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Insulator-to-Metal Transition and Isotropic Gigantic Magnetoresistance in Layered Magnetic Semiconductors

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Magnetotransport, the response of electrical conduction to external magnetic field, acts as an important tool to reveal fundamental concepts behind exotic phenomena and plays a key role in enabling spintronic applications. Magnetotransport is generally sensitive to magnetic field orientations. In contrast, efficient and isotropic modulation of electronic transport, which is useful in technology applications such as omnidirectional sensing, is rarely seen, especially for pristine crystals. Here a strategy is proposed to realize extremely strong modulation of electron conduction by magnetic field which is independent of field direction. GdPS, a layered antiferromagnetic semiconductor with resistivity anisotropies, supports a field-driven insulator-to-metal transition with a paradoxically isotropic gigantic negative magnetoresistance insensitive to magnetic field orientations. This isotropic magnetoresistance originates from the combined effects of a near-zero spin–orbit coupling of Gd3⁺-based half-filling *f***-electron system and the strong on-site** *f***–***d* **exchange coupling in Gd atoms. These results not only provide a novel material system with extraordinary magnetotransport that offers a missing block for antiferromagnet-based ultrafast and efficient spintronic devices, but also demonstrate the key ingredients for designing magnetic materials with desired transport properties for advanced functionalities.**

1. Introduction

Electronic transport is one of the most widely adopted techniques to uncover physical properties of materials. Owing to its sensitivity to density of states and carrier scatterings, applying a magnetic field modulates electronic transport via various mechanisms that are directly associated with the fundamental properties of materials. The competition of diverse contributing mechanisms leads the electrical resistivity to change with the direction and strength of the applied magnetic field. A variety of magnetotransport phenomena have been well-explored, such as weak (anti)localization,^{[\[1\]](#page-9-0)} quantum oscillations,[\[2\]](#page-9-0) giant magnetore-sistance (GMR),^{[\[3\]](#page-9-0)} and colossal magnetoresistance (CMR) .^{[\[4\]](#page-9-0)} This venerable subject remains rather vibrant in contemporary condensed matter and materials physics research, not only because it provides an effective approach to understand the electronic properties of materials, but also

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owing to emergence of novel phenomena, such as chiral magnetic effect^[5-7] and extremely large positive magnetoresistance $(MR)^{[6-9]}$ in topological quantum materials.

Efficient modulation of electron conduction by magnetic field, which is manifested into resistance increase or decrease and characterized by MR, is scientifically interesting and technologically useful. Generally, MR is sensitive to the direction of the magnetic field with respect to the applied current. Examples include ordinary orbital MR due to Lorentz force that is sensitive to the angle between field and current,^{[\[10\]](#page-10-0)} GMR caused by spindependent scattering in magnetic multilayers and hence naturally anisotropic, $[11]$ and chiral magnetic effects induced by the magnetic field component parallel to the current orientation.^[5-7] In contrast, strong MR independent of magnetic field direction is much less common. Ferromagnetic metals and alloys show small MR anisotropy of a few percent, but the amplitude of MR is too small to be technologically useful.^{[\[12\]](#page-10-0)} The traditional CMR effect in correlated materials is considered to be more isotropic than GMR effects, but it generally varies by at least 30% percent depending on magnetic field direction.^{[\[13,14\]](#page-10-0)} Recently, a few non-oxide materials with strong negative MR have been discovered, such as EuMnSb_{2} ,^{[\[15,16\]](#page-10-0)} EuTe_{2} ,^{[\[17,18\]](#page-10-0)} $\text{Mn}_{3}\text{Si}_{2}\text{Te}_{6}$,^{[\[19,20\]](#page-10-0)} and $EuCd₂P₂.^[21]$ $EuCd₂P₂.^[21]$ $EuCd₂P₂.^[21]$ These materials all display substantial fielddirection dependent MR.

Magnetotransport phenomena typically arise from modulations in carrier density and/or scattering by the applied magnetic field. Therefore, the dominance of anisotropic magnetotransport is not surprising, given that both the magnetic field and electron transport are directional. Anisotropic and strong MR has already found applications in electronic compasses, position and motion sensing, and hard disk drives. On the other hand, isotropic MR, if realized, can provide omnidirectional sensing capabilities,

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serving as a valuable complement to anisotropic MR-based devices. This synergy allows for the acquisition of comprehensive information for magnetic field sensing and magnetic imaging. Here we propose a strategy to create isotropic colossal MR independent of magnetic field direction in compounds with vanishing magnetic anisotropy, through an insulator-to-metal transition driven by the same-atom *f*–*d* exchange splitting. We present a magnetic semiconductor GdPS as such a case, in which the zero orbital angular momentum for *f*-electrons yields a near-zero magnetic anisotropy. The substantial magnetization from halffilled *f*-orbitals strongly splits the Gd *d*-bands through an on-site exchange interaction, and results in a metallic state with welldefined Fermi Surface with unprecedented negligible anisotropy for MR.

2. Results and Discussion

2.1. Strategy Toward Colossal Isotropic MR

Isotropic magnetotransport is challenging from the perspective of electron dynamics as mentioned above. A more feasible approach is to couple magnetism with carrier scattering and/or scattering density in an isotropic manner. This approach requires isotropic magnetization which is usually challenging in crystalline materials because magnetic orders are generally characterized by magnetic easy and hard axes. Fortunately, this task is addressable since the mechanism of magnetic anisotropy has been well explored. Spin–orbital coupling (SOC) and crystal field interactions play key roles in determining magnetic anisotropy of bulk crystalline materials.[\[22\]](#page-10-0) Electron orbitals are oriented by the lattice due to electrostatic interaction, so that the coupling between electron orbits and spins gives rise to easy-axis or easy-plane anisotropy. Strong SOC and field angledependent magnetization can be utilized to create anisotropic MR.[\[20\]](#page-10-0) Here, minimizing SOC and the crystal field effect may lead to isotropic magnetism. Since this work aims to develop a universal approach to create isotropic magnetotransport, we look into the commonly seen magnetism arising from *d*- or *f*-electrons. *d*-electrons are generally more sensitive to crystal fields. Even though orbital angular momentum for d orbitals is usually quenched or partially quenched, SOC still creates resid-ual orbital moment and strong magnetic anisotropy.^{[\[22,23\]](#page-10-0)} For *f* electrons that are well-screened from the crystal field, however, they are characterized by even stronger SOC and hence significant magnetic anisotropy is expected, which has been known to be a key enabling factor of rare earth-based permanent magnet.[\[22\]](#page-10-0) An exception applies to the vanishing orbital angular momentum L for half-filled *f*-orbitals, where SOC is effectively zero and the magnetic anisotropy is mainly attributed to shape anisotropy.[\[22\]](#page-10-0) Therefore, starting from the common +3 valence for rare earth ions, Gd^{3+} with a half-filled $4f^7$ configuration offers the opportunity to realize isotropic magnetism.

Indeed, small magnetic anisotropic energy (MAE) has been found in some Gd compounds such as ferromagnetic GdN (0.45 μeV per Gd) and GeFe₂ (9 μeV per Gd),^{[\[24\]](#page-10-0)} but strong isotropic MR have yet been discovered. The next challenge is to establish an approach to couple transport with magnetism to realize isotropic and strong MR. In magnetic materials, in addition **www.advancedsciencenews.com www.advmat.de**

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Figure 1. Strategy for realization isotropic magnetoresistance. a) Schematic band structure for the ground state (zero magnetic field) of an AFM semiconductor insulator, showing a gap opening at the Fermi energy E_F . Exchange energy E_x is zero due to the vanished net magnetic moment in AFM state, as illustrated in the bottom. b) Applying magnetic field polarizes the moments to form an FM state, creating strong net moment and turn on exchange interaction that split the band. Sufficiently strong exchange splitting drives a band to cross the E_F , resulting in a metallic state. c) Isotropic magnetization. The magnetization (*M*) of the sample is induced and parallel to the external magnetic field (*H*), rotating in-phase with the field rotation.

to CMR and GMR in a few materials that do not demonstrate isotropic MR as mentioned above, weak MR due to the fieldmodulation of spin scattering for charge carriers is more gener-ally seen.^{[\[25,26\]](#page-10-0)} To realize strong and isotropic MR, field-induced metal-to-insulator (or vice versa) transition originating from electronic band reconstruction provides an effective approach. This can be realized via exchange splitting of electronic bands that results in gap closing or opening. Exchange interaction is sensitive to the net moment but not the direction of the moments. Therefore, as illustrated in **Figure 1**, starting from an antiferromagnetic (AFM) ground state with zero net moment, finite magnetization induced by magnetic field turns on exchange splitting and modulates electronic bands. For isotropic magnetization, this band structure modulation would be isotropic with respect to the direction of the applied magnetic field. Because *f*electrons are usually localized, here we pursue exchange interaction between *f* and orbitals near the Fermi energy (E_F) from the same Gd^{3+} ions to maximize exchange splitting.^{[\[27\]](#page-10-0)} Since the exchange interaction splits spin up and down bands, it is easier to start from a gapped system and close the bandgap by splitting bands near the E_F , as opposed to opening a gap for a metallic system which may involve band structure complications. With this approach, field driven insulator-to-metal transition is expected.

2.2. Isotropic Magnetism in GdPS

To realize the above scenario, the bandgap of the candidate material should be within the band shifts achievable by reasonable exchange splitting, while sufficiently large to minimize the effect of inevitable defects, self-doping, and thermally excited carriers. Therefore, the candidate material should be an AFM semiconductor possessing a moderate bandgap, *f*-electron magnetism and *d*-bands near E_F . GdPS is a system meeting all the above needs with Gd^{3+} to ensure isotropic magnetism. $GdPS$ was dis-covered decades ago^{[\[28\]](#page-10-0)} but only the electronic structure has been explored theoretically.^{[\[29\]](#page-10-0)} This material adopts a layered crystal structure consisting of flat phosphorus planes sandwiched by Gd-S layers (**Figure 2**[a\)](#page-4-0). It is structurally similar to the tetrag-onal ZrSiS^{[\[29\]](#page-10-0)} with Si square-net planes,^{[\[30,31\]](#page-10-0)} with the difference that GdPS undergoes an orthorhombic structure distortion where the P plane fragments into an arm-chair chain-like struc-ture, as shown in Figure [2b.](#page-4-0) More details on structure determination are provided in Supporting Information. Earlier theoretical studies[\[29\]](#page-10-0) have shown that such a structural distortion plays a vital role in modulating the electronic phase. As shown in Figure $2c$, in tetragonal structure (such as $ZrSiS$), the $C₂$ symmetry leads to a Dirac nodal-line state that is characterized by linearly dispersed bands crossing near the E_F .^{[\[29–31\]](#page-10-0)} In GdPS (Figure [2d\)](#page-4-0),

Square-net, Dirac crossing Arm-chair chain, gapped

Figure 2. Electronic structure of GdPS. a) The orthorhombic crystal structure of GdPS. b) Arm-chair chain structure of the P layers. c) The tetragonal structure with square-net layer leads to Dirac crossings. d) the orthorhombic distorted arm-chair chain-type P layer opens a gap at the Dirac point (see Figure S4 in Supporting Information for detailed band structures). e) Electronic band structure of GdPS in the AFM ground state. The right panel shows the checkerboard (Néel)-type AFM magnetic structure for the ground state determined by calculations (see Figure S6, Supporting Information). While the magnetic structure is determined, the Gd moments' orientations are schematic, which is difficult to determine owing to the very weak MAE. f) Electronic band structure of GdPS in the polarized FM state. Magnetic structure for the FM state shows the polarization of Gd moments along the direction of magnetic field. g) Exchange splitting between spin up and down bands for the Gd *d*-band near Γ, which can be seen by overlaying the electronic structures of the AFM state (solid line) and the FM state (dashed line). The comparison of the entire electronic band structure is shown in Figure S5 in Supporting Information.

however, the arm-chair chain configuration of P atoms loses one mirror plane as compared to the square lattice, which strongly gaps the Dirac crossing^{[\[29\]](#page-10-0)} (see Figure $S4$ in Supporting Information for more details) and eventually leads GdPS to be a semiconductor.

DFT calculations (Figure S6, Supporting Information) reveal that GdPS adopts an AFM ground state with a checkerboard (Néel)-type magnetic structure. Though the predicted magnetic structure is difficult to verify by neutron scattering due to the strong neutron absorption of Gd, the AFM ground state is consistent with our magnetization measurements that reveal the absence of irreversibility between zero-field-cooling and fieldcooling susceptibility (**Figure 3**[a.](#page-5-0) More data shown in Figure S7 in Supporting Information), and the linear field dependence of isothermal magnetization at low fields (Figure [3b\)](#page-5-0). At high field around 15 T, isothermal magnetization saturates quickly (Figure [3b\)](#page-5-0), which should be attributed to a spin-flip transition to a polarized ferromagnetic (FM) state, as will be discussed below. The AFM transition temperature T_N displays a slight sample variation between 7 and 7.5 K, which should be ascribed to the light P vacancies (see Experimental Section). At the AFM ground state, DFT calculations estimate a moderate bandgap of \approx 0.5 eV (Figure 2e), which is promising for realizing the desired insulator-to-metal transition as will be shown below.

The vanishing SOC of Gd^{3+} , as well as the very weak ligand SOC from the light elements P and S, leads to the needed isotropic magnetism. As shown in Figure [3b,](#page-5-0) despite the layered crystal structure, magnetizations measured under in-plane (*H*//*ab*) and out-of-plane (*H*//*c*) magnetic fields are almost identical for the temperature below (1.5 K) and above (10 K) T_N . The

isotropic magnetization persists to higher magnetic fields and saturates above \approx 15 T for both field orientations, implying a transition from an AFM to a polarized FM state. The saturation moment of ≈7.2 μ_B per Gd is consistent with the ⁸S^{7/2} multiplet for Gd^{3+} . The moment polarization persists even above T_N , with a nearly unchanged saturation moment at 10 K, as shown in the lower panel of Figure [3b.](#page-5-0) The large magnetization is important for the desired exchange splitting. Because sample rotation is difficult during high field magnetization measurements, tunnel diode oscillator (TDO) was adopted to examine the moment polarization with field orientations (Figure [3c\)](#page-5-0), from which the moment polarization field H_n at various magnetic field orientations can be extracted. As shown in Figure $3d$, H_p at 0.5 K displays very weak angular dependence. The observed isotropic magnetism in GdPS is consistent with the calculated small MAE of 30 μeV per Gd for the AFM ground state (see Experimental Section). The MAE can be further reduced to only 2 μeV per Gd in the polarized FM state. The combined experimental and theoretical results indicate that the Gd moments rotate freely with the external magnetic field, which is important for the isotropic MR as shown below.

2.3. Isotropic Gigantic MR and Insulator-to-Metal Transition in GdPS

Given the large Gd^{3+} moment, the field-driven spin-polarization leads to strong exchange splitting that modifies the electronic structure. As shown in Figure 2e, in the AFM ground state, the conduction band edge of the semiconducting GdPS consists of

Figure 3. Magnetic properties of GdPS. a) Zero-field-cooling (ZFC) and field-cooling (FC) temperature dependent magnetic susceptibility for GdPS measured with out-of-plane (*H*//*c*) magnetic fields of 1 T. Data taken under in-plane field is provided in Figure S7 in Supporting Information. b) Fielddependent magnetization of GdPS at (top) 1.5 K and (bottom) 10 K, measured with field along the *c*-axis (perpendicular to the P plane) and within the *ab*-plane (parallel to the P plane). Magnetization exhibits saturation behavior at high field, indicating a polarized FM state. c) TDO measured at various magnetic field orientations. The transition-like feature near 15 T (indicated by the black arrow) signatures the moment polarization to an FM state. At high field, dHvA oscillation in TDO can be observed. d) Critical field for moment polarization, as a function of the magnetic field orientation, extracted from TDO measurement. The polarization field H_p is essentially field-orientation-independent.

Gd *d*-bands around Γ and P *p*-bands around Y and T points of the Brillouin zone. In the polarized FM state, the strong net moment from Gd sublattice leads to exchange interactions that split both bands, where the Gd *d*-band splitting at the conduction band edge is as large as 0.5 eV (Figure $2f$,g, and Figure $S5$ in Supporting Information) owing to the same-atom *f*–*d* exchange, as compared to a much smaller band splitting of 0.1 eV for the P *p*-bands. The strong Gd d-band splitting at Γ lowers the energy of the spinpolarized band, replacing the P *p*-band edge at Y as the new conduction band minimum that approaches the E_F . In real materials, when P vacancies increase E_F , a Fermi surface from Gd *d*-band centered at Γ can emerge, leading to an insulator-to-metal transition which has indeed been observed in our magnetotransport experiments described below. Such a process is only determined by the strength of the exchange splitting, thus it is isotropic with respect to magnetic field due to isotropic magnetization of GdPS.

As shown in **Figure** 4[a,](#page-6-0) the in-plane resistivity (ρ_{xx}) measured with current following parallel to the phosphorus plane of GdPS (*I* // *ab*) increases with cooling, a typical non-metallic behavior for the gapped electronic structure in the AFM ground state. Applying a perpendicular magnetic field (*H*⊥*ab*) changes resistivity drastically, and eventually leads to a metallic-like temperature dependence with decreased resistivity upon cooling. The metal state is further verified by quantum oscillations under high magnetic fields as will be discussed later. Such an insulator-to-metal transition is accompanied by gigantic negative MR. At *T* = 2 K, resistivity drops drastically up on increasing field as shown in Figure [4b.](#page-6-0) The corresponding MR, quantified by the field-induced resistivity change normalized to its zero-field value, reaches 97.3%. When normalized to the 9T resistivity value following previous reports,[\[15,17,32\]](#page-10-0) the MR is as high as 3659%. The gigantic negative MR is reproducible in all samples we measured, reaching up to 99%. Transport data and summary on more samples are provided in Figure S8 and Table S3 in Supporting Information. The strong negative MR occurs in a wide temperature and field region. As shown in Figure [4b,](#page-6-0) at 2K, 50% reduction of resistivity is easily achievable under a magnetic field of 3 T, which is much lower than the spin polarization field of 15 T. Also, heating up to 100 K which is well above the magnetic ordering temperature, MR remains as high as nearly 50% at 9 T. This further confirms the exchange splitting scenario that is driven by net magnetization without a need for magnetic ordering or full moment polarization. As shown in Figure 3a, magnetic susceptibility at 50 K remains as significant as 1/3 of the maximum value at low temperatures. In addition, as shown in Figure 3b, at 10 K which is already above T_N , magnetization still displays polarization behavior with saturation magnetization almost identical to that at 1.5 K. Therefore, exchange splitting remains significant due to the strong magnetization at escalated temperatures, leading to strong MR at high temperatures and making GdPS useful for practical applications where operations at high temperature and low field are preferable.

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Figure 4. Gigantic, isotropic negative MR in GdPS. a) Temperature dependence for in-plane resistivity ρ_{xx} , measured under various magnetic field from 0 to 9 T. Inset: measurement setup. b) Magnetic field dependence for ρ_{xx} , measured at various temperatures from 2 to 300 K. c) Angular dependence for MR at $T = 5$ K for in-plane resistivity ρ_{xx} , measured by rotating field perpendicular to (θ -dependence, left) and within (φ -dependence, right) the phosphorus plane. Inset: schematic of magnetic field rotation. d) Temperature dependence for out-of-plane resistivity ρ_{zz} , measured under various magnetic field from 0 to 9 T. Inset: measurement setup. e) Magnetic field dependence of ρ_{zz} , measured at various temperatures from 2 to 300 K. f) Angular dependence for MR at $T = 5$ K for out-of-plane resistivity ρ_{zz} , measured by rotating field perpendicular to (θ -dependence, left) and within (φ -dependence, right) the phosphorus plane. Inset: schematic of magnetic field rotation.

As expected, MR in GdPS is isotropic. Figure 4c depicts the angular dependent MR at $T = 5$ K measured by rotating magnetic field both perpendicular to (θ -dependence) and within (φ dependence) the phosphorus plane, which reveals that MR is essentially independent of the field orientation. This isotropic MR is reproducible and persists at high temperatures beyond T_N , as shown in Figure $S9$ in Supporting Information. To better quantify the isotropy of MR, we define the isotropic index by the difference of MR maximum and minimum in the angular dependent measurements, normalized to the average MR (see Experimental Section). Such definition takes the strength of MR into account so that it characterizes both isotropy and amplitude of MR. The calculated isotropic index is as small as 0.8% and 0.2% at 9 T for out-of-plane (θ -) and in-plane (φ -) field rotation, respectively.

Similar isotropic and gigantic MR is also present in outof-plane electronic transport, where the electrical current following perpendicularly to the phosphorus plane, as shown in Figure 4d,e. Compared to in-plane transport, the zero-field outof-plane resistivity ρ_{zz} at low temperatures is much larger than ρ_{xx} by a factor of 35 (Figure 4d), consistent with the layered crystals structure of GdPS. However, ρ_{zz} is much more sensitive to magnetic field and drops strongly under a small magnetic field. As shown in Figure 4e, at 2 K, negative MR up to 90% for ρ_{zz} can be induced by a small field of 1 T. Further increasing field to 9 T boosts MR to 99.99%, which converts to 740167% when normalizing to the 9 T resistivity value. Similar to ρ_{xx} , strong negative MR in ρ_{zz} persists to high temperatures with a much greater value of 93.8% at 100 K and 9 T (Figure 4e), and displays isotropic field orientation dependence with isotropic index *<* 0.01% for both out-of-plane and in-plane rotations (Figure 4f). The isotropic MR in ρ_{zz} is reproducible and also persists at high temperatures, as shown in the Figure S10 in Supporting Information. Such extremely strong modulation of the out-of-plane transport is associated with the evolution of the electronic structure due to exchange splitting, as will be discussed later.

The isotropic and gigantic MR in both in-plane and out-ofplane magnetotransport drastically differs from other strong but anisotropic MR compounds such as EuMnSb_{2} , $^{[15,16]}$ $^{[15,16]}$ $^{[15,16]}$ EuTe_{2} , $^{[17,18]}$ $^{[17,18]}$ $^{[17,18]}$ $Mn_3Si_2Te_6$, $[19,20]$ $[19,20]$ and $EuCd_2P_2$. $[21]$ The only known example of large, isotropic MR material is a disordered topological insulator with delicate composition tuning to allow a formation of a percolation of the current paths under field, and hence the MR isotropy is sample dependent.^{[\[33\]](#page-10-0)} Therefore, GdPS represents the first example of isotropic gigantic MR in pristine crystals which is driven by a well-defined insulator-to-metal transition from exchange splitting. The isotropic magnetism of GdPS ensures isotropic MR at various temperatures (above or below T_N) and magnetic fields, which is distinct from other known materials showing strong negative MR as summarized in Figure S11 in Supporting Information. Such a unique property again highlights the potential of GdPS for applications.

Quantitatively comparing in-plane and out-of-plane transport at low temperatures, one can find that ρ_{zz} is much larger at 0 T but becomes comparable with ρ_{xx} at 9 T. This suggests that electronic transport becomes more isotropic under a magnetic field, which can be understood by the evolution of electronic structure during the insulator-to-metal transition. Electronic structure at zero magnetic field can be probed by angle-resolved **www.advancedsciencenews.com www.advmat.de**

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Figure 5. Electronic structure of GdPS. a–d) Electronic structure at zero field. APRES observation of the electronic band dispersion at 90 K along a) Γ-X and b) Γ-Y directions. An electron band slightly crosses the *E_F* at Y, leading to a small Fermi surface at Y point as shown in (c). d) Brillouin zone for GdPS. The black dotted square in c marks the surface Brillouin zone. e) Angular dependence for MR for in-plane resistivity ρ_{xx} , measured by rotating field perpendicular to (θ -dependence, left) and within (φ -dependence, right) the phosphorus plane. Inset: schematic of magnetic field rotation. e–i) Electronic structure of the polarized FM state probed by quantum oscillations. e) High field MR at *T* = 0.5 K for GdPS, measured at various magnetic field orientations. Upper inset shows the measurement schematic. Lower inset shows the SdH oscillation at high field. f) Field-angular dependence for SdH oscillation frequency obtained from data in (e) (see Figure S12, Supporting Information). The observed angular dependence matches well with the calculated Fermi surface formed from the exchange-split Gd d-band at Γ, assuming shifted *E_F* by 0.245 eV (see Experimental Section). g) Contour plot showing the dependence of the SdH oscillatory component ρ_{xx} ^{osc} with temperature and inverse magnetic field, constructed from SdH effect at various temperatures shown in Figure S13a in Supporting Information. Inset: Oscillation pattern at 2 and 12 K. h) Temperature- and i) field-dependence for SdH oscillation frequency of the same sample (see Experimental Section for extracting frequencies). Results in (e–i) are from the identical sample measured using e,f) a 41 T magnet and g–i) a 31 T magnet.

photoemission spectroscopy (ARPES). As shown in **Figure 5**a–d, ARPES observations are qualitatively consistent with the calculated band dispersions in Figure [2e.](#page-4-0) At the Y point, a small electron pocket is seen, which can be attributed to P vacancies that electron-dopes GdPS. Despite its presence, this electron pocket is not large enough to support full metallicity, as the non-metallic transport up to room temperature at zero field (Figure [4a\)](#page-6-0) implies our GdPS is a lightly doped semiconductor.

The electronic structure under magnetic fields, though unattainable by ARPES, can be resolved by quantum oscillations. Both Shubnikov-de Haas (SdH) oscillations in magnetotransport (Figure 5e) and de Haas-van Alphen (dHvA) oscillations in TDO measurements (Figure [3c](#page-5-0) and Figure S14d–f in Supporting Information) have been observed in our high field experiments. One typical example is shown in Figure 5e, from which SdH effects in MR data can be clearly seen above \approx 17 T at T = 0.5 K. Importantly, SdH oscillation is observed for any magnetic field orientations ranging from out-of-plane ($\theta = 0^{\circ}$) to nearly in-plane ($\theta = 82.6^{\circ}$) directions, which indicates a well-defined three-dimensional (3D) Fermi surface and rule out the possibility of quantum oscillation from surface states in insulators^{[\[34,35\]](#page-10-0)} The oscillation frequency does not vary strongly with field orientations, growing by only 16% from 500 T for $\theta = 0^\circ$ to 580 T for $\theta = 82.6^\circ$, as shown in Figure 5f. Since the oscillation frequency is proportional to the extremal Fermi surface cross-sectional area perpendicular to the

magnetic field orientation, such angular dependence implies a nearly spherical Fermi surface, which is consistent with the 3D Fermi surface at Γ from the DFT band structure. As discussed above (Figure [2f\)](#page-4-0), in the polarized FM state, the strong *f*–*d* exchange splitting may lead to a Fermi surface at Γ from the spinpolarized Gd d -band. Indeed, assuming a shifted E_F by 0.245 eV attributed to self-doping from the P vacancies, a nearly spherical Fermi surface at Γ has been discovered by our DFT calculations. The calculated angular dependence of oscillation frequency based on that Fermi surface matches well with the experimental observations, as shown in Figure 5f. Aside from the Gd *d*-band, other possible bands such as the P *p*-band are not resolvable by quantum oscillations.

The nearly spherical Fermi surface at Γ emerging with increasing magnetic fields is further supported by the surprising quantum oscillation anomalies. Figure 5g depicts the contour plot showing the temperature and field dependencies of the SdH effect under an in-plane magnetic field. Interestingly, the locations of the oscillation maxima and minima shift systematically with varying temperature, as illustrated by the bent stripes in Figure 5g. Typical data taken at 2 K and 12 K are shown in the inset, where misaligned oscillations are clearly seen. This behavior is observed in multiple samples and occurs in both SdH and dHvA effects (Figure S14, Supporting Information). Such strong shift of oscillation pattern with temperature has not been

observed in other materials. Further analysis reveals that it is caused by the temperature dependent oscillation frequency that drops from 560 T at 1.5 K to 450 T at 18 K, as shown in Figure [5h.](#page-7-0) This appears inconsistent with the standard quantum oscillation theory based on Landau quantization of electronic systems with unchanged charge densities, but agrees well with the scenario of exchange splitting-driven insulator-to-metal transition: The emergence of Fermi surface is due to the exchange-split Gd *d*-bands that crosses the E_F , so the size of the Fermi surface is determined by the strength of the sample magnetization which reduces upon warming. Hence, quantum oscillation frequency also evolves with magnetic field in the same manner as magnetization. As shown Figure [5h,](#page-7-0) the oscillation frequency at 1.5 K grows with field with a saturation behavior. At elevated temperatures (6 and 10 K), the frequency is lower but the field dependence becomes even stronger. All of these match well with the magnetization of GdPS that display saturation behavior at low temperatures and weakens at higher temperatures (Figure [3b\)](#page-5-0).

With the electronic structures revealed by ARPES and quantum oscillation, the much stronger MR in out-of-plane transport can now be understood. At zero magnetic field, transport is governed by P layers as revealed by ARPES observations, so that the transport is more two-dimensional-like and ρ_{zz} is much greater than ρ_{xx} . In addition, the out-of-plane transport through P and Gd-S layers occurs in a manner like tunneling because the P-layer is more conductive than Gd-S. Applying magnetic field reduces bandgap efficiently as discussed above, which strongly enhances out-of-plane transport. When the field is sufficiently strong, the exchange splitting induces a 3D spherical-like Fermi surface from Gd *d*-bands. Because carrier scattering is relatively less important in this insulator-to-metal transition process, electronic transport reflects the evolution of the dimensionality of electronic structure, leading ρ_{zz} to drops more strongly than ρ_{xx} and eventually becomes comparable with ρ_{xx} .

The layered crystal structure of GdPS allows exfoliation down to thin flakes (Figure S15, Supporting Information). Thus, this material offers an excellent platform for device integration, especially for novel omnidirectional sensing that is complementary to the established directional sensing based on materials with strong anisotropy. In addition to isotropy, compared to other materials showing transition between insulator and metal and even superconductor such as cuprates, some of which also present metal/superconductor-insulator crossover at relatively low field,^{[\[36\]](#page-10-0)} GdPS still present certain advantages. The AFM ground state with vanishing stray fields further enables ultra-fast and high-efficient device response.^{[\[37,38\]](#page-10-0)} In addition, in the field-induced metallic state, the Fermi surface formed from the exchange-split Gd *d*-band is spin polarized. Therefore, controlling E_F by chemical or electrostatic doping to remove possible contribution from the phosphorus bands may lead to a fully spin polarized Fermi surface that is highly desired for spintronics development. The above properties, together with the fact that the gigantic, isotropic MR occurs in pristine GdPS without the need for fine composition control, may open vast opportunities for its applications. Our finding highlights the effectiveness of utilizing exchange splitting-driven insulator-to-metal transition to realize isotropic and gigantic modulation of transport by magnetic field, which is enabled by suppressed crystal field effect in *f*electron systems and near-zero SOC in half-filled *f*-orbitals to create isotropic magnetism, as well as an AFM and semiconducting ground state. Such a strategy can be generalized to discover more candidate materials by computational high-throughput screening followed by experimental optimization of material parameters by doping or strain, to ensure band edge states by *d*-orbitals experiencing strong *f*–*d* exchange, as well as appropriate spin polarization fields and magnetizations at elevated temperatures to extend the phenomena towards high temperatures and low fields for technology applications.

3. Experimental Section

Sample Preparation: Single crystals of GdPS were grown by a direct chemical vapor transport with a stoichiometric ratio of Gd, P, and S as source materials and $TeCl₄$ as the transport agent. The growths were performed in a dual heating zone furnace with a temperature gradient from 1075 to 975 °C for 3 weeks. The produced crystals typically have either needle-like or plate-like structures (Figure S1, Supporting Information), both of which were confirmed to be the GdPS phase by the composition analysis using energy-dispersive X-ray spectroscopy and the structure analysis using X-ray diffraction.

The structure of the GdPS single crystal was determined by single crystal XRD. The structure parameters obtained from single crystal XRD refinement were given in Table S1 in Supporting Information. A careful structure investigation found that the atomic displacement parameter of P atoms (0.0116(10) \AA^2) is greater than other atoms including Gd1 (0.0065(5) \AA^2), Gd2 (0.0070(5) Å²), S1 (0.0060(12) Å²), and S2 (0.0050(12) Å²) under the same refinement conditions, which indicated the possible presence of vacancy at P atomic sites. The refinement of atomic occupancy of P atoms revealed the occupancy of 0.95, which indicates the presence of P vacancy that is consistent with the electron doping scenario revealed by ARPES.

Angle-Resolved Photoemission Spectroscopy: ARPES measurements were conducted out at Beamline 10.0.1 of the Advanced Light Source (ALS) in Berkeley, California, USA. The energy resolution was better than 30 meV and the temperature for measurement was 90 K. The incident photon energy was 55 eV. Samples were cleaved in situ and measured under vacuum better than 5×10^{-11} Torr.

Band Structure Calculations: The band structure was calculated us-ing density-functional theory (DFT) as implemented in VASP^{[\[39\]](#page-10-0)} using the Perdew-Burke-Ernzerhof (PBE) exchange-correlation functional^{[\[40\]](#page-10-0)} and projector augmented wave (PAW) pseudo-potentials. The cut-off for the kinetic energy of plane was to be 520 eV and a 4×4×1 k-point mesh was used in the Brillouin zone. The Hubbard U parameter was used for the Gd-4*f* electron (U_{eff} ^{Gd,5f} = 7 eV). The total energy was converged at least 10−⁶ eV (10−⁷ eV for MAE) during electron configuration relaxation. Spin– orbit coupling (SOC) was included in all calculations unless otherwise stated. To model the paramagnetic state for GdPS (Figure S4, Supporting Information), the calculation was performed by using a non-magnetic state.

Magnetic Anisotropy Energy Calculations: The calculation of the variation of MEA with polar angle θ and azimuthal angle φ provides stable structures as references for MAE calculations. As shown in Table S2 in Supporting Information, it is observed that in the case of AFM GdPS, the configuration with the magnetic moment oriented at $\theta = 45^{\circ}$ and $\varphi = 45^{\circ}$ is slightly more stable than other AFM configurations. Similarly, in the case of FM GdPS, the configuration with the magnetic moment oriented at $\theta =$ 0° and $\varphi = 0$ ° is slightly more stable than other FM configurations. These stable structures are considered as references for MAE calculations.

Extremal Fermi Surface Cross-Section Area and Associated Quantum Oscillation Frequency: To calculate the Fermi surface with the interpolation of the energy bands, the maximally localized Wannier functions (MLWF) for the ferromagnetic state of GdPS were obtained using the Wannier90 package.[\[41\]](#page-10-0) Moreover, the dHvA frequencies were calculated by the SKEAF code^{[\[42\]](#page-10-0)} with the Fermi surface information.

Measurements: The magnetization and magnetotransport up to 9 T were measured using a physical property measurement system

(PPMS DynaCool, Quantum Design). The standard 4-probe measurement was used for magnetotransport. The high field magnetization measured with VSM (up to 35 T), magnetotransport measurements (up to 31 T and 45 T), and TDO measurements (up to 45 T) were performed at the National High Magnetic Field Laboratory (NHMFL) at Tallahassee.

Anisotropic Index: Anisotropic index characterizes the degree of anisotropy. It is defined as the difference between the maximal and minimal MR at different magnetic field orientations $[MR(\theta)_{max} - MR(\theta)_{min}]$, normalized by the average MR [MR(θ)_{avg}] calculated by averaging the maximal and minimal MR:

$$
\text{Anisotropic index} \equiv \frac{\text{MR}(\theta)_{\text{max}} - \text{MR}(\theta)_{\text{min}}}{\text{MR}(\theta)_{\text{avg}}} = \frac{\text{MR}(\theta)_{\text{max}} - \text{MR}(\theta)_{\text{min}}}{[\text{MR}(\theta)_{\text{max}} + \text{MR}(\theta)_{\text{min}}]/2} (1)
$$

Therefore, anisotropic index takes a value between 0 and 2. A value of 0 means perfect isotropic when $MR(\theta)_{max} = MR(\theta)_{min}$, and a value of 2 refers to completely anisotropic when $\text{MR}(\theta)_{\text{min}} = 0$.

MR changes with temperature and magnetic field, which allows for extracting the anisotropic index as a function of the amplitude of MR, as shown in Figure S11 in Supporting Information. Other materials with strong negative MR, including EuMnSb₂,^{[\[15\]](#page-10-0)} EuTe₂,^{[\[18\]](#page-10-0)} Mn₃Si₂Te₆,^{[\[19\]](#page-10-0)} EuCd₂P₂,^{[\[21\]](#page-10-0)} disordered topological insulator TlBi_{0.15}Sb_{0.85}Te₂,^{[\[33\]](#page-10-0)} and CeCuAs₂[^{43]} are provided in Figure S11 in Supporting Information for comparison.

Extracting Temperature- and Field-Dependent Quantum Oscillation Frequency: Because the Fermi surface of the polarized FM state is formed from the exchange-split Gd d-bands, it evolves with magnetization and thus temperature- and field-dependent. Therefore, the corresponding quantum oscillation frequency, which is proportional to the extremal Fermi surface cross-section area perpendicular to magnetic field direction, also changes with temperature and magnetic field. For field dependence pre-sented in Figure [5i,](#page-7-0) the oscillation frequency cannot be extracted from the conventional Fourier transform because the oscillation data is in the field domain. Therefore, to determine oscillation frequency at various magnetic fields, the distance between neighboring oscillation peaks is evaluated, which is the oscillation period. The inverse of the period yields frequency for that field range, as shown in Figure S13c in Supporting Information.

For temperature dependence, the oscillation frequency is obtained from the Fourier transform of the oscillation pattern, as shown in Figure S13b in Supporting Information, which is the convention method for quantum oscillation analysis. For temperature-dependent Fermi surface in GdPS, this method bears small errors. That is because Fourier transform is applied to a certain field range containing multiple oscillation periods, but the frequency changes with field in GdPS. Nevertheless, given that the oscillation amplitude damps with reducing magnetic field, the peak position in Fourier spectrum is mainly determined by the oscillation in the high field range. Indeed, frequency extracted from Fourier transform in Figure [5h](#page-7-0) agrees with the high field (29– 31T) frequency obtained by the distance between neighboring oscillation peaks in Figure [5i.](#page-7-0)

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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Conflict of Interest

The authors declare no conflict of interest.

Data Availability Statement

The data that support the findings of this study are available from the corresponding author upon reasonable request.

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- [1] J. J. Lin, J. P. Bird, *J. Phys.: Condens. Matter* **2002**, *14*, R501.
- [2] D. Shoenberg, in *Magnetic Oscillations in Metals*, Cambridge University Press, Cambridge **2009**.
- [3] P. A. Grünberg, *Rev. Mod. Phys.* **2008**, *80*, 1531.
- [4] A. P. Ramirez, *J. Phys.: Condens. Matter* **1997**, *9*, 8171.

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- [5] A. A. Burkov, *J. Phys.: Condens. Matter* **2015**, *27*, 113201.
- [6] N. P. Armitage, E. J. Mele, A. Vishwanath, *Rev. Mod. Phys.* **2018**, *90*, 015001.
- [7] A. Bernevig, H. Weng, Z. Fang, X. Dai, *J. Phys. Soc. Jpn.* **2018**, *87*, 041001.
- [8] T. Liang, Q. Gibson, M. N. Ali, M. Liu, R. J. Cava, N. P. Ong, *Nat. Mater.* **2015**, *14*, 280.
- [9] J. Hu, S.-Y. Xu, N. Ni, Z. Mao, *Annu. Rev. Mater. Res.* **2019**, *49*, 207.
- [10] A. B. Pippard, in *Magnetoresistance in Metals*, Cambridge University Press, Cambridge **1989**.
- [11] L. B. Steren, A. Barthélémy, J. L. Duvail, A. Fert, R. Morel, F. Petroff, P. Holody, R. Loloee, P. A. Schroeder, *Phys. Rev. B* **1995**, *51*, 292.
- [12] T. McGuire, R. Potter, *IEEE Trans. Magn.* **1975**, *11*, 1018.
- [13] J. O'Donnell,, J. N. Eckstein, M. S. Rzchowski, *Appl. Phys. Lett.* **2000**, *76*, 218.
- [14] R.-W. Li, H. Wang, X. Wang, X. Z. Yu, Y. Matsui, Z.-H. Cheng, B.-G. Shen, E. W. Plummer, J. Zhang, *Proc. Natl. Acad. Sci. USA* **2009**, *106*, 14224.
- [15] Z. L. Sun, A. F. Wang, H. M. Mu, H. H. Wang, Z. F. Wang, T. Wu, Z. Y. Wang, X. Y. Zhou, X. H. Chen, *npj Quantum Mater.* **2021**, *6*, 94.
- [16] C. Yi, S. Yang, M. Yang, L. Wang, Y. Matsushita, S. Miao, Y. Jiao, J. Cheng, Y. Li, K. Yamaura, Y. Shi, J. Luo, *Phys. Rev. B* **2017**, *96*, 205103.
- [17] J. Yin, C. Wu, L. Li, J. Yu, H. Sun, B. Shen, B. A. Frandsen, D.-X. Yao, M. Wang, *Phys. Rev. Mater.* **2020**, *4*, 013405.
- [18] H. Yang, Q. Liu, Z. Liao, L. Si, P. Jiang, X. Liu, Y. Guo, J. Yin, M. Wang, Z. Sheng, Y. Zhao, Z. Wang, Z. Zhong, R.-W. Li, *Phys. Rev. B* **2021**, *104*, 214419.
- [19] Y. Ni, H. Zhao, Y. Zhang, B. Hu, I. Kimchi, G. Cao, *Phys. Rev. B* **2021**, *103*, L161105.
- [20] J. Seo, C. De, H. Ha, J. E. Lee, S. Park, J. Park, Y. Skourski, E. S. Choi, B. Kim, G. Y. Cho, H. W. Yeom, S.-W. Cheong, J. H. Kim, B.-J. Yang, K. Kim, J. S. Kim, *Nature* **2021**, *599*, 576.
- [21] Z.-C. Wang, J. D. Rogers, X. Yao, R. Nichols, K. Atay, B. Xu, J. Franklin, I. Sochnikov, P. J. Ryan, D. Haskel, F. Tafti, *Adv. Mater.* **2021**, *33*, 2005755.
- [22] R. Skomski, D. J. Sellmyer, *J. Rare Earths* **2009**, *27*, 675.
- [23] P. S. Riseborough, *Phys. Rev. B* **1979**, *20*, 1362.
- [24] S. Abdelouahed, M. Alouani, *Phys. Rev. B* **2009**, *79*, 054406.
- [25] H. Yamada, S. Takada, *J. Phys. Soc. Jpn.* **1973**, *34*, 51.
- [26] H. Yamada, S. Takada, *Prog. Theor. Phys.* **1972**, *48*, 1828.
- [27] P. Larson, W. R. L. Lambrecht, *Phys. Rev. B* **2006**, *74*, 085108.
- [28] F. Hulliger, *Nature* **1968**, *219*, 373.
- [29] W. Tremel, R. Hoffmann, *J. Am. Chem. Soc.* **1987**, *109*, 124.
- [30] L. M. Schoop, M. N. Ali, C. Straßer, A. Topp, A. Varykhalov, D. Marchenko, V. Duppel, S. S. P. Parkin, B. V. Lotsch, C. R. Ast, *Nat. Commun.* **2016**, *7*, 11696.
- [31] Q. Xu, Z. Song, S. Nie, H. Weng, Z. Fang, X. Dai, *Phys. Rev. B* **2015**, *92*, 205310.
- [32] P. Rosa, Y. Xu, M. Rahn, J. Souza, S. Kushwaha, L. Veiga, A. Bombardi, S. Thomas, M. Janoschek, E. Bauer, M. Chan, Z. Wang, J. Thompson, N. Harrison, P. Pagliuso, A. Bernevig, F. Ronning, *npj Quantum Mater.* **2020**, *5*, 52.
- [33] O. Breunig, Z. Wang, A. A. Taskin, J. Lux, A. Rosch, Y. Ando, *Nat. Commun.* **2017**, *8*, 15545.
- [34] L. Li, K. Sun, C. Kurdak, J. W. Allen, *Nat. Rev. Phys.* **2020**, *2*, 463.
- [35] Y. Ando, *J. Phys. Soc. Jpn.* **2013**, *82*, 102001.
- [36] A. Yurgens, D. Winkler, N. V. Zavaritsky, T. Claeson, *Phys. C* **1997**, *282-287*, 2293.
- [37] V. Baltz, A. Manchon, M. Tsoi, T. Moriyama, T. Ono, Y. Tserkovnyak, *Rev. Mod. Phys.* **2018**, *90*, 015005.
- [38] T. Jungwirth, X. Marti, P. Wadley, J. Wunderlich, *Nat. Nanotechnol.* **2016**, *11*, 231.
- [39] G. Kresse, J. Furthmüller, *Phys. Rev. B* **1996**, *54*, 11169.
- [40] J. P. Perdew, K. Burke, M. Ernzerhof, *Phys. Rev. Lett.* **1996**, *77*, 3865.
- [41] N. Marzari, D. Vanderbilt, *Phys. Rev. B* **1997**, *56*, 12847.
- [42] P. M. C. Rourke, S. R. Julian, *Comput. Phys. Commun.* **2012**, *183*, 324.
- [43] L. Chen, Y. Gu, Y. Wang, Y. Zhou, K. Liao, Y. Pan, X. Wu, Y. Li, Z. Wang, Y. Ma, Z. Guo, J. Ma, D. Su, J. Hu, G. Wang, *Innovation Mater.* **2023**, *1*, 100011.

