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# Single-Molecule Stretching Shows Glycosylation Sets Tension in the Hyaluronan-Aggregan Bottlebrush

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**ABSTRACT** Large bottlebrush complexes formed from the polysaccharide hyaluronan (HA) and the proteoglycan aggregan contribute to cartilage compression resistance and are necessary for healthy joint function. A variety of mechanical forces act on these complexes in the cartilage extracellular matrix, motivating the need for a quantitative description that links their structure and mechanical response. Studies using electron microscopy have imaged the HA-aggregan brush but require adsorption to a surface, dramatically altering the complex from its native conformation. We use magnetic tweezers force spectroscopy to measure changes in extension and mechanical response of an HA chain as aggregan monomers bind and form a bottlebrush. This technique directly measures changes undergone by a single complex with time and under varying solution conditions. Upon addition of aggregan, we find a large swelling effect manifests when the HA chain is under very low external tension (i.e., stretching forces less than  $\sim 1$  pN). We use models of force-extension behavior to show that repulsion between the aggregans induces an internal tension in the HA chain. Through reference to theories of bottlebrush polymer behavior, we demonstrate that the experimental values of internal tension are consistent with a polydisperse aggregan population, likely caused by varying degrees of glycosylation. By enzymatically deglycosylating the aggregan, we show that aggregan glycosylation is the structural feature that causes HA stiffening. We then construct a simple stochastic binding model to show that variable glycosylation leads to a wide distribution of internal tensions in HA, causing variations in the mechanics at much longer length scales. Our results provide a mechanistic picture of how flexibility and size of HA and aggregan lead to the brush architecture and mechanical properties of this important component of cartilage.

**SIGNIFICANCE** We use single-molecule stretching experiments to study a macromolecular complex of hyaluronan and aggregan whose structure is crucial to maintaining the mechanical strength of articular cartilage. We experimentally validate a model that quantitatively describes how the extension of a polysaccharide hyaluronan chain is affected by binding aggregan side chains. This model yields information about the sensitivity of the complex size to different features of the bottlebrush architecture and predicts when and how aggregan damage leads to collapse of the complex.

## INTRODUCTION

Large bottlebrush complexes formed from the polysaccharide hyaluronan (HA) and the proteoglycan aggregan constitute a major component of cartilage (1). HA is a long, linear polyelectrolyte with a charge density of  $1 e/\text{nm}$ , a persistence length of around 5 nm, and a native size of  $\sim 1$ – $10$  MDa, corresponding to contour lengths of  $\sim 2$ – $20 \mu\text{m}$  (2).

In cartilage, HA is secreted by chondrocytes and anchored to the cell surface by HA synthases or the receptor CD44 (3), forming the sugar-rich “glycocalyx” coating around the cell. Each HA chain in the glycocalyx is complexed with many noncovalently bound aggregans to form a large macromolecular bottlebrush that is known to be a major contributor to cartilage compression resistance (4). In cartilage, the complex involves a third participant, link protein (LP), which binds to both HA and aggregan to stabilize the interaction (4). As illustrated in Fig. 1 A, each aggregan monomer is itself a bottlebrush, consisting of a  $\sim 300$ -kDa protein core that is densely decorated by the glycosaminoglycan side chains keratan sulfate (KS) and the more abundant chondroitin sulfate (CS) (4). The core protein includes

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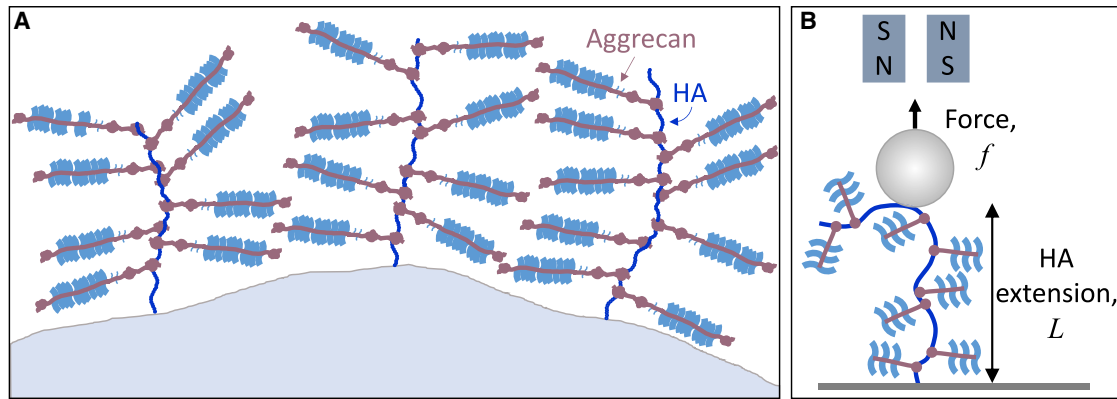


FIGURE 1 (A) Cartoon of the chondrocyte glycoalyx, in which HA chains are tethered to the cell surface and decorated with the proteoglycan aggrecan, forming a microns-thick coating around the cell. (B) The experimental setup is as follows: a single HA chain, decorated by aggrecans, is tethered between a glass surface and a magnetic bead. A magnet applies stretching forces to the HA chain at the center of the bottlebrush. To see this figure in color, go online.

three globular domains: two at the N-terminus (G1 and G2) and one at the C-terminus (G3) (1). G1 is the HA-binding domain. In between G2 and G3 is an extended domain decorated by  $\sim 100$  CS chains (1). The high charge and large size of aggrecan result primarily from this CS-rich region. This structure-defining domain changes with age, showing a reduction in the number and size of CS side chains (5–10).

Previous studies have used electron microscopy and atomic force microscopy (AFM) to visualize the structure of individual aggrecans and aggrecan-HA complexes (10–14). These studies have provided estimates of the length and diameter of aggrecan, as well as the density of bound aggrecans on an HA chain. However, imaging requires adsorption to a surface that significantly distorts the conformation of the complex. To learn about the solution structure of the complex, including the conformation adopted by the central HA chain, it is necessary to probe the structure of individual complexes by a method that does not confine the aggregates to 2D. To this end, others have used single-molecule force spectroscopy to measure properties of the complexes, such as interaggrecan interactions (15,16). AFM and laser tweezers, which can access forces on the order of 10–1000 and 1–100 pN, respectively (17,18), have also been used to test the strength of the bond connecting aggrecan and HA (19–22). Liu et al. used laser tweezers to probe moderate-force tensile behavior ( $\sim 10$ –40 pN) and compressibility of single HA-aggrecan complexes (23,24).

To our knowledge, no prior work has conducted low-force tensile testing on single bottlebrush complexes with subpico-newton resolution, despite this likely being the biologically relevant force range. A study on osteocytes estimated that flow forces can lead to tension in the cell-attached components (e.g., HA) in the range of  $\sim 1$ –10 pN (25). Here, we aim to quantify the structure of the complex based on its response to forces in this range. To do this, we use magnetic tweezers force spectroscopy. Prior work has shown that magnetic tweezers are particularly useful for quantifying structural changes that manifest at low forces, such as intrachain

electrostatic repulsion (26). Thermally dominated structure at a particular length scale  $x$  is probed by forces of the scale  $k_B T/x$  (18,27). Magnetic tweezers typically apply forces on the order of 0.01–10 pN and thus are sensitive to structure on length scales between  $\sim 1$  and 100 s of nm. Because of this long length scale sensitivity, magnetic tweezers have succeeded in measuring the structural changes induced by side chains in a synthetic polymer bottlebrush (28).

By applying low-force magnetic tweezers stretching to study the formation and properties of the HA-aggrecan bottlebrush complex, we arrive at a physical description of this important component of cartilage which is, to our knowledge, novel. Prior work has shown that aggrecan swells HA chains in pericellular coats (29) and pure HA brushes (30); we find this effect is also detectable at the single-chain level. Thus, we can observe in real time the assembly of the complexes by tracking the increase in chain extension. We find our data are well described by a model in which the effect of all interaggrecan repulsions can be combined into a single internal tension, which acts in combination with the externally applied stretching force to extend the HA chain. We show that, on average, the internal tension generated in the central HA chain is  $\sim 0.4$  pN but with large variability between measurements. We use an analytical theory of bottlebrush physics to show that the spread in the data is consistent with variability in degree of glycosylation of the bound aggrecan.

## MATERIALS AND METHODS

Hyaluronic acid (2.5 MDa), chemically functionalized to allow tethering to two surfaces, was purchased from Creative PEGworks (Durham, NC). Each HA chain has a biotin group at the reducing end and randomly situated thiol groups with a stoichiometry of one thiol per chain. Microfluidic sample chambers were constructed using maleimide-functionalized PEG-grafted glass coverslips purchased from MicroSurfaces (Englewood, NJ). HA chains were attached to the surface via reaction of the thiol groups with the maleimide, carried out in 50 mM sodium phosphate buffer (pH 7.2), 50 mM NaCl, 0.03% Tween 20, and 100 mM TCEP. After attachment, the excess polymer was removed by rinsing with 10 mM MOPS buffer

(pH 7, with an ionic strength of 4 mM). Streptavidin-coated magnetic beads 1  $\mu\text{m}$  in diameter (MyOne C1 paramagnetic beads; Invitrogen, Carlsbad, CA) were attached to the biotin-labeled chain ends.

This experimental protocol leads to polydispersity in the tether lengths due to the random locations of the thiols and necessitates normalization by contour length to compare measurements on different tethers. We expect an approximately uniform distribution of tether sizes up to  $\sim 6 \mu\text{m}$ . However, because of the potential for bead-surface interactions at low forces, we do not collect data on tethers shorter than  $\sim 0.6 \mu\text{m}$ . The stochastic thiol labeling strategy means that the magnetic bead is attached internally within the chain and thus that there is a second aggregan-decorated HA (not pinned to the surface, and so under no tension) in close proximity to the elastically tested tether (see Fig. 1 B). The excluded volume presented by this second chain is small compared with the bead and surface, and we do not expect it to have an effect at the forces probed here.

Aggregan ( $\geq 2.5$  MDa, purified from bovine articular cartilage) was purchased from Sigma-Aldrich (St. Louis, MO). It was centrifuged to remove large aggregates, and small contaminants were removed by filtration with a 100-kDa spin column. Recombinant aggregan G1-IGD-G2 (with C-terminal 10-His tag) was purchased from BioLegend (San Diego, CA). Experiments were conducted in 100 mM NaCl, 1–10 mM MOPS (pH 7), and 0.03% Tween 20.

Experiments were conducted using a custom-built magnetic tweezers instrument, as described in detail elsewhere (31). Briefly, HA-tethered beads are imaged, and bead position is measured using an image analysis routine based on the bead's diffraction pattern. The output of that routine includes the tether length and lateral bead fluctuations; the latter are analyzed to estimate the applied tension (32). To ensure each bead is tethered by a single HA chain, the tether length is measured as the bead is rotated. Multiple tethers become interwound during rotation, leading to a characteristic decrease in bead height (33).

Experiments on aggregan-induced swelling used an aggregan concentration of 2 mg/mL ( $\sim 0.8 \mu\text{M}$ ). Above  $\sim 1$  mg/mL, the HA chain extension does not change appreciably with additional aggregan (see Fig. S1). A concentration of 2 mg/mL is well into the concentration-insensitive regime and is close to the overlap concentration at which the distance between aggregans in solution approaches the aggregan radius of gyration (34). This relatively high solution concentration is still well below the density of aggregan bound in cartilage, which is thought to be on the order of 10 s of mg/mL (35). For experiments using the G1-IGD-G2 fragment, the protein was dissolved at  $\sim 0.8 \mu\text{M}$ .

To obtain deglycosylated aggregan core protein, aggregan (Sigma-Aldrich) was incubated overnight with chondroitinase ABC (recombinant *Proteus vulgaris* chondroitinase ABC with N-terminal Met and 6-His tag; purchased from R&D Systems (Minneapolis, MN)). The reaction took place at 37°C in 50 mM Trizma HCl (pH 8), 60 mM sodium acetate, and 0.02% BSA. Chondroitinase ABC digests HA in addition to CS, so it was necessary to separate it from the deglycosylated aggregan before experiments could proceed. Separation was accomplished using Dynabeads His-Tag Isolation and Pulldown, which were then removed magnetically. The presence of aggregan core protein ( $\sim 300$  kDa) at the expected concentration was confirmed with SDS-PAGE (Fig. S2) and dynamic light scattering on denatured aggregan (Fig. S3), using a Malvern Zetasizer Nano ZS instrument. HA force-extension curves in the presence of aggregan core protein were obtained in 100 mM NaCl, 10 mM MOPS (pH 7), 0.03% Tween 20, and 1 mg/mL free CS (CS sodium salt from shark cartilage, purchased from Sigma-Aldrich). The free CS was included to protect the HA tethers from digestion by any residual enzyme and was shown to have no effect on the HA mechanical response (Fig. S4).

## RESULTS

With magnetic tweezers, we measure the extension of tethered HA chains as aggregan is added to solution. Our data show that aggregan expands the HA chain upon binding

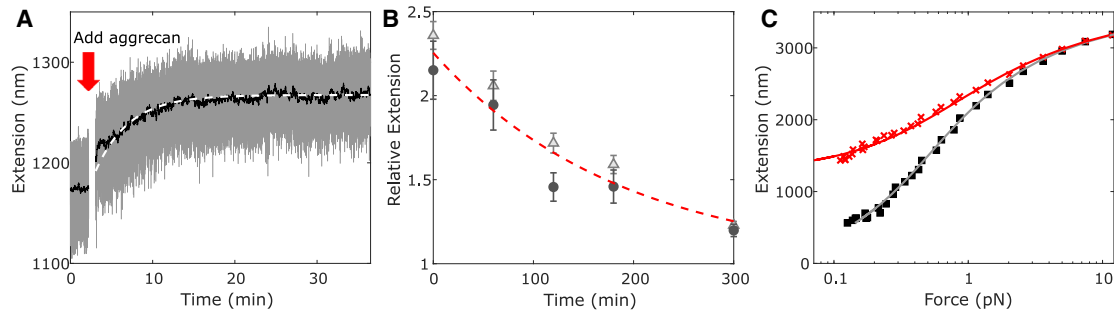
(Fig. 2). When the chain is minimally stretched, we can track the progress of aggregan binding by monitoring the chain end-to-end extension. As can be seen in Fig. 2 A, the length increases upon addition of 2 mg/mL aggregan and levels off after  $\sim 15$  min, indicating the system has equilibrated. We can subsequently rinse free aggregan out of the bulk solution and observe the HA length change as aggregan unbinds (Fig. 2 B). This process takes several hours. Comparing time-scales obtained by fitting exponential models to the data, we estimate that the timescale of aggregan binding is at least 50 times faster than aggregan unbinding. Although aggregan interactions may lead to anticoperative behavior and nonexponential kinetics, the data (within the experimental resolution) appear exponential. Thus, to good approximation, a simple independent binding picture can be used, which here indicates that, because the off-rate is 50 times slower than the on-rate, the maximal aggregan density on HA has been achieved. We note that in cartilage, LP would further stabilize the bound state, likely causing higher aggregan density and even slower unbinding.

Relative to the mechanical behavior measured before the addition of aggregan, the ligand binding dramatically alters the low-force response (see example data in Fig. 2 C). Repeated experiments on 10 HA tethers, with and without aggregan, all show similar elasticity changes (Fig. 3). With the addition of aggregan (2 mg/mL), all tethers undergo swelling at low forces; this is reflected by the shallower slopes of the force-extension curves. The slope change indicates that aggregan binding stiffens the chain, causing it to become less responsive to the externally applied force. The observed low-force elasticity change is consistent with long length scale structural effects, such as swelling and stiffening of the central HA chain due to repulsion between bound aggregans, as expected from the charged brush structure of the aggregan monomers (36).

We quantify the low-force swelling using an internal tension model (26), in which the long length scale stiffening (caused by the bottlebrush architecture) is described by a mean field internal tension  $f_{int}$ , which acts in addition to the applied force  $f$  to straighten the polymer. This first-order correction captures the deviation from worm-like chain (WLC) elasticity, which is most prominent at low forces, at which the chain physics is dominated by long-range side-chain interactions. The internal tension term is incorporated into the Marko-Siggia WLC model (37), giving the following:

$$f = \frac{k_B T}{l_p} \left( \frac{1}{4} \left( 1 - \frac{L}{L_C} \right)^{-2} - \frac{1}{4} + \frac{L}{L_C} \right) - f_{int}, \quad (1)$$

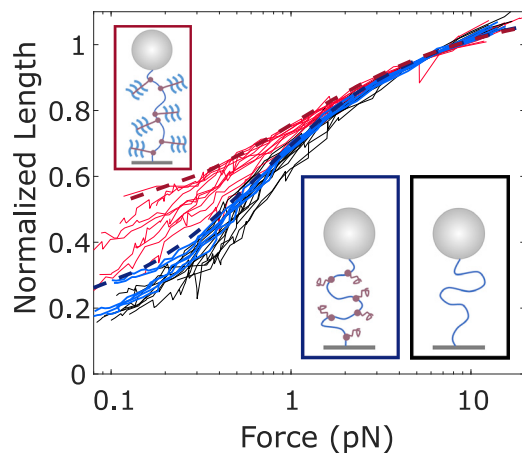
where  $l_p$  is the intrinsic HA persistence length,  $L$  is the chain extension, and  $L_C$  is the contour length. Equation 1 is used to fit force-extension data, with  $L_C$  and  $l_p$  fitted globally for pairs of curves on the same molecule.  $f_{int}$  is fixed to zero for the bare HA force-extension data (see Fig. 2 C). We



**FIGURE 2** (A) HA extension is shown while aggrecan binds under a constant applied force of 1.8 pN (*gray*: all points; *black*: moving average over 4 s). The data are fitted by an exponential with a characteristic timescale of  $244 \pm 2$  s (*white dashed line*). (B) HA extension during aggrecan unbinding is shown. After removing free aggrecan from the bulk solution, extension of two aggrecan-decorated tethers at 0.2 pN (first tether: *light gray triangles*, second tether: *dark gray circles*), relative to aggrecan-free extension of the same tether at the same force, slowly decreases toward 1. Error bars show uncertainty arising from interpolation of force-extension curves at 0.2 pN. The data are fitted by an exponential with a characteristic timescale of  $190 \pm 50$  min (*red dashed line*). (C) Example force-extension data on a single HA tether is shown before (*black squares*) and after (*red x symbols*) addition of 2 mg/mL aggrecan and equilibration of at least 15 min. Both force-extension curves are globally fitted by Eq. 1, with common contour length  $L_C$  and persistence length  $l_p$ . Best-fit parameters are  $L_C = 3573 \pm 14$  nm,  $l_p = 7.4 \pm 0.4$  nm, and  $f_{int} = 0.40 \pm 0.03$  pN, where  $f_{int}$  is the internal tension generated by aggrecan ( $f_{int}$  is set to zero for the data measured without aggrecan). Errors reflect 95% confidence intervals for the fitted parameters. To see this figure in color, go online.

repeated such analyses on each of the 10 tethers, measuring force-extension curves before and after the addition of 2 mg/mL aggrecan and extracting  $f_{int}$  as a fitted parameter. The average was  $0.35 \pm 0.21$  pN (standard deviation (SD)). The spread in  $f_{int}$  reflects variability in the response of the aggrecan-decorated chains, demonstrated by the scatter of the red traces in Fig. 3. We hypothesize that this variability arises because of differences in glycosylation of the aggrecan samples, which originated in biological tissue (7,38,39).

For comparison, we digested the CS side chains of the aggrecan using chondroitinase ABC and found that the resulting deglycosylated protein induces almost no swelling in HA (*blue lines* in Fig. 3). Although the less abundant KS side chains are thought not to be affected by this enzyme, others have shown that, in contrast to CS, removal of the KS side chains has little to no effect on the aggrecan monomer structure (5).



**FIGURE 3** Thin red lines show force-extension curves on 10 HA tethers with 2 mg/mL aggrecan. Extensions are normalized by multiplicative adjustment, with factors determined by comparing each curve to the median of the 10 and minimizing the sum of squared residuals in length for forces between 3 and 10 pN. Black lines show force-extension curves on the same 10 tethers in aggrecan-free conditions. Blue lines show force-extension curves of six HA tethers in the presence of deglycosylated aggrecan core protein. Thick dashed lines are predictions from the WLC internal tension model for fully glycosylated aggrecan (*dark red*) and aggrecan core protein (*dark blue*). Cartoons (*inset*) illustrate the conditions for each set of force-extension curves (full aggrecan: *red box, upper left*; chABC-digested aggrecan: *blue box, bottom middle*; bare HA: *black box, bottom right*). To see this figure in color, go online.

## DISCUSSION

The HA-aggrecan force-extension data highlight that properties of the side chains play a crucial role in defining the bottlebrush architecture. At low forces ( $\sim 0.1$  pN), the end-to-end extension of HA molecules often increases more than twofold upon introduction of aggrecan. However, when HA is decorated instead by deglycosylated aggrecan, this bottlebrush-induced stiffening is dramatically reduced, demonstrating the importance of the chemical details of aggrecan. Even without artificial reduction of aggrecan glycosylation, the experiments display a wide range of responses. These observations can be explained using established polymer bottlebrush theory. We show the full range of stiffening observed is consistent with predictions from scaling theory combined with previous experimental measurements of HA and aggrecan structure. We further find that the variance in the glycosylated aggrecan data is consistent with an aggrecan population of varying degrees of glycosylation. Such results demonstrate the interplay between the branched architecture of the complex and the conformational structure of its constituent chains.

Following the logic of Berezney et al. (28), we use a model developed by Panyukov et al. (40) to predict, independently of our own measurements, the range of internal tensions in the HA backbone of bottlebrush complexes

extended by glycosylated and deglycosylated aggregan. The model treats the side chains as random-walk polymers, whose size, flexibility, and binding density control the internal tension generated in the central chain as follows:

$$f_{int} = \alpha k_B T d^\mu b_a^{-(\mu+1)} n^\nu |\tau|^t, \quad (2)$$

where  $\alpha$  is a prefactor that we take to be 1;  $\mu$ ,  $\nu$ , and  $t$  are scaling exponents (described below);  $b_a$  is the statistical segment length of aggregan;  $n$  is the number of statistical segments within each aggregan side chain;  $\tau$  describes the solvent quality and is 1 in good solvent; and  $d$  is the distance between bound aggregans. Estimates of  $d$  range from 12 to 40 nm (13,14); we choose the intermediate value of this range (26 nm). If LP were present, a smaller value ( $\sim 10$  nm) would be appropriate because LP is known to increase aggregan density (13). Based on our experiments showing that unbinding is very slow, we assume the number and spacing of bound aggregans are constant over experimental timescales (measurement of each force-extension curve takes 20 min). We assume that deglycosylation will be reflected as a change in aggregan's Kuhn length  $b_a$ . The model (Eq. 2) considers the athermal limit and takes the excluded volume to be  $b_a^3$ . For bottlebrushes, Kuhn length typically scales with the diameter of the complex (41). Thus, for fully glycosylated aggregan, we will approximate  $b_a \approx 50$  nm, roughly the diameter of the cylindrical proteoglycan (14). In the opposite limit of complete deglycosylation, the side chains are aggregan core protein alone. Based on the disordered nature of much of the core protein (42), we estimate  $b_a$  as 1.6 nm, twice the persistence length of an unfolded polypeptide (43). This value represents a physical lower limit; in reality, it is likely that some residual structure leads to a slightly larger Kuhn length (42), but the model is not very sensitive to this assumption (Fig. S6). The number of segments  $n$  is obtained by dividing the aggregan contour length, estimated as 375 nm following AFM studies (10,14), by the Kuhn length. SDS-PAGE of the deglycosylated protein (Fig. S2) shows a number of bands near the expected weight, suggesting there may be a spread of aggregan sizes present; we do not incorporate this into our model but recognize that it may be an additional source of variability.

We assume the central HA chain is always strongly stretched because of its stiffness ( $b_{HA} \approx 10$  nm), side chains, and the external force; thus, the random coil structure of the side chains will be the dominant contributor to the HA internal tension. It is expected that deglycosylation will significantly perturb aggregan structure; this will change not only its Kuhn length, as discussed above, but also the scaling laws describing its solution structure. Fully glycosylated aggregan is strongly extended because of repulsion between its own CS branches. Following Panyukov et al. (40), the case of swollen aggregan side chains dictates  $\mu = -13/8$ ,  $\nu = 3/8$ , and  $t = 1/8$  in Eq. 2.

The deglycosylated core protein structure will also likely be dominated by the large CS attachment region, now devoid of CS. Prior evidence has shown that the CS attachment domain of aggregan is structurally disordered (42). The solution structure of intrinsically disordered proteins reflects a trade-off between conformational flexibility and transient short length scale interactions and is frequently modeled by ideal scaling (44). Thus, we assume that without CS to provide stretching, the core protein will behave ideally, dictating scaling exponents  $\mu = -5/3$ ,  $\nu = 1/3$ , and  $t = 0$ . We note that the model of Panyukov et al. is formally correct for a bottlebrush whose backbone and side chains have the same chemistry, which is not the case here. However, as discussed above, HA elasticity is not expected to have significant effects because the chain is always in a regime of strong stretching. Further, if we instead assume the core protein is swollen, the predicted tension does not change drastically (Fig. S6).

This model (Eq. 2 with parameters listed above) allows us to estimate the range of  $f_{int}$  values expected for aggregan in different conditions: 0.15 pN in the limit of complete deglycosylation and 0.51 pN for a monodisperse sample of undamaged aggregan. Combining these values with the WLC model (Eq. 1), we obtain expected force-extension curves for HA decorated by whole aggregan (*dark red dashed curve* in Fig. 3) and HA decorated by deglycosylated aggregan core protein (*dark blue dashed curve* in Fig. 3). Our experimental force-extension curves for aggregan-decorated HA (Fig. 3) mostly fall between these limits. The experimental curves showing the largest swelling are consistent with the undamaged aggregan prediction of  $f_{int} \approx 0.5$  pN. The force-extension curves on HA with deglycosylated aggregan core protein (*blue traces* in Fig. 3) nearly coincide with the model's prediction for deglycosylated aggregan. As noted above, in cartilage, the complex is often stabilized by LP, which reduces the aggregan spacing to  $d \approx 10$  nm. Because  $f_{int}$  depends sensitively on  $d$ , LP's contribution will likely lead to a much higher tension in healthy physiological complexes ( $\sim 2$  pN, instead of  $\sim 0.5$  pN as reported here).

### Aggregan polydispersity

The above model makes predictions for extreme cases, such as monodisperse populations of either undamaged aggregan or completely deglycosylated core protein. It is clear that a monodisperse population of aggregan with intermediate glycosylation (and thus  $b_a$  between 1.6 and 50 nm) would generate a tension intermediate between these cases. In practice, however, there will usually be a mixture of aggregan at varying levels of glycosylation, ranging from the bare core protein to a dense bottlebrush. In the following, we consider a two-species mixture of undamaged and deglycosylated aggregan. In neglecting intermediate levels of glycosylation, this scenario is still a significant simplification. Nonetheless, it illustrates how polydispersity in the

aggregran population can lead to variability in the extension of aggregran-decorated HA chains.

We consider a simple model in which the HA chain is represented by a 1D array of binding sites, spaced by  $d = 26$  nm. Each site can be occupied either by an undamaged aggregran (A) or a deglycosylated core protein (B). Each bound side chain interacts only with its nearest neighbors, stretching the intervening HA segment. If stretched by an A aggregran, the segment will experience  $f_{int}^{AA} = 0.51$  pN, as calculated above, whereas a segment stretched by two B proteins will experience  $f_{int}^{BB} = 0.15$  pN. We assume that the tension in a segment with one A neighbor and one B neighbor,  $f_{int}^{AB}$ , is closer to the completely deglycosylated case because of the lack of charged CS on the core protein, as well as its presumed ideal (i.e., self-intersecting) behavior. Thus, we take  $f_{int}^{AB} = 0.15$  pN as well. We define  $\Phi$  as the fraction of undamaged A aggregran in the bulk solution and populate each binding site stochastically with probability  $P(A) = \Phi$ . We use the asymptotic form of the MS-WLC expression (37) to predict the relative length of a segment as a function of the total force it experiences (external force plus internal tension) as follows:

$$\frac{L_c^{seg}}{L_c} = 1 - \sqrt{\frac{k_B T}{4(f + f_{int})l_p}}. \quad (3)$$

We count the number of each type of segment (A-A, A-B, and B-B) to approximate force-extension curves for HA decorated by a heterogeneous aggregran population.

In our model, variability can arise because of two factors. The quality of aggregran may differ between samples; thus, we make  $\Phi$  a random variable. We will set the mean  $\bar{\Phi} = 0.8$  and assume  $\Phi$  is normally distributed with an SD of 10%. Additionally, even without noise in  $\Phi$ , the randomness inherent in the binding process leads to fluctuations in the actual fraction of A aggregrans bound, as well as their arrangement on the chain. For short HA tethers (e.g.,  $\sim 600$  nm or  $\sim 20$  binding sites), these fluctuations from the average are significant. Our experimental setup invariably leads to polydispersity in tether length; the tethers included in Fig. 3 range in length from  $\sim 600$  to  $\sim 5000$  nm ( $\sim 200$  binding sites). Thus, in our model, we select a contour length at random from the set of 10 experimentally measured tethers.

Our model results qualitatively show that sample-to-sample variation in aggregran quality, along with random fluctuations in the actual bound population and arrangement on short tethers, can lead to the kind of noisy data we observe experimentally. We simulate  $1 \times 10^5$  tethers, each with random  $\Phi$  and  $L_c$  as described above. Force-extension curves for the first 10 trials are plotted on the right side of Fig. 4. The histogram on the left of Fig. 4 shows the low-force (0.1 pN) extension of all  $1 \times 10^5$  tethers. With no sample-to-sample variation ( $\Phi$  fixed to 0.8), this distribution

narrows significantly, suggesting that sample quality variation is needed to explain the data. Further, our experiments include two pairs of curves from tethers that were measured simultaneously and thus experienced the same aggregran sample (same  $\Phi$ ). These pairs of tethers show nearly identical mechanical responses (Fig. S5), suggesting that sample-to-sample differences in aggregran quality are the major cause of the variability in the force-extension curves.

## HA structure on short length scales

Prior work has shown that some HA-binding proteins, such as the proteoglycan versican G1, bind cooperatively and may induce higher-order helical structure in HA (45). Such a structure would affect the HA chain on relatively short length scales and would be measured as a decrease in extension under relatively high tension. It is known that the aggregran G1 domain binds to five disaccharide units (46), approximately equal to the intrinsic HA persistence length. Bends on this length scale would likely manifest as a decrease in the fitted persistence length (47).

We see no evidence that aggregran binding alters the short length scale structure of HA by inducing bends or helical superstructure. At sufficiently high force (significantly greater than  $\sim 1$  pN, probing structural features smaller than  $l_p$ ), we see no difference between force-extension curves with and without aggregran (see example in Fig. 2 C). Furthermore, in agreement with Liu et al. (21), if we perform fits with persistence length as a free parameter, we do not find a statistically

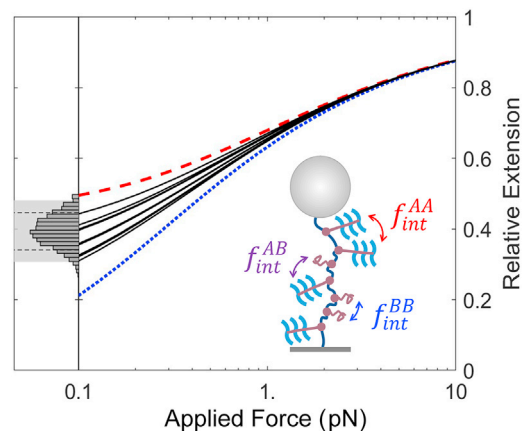


FIGURE 4 Predicted force-extension curves for HA in a mixture of undamaged and deglycosylated aggregran. Right: red dashed (blue dotted) curve shows the predicted force-extension behavior for  $\Phi = 1$  (0). Black curves show predicted force-extension for 10 tethers, each populated randomly with whole aggregran and deglycosylated core protein, according to probability  $\Phi$ , with  $\Phi$  drawn from a distribution of mean  $\bar{\Phi} = 0.8$ . For each tether,  $L_c$  is selected randomly, as described in the text. Left: histogram shows the length at 0.1 pN for 100,000 simulated tethers. The shaded gray region includes 95% of the data. The dashed lines indicate how this region would narrow if there were no noise in  $\Phi$ . Inset: schematic illustrates the 1D binding model and the heterogeneous tensions induced in segments throughout the chain. To see this figure in color, go online.

significant difference in fitted HA persistence length with and without aggregan when swelling is accounted for by an internal tension (bare HA:  $l_p = 6.6 \pm 0.4$  nm; with aggregan:  $l_p = 6.1 \pm 0.7$  nm,  $N = 10$ , reported errors are standard error of the mean). We note that these estimates of the HA persistence length are consistent with prior magnetic tweezers experiments (28) but are systematically larger than estimates from AFM (22); we attribute this to the general principle that high-tension stretching systematically biases measurements to lower persistence lengths (48) (e.g., by force disrupting local interactions and secondary structure in the chain).

To confirm this, we performed experiments using an HA-binding aggregan fragment, G1-IGD-G2, which lacks the large CS-rich region that constitutes most of aggregan's size and charge and is responsible for the large degree of swelling. Over the range of forces that probe short length scale structure, there is no discernible difference in the response with and without G1-IGD-G2 (see example data in Fig. S7). Using SDS-PAGE, we qualitatively confirmed that the fragment can bind to HA (Fig. S8). These results are in agreement with others who have found that, unlike versican G1, aggregan G1 does not bind cooperatively, does not pack tightly on HA, and does not cause the same superstructures (49). In vivo, aggregan binds HA alongside LP in a ternary complex (4). It remains a possibility that LP could induce higher-order structure in the HA at the center of these complexes, even if aggregan binding alone does not, because it has been suggested that LP induces cooperativity in aggregan G1 binding (13).

## CONCLUSIONS

We establish low-force stretching as a useful technique to detect and study proteoglycan binding to HA. Side-chain-induced swelling affects HA's chain structure on long length scales (10–100 s of nm). Structure on these scales is sensitive to subpiconewton forces, accessible using magnetic tweezers. Here, we have used this technique to quantify the expansion undergone by a single HA chain at the center of an HA-aggregan bottlebrush complex (see Fig. 2) and found that interaggregan repulsion generates an internal tension of  $\sim 0.5$  pN. We have also examined the complexes under larger stretching forces ( $\sim 1$ – $10$  pN), in which all long length scale structure has been pulled out and we are sensitive to small structures ( $\sim$ nm). At these high forces (short length scales), we find no difference in HA chain structure with and without aggregan, indicating that, whereas aggregan causes significant swelling of the random coil structure of HA, its binding does not cause any local structure changes. Although all experiments show swelling, we observe significant variability in the force response of the aggregan-decorated HA chains, which is likely caused by polydispersity of the aggregan. We use a theory of bottlebrush polymers with previously measured physical parameters for aggregan and find that our data are bounded by the expected limits for fully glyco-

sylated and deglycosylated aggregan. In doing so, we demonstrate the sensitivity of the magnetic tweezers technique to the molecular level degradation of proteoglycans and connect the chemistry of aggregan to the mechanics of larger-scale structures in cartilage.

Our experimentally validated model quantitatively predicts how bottlebrush architecture leads to the expanded nature of the HA-aggregan complex, which is crucial in maintaining the structural integrity of cartilage. Although from previous work (29,30), swelling is expected, developing a firm physical basis for the system improves understanding of the basic processes and offers quantitative methods that can later be extended to answer other biological questions. Aggregan glycosylation undergoes slight changes with age, but the more dramatic alterations to aggregan structure occur in pathologic conditions, in which aggreganase enzymes cleave the core protein, at times removing some or all of the CS attachment domain (9,10). The model presented here makes predictions for how the complex structure will change in the case of reduced core protein size; future work will explore aggregan proteolysis and evaluate these predictions. Further, the internal tension/brush model will likely be useful to describe complexes of HA with other proteoglycans (e.g., versican and brevican), which have a similar bottlebrush geometry but drastically different biological functions. In these various complexes, changes in the bottlebrush structure will set the tension; this, in turn, will affect not only the extracellular matrix mechanics but also the force signals transmitted from the exterior to the interior of cells by means of the mechanosensitive HA receptor CD44 (50).

## SUPPORTING MATERIAL

Supporting Material can be found online at <https://doi.org/10.1016/j.bpj.2020.08.016>.

## AUTHOR CONTRIBUTIONS

O.A.S., J.P.B., and S.N.I.-G. designed the research. S.N.I.-G. and J.P.B. designed and performed experiments and analyzed data. S.N.I.-G., J.P.B., and O.A.S. wrote the article.

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