

Cite this: *Dalton Trans.*, 2017, **46**, 7309

Rapid self-healing and anion selectivity in metallosupramolecular gels assisted by fluorine–fluorine interactions†

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Simple ML_2 [$M = Fe(II), Co(II), Ni(II)$] complexes obtained from a perfluoroalkylamide derivative of 4-amino-phenyl-2,2',6,2'-terpyridine spontaneously, yet anion selectively, self-assemble into gels, which manifest an unprecedented rapid gel strength recovery, viz. self-healing, and thermal rearrangement in aqueous dimethyl sulfoxide. The key factor for gelation and rheological properties emerges from the fluorine–fluorine interactions between the perfluorinated chains, as the corresponding hydrocarbon derivative did not form metallogels. The perfluoro-terpyridine ligand alone formed single crystals, while its $Fe(II)$, $Co(II)$ or $Ni(II)$ complexes underwent rapid gelation leading to highly entangled fibrillar networks visualized by electron microscopy. The thermodynamic parameters of gelation based on variable temperature NMR 1H and ^{19}F resonances showed that gelation was enthalpically favourable and entropically disfavourable. The step strain rheological experiments revealed that the gels undergo rapid self-healing and the morphological features, thermal stability and mechanical properties were found to depend on the nature of the metal ion.

Received 18th March 2017,
Accepted 8th May 2017

DOI: 10.1039/c7dt00983f

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Introduction

Metallosupramolecular self-assembly has continued to gain a new dimension due to its ability to control the shape and size of molecular, as well as, supramolecular superstructures.^{1–5} Examples include self-assembly into cages,⁶ cubes,⁷ spheres,⁸ grids,⁹ helicates,¹⁰ and metal–organic frameworks.¹¹ In addition, the above examples also provide a dynamic control for host–guest chemistry,¹² size-selective entrapment,¹³ stereo-controlled chemical reactions,¹⁴ and supramolecular polymers.¹⁵ Recently, the self-assembly process of luminescent platinum complexes has been visualized and controlled using

the change in the photophysical properties of the compounds upon aggregation.¹⁶ Supramolecular gelation, on the other hand, is continuously an evolving area of research with new insights and novel application potentials. Non-covalent interactions such as hydrogen bonding, π – π stacking, charge transfer interactions, and hydrophobic effects have been extensively utilized towards the rational design of low molecular mass organo- and hydrogelators with new functional properties.^{17–20} Peptides,²¹ carbohydrates,²² steroids,²³ urea-derivatives,²⁴ and simple fatty acids²⁵ have been shown to exhibit remarkable ability to self-assemble into highly entangled fibrillar networks thereby being able to encapsulate and immobilize solvent molecules. While a majority of the gelators have been discovered serendipitously,^{26,27} continuing efforts have been made towards the rational design of gelators with tunable properties, for example thermotropic polymorphism²⁸ or stimuli-responsiveness²⁹ including anion dependent gels³⁰ with interesting applications in molecular sensing.³¹ Recently, metal coordination induced self-assembly of small molecules leading to solvent immobilization and gelation has emerged as a fascinating area of research in the field of supramolecular gels.^{32,33} In order to promote supplementary interactions and achieve supramolecular gelation, more than one non-covalent interaction is needed, thereby facilitating self-assembly at multiple length scales. As a result, additional interaction motifs such as peptide bonds, urea-linkers or steroidal units have been

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† Electronic supplementary information (ESI) available: Details on the synthesis and purification of ligands **3**, **3H** and **4**, 1H , ^{13}C and ^{19}F NMR spectra, FT-IR spectra, single crystal data of ligand **3**, its solvates and complex $[Ni_4]Cl_2$, and powder X-ray diffraction patterns of solids and xerogels. CCDC 1533110–1533114. For ESI and crystallographic data in CIF or other electronic format see DOI: 10.1039/c7dt00983f

appended to the basic ligand units. Moreover, it has been shown that by combining the metal–ligand coordination bonds, along with additional supramolecular interactions and ligand design, the concept of subcomponent self-assembly can be used in molecular gelation.³⁴ Among several ligands studied for gelation, terpyridine and its derivatives were shown to be attractive candidates because of their easy synthesis and functionalization as well as for their ability to bind a diverse number of metal ions.^{35,36} On the other hand, less studied interactions such as halogen bonding have been reported to control the self-assembly of a variety of motifs such as ureas³⁷ and peptides³⁸ leading to hydrogelation. Weak interactions, including fluorine–fluorine (F...F) interactions, are also gaining attention in the field of contemporary supramolecular chemistry.^{39–43} Previously, fluorinated solvents (fluorous)⁴⁴ and pharmaceutically important molecules^{45,46} have been extensively studied. Beyond being a consequence of crystal packing, recent experimental and theoretical studies have supported the existence of F...F interactions and opened up new applications in supramolecular chemistry.^{47,48} The fluorinated analogues of hydrocarbon compounds possess exceptional properties such as steric bulk, stiff and stable nature, simultaneous hydro- and lipophobicity, slower aggregation behaviour and lower critical gelation/aggregation concentration. These features of fluorinated counterparts offer a unique opportunity to design materials with unprecedented properties.⁴⁹ In polymer science, polyethylene glycol derivatives with terminal fluoroalkyl groups have shown to exhibit peculiar rheological properties as “associative thickeners”.⁵⁰ Similarly, polyacrylamides with a certain number of fluorocarbon groups have shown to associate much stronger than those of the simple hydrocarbon derivatives.⁵¹ In 1999 Hamilton *et al.* showed the gelation of partially fluorinated urea derivatives in supercritical carbon dioxide.⁵² Supramolecular gels derived from fluorinated gallic acid,³⁷ peptides,⁴⁰ bile acids,⁴¹ bis-urea,⁴² *N*-alkylamides,⁵³ gemini phenoxy units⁵⁴ and biphenyl⁵⁵ have been reported in the literature. This work shows how a simple perfluoroalkylamide of 4-aminophenyl-2,2',6,2'-terpyridine can act as a metallosupramolecular gelator. The metal–ligand coordination leads to the ML₂ [M =

Fe(II), Ni(II) and Co(II)] coordination complex and fluorine–fluorine interactions between the side chain drive an extended lateral assembly, furnishing highly entangled fibrillar networks. The gelation is anion selective and the mechanical and thermal properties can be tuned by changing the metal cation. Furthermore, the rheological step strain experiments show that the gels undergo a rapid self-healing⁵⁶ by recovering the gel strength after the release of the step strain.

Results and discussion

Synthesis and gelation studies

The starting material 4-aminophenyl-2,2':6',2''-terpyridine **2** was synthesized according to the literature procedure reported previously.⁵⁷ The gelating ligand *N*-(4-([2,2':6',2''-terpyridin]-4'-yl)phenyl)-2,2,3,3,4,4,5,5,6,6,7,7,8,8,8-penta-decafluorooctanamide **3** was prepared by the dropwise addition of penta-decafluorooctanoyl chloride **1** to a solution of **2** in anhydrous dichloromethane and in the presence of *N,N*-diisopropylethyl amine (Fig. 1, see the ESI† for complete synthesis procedure and characterization). The ligand **3** upon recrystallization from chloroform formed single crystals suitable for X-ray diffraction analysis. A similar procedure was followed for the synthesis of the hydrocarbon analogue **3H** (see the ESI†). However, and not surprisingly, efforts to obtain single crystals remained unsuccessful. In addition, ligand **4** was prepared (Fig. 1 and ESI†), which includes a partially perfluorinated chain with an ethylene moiety between the carbonyl and perfluoro groups. When a known amount of ligand **3** was dissolved in dimethyl sulfoxide (DMSO) by gentle heating, the addition of a known volume of water reaching a DMSO:water ratio of 5:1 (v/v), resulted in a viscous precipitate. However, the viscous material did not survive during the gelation test (test tube inversion). Interestingly, the addition of 0.5 molar equivalents of aqueous metal salt (CoCl₂, NiCl₂ or FeCl₂) into a solution of **3** in DMSO resulted in an immediate colour change followed by precipitation. The complexes upon heating led to a clear solution and upon cooling to room temperature stable coloured gels were obtained (Fig. 1, see the ESI† for details) showing resistance to

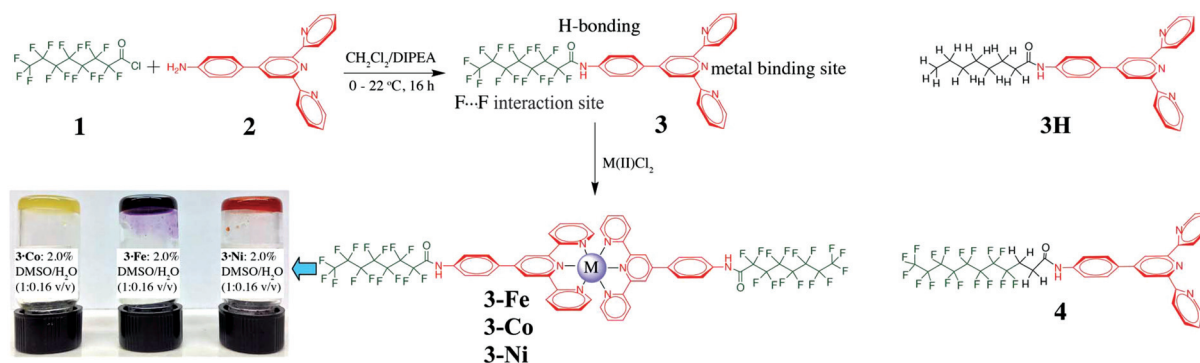


Fig. 1 Synthetic scheme of ligand **3** and its complexation with M(II)Cl₂ (M = Ni, Fe, Co). Photographs of **3**-M gels. Chemical structure of ligands **3H** and **4**.

flow upon test tube inversion. The gels underwent repeated sol–gel transition upon heating–cooling cycles. The ESI-TOF mass spectrometry on positive mode revealed the formation of ML_2 type complexes (see the ESI†), supported by NMR spectroscopy studies, which suggest the presence of a single product. In the control experiments analogous metal complexes containing ligand **3H** were found not to be gelators, thereby highlighting the importance of the perfluoroalkyl unit. The comparison of powder X-ray diffraction (PXRD) patterns of **3H** with the simulated PXRD patterns using the single crystal X-ray data of **3** (Fig. S9†) revealed a clear difference in the packing of the two ligands.

Similarly, with the partially fluorinated side chain in **4**, no gelation was observed by the ligand itself or upon complexation with $FeCl_2$. Interestingly, ligand **4** leads to weak gels upon complexation with 0.5 eq. of $CoCl_2$ (**4-Co**) or $NiCl_2$ (**4-Ni**). The metallogels **4-M** [$M = Co(II)$ or $Ni(II)$] are much less stable than their **3-M** analogues. Indeed, **4-Co** precipitates out after 6 hours and **4-Ni** crystallizes out after 36 hours. We were pleased to observe that **4-Ni** gel evolved into good quality single crystals of $[Ni_4_2]Cl_2$. Such an intriguing gel to the crystal transition phenomenon has been previously reported and provides an excellent opportunity to investigate the potential supramolecular interactions that cause the gelation.⁵⁸ A careful examination of the crystal supports our hypothesis on supramolecular polymerization through the $F\cdots F$ interactions during the gelation and will be discussed in detail later. Furthermore, in order to promote polymerization and increase the strength of the supramolecular structure by the potential formation of hydrogen bonds, the perfluorinated chain was attached *via* amide bonds to the phenyl-terpyridine core. The anion seems to play also a crucial role as only chloride anions induce the gelation, whereas the use of other counter anions such as bromide, tetrafluoroborate, perchlorate, triflate or sulphate did not lead to the gel formation. More importantly, further addition of chloride ions to the non-gelling complexes turned on the gelation. This directly provides evidence that the gelation is Cl^- specific. The chloride salts of $Fe(II)$, $Co(II)$ and $Ni(II)$ resulted in instantaneous gelation at 2.0 w/v% (mass of gelator by the volume of solvent) and with a minimum gelation concentration (MGC) of <1.0%. The gel melting temperature (T_{gel}) and thermal stability measured using an inverted test tube method showed a linear increase in the T_{gel} with increasing concentration. However, for a given concentration the gel melting trend followed the order $Fe > Ni > Co$ (see Table 1 and ESI†). The T_{gel} was sharp for **3-Fe** and **3-Co** but **3-Ni** showed a broad range of melting temperatures (see Table S1 and Fig. S11†).

Single crystal X-ray diffraction study

Single crystals of good quality for X-ray diffraction studies were obtained either in their pure form or as solvates for ligand **3** upon the slow evaporation of chloroform (**3** and **3-CHCl₃**), tetrahydrofuran (**3-THF**) and moist methanol (**3-CH₃OH·H₂O**) solutions. A systematic analysis of the X-ray structures (see Fig. 2A and B and Fig. S12–S14†) revealed a slightly twisted,

Table 1 Gelation response of ligands **3** and **3H** towards different metal salts at 2% DMSO : water (5 : 1 v/v). G = gel, S = solution

Metal salt	Ligand 3	Ligand 3H
$FeCl_2$	G (70 °C) ^a	S
$CoCl_2$	G (47 °C) ^a	S
$NiCl_2$	G (60 °C) ^a	S
$FeBr_2$	S	S
$Fe(ClO_4)_2$	S	S
$Fe(BF_4)_2$	S	S
$FeSO_4$	S	S
$Fe(OTf)_2$	S	S

^a T_{gel} of 2% gels.

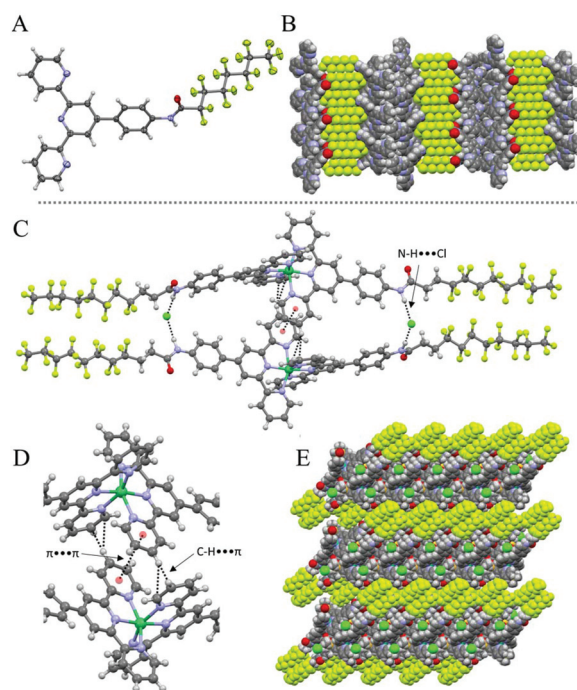


Fig. 2 (A) Crystal structure of ligand **3** with thermal displacement parameters at a 50% probability level. (B) Crystal packing of ligand **3**. The lattice is constructed from $F\cdots F$ interactions between the perfluorinated chains and are reinforced by the $N-H\cdots O=C$ hydrogen bonds between adjacent molecules of **3**. (C) Crystal structure of $[Ni_4_2]Cl_2$ stabilized by $N-H\cdots Cl$, $\pi\cdots\pi$ and $C-H\cdots\pi$ interactions. (D) Closer view of $[Ni_4_2]Cl_2$ around the terpyridine core to show $\pi\cdots\pi$ and $C-H\cdots\pi$ interactions. (E) Crystal packing of complex $[Ni_4_2]Cl_2$ viewed along the a -axis highlighting the $F\cdots F$ and $C-F\cdots\pi$ interactions.

non-linear nature for the ligand **3** and its ability to form hydrogen bonds through the amide group. More importantly, the molecular packing also showed the extended tail to tail interactions of the perfluoroalkyl chains in addition to π – π stacking between the aromatic moieties. Therefore, in the solid-state, the $F\cdots F$ interaction acts as one of the key interactions in the formation of the supramolecular structures. The packing of **3** and its solvates manifests similar $F\cdots F$ interactions to those shown by the $[Ni_4_2]Cl_2$ crystal (Fig. 2B and E). Additionally, the anion selectivity gelation is highly supported by the critical

role that chloride anions play in furnishing the hydrogen bond network as shown in Fig. 2E for $[\text{Ni}_4\text{Cl}_2]$.

Morphology of the gels

The electron microscopy images of dry precipitates of ligand **3** and xerogels for metalloids **3-M** [$\text{M} = \text{Fe}(\text{II}), \text{Ni}(\text{II})$ or $\text{Co}(\text{II})$] are shown in Fig. 3 and 4. The scanning electron microscopy (SEM) images showed that ligand **3** forms microcrystalline sheets several micrometers long and 1–2 μm wide (Fig. 3A). The absence of interconnected networks is in good agreement with the non-gelling nature of **3**. The SEM micrographs of gels derived from metal complexes showed highly entangled fibrillar networks. The diameter of the fibers ranges from 50 to 200 nm with an indefinite length, one of the features of gels from low molecular weight gelators. The transmission electron microscopy (TEM) analysis of ligand **3** showed the presence of microcrystalline strands (Fig. 4A). The xerogels obtained from metalloids showed the highly entangled fibrillar network wherein the nature and size of the fibers depend on the metal ion used in the study (Fig. 4B–D). The fiber diameter varied

from 100–125 nm for **3-Fe**, 80–100 nm for **3-Co** and to 50–100 nm for **3-Ni**.

Variable temperature (VT) NMR

Solution,⁵⁹ and solid state,⁶⁰ NMR spectroscopy experiments have been extensively used to study the interactions, structural changes and dynamics of gelation. However, NMR spectroscopic characterization of certain metal complexes is a challenging task due to the paramagnetic nature of these species. Therefore, out of the three metal complexes, the **3-Fe** gel was selected for NMR studies due to the diamagnetic character of the $\text{Fe}(\text{II})$ -terpyridine low spin complexes. In more detail, a 2.0% $\text{DMSO-}d_6$ - D_2O gel of **3-Fe** was analysed by variable temperature (VT) ^1H NMR in a temperature interval from 30 to 90 $^\circ\text{C}$ with 5 $^\circ\text{C}$ increments. The broad signals in the gel state at room temperature turned sharper until the temperature reached 50 $^\circ\text{C}$. Surprisingly, at 55 $^\circ\text{C}$ the signals turned broad again, however, a further increase in temperature above 55 $^\circ\text{C}$ led to sharp peaks. A more accurate study was performed by heating the gel sample from 40 to 70 $^\circ\text{C}$ with 2 $^\circ\text{C}$ increments. It was found that the transition appeared at 58 $^\circ\text{C}$. Furthermore, a controlled cooling ramp (sol-gel) was performed from 85 to 30 $^\circ\text{C}$. Upon cooling, the signals became broader until 55 $^\circ\text{C}$ and below 55 $^\circ\text{C}$ they turned sharper and once again continued to broaden (see Fig. S17–S20†). This behaviour suggests a thermally induced supramolecular rearrangement of the metal complexes. These results strongly support the hypothesis of a phase transition observed in temperature sweep rheological measurements (see the next section). The VT ^1H NMR of the gel showed an upfield shift ($\Delta\delta = 0.45$ ppm) for the amide proton ($-\text{NH}$) signal from 11.79 ppm at room temperature (in gel state) to 11.34 ppm upon increasing the temperature (in solution state) (Fig. 5). It can be con-

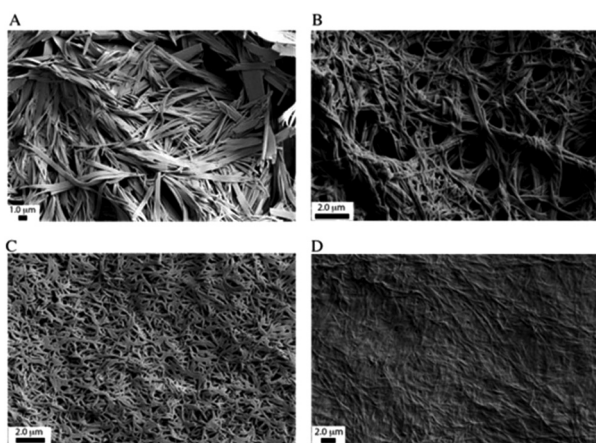


Fig. 3 Scanning electron micrographs of the dry precipitate for ligand **3** (A) and xerogels **3-Fe** (B); **3-Co** (C) and **3-Ni** (D) obtained from 2% $\text{DMSO}/\text{H}_2\text{O}$ samples.

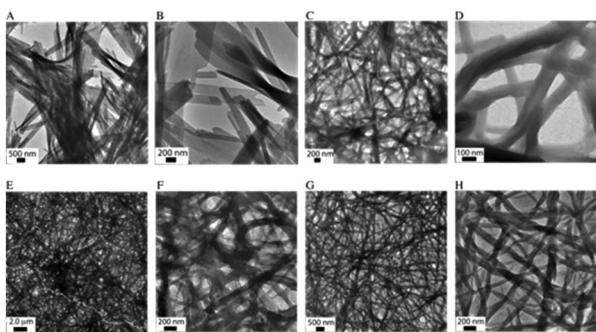


Fig. 4 Transmission electron micrographs of the dry precipitate for **3** (A & B) and xerogels for **3-Fe** (C & D); **3-Co** (E & F) and **3-Ni** (G & H) obtained from 2% $\text{DMSO}/\text{H}_2\text{O}$ samples.

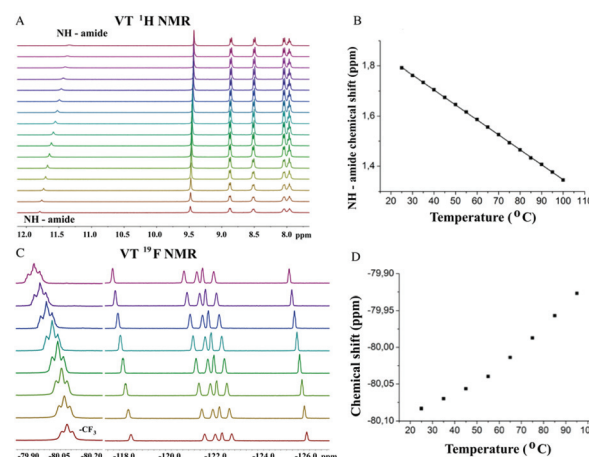


Fig. 5 Variable temperature NMR spectroscopy experiments. (A) Partial VT ^1H NMR spectra of 2% $\text{DMSO-}d_6$ - D_2O gel of **3-Fe**. (B) Plot showing change in the chemical shift values of amide proton signal as a function of temperature. (C) VT ^{19}F NMR of 2% $\text{DMSO-}d_6$ + D_2O gel of **3-Fe**. (D) Change in the chemical shift value of fluorine signals as a function of temperature.

cluded that hydrogen bonding is one of the driving forces involved in self-assembly.⁵⁶ Furthermore, the Fourier transform infrared (FT-IR) spectroscopy studies support this observation. The FT-IR spectrum of the bulk solid **3** was compared to the xerogel of aggregates of **3** and metalogels **3-M** (M = Fe, Co, Ni). The spectrum of bulk solid **3** showed amide -NH stretching at 3323 cm⁻¹ which shifts to 3245 cm⁻¹ for the xerogels, clearly indicating the presence of a hydrogen bond through the amide moieties (see Fig. S26†). In order to investigate the role of fluorine, a VT ¹⁹F NMR experiment was performed. The ¹⁹F signals underwent a progressive broadening and downfield shift upon temperature increase (Fig. 5C). Even though, the change in the chemical shift value is only 0.2 ppm, it is significant considering the strength of the F...F interaction.^{40,61} Furthermore, FT-IR was employed to investigate the influence of gelation on the C-F stretching frequency, which is usually found in the range of 1000–1400 cm⁻¹. The C-F stretching frequency for bulk solid **3** exhibited two narrow peaks at 1200 and 1144 cm⁻¹. However, for the case of the metalogels **3-M** (M = Fe, Co, Ni), the former signal led to a broad band peaking at 1198 cm⁻¹, meanwhile, the latter resolved into two partially merged signals at 1146 and 1130 cm⁻¹ (see Fig. S27†). These results certainly indicate that the fluorine atoms play an essential role in the polymerization process,⁴⁰ in excellent agreement with the conclusions obtained by VT ¹⁹F NMR. This strongly suggests the existence of the robust F...F interactions in the gel state.

A carefully designed NMR spectroscopy experiment can provide, not only the gel melting temperature and structural information about supramolecular interactions responsible for gelation, but also the concentration of the aggregated and non-aggregated components, enthalpy, entropy and free energy of gelation. It has been argued that the NMR resonances observed in a gel-phase at room temperature are largely resulting from the free molecules (mobile, non-aggregated components), whereas the aggregated components are NMR silent.^{59,62} Experimental quantification is often achieved using a standard and monitoring the intensity of the observed gelator signal. In our experiment, fluorobenzene was used as an external standard in order to quantify the ratio between the aggregated and non-aggregated fraction in the **3-Fe** gel upon increasing the temperature. The experiments were performed from 20 to 100 °C with 5 °C increments. When the concentration of the free molecules (in mM) was plotted against the temperature a steady increase until 60 °C was observed (Fig. 6A). Above 60 °C the concentration remained constant indicating that a gel-sol transition occurs at 60 °C and all the components are disassembled. This temperature is slightly lower than the T_{gel} value (T_{gel} **3-Fe** = 69 °C) obtained by placing a sealed gel sample upside-down in a thermally controlled oil bath, increasing the temperature and assigning the T_{gel} as the temperature at which the gel starts to flow. In such experiments, done in an oil bath, the oil temperature might slightly differ from the temperature of the gel inside the test tube. Additionally, the conditions in which the gel undergoes transformation into a solution are different, therefore enabling a

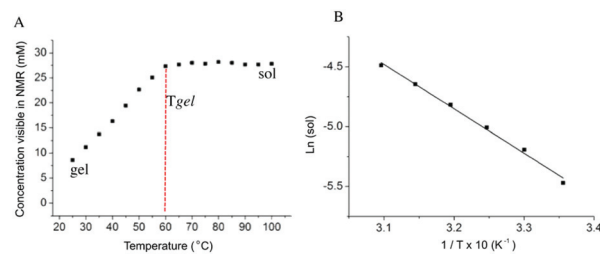


Fig. 6 Determination of thermodynamic parameters. (A) Concentration of the liquid-like phase free gelator as a function of temperature determined using fluorobenzene as an external standard in a 2% DMSO-*d*₆ + D₂O gel of **3-Fe**. (B) van't Hoff's treatment of VT NMR data, ln(solubility) against inverse of temperature.

small deviation within the T_{gel} measurements through different methods. With increasing temperature the gel fibers disassemble due to the thermal motion forming a non-aggregated material and therefore sharper peaks. The process of disassembly can be considered as a solubilization process; therefore, the van't Hoff treatment of the VT NMR data can be applied (see Fig. 6B and ESI†).^{17,63} The gelator molecules are considered to be soluble if they are observable by NMR and form insoluble aggregates if they are NMR silent. The data suitable for the van't Hoff treatment come from the measurements made well below T_{gel} and above MGC in order to avoid the initial steps of gelation. ΔH_{diss} and ΔS_{diss} are assumed to be temperature independent. For an ideal solution, the solubility (sol) at a certain temperature can be fitted to the van't Hoff equation (eqn 1).

$$\ln(\text{Sol}) = (-\Delta H_{\text{diss}}/RT_{\text{eq}}) + (\Delta S_{\text{diss}}/R) \quad (1)$$

where ΔH_{diss} and ΔS_{diss} are the enthalpy and entropy of dissolution, respectively. The T_{eq} is the equilibrium temperature and R is the gas constant. The gelation process (which is the opposite of the dissolution process) is enthalpically favourable ($\Delta H_{\text{diss}} = 31(1)$ kJ mol⁻¹) and entropically disfavoured ($\Delta S_{\text{diss}} = 58(4)$ J mol⁻¹ K⁻¹) as expected for an assembly of ordered fibers through hydrogen bonding and F...F interactions.⁵⁵ The value of the free energy is $\Delta G_{\text{diss}} = 13(2)$ kJ mol⁻¹. Moreover, a parallel van't Hoff study was performed by monitoring the signal intensity of the gelator using VT ¹⁹F NMR. The thermodynamical values obtained by applying van't Hoff treatment to VT ¹⁹F NMR ($\Delta H_{\text{diss}} = 29(1)$ kJ mol⁻¹, $\Delta S_{\text{diss}} = 51(4)$ J mol⁻¹ K⁻¹ and $\Delta G_{\text{diss}} = 13(2)$ kJ mol⁻¹) are in good agreement with the values obtained by VT ¹H NMR (see the ESI†). To the best of our knowledge, this is the first example where thermodynamic parameters of the sol-gel transition are obtained through VT ¹⁹F NMR spectroscopy.

Rheological properties

In order to understand the mechanical properties and response to mechanical stimuli rheological measurements for aggregates of **3** and gels **3-Fe**, **3-Co** and **3-Ni** were carried out (Fig. 7). First, the sol-gel transition was investigated by using an oscillatory time sweep experiment with 2% gels from

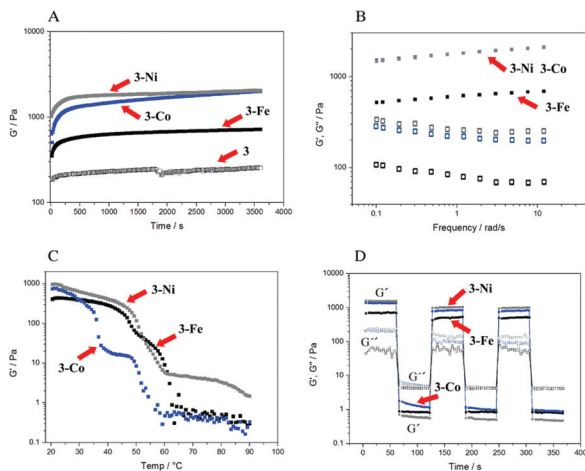


Fig. 7 Rheological properties of 2% DMSO/H₂O gels derived from gelators. (A) Time sweep experiments. (B) Frequency sweep. (C) Temperature sweep experiments. (D) Step strain experiments demonstrating fast self-healing.

20 min to 2 h. The time sweep experiments were performed within the viscoelastic regime (10% strain) with 6.2 rad s^{-1} under controlled temperature (Fig. 7A). This experiment allows the gelation kinetics as there will be a rapid change in the storage (G') and loss modulus (G'') upon gel formation. On the other hand, it also provides information related to the time required to form a stable gel. The elastic modulus G' for free ligand **3** was found to be 0.2 kPa which is an order of magnitude lower than that of the metallogels. Among metallogels the gel formation follows the order $\text{Fe} > \text{Co} \sim \text{Ni}$ with elastic moduli $\mathbf{3-Ni}$ (1.6 kPa) $>$ $\mathbf{3-Co}$ (1.4 kPa) $>$ $\mathbf{3-Fe}$ (0.7 kPa) (Fig. 7A). The frequency sweep experiments showed that G' of the gels is higher than G'' , confirming that the materials under study are viscoelastic solids (Fig. 7B). The gel-sol transition was followed using a temperature sweep rheological experiment (Fig. 7C). Temperature ramps from 20 to 90 °C were performed with a 5 °C min^{-1} heating rate. Unlike many supramolecular gels, which show rapid changes in their elastic moduli upon gel-sol transition, the gels here showed some unique properties: they manifested a two-step melting behaviour, where the elastic moduli of gels decreased rapidly by two orders of magnitude at around 45 °C, 35 °C and 55 °C for **3-Fe**, **3-Co** and **3-Ni** respectively (Fig. 7C). Further heating resulted in plateaus for 5–10 °C before complete melting. This can be ascribed to the structural re-organization of the gel network and the role of several weak interactions responsible for gelation as shown by the VT NMR studies. The initial response can be attributed to the weakening of the F...F interactions, followed by a disruption of the H-bonding network. Then, the step-strain experiment was performed in order to investigate the gel-sol transition and self-healing understood as the gel strength recovery, under several cycles. For the step strain experiments controlled strains of 0.1% and 150% were cycled for 60 s, respectively. The gels showed a rapid response to increased strain by turning into viscoelastic liquids indicated by the rapid

decrease in G' by three orders of magnitude and well below that of G'' (Fig. 7D). The application of increased strain also appears to break the structure further during the 60 s experiment as shown by the gradually decreasing elastic moduli values. Upon switching to lower strain the gels recover most of their original mechanical strength almost instantly *i.e.* rapid self-healing. Additionally, the process can be repeated as shown in Fig. 7D. However, slightly lower elastic moduli after the first high-low strain cycle and gradual build-up are observed, therefore indicating that the structure build-up to the equilibrium state would require longer periods of “rest” (low strain).

Conclusions

In conclusion, we have presented a simple terpyridine ligand decorated with a perfluoroalkyl chain that leads to metallo-supramolecular gels in aqueous dimethyl sulfoxide. Our work demonstrates the importance of not only the fluorine-fluorine interactions but also the hydrogen bonding and the counter anion. Therefore, being an anion specific process with potential use in molecular recognition, only chloride induces the gelation. Furthermore, we showed that morphological, thermal and mechanical properties depend on the nature of the metal ion. Indeed, SEM and TEM images show clear differences in the size and morphology of the gel fibers. Additionally, **3-Fe** and **3-Ni** are thermally more stable than **3-Co**, meanwhile rheological experiments showed that G' follows the trend $\mathbf{3-Ni} > \mathbf{3-Co} > \mathbf{3-Fe}$. Thermodynamic parameters of supramolecular gelation based on ^1H and, for the first time, on ^{19}F VT NMR resonances reveal that gelation is enthalpically favourable and entropically disfavourable. The excellent agreement between the results obtained by ^1H and ^{19}F VT NMR demonstrates the versatility of different spin active nuclei in determining thermodynamic parameters and could be further employed in supramolecular gels lacking hydrogen atoms. Metallogels undergo step-wise melting behaviour as suggested by ^1H NMR and supported by rheological measurements. More interestingly, metallogels exhibit self-healing properties under step-strain rheological experiment. Repeated cycles of high-low strain were applied to metallogels. A decrease of three orders of magnitude in the values of G' turns into a fast strength recovery (self-healing) upon removing the strain. Our results suggest that the incorporation of fluorine atoms induces metallogelation with novel thermal and mechanical properties.

Acknowledgements

The Academy of Finland (KR grant no. 263256, 265328 and 292746), the Centre of Excellence in Molecular Engineering of Biosynthetic Hybrid Materials (HYBER, 2014-2019), the ERC Advanced Grant MIMEFUN, AXA funds and the University of Jyväskylä are kindly acknowledged for the financial support.

This work made use of the Aalto University Nanomicroscopy Center (Aalto-NMC) facilities.

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