overpotential for CO₂RR.² During the CO₂RR, the structure 22 and chemical state of the Cu surface has shown dynamic 23 changes in response to the local environment and applied 24 potential.³ The Cu valence state at the surface is of particular 25 interest since recent computational studies have shown that the 26 presence of subsurface oxygen or an oxide phase can improve $_{27}$ CO₂RR activity and steer the selectivity into C₂₊ products.^{4–7} 28 Based on such a hypothesis, several studies have tried to probe 29 the oxidation state of Cu-based catalysts by utilizing ex situ and 30 in situ/operando spectroscopy or microscopy.⁸⁻¹² However, 31 these studies have not reached a consensus on whether the 32 oxide phase is present and stable at the catalyst surface under $_{33}$ realistic operating conditions where the CO₂RR performance is 34 measured. The rapid reoxidation of the Cu surface upon 35 exposure to O₂ makes it difficult to reach a robust conclusion 36 on this topic, and most of the in situ/operando techniques 37 employed struggle to distinguish whether the oxygen is located 38 at the surface or buried in the bulk. Therefore, measurements 39 that can specifically probe the active catalytic surface to 40 determine the oxidation states of Cu present under CO₂RR 41 rates similar to those used for performance evaluation are 42 needed.

⁴³ In addition to the Cu valence state, the surface can ⁴⁴ dynamically reconstruct through its interaction with the local ⁴⁵ environment.^{13,14} Since Cu has low cohesive energy and high ⁴⁶ surface mobility, the Cu atoms at the surface can easily

migrate. CO, which is a key intermediate in the CO₂RR, has 47 been shown to exacerbate this reconstruction in near-ambient 48 pressure conditions.¹⁵ Surface reconstructions can affect 49 product selectivity because the Cu(111) surface preferentially 50 yields CH_4 , whereas the Cu(100) surface produces C_2H_4 with 51 a lower onset potential.¹⁶ To probe the surface structure under 52 CO₂RR conditions, electrochemical scanning tunneling mi- 53 croscopy (ECSTM) has been utilized to image Cu surfaces 54 with atomic resolution and has successfully demonstrated that 55 polycrystalline Cu (hereafter referred to as Cu(pc)) 56 reconstructs into Cu(100) surfaces in N₂-purged electrolytes.¹⁷ 57 However, one of the limitations of ECSTM is its limited field 58 of view, and it is unclear whether these changes occur globally. 59 Therefore, to understand the structural dynamics of Cu 60 surfaces more fully, it is imperative to elucidate both the local 61 atomic structure and long-range order under realistic CO₂RR 62 conditions. Here, we characterize the near-surface structure of 63 a Cu(pc) thin film (50 nm thickness) under CO₂RR $_{64}$ conditions by utilizing in situ grazing incidence X-ray 65 absorption spectroscopy (GIXAS) and X-ray diffraction 66 (GIXRD). The Cu(pc) thin film is utilized as an electrocatalyst 67 because it has been demonstrated that the roughness of the Cu 68 thin film is low enough to allow sensitivity to a few nm of the 69

Received: September 18, 2020

73 densities by allowing a grazing incidence geometry and 74 improving mass transport of reactive species to the catalyst.¹⁹ 75 Our results confirmed that no detectable oxide remains at the 76 near-surface at potentials relevant to the CO₂RR, and, 77 simultaneously, that potential-dependent surface reconstruc-78 tion from Cu(pc) to Cu(100) can be observed across μ m-mm 79 areas of the surface in a CO₂ saturated electrolyte.

We first characterized as-prepared Cu(pc) films by using 80 81 GIXRD, GIXAS, and X-ray photoelectron spectroscopy 82 (Figures S1 and S2), and the probe depths were calculated 83 based on the incidence angles (Figure S3). The CO₂RR 84 performance of the Cu(pc) thin film is similar to that of Cu foil 85 reported previously (Figure S4).²⁰ To investigate the changes 86 of oxidation states and local atomic structure during the 87 CO₂RR, we performed in situ GIXAS measurements on the Cu 88 thin film in a 0.1 M KHCO₃ electrolyte saturated with CO₂. 89 The total current densities exceeded 20 mA/cm² at -1.1 V vs 90 RHE (all the potentials are hereafter noted vs RHE). 91 conditions under which Cu is known to form CO₂RR products 92 and which are often employed when reporting CO2RR 93 efficiencies (Figure S5). The XANES spectrum at open-circuit 94 potential (OCP) showed rising edge peaks at both 8982 and 95 8996 eV, indicating that the surface was composed of a mixture 96 of metallic Cu and Cu₂O (Figure 1a). These Cu(I) features 97 disappeared at -0.3 V, which is before the onset potential of 98 $CO_2 RR$ on Cu(pc) (generally ~-0.63 V).²¹ Over the applied

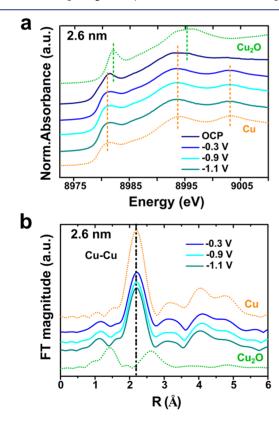


Figure 1. In situ GIXAS of Cu(pc) thin-film electrode in CO₂saturated 0.1 M KHCO₃. XANES (a) and EXAFS (b) spectra of Cu (pc) at a probe depth of 2.6 nm as a function of the applied potentials. The Cu (orange) and Cu₂O (green) reference spectra (dotted line) are plotted for comparison.

potential range relevant to the CO_2RR , we observed only 99 metallic Cu at the surface. The corresponding extended X-ray 100 absorption fine structure (EXAFS) spectra displayed a 101 prominent Cu–Cu scattering peak at 2.2 Å, which matches 102 with the metallic Cu reference (Figure 1b). Although angle-103 dependent overabsorption distorted and dampened XANES 104 and EXAFS spectra,¹⁹ it did not affect the phase information 105 (Figure S6). We obtained similar results in the Ar-purged 106 phosphate and bicarbonate electrolytes, suggesting that the 107 reduction of oxide is primarily driven by the polarization to 108 negative potential (Figure S7). As the GIXAS is highly 109 sensitive to the disordered and local surface structure, our 110 results demonstrate that an oxide phase in the near-surface 111 region was either not present or below our ability to measure. 112

To elucidate the crystalline structure during CO₂RR, we 113 performed in situ GIXRD as a function of probe depth and 114 applied potential in CO₂- and Ar-purged electrolytes (Figure 115 S8). The obtained diffraction measurements were corrected for 116 refraction at the electrode-electrolyte interface (Figure S9a,b) 117 and fit with a pseudo-Voigt function without contribution from 118 electrolytes (Figure S9c). To estimate the uncertainty in 119 determining the Cu(111) peak position, we calculate errors 120 based on the peak asymmetry and sample variations (Figure 121 S10). In the case of the as-synthesized Cu thin film, the d- 122 spacing of Cu(111) showed a 0.15% expansion near the surface 123 compared to that of bulk, possibly originating from the 124 existence of the oxide phase (Figure S11). However, at 125 potentials where the CO₂RR occurs, no changes larger than the 126 estimated error in the Cu(111) d-spacing values were observed 127 depending on the applied potential and probe depth (Figure 128 S12a). The result with Ar-purged electrolytes is quite similar, 129 except for 0.14% surface expansion at -1.1 V, possibly through 130 hydrogen-induced expansion (Figure S12b).²² We note that 131 the metallic Cu surface rapidly oxidized into Cu₂O after 132 releasing the applied potential (Figure 2). The Cu₂O phase 133 f2

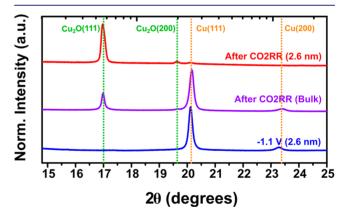


Figure 2. GIXRD of Cu(pc) at a probe depth of 2.6 nm and bulk before and after releasing the applied potential.

was even detected in the bulk within 5 min, demonstrating that 134 the metallic Cu electrode can easily reoxidize at the OCP 135 (generally ~ 0.55 V in our case). This result suggests that *ex* 136 situ or quasi in situ characterization could contain Cu(I) 137 artifacts on the sample surfaces even without exposure to 138 ambient conditions.

While we observe that the valence state and *d*-spacing of the 140 Cu surface remain unchanged, we do find evidence for surface 141 reconstruction as a function of applied potential. We calculate 142 f3 the area ratio of Cu(200) to Cu(111) Bragg peaks at a probe 143

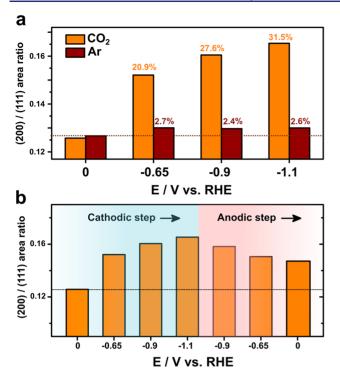


Figure 3. (a) Ratio of the area of Cu(200) to Cu(111) Bragg peaks at a probe depth of 2.6 nm in CO_{2^-} and Ar-purged electrolytes as a function of the applied potentials. Each percentage above the bars indicates an increase of ratio compared to the value at 0 V. The dashed line indicates the ratio in Ar-purged electrolyte at 0 V. (b) Changes of Cu(200)/(111) area ratio at a probe depth of 2.6 nm in cathodic and anodic steps in CO_2 -purged electrolyte. The dashed line indicates the ratio at 0 V.

144 depth of 2.6 nm according to the applied potentials under both 145 CO₂ and Ar-purged electrolytes (Figure 3a). Although peak 146 area is not a rigorously quantitative measure of the amount of 147 specific facet, relative area ratios reflect changes in the 148 preferentially exposed facets on the surface.²³ At 0 V, where 149 no CO₂RR or hydrogen evolution reaction (HER) occurs on 150 Cu(pc), the area ratio in the surface exhibited similar values in 151 CO₂- and Ar-purged electrolytes. As the applied potential 152 becomes more negative, the (200)/(111) area ratio in the 153 surface increases (20.9–31.5%) under CO₂-purged electrolyte, 154 while the area ratio in the Ar-purged electrolyte showed only a 155 slight increase (2.4-2.7%) across the applied potentials. The 156 increase in area ratio was only observed at the probe depth of 157 2.6 nm, indicating the reconstruction is a surface phenomenon 158 (Figure S13). Both the intensity and integral breadth of the (200)/(111) ratio also increase as a function of the applied 160 potential under CO_2 -purged electrolytes (Figure S14). The 161 increase in the area ratio is significantly higher than that of the 162 estimated errors (Figure S15), and the trend is reproduced in 163 another Cu sample (Figure S16). Therefore, these results 164 indicate that a surface reconstruction from Cu(pc) to 165 Cu(100)-like surface occurs during CO_2RR . The differences 166 in CO₂ and Ar-purged electrolytes suggest that surface 167 hydrogen and hydroxide species generated from HER have a 168 limited effect on surface reconstruction. Previous theoretical 169 calculations and spectroscopic observations indicate that the 170 dominant intermediate during CO₂RR is adsorbed CO 171 (CO*).^{24,25} We found that the surface reconstruction is 172 enhanced at more negative electrode potentials. Computa-173 tional results indicate that similar trends in the CO* coverage

at these potentials.²⁶ We also observed a similar reconstruction 174 under CO reduction conditions under alkaline electrolytes in a 175 previous study.²³ Considering that the local pH at the Cu 176 surface increases due to the OH⁻ ions generated from CO₂RR, 177 we conjecture that the microenvironment under CO₂-purged 178 electrolytes could be similar to the situation under CO-purged 179 alkaline electrolytes. This suggests that CO^{*} is likely a key 180 reaction intermediate in promoting surface reconstruction (see 181 SI for a detailed discussion). 182

For a more quantitative description of the surface 183 reconstruction, we analyzed the relative changes of the area 184 ratio depending on the applied potential. The area ratio at a 185 probe depth of 2.6 nm increases by 20.9% at -0.65 V 186 compared to the value at 0 V. It is reported that the CO 187 adsorption starts to appear at -0.5 V in a CO₂-saturated 0.1 M 188 KHCO₃ electrolyte, and its coverage gradually increases with 189 increasingly negative applied potentials.²⁵ This accumulated 190 CO^* on the surfaces could reconstruct Cu(pc) to Cu(100) 191 before CO_2RR occurs. At -0.9 V, where the C_{2+} products 192 begin to emerge on Cu(pc), the area ratio shows an increase of 193 27.6% and this change helps to explain why the hydrocarbon 194 formation on Cu(pc) is invariant to that on Cu(100).²⁷ The 195 area ratio further increases to 31.5% at -1.1 V, where the 196 selectivity toward HER has been reported to surpass the 197 CO₂RR products due to the concentration polarization for 198 CO₂ (i.e., mass transport limitation) in the H-cell config- 199 uration.²⁰ This result implies that enhanced mass transport in 200 our cell allows a higher degree of surface reconstruction to 201 (100) at -1.1 V, possibly through an increase of CO* coverage 202 on the surface. We collected the GIXRD data after reaching a 203 steady state; however, further work is needed to understand 204 the kinetics of surface reconstruction which has been observed 205 on various time scales previously.^{17,28} We expect that the high 206 mass transport rates and current densities in our cell translate 207 to fast reconstruction kinetics. We also found that the surface 208 reconstruction is only partially reversible. When the applied 209 potential returns to more positive values, at 0 V the surface 210 displays more (100) faceting than the original surface at 0 V 211 (Figure 3b). The irreversible change of surface morphology has 212 been observed during HER on the Cu(100), leaving behind 213 structural defects such as small holes and clusters.²⁹ This 214 observation likely explains the previously reported hysteresis of 215 product distribution during cyclic voltammetry.^{27,30} In these 216 reports, the enhanced generation of multi-carbon products 217 during cyclic voltammetry has been observed only during the 218 anodic sweep. Our results suggest this may be because the 219 applied potential during the cathodic sweep leads to a 220 significant surface reconstruction to a Cu(100)-like surface 221 that can steer the CO₂RR selectivity toward hydrocarbon 222 formation. During the anodic sweep, the surface is already 223 reconstructed to be more Cu(100)-like, enabling the increased 224 production of C2+ products. These results can provide 225 fundamental insights into the previously observed processes 226 that did not take account of the surface reconstruction during 227 CO₂RR. 2.2.8

In summary, we investigated the changes in the valence state 229 and crystallographic structure in the near-surface region of 230 Cu(pc) thin films under realistic CO_2RR conditions. By using 231 an electrochemical flow cell that allows for *in situ* GIXAS and 232 GIXRD with improved CO_2 mass transfer, we successfully 233 demonstrated that the surface copper oxide is reduced to 234 metallic Cu prior to the onset of CO_2RR and that metallic Cu 235 is the only detectable phase during CO_2RR . We also showed 236

237 that the surface reconstruction from Cu(pc) to Cu(100) takes $_{238}$ place only in the presence of CO₂ molecules, suggesting it is a 239 CO-driven phenomenon. The surface reconstruction increases 240 as the applied potential becomes more negative, and the 241 reconstructed surface partially persists in the anodic steps. Our 242 in situ measurements of surface oxidation state and 243 reconstruction behavior provide new insights for the atomic-244 scale understanding of Cu-based electrocatalysts.

ASSOCIATED CONTENT 245

246 Supporting Information

247 The Supporting Information is available free of charge at 248 https://pubs.acs.org/doi/10.1021/jacs.0c10017.

- Experimental details, detailed in situ GIXAS and GIXRD 2.49
- supplementary figures, and estimation of error margins 250 (PDF) 251
- Raw data in .txt format for all figures (ZIP) 252

AUTHOR INFORMATION 253

Corresponding Authors 254

- Apurva Mehta Stanford Synchrotron Radiation Lightsource, 255
- SLAC National Accelerator Laboratory, Menlo Park, 256
- California 94025, United States; @ orcid.org/0000-0003-257 0870-6932; Email: mehta@slac.stanford.edu 2.58
- Ryan C. Davis Stanford Synchrotron Radiation Lightsource, 2.59 SLAC National Accelerator Laboratory, Menlo Park, 260
- California 94025, United States; Email: rydavis@ 261
- slac.stanford.edu 262
- Thomas F. Jaramillo Department of Chemical Engineering, 263
- Stanford University, Stanford, California 94305, United 264
- States; SUNCAT Center for Interface Science and Catalysis, 265
- SLAC National Accelerator Laboratory, Menlo Park, 2.66
- California 94025, United States; O orcid.org/0000-0001-267 9900-0622; Email: jaramillo@stanford.edu 2.68
- Christopher Hahn SUNCAT Center for Interface Science 269
- and Catalysis, SLAC National Accelerator Laboratory, 270
- Menlo Park, California 94025, United States; 271
- Email: chahn@slac.stanford.edu 272
- Walter S. Drisdell Joint Center for Artificial Photosynthesis 273
- and Chemical Sciences Division, Lawrence Berkeley National 274
- Laboratory, Berkeley, California 94720, United States; 2.75
- orcid.org/0000-0002-8693-4562; Email: wsdrisdell@ 276
- lbl.gov 277

278 Authors

- 279 **Soo Hong Lee** – Joint Center for Artificial Photosynthesis and Chemical Sciences Division, Lawrence Berkeley National 280
- Laboratory, Berkeley, California 94720, United States; 281 orcid.org/0000-0002-2734-9654 282
- John C. Lin Department of Chemical Engineering, Stanford 2.83 University, Stanford, California 94305, United States; 284
- SUNCAT Center for Interface Science and Catalysis, SLAC 285
- National Accelerator Laboratory, Menlo Park, California 286
- 94025, United States 287
- Maryam Farmand Joint Center for Artificial Photosynthesis 288 and Chemical Sciences Division, Lawrence Berkeley National 289
- Laboratory, Berkeley, California 94720, United States 290
- Alan T. Landers SUNCAT Center for Interface Science and 291 Catalysis, SLAC National Accelerator Laboratory, Menlo 2.92
- Park, California 94025, United States; Department of 293
- Chemistry, Stanford University, Stanford, California 94305, 294
- United States; orcid.org/0000-0001-7290-711X 295

Jeremy T. Feaster – Department of Chemical Engineering,	296
Stanford University, Stanford, California 94305, United	297
States; SUNCAT Center for Interface Science and Catalysis,	298
SLAC National Accelerator Laboratory, Menlo Park,	299
California 94025, United States	300
Jaime E. Avilés Acosta – SUNCAT Center for Interface	301
Science and Catalysis, SLAC National Accelerator	302
Laboratory, Menlo Park, California 94025, United States;	303
Department of Materials Science and Engineering, Stanford	304
University, Stanford, California 94305, United States	305
Jeffrey W. Beeman – Joint Center for Artificial Photosynthesis	306
and Chemical Sciences Division, Lawrence Berkeley National	307
Laboratory, Berkeley, California 94720, United States	308
Yifan Ye – Joint Center for Artificial Photosynthesis, Chemical	309
Sciences Division, and Advanced Light Source, Lawrence	310
Berkeley National Laboratory, Berkeley, California 94720,	311
United States	312
Junko Yano – Joint Center for Artificial Photosynthesis and	313
Molecular Biophysics and Integrated Bioimaging Division,	314
Lawrence Berkeley National Laboratory, Berkeley, California	315
94720, United States; 💿 orcid.org/0000-0001-6308-9071	316
Complete contact information is available at:	317
https://pubs.acs.org/10.1021/jacs.0c10017	318
Author Contributions	319

^{II}S.H.L. and J.C.L. contributed equally. 320 Notes 321 The authors declare no competing financial interest. 322

ACKNOWLEDGMENTS

The material for synthesis of Cu thin films, electrochemical 324 flow cell assembly, and GIXAS and GIXRD measurements is 325 based on work performed by the Joint Center for Artificial 326 Photosynthesis, a DOE Energy Innovation Hub, supported 327 through the Office of Science of the U.S. Department of 328 Energy, under Award No. DE-SC0004993. The material for 329 electrochemistry experiments with product analysis is based on 330 work performed by the Liquid Sunlight Alliance, which is 331 supported by the U.S. Department of Energy, Office of 332 Science, Office of Basic Energy Sciences, Fuels from Sunlight 333 Hub under Award No. DE-SC0021266. Use of the Stanford 334 Synchrotron Radiation Lightsource, SLAC National Accel- 335 erator Laboratory, is supported by the U.S. Department of 336 Energy, Office of Science, Office of Basic Energy Sciences 337 under Contract No. DE-AC02-76SF00515. Part of this work 338 was performed at the Stanford Nano Shared Facilities (SNSF) 339 and the Stanford Nanofabrication Facility (SNF), supported by 340 the National Science Foundation under Award ECCS- 341 1542152. 342

REFERENCES

343

323

(1) Nitopi, S.; Bertheussen, E.; Scott, S. B.; Liu, X.; Engstfeld, A. K.; 344 Horch, S.; Seger, B.; Stephens, I. E. L.; Chan, K.; Hahn, C.; Nørskov, 345 J. K.; Jaramillo, T. F.; Chorkendorff, I. Progress and Perspectives of 346 Electrochemical CO₂ Reduction on Copper in Aqueous Electrolyte. 347 Chem. Rev. 2019, 119 (12), 7610-7672. 348

(2) Peterson, A. A.; Abild-Pedersen, F.; Studt, F.; Rossmeisl, J.; 349 Nørskov, J. K. How copper catalyzes the electroreduction of carbon 350 dioxide into hydrocarbon fuels. Energy Environ. Sci. 2010, 3 (9), 351 1311-1315. 352

(3) Handoko, A. D.; Wei, F.; Jenndy; Yeo, B. S.; Seh, Z. W. 353 Understanding heterogeneous electrocatalytic carbon dioxide reduc- 354 355 tion through operando techniques. *Nat. Catal.* **2018**, *1* (12), 922–356 934.

(4) Xiao, H.; Goddard, W. A.; Cheng, T.; Liu, Y. Cu metal
seembedded in oxidized matrix catalyst to promote CO₂ activation and
CO dimerization for electrochemical reduction of CO₂. *Proc. Natl. Acad. Sci. U. S. A.* 2017, *114* (26), 6685–6688.

(5) De Luna, P.; Quintero-Bermudez, R.; Dinh, C.-T.; Ross, M. B.;
Bushuyev, O. S.; Todorović, P.; Regier, T.; Kelley, S. O.; Yang, P.;
Sargent, E. H. Catalyst electro-redeposition controls morphology and
axidation state for selective carbon dioxide reduction. *Nat. Catal.*2018, 1 (2), 103-110.

366 (6) Liu, C.; Lourenço, M. P.; Hedström, S.; Cavalca, F.; Diaz-367 Morales, O.; Duarte, H. A.; Nilsson, A.; Pettersson, L. G. M. Stability 368 and Effects of Subsurface Oxygen in Oxide-Derived Cu Catalyst for 369 CO₂ Reduction. *J. Phys. Chem. C* **2017**, *121* (45), 25010–25017.

370 (7) Favaro, M.; Xiao, H.; Cheng, T.; Goddard, W. A.; Yano, J.; 371 Crumlin, E. J. Subsurface oxide plays a critical role in CO₂ activation 372 by Cu(111) surfaces to form chemisorbed CO₂, the first step in 373 reduction of CO₂. *Proc. Natl. Acad. Sci. U. S. A.* **2017**, *114* (26), 374 6706–6711.

375 (8) Zhao, Y.; Chang, X.; Malkani, A. S.; Yang, X.; Thompson, L.; 376 Jiao, F.; Xu, B. Speciation of Cu Surfaces During the Electrochemical 377 CO Reduction Reaction. *J. Am. Chem. Soc.* **2020**, *142* (21), 9735– 378 9743.

(9) Firet, N. J.; Burdyny, T.; Nesbitt, N. T.; Chandrashekar, S.;
180 Longo, A.; Smith, W. A. Copper and silver gas diffusion electrodes
181 performing CO₂ reduction studied through operando X-ray
182 absorption spectroscopy. *Catal. Sci. Technol.* 2020, 10 (17), 5870–
183 5885.

184 (10) Mistry, H.; Varela, A. S.; Bonifacio, C. S.; Zegkinoglou, I.; 185 Sinev, I.; Choi, Y.-W.; Kisslinger, K.; Stach, E. A.; Yang, J. C.; Strasser, 186 P.; Cuenya, B. R. Highly selective plasma-activated copper catalysts 187 for carbon dioxide reduction to ethylene. *Nat. Commun.* **2016**, 7 (1), 188 12123.

(11) Cavalca, F.; Ferragut, R.; Aghion, S.; Eilert, A.; Diaz-Morales,
O.; Liu, C.; Koh, A. L.; Hansen, T. W.; Pettersson, L. G. M.; Nilsson,
A. Nature and Distribution of Stable Subsurface Oxygen in Copper
Electrodes During Electrochemical CO₂ Reduction. *J. Phys. Chem. C*2017, *121* (45), 25003–25009.

³⁹⁴ (12) Velasco-Velez, J.-J.; Mom, R. V.; Sandoval-Diaz, L.-E.; Falling, ³⁹⁵ L. J.; Chuang, C.-H.; Gao, D.; Jones, T. E.; Zhu, Q.; Arrigo, R.; ³⁹⁶ Roldan Cuenya, B.; Knop-Gericke, A.; Lunkenbein, T.; Schlögl, R. ³⁹⁷ Revealing the Active Phase of Copper during the Electroreduction of ³⁹⁸ CO₂ in Aqueous Electrolyte by Correlating *In Situ* X-ray Spectroscopy ³⁹⁹ and *In Situ* Electron Microscopy. *ACS Energy Lett.* **2020**, *5* (6), 2106– ⁴⁰⁰ 2111.

401 (13) Kondrat, S. A.; van Bokhoven, J. A. A Perspective on Counting 402 Catalytic Active Sites and Rates of Reaction Using X-Ray Spectros-403 copy. *Top. Catal.* **2019**, *62* (17), 1218–1227.

404 (14) Huang, J.; Hörmann, N.; Oveisi, E.; Loiudice, A.; De Gregorio, 405 G. L.; Andreussi, O.; Marzari, N.; Buonsanti, R. Potential-induced 406 nanoclustering of metallic catalysts during electrochemical CO_2 407 reduction. *Nat. Commun.* **2018**, 9 (1), 3117.

408 (15) Eren, B.; Zherebetskyy, D.; Patera, L. L.; Wu, C. H.; Bluhm, H.; 409 Africh, C.; Wang, L.-W.; Somorjai, G. A.; Salmeron, M. Activation of 410 Cu(111) surface by decomposition into nanoclusters driven by CO 411 adsorption. *Science* **2016**, *351* (6272), 475–478.

412 (16) Hori, Y.; Wakebe, H.; Tsukamoto, T.; Koga, O. Adsorption of 413 CO Accompanied with Simultaneous Charge Transfer on Copper 414 Single Crystal Electrodes Related with Electrochemical Reduction of 415 CO₂ to hydrocarbons. *Surf. Sci.* **1995**, 335, 258–263.

416 (17) Kim, Y.-G.; Baricuatro, J. H.; Soriaga, M. P. Surface 417 Reconstruction of Polycrystalline Cu Electrodes in Aqueous 418 KHCO₃ Electrolyte at Potentials in the Early Stages of CO_2 419 Reduction. *Electrocatalysis* **2018**, 9 (4), 526–530.

420 (18) Hahn, C.; Hatsukade, T.; Kim, Y.-G.; Vailionis, A.; Baricuatro, 421 J. H.; Higgins, D. C.; Nitopi, S. A.; Soriaga, M. P.; Jaramillo, T. F. 422 Engineering Cu surfaces for the electrocatalytic conversion of CO₂: Controlling selectivity toward oxygenates and hydrocarbons. Proc. 423 Natl. Acad. Sci. U. S. A. 2017, 114 (23), 5918–5923. 424

(19) Farmand, M.; Landers, A. T.; Lin, J. C.; Feaster, J. T.; Beeman, 425
J. W.; Ye, Y.; Clark, E. L.; Higgins, D.; Yano, J.; Davis, R. C.; Mehta, 426
A.; Jaramillo, T. F.; Hahn, C.; Drisdell, W. S. Electrochemical flow cell 427
enabling operando probing of electrocatalyst surfaces by X-ray 428
spectroscopy and diffraction. *Phys. Chem. Chem. Phys.* 2019, 21 429
(10), 5402–5408.

(20) Clark, E. L.; Resasco, J.; Landers, A.; Lin, J.; Chung, L.-T.; 431 Walton, A.; Hahn, C.; Jaramillo, T. F.; Bell, A. T. Standards and 432 Protocols for Data Acquisition and Reporting for Studies of the 433 Electrochemical Reduction of Carbon Dioxide. *ACS Catal.* **2018**, 8 434 (7), 6560–6570. 435

(21) Kuhl, K. P.; Cave, E. R.; Abram, D. N.; Jaramillo, T. F. New 436 Insights into the Electrochemical Reduction of Carbon Dioxide on 437 Metallic Copper Surfaces. *Energy Environ. Sci.* **2012**, *5*, 7050–7059. 438 (22) Huynh, T. M. T.; Broekmann, P. From In Situ towards In 439 Operando Conditions: Scanning Tunneling Microscopy Study of 440 Hydrogen Intercalation in Cu(111) during Hydrogen Evolution. 441 *ChemElectroChem* **2014**, *1* (8), 1271–1274. 442

(23) Scott, S. B.; Hogg, T. V.; Landers, A. T.; Maagaard, T.; 443 Bertheussen, E.; Lin, J. C.; Davis, R. C.; Beeman, J. W.; Higgins, D.; 444 Drisdell, W. S.; Hahn, C.; Mehta, A.; Seger, B.; Jaramillo, T. F.; 445 Chorkendorff, I. Absence of Oxidized Phases in Cu under CO 446 Reduction Conditions. *ACS Energy Lett.* **2019**, *4* (3), 803–804. 447

(24) Tang, M. T.; Ulissi, Z. W.; Chan, K. Theoretical Investigations 448 of Transition Metal Surface Energies under Lattice Strain and CO 449 Environment. J. Phys. Chem. C 2018, 122 (26), 14481–14487. 450

(25) Zhu, S.; Jiang, B.; Cai, W.-B.; Shao, M. Direct Observation on 451 Reaction Intermediates and the Role of Bicarbonate Anions in CO_2 452 Electrochemical Reduction Reaction on Cu Surfaces. J. Am. Chem. 453 Soc. 2017, 139 (44), 15664–15667. 454

(26) Luo, W.; Nie, X.; Janik, M. J.; Asthagiri, A. Facet Dependence 455 of CO_2 Reduction Paths on Cu Electrodes. ACS Catal. **2016**, 6 (1), 456 219–229. 457

(27) Schouten, K. J. P.; Qin, Z.; Pérez Gallent, E.; Koper, M. T. M. 458 Two Pathways for the Formation of Ethylene in CO Reduction on 459 Single-Crystal Copper Electrodes. J. Am. Chem. Soc. **2012**, 134 (24), 460 9864–9867. 461

(28) Gunathunge, C. M.; Li, X.; Li, J.; Hicks, R. P.; Ovalle, V. J.; 462 Waegele, M. M. Spectroscopic Observation of Reversible Surface 463 Reconstruction of Copper Electrodes under CO₂ Reduction. *J. Phys.* 464 *Chem. C* 2017, 121 (22), 12337–12344. 465

(29) Matsushima, H.; Taranovskyy, A.; Haak, C.; Gründer, Y.; 466 Magnussen, O. M. Reconstruction of Cu(100) Electrode Surfaces 467 during Hydrogen Evolution. J. Am. Chem. Soc. **2009**, 131 (30), 468 10362–10363. 469

(30) Clark, E. L.; Bell, A. T. Direct Observation of the Local 470 Reaction Environment during the Electrochemical Reduction of CO₂. 471 J. Am. Chem. Soc. **2018**, 140 (22), 7012–7020. 472