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#### Neutron-diffraction studies of UBe<sub>13</sub> and ThBe<sub>13</sub>

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Neutron-diffraction measurements have been made on polycrystalline samples of UBe<sub>13</sub> and ThBe<sub>13</sub> at several temperatures between 10 and 250 K. Refinement of the powder patterns using the Rietveld method of profile analysis yields structural parameters in substantial agreement with previous work [space group  $O_h^{\beta}$  (*Fm3c*); 8 U in 8(*a*); 8 Be(I) in 8(*b*); 96 Be(II) in 96(*i*)]. Values for the (*y,z*) coordinates of the 96 Be(II) are (0.1763, 0.1150) and (0.1745, 0.1129) for UBe<sub>13</sub> and ThBe<sub>13</sub>, respectively. They correspond very closely to those values required for angular snub-cube coordination of the Be(II) around the uranium atoms.

#### I. INTRODUCTION

The discovery<sup>1</sup> of superconductivity in the heavy-fermion system UBe<sub>13</sub> has stimulated new investigations of both the normal and superconducting states of this material. Previous structural studies of the  $MBe_{13}$  (M = Ce, Th, U, Zr) intermetallic compounds have been reported by Baenziger and Rundle<sup>2</sup> and Koehler, Singer, and Coffinberry.<sup>3</sup> Qualitative agreement between the observed x-ray and neutron profiles was obtained by use of the parameters known for NaZn<sub>13</sub>.<sup>4</sup> In view of the recent interest in UBe<sub>13</sub>, and the fact that a quantitative detailed analysis of its structure has not yet been performed, we have carried out a detailed structural analysis by Rietveld refinement<sup>5</sup> of neutron powder data of  $UBe_{13}\ \text{and}\ ThBe_{13}.$  In addition, our measurements of the UBe<sub>13</sub> and ThBe<sub>13</sub> powder spectra were made at several temperatures, providing information about the temperature dependence of the lattice constant, positional parameters, and temperature factors.

#### **II. EXPERIMENTAL**

Samples of  $UBe_{13}$  and  $ThBe_{13}$  were prepared by use of a method described elsewhere,<sup>1</sup> powdered, and passed

through a sieve with  $420 \ \mu m$  openings. The UBe<sub>13</sub> powder was loaded into an aluminum cylinder 1 cm in diameter and 2.5 cm long. This sample holder was placed in a second aluminum can, filled with He gas, sealed, and attached to the cold finger of a Displex refrigerator. The ThBe<sub>13</sub> sample was similarly handled, however, a 2.5-cm-diam sample holder was used. The measurements were performed at several temperatures between 10 and 250 K as indicated in Table I. The temperature was regulated to within  $\pm 1$  K.

The powder-diffraction measurements were made on a triple-axis spectrometer at the Brookhaven National Laboratory High-Flux Beam Reactor (HFBR) using a bent pyrolytic graphite (PG)(002) monochromator and a bent PG(004) analyzer both set to diffract 14.7 meV neutrons ( $\lambda = 2.359$  Å). The horizontal collimation was 40'-open-40'-20' between the reactor and monochromator, the monochromator and sample, the sample and analyzer, and the analyzer and detector, respectively. A PG filter was used before the monochromator to eliminate  $\lambda/2$  contamination. Data were taken over a range in scattering angle of 20°-130° encompassing thirteen reflections. A few small ( << 1% of the most intense reflection) impurity peaks were observed in the diffraction spectra and could not be identified.

The samples were examined for the presence of large

Т (К)	<i>a</i> <sub>0</sub> (Å)	У	Z	B <sub>U</sub>	B <sub>Be(I)</sub>	B <sub>Be(II)</sub>	$R_I^a$	R <sub>wp</sub> a	$R_E^{a}$
				UBe <sub>13</sub>					
10	10.248 87(2) <sup>b</sup>	0.1763(1) <sup>b</sup>	0.1150(1)	0.7(1)	0.1(1)	1.1(1)	2.06	5.90	3.83
50	10.24987(3)	0.1763(1)	0.1148(1)	0.3(1)	0.2(1)	0.8(1)	1.54	6.38	3.83
100	10.250 39(2)	0.1761(1)	0.1150(1)	0.7(1)	0.2(1)	1.1(1)	1.23	5.84	3.81
175	10.25405(3)	0.1765(1)	0.1152(1)	0.9(1)	0.8(1)	1.4(1)	1.12	6.61	3.96
250	10.26019(3)	0.1763(1)	0.1150(1)	0.9(1)	0.4(1)	1.3(1)	1.41	6.67	3.86
				ThBe <sub>13</sub>					
10	10.41005(3)	0.1745(1)	0.1129(1)	0.2(1)	0.6(1)	1.0(1)	3.41	6.52	2.3
100	10.410 55(3)	0.1747(1)	0.1130(1)	0.2(1)	0.4(1)	0.8(1)	4.27	7.12	2.31
250°	10.41727(4)	0.1751(2)	0.1175(2)	• • •	• • •		3.37	5.35	2.13

TABLE I. Profile refinement results.

<sup>a</sup>See Ref. 5 for definitions.

<sup>b</sup>Numbers in parenthesis refer to the estimated standard deviation referred to the last digit.

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<sup>c</sup>Incomplete data set.

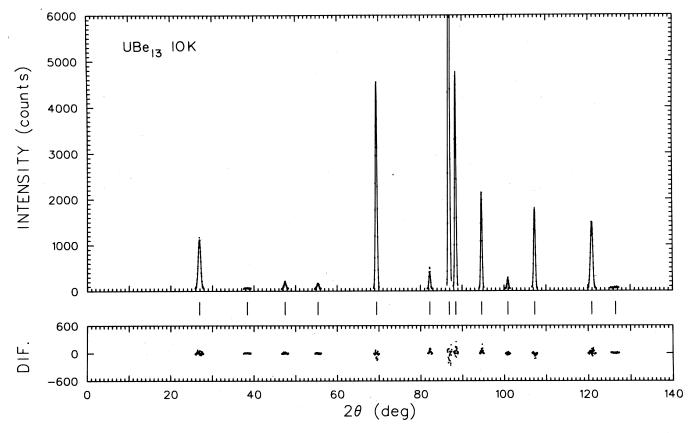


FIG. 1. Fit to the UBe<sub>13</sub> powder pattern at T = 10 K and residuals of the fit. The short vertical lines indicate allowed reflections.

grains by setting the spectrometer to the correct scattering angle for the (531) reflection and rotating the sample. Variations in intensity of approximately 30% indicated the presence of grains in the UBe<sub>13</sub> sample, while no significant intensity variations were noted for the ThBe<sub>13</sub> powder. In order to minimize these effects, powder patterns for UBe<sub>13</sub> were recorded while the sample table was rotated between  $-45^{\circ}$  and  $+45^{\circ}$  at each data point. The absorption of both samples was determined by measuring the transmitted in-

TABLE II. Nearest-neighbor distances in UBe<sub>13</sub> at 10 K.

Bond to atom	Number of bonds	d (Å)	
A. Bonds involving U			
Be(11) $(O, y, z)$	24	3.01	
B. Bonds involving Be(I)			
Be(II) (O, y, z)	12	2.16	
C. Bonds involving Be(II)			
Be(I) (0,0,0)	- 1.	2.16	
U $(\frac{1}{4}, \frac{1}{4}, \frac{1}{4})$	2	3.01	
Be(II) $(0, z, \frac{1}{2} - y)$	2	2.23	
Be(II) $(z, 0, y)$	4	2.25	
Be(II) $(0, y, \overline{z})$	1	2.36	
Be(II) $(z, \frac{1}{2} - y, 0)$	2	2.25	

tensity, enabling self-shielding corrections<sup>6</sup> to the temperature factors to be made.

The results of the refinement are summarized in Table I. In the refinement procedure, two positional structural parameters [the Be(II) atoms 0,y,z], the lattice constant  $a_0$ , and three isotropic temperature factors  $(B_M, B_{Be(II)}, B_{Be(II)})$ , were optimized. For details of the Rietveld refinement procedure, the reader is directed to Ref. 5. The data, fit, and residuals of the UBe<sub>13</sub> (T = 10 K) profile are shown in Fig. 1. In all cases reasonable fits to the observed data were obtained. The three *R*-factor entries in Table I are a measure of the goodness of fit of the calculated profile to the data, and are described in Ref. 5.

As indicated in Table I, the UBe<sub>13</sub> lattice parameter increases smoothly from about 10.2489 Å at 10 K to 10.2602 Å at 250 K. Comparing the lattice constants of UBe<sub>13</sub> and ThBe<sub>13</sub> at 10 and 250 K, we find that the thermal expansion of the UBe<sub>13</sub> lattice is nearly twice as large. The positional coordinates (0,y,z) of the Be(II) atoms are temperature independent throughout the range investigated. Based on these values for  $a_0$ , y, and z, Table II lists the calculated nearest-neighbor distances in UBe<sub>13</sub>.

#### **III. DISCUSSION**

Our values for  $a_0$ , y, and z agree substantially with those of previous investigations.<sup>2,3</sup> The structure of the MBe<sub>13</sub> (M = U, Th) compounds belongs to the cubic NaZn<sub>13</sub>-type family with space group  $O_h^6$  (Fm3c) with

U 8(a): 
$$\pm \frac{1}{4}, \frac{1}{4}, \frac{1}{4}$$
,  
Be(I) 8(b):0, 0, 0;  $\frac{1}{2}, \frac{1}{2}$ ,

Be(II) 96(*i*):  $\pm O, y, z$ ; etc.

The average values for (y,z) in UBe<sub>13</sub> and ThBe<sub>13</sub> were determined to be (0.1763, 0.1150) and (0.1745, 0.1129), respectively. The structure of the NaZn<sub>13</sub>-type compounds has been discussed in detail by Shoemaker et al.<sup>7</sup> The eight U ions and eight Be(I) atoms form a CsCl-type lattice, each U being coordinated by 24 Be(II) atoms at the corners of a nearly regular snub cube. The snub cube may be visualized by snipping off the eight corners of a cube with a U ion at the center. The Be(I) sites are surrounded by 12 Be(II) atoms on a nearly regular icosahedron. However, the conditions for regular snub-cube coordination (y = 0.1761, z = 0.1141) and icosahedral coordination (y = 1.618z) are incompatible. Figure 2 is a parameter plot<sup>7</sup> which graphically displays the constraints for regular snubcube and regular icosahedral coordination. Also plotted are the values of y and z determined for UBe<sub>13</sub> and ThBe<sub>13</sub> in this work, along with those determined for CeBe<sub>13</sub>,<sup>2</sup>  $NaZn_{13}$ , <sup>7</sup> and  $KCd_{13}$  (Ref. 8) measured in previous studies. Uncertainties, where known, are indicated by the radii of circles drawn around the points. While the positional parameters derived for the NaZn13 and KCd13 compounds differ from those appropriate to either snub-cube or icosahedral regularity, y and z for UBe<sub>13</sub> correspond closely to the values required for regular snub-cube coordination of the U ions by the Be(II) atoms. These values are also close to those used by Bucher et al.<sup>9</sup> in their crystal-field calculations based on the CeBe<sub>13</sub> positional parameters. In general, it appears that in the MBe<sub>13</sub> compounds, the near-neighbor environment of the M ions is more regular than that found in other compounds in the  $NaZn_{13}$ -type structural family.

Table I also lists the values obtained for the isotropic temperature factors (B) for each site in the structure, corrected for shielding (absorption) effects for each sample.<sup>6</sup> Here, the thermal parameter  $B = 8\pi^2 \langle u_{\parallel}^2 \rangle$ , where  $\langle u_{\parallel}^2 \rangle$  is the mean-square displacement along the scattering vector. There is clearly some scatter in the values in excess of the estimated standard deviations probably caused by systematic effects. Nevertheless, we can make some qualitative conclusions. First, we note that  $B_{\rm Be(II)}$  is consistently larger than  $B_{\rm Be(1)}$  for both UBe<sub>13</sub> and ThBe<sub>13</sub>, and no significant temperature dependence for these factors is observed. This indicates that there is substantially greater static disorder on the Be(II) sites than on the Be(I) sites. Refinements in which anisotropic temperature factors were included, or the

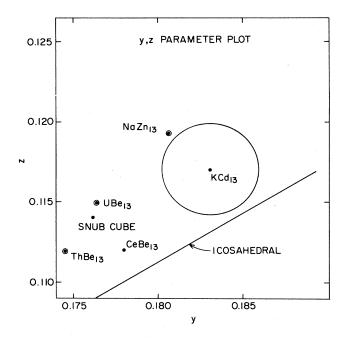


FIG. 2. Parameter plot of y and z for compounds with the NaZn<sub>13</sub> family-type structure. Radii around points indicate calculated errors.

Be(II) site occupancy was varied, gave no significant change. Second, it can be seen from Table I that  $B_U$  is larger than  $B_{Th}$  at 10 and 100 K. Recently, Overhauser and Appel<sup>10</sup> have proposed that if there is significant hybridization between the localized f and band-like s electrons, the U-Be spring constant should be substantially smaller than the Th-Be spring constant. This should result in an effective Debye temperature which is smaller for U in UBe<sub>13</sub> than for Th in ThBe<sub>13</sub>, and is therefore qualitatively consistent with our observation. This point can be checked much more accurately with x rays, and measurements of the Debye-Waller factor for the U and Th atoms are currently underway.

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