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Designed Spin-Texture-Lattice to Control Anisotropic Magnon Transport in Antiferromagnets

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Spin waves in magnetic materials are promising information carriers for future computing technologies due to their ultra-low energy dissipation and long coherence length. Antiferromagnets are strong candidate materials due, in part, to their stability to external fields and larger group velocities. Multiferroic antiferromagnets, such as BiFeO₃ (BFO), have an additional degree of freedom stemming from magnetoelectric coupling, allowing for control of the magnetic structure, and thus spin waves, with the electric field. Unfortunately, spin-wave propagation in BFO is not well understood due to the complexity of the magnetic structure. In this work, long-range spin transport is explored within an epitaxially engineered, electrically tunable, 1D magnonic crystal. A striking anisotropy is discovered in the spin transport parallel and perpendicular to the 1D crystal axis. Multiscale theory and simulation suggest that this preferential magnon conduction emerges from a combination of a population imbalance in its dispersion, as well as anisotropic structural scattering. This work provides a pathway to electrically reconfigurable magnonic crystals in antiferromagnets.

1. Introduction

In the push to design next-generation memory and logic technologies that utilize spin, antiferromagnets are particularly attractive due to their stability against external magnetic fields and large magnon propagation group velocities.^[1–3] One pathway to control magnon transport is by lithographically patterning periodic structures to create a so-called magnonic crystal.^[4–9] A disadvantage of these extrinsic structures, however, is that the periodic structure is fixed. Their utility would be enhanced if they were tunable or reconfigurable.^[10] To this end, the interaction of spin waves with magnetic domain walls has been studied in ferromagnets,^[11–14] since domain walls can be written and moved using

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current-induced effects.^[15–17] In order to design and create a magnonic crystal, however, periodic arrays of domain walls would need to be created. Here, we investigate magnon-transport behavior in a room temperature magnetoelectric with an intrinsically periodic, 1D-cycloidal magnetic structure. Because this spin-texture-lattice is intrinsic to a multiferroic material and governed by the directionality of the polarization, we demonstrate that electric fields can be used to manipulate the magnonic crystal and control transport behavior.

For this study, BiFeO₃ (BFO) is used as a model system. BFO is a well-studied antiferromagnetic oxide due to the strong coupling between polarization (ferroelectricity) and magnetic (antiferromagnetism) order at room temperature, enabling all-electric-field control of magnetization.^[18–21] Because of this interaction, BFO has garnered interest in the field of spintronics, especially with recent observations showing that electric-field control of spin transport is possible.^[22] The fundamental magnetic order in BFO consists of G-type antiferromagnetism, which is modulated by broken symmetries stemming from the polarization and from antiferrodistortive FeO₆ octahedral rotations. These phenomena give rise to a long-period ($\lambda \approx 65$ nm) spin cycloid.^[23] In its equilibrium (bulk, single-crystal) structure, this antiferromagnetic cycloid propagates along three symmetry-equivalent directions: k_1 ($[\bar{1}10]$), k_2 ($[\bar{1}01]$), and k_3 ($[0\bar{1}1]$), when polarization, P , is parallel to $[111]$,^[24] with Fe atomic moments rotating in the $(11\bar{2})$, $(1\bar{2}1)$, and $(\bar{2}11)$, for k_1 , k_2 , and k_3 , respectively.^[25] The antiferromagnetic cycloid is then further perturbed by the Dzyaloshinskii–Moriya interaction (DMI), canting the moments slightly out of the $P - k$ plane and resulting in a spin-density wave (M) with the same period as the antiferromagnetic cycloid.^[26,27] This intrinsic, periodic magnetic structure in BFO can be modulated by an electric field, a capability attributed to its strong magnetoelectric coupling.^[28,29] Despite its fundamental nature, the behavior of spin propagation within such a complex spin structure is not fully understood. This gap in knowledge suggests that the cycloid's periodicity could have significant, yet unexplored, effects on the magnon dispersion.

The intimate coupling between magnetic order and ferroelectric polarization in BFO allows for the use of its ferroelectric

structure as a tool to design periodic magnetic lattices, facilitating spin-wave transport. Previous studies have shown that the ferroelectric domain structure of thin-film BFO can be controlled using both electrostatics and epitaxial strain,^[30] leading to quasi-ordered arrays of either 71° or 109° domains.^[31] This directly influences the magnetic structure.^[32,33] Our current study demonstrates that by engineering these phenomena through the boundary conditions imposed by electrostatics and lattice-mismatch strain, we can effectively guide the ferroelectric domain structure to modulate the spin cycloid. This manipulation results in the creation of a long-range, intrinsic, quasi-1D-ordered lattice. Such a lattice structure creates a gap in the magnon dispersion and introduces a significant spin-transport anisotropy, which is further tunable with an electric field.

2. Designing the Multiferroic Structure

To control the ferroelectric and magnetic structure of BFO, we synthesize epitaxial thin films on both DyScO₃ (DSO, $\approx 0.3\%$ compressive strain) and TbScO₃ (TSO, $\approx 0\%$ strain) substrates (Experimental Section). From X-ray diffraction studies (Figure S1, Supporting Information), films are confirmed to be ≈ 100 nm thick and of excellent crystalline quality. In addition to slightly different epitaxial strain states from the choice of substrate, these systems are chosen to engineer films with different ferroelectric domain structures. These films host characteristic stripe-like 71° ferroelastic domains on DSO and 109° ferroelastic domains on TSO,^[34] shown in Figure 1a,b using Piezoresponse force microscopy (PFM). The nanoscale magnetic structures of these different strain and domain configurations can then be investigated in real space using scanning nitrogen-vacancy (NV) magnetometry (Experimental Section).^[24,29,33,35]

In films deposited on DSO (Figure 1c), nitrogen vacancy (NV) magnetometry shows a chevron-like magnetic texture of the stray magnetic field, revealing a k_1 -oriented cycloid that switches direction with the change of the P directions across the 71° ferroelastic domain walls. This is consistent with previous reports^[29] and is a result of the magnetoelastic anisotropy from the substrate lattice mismatch^[32,35]. Strikingly, in the sample deposited on TSO (Figure 1d), only a single k variant of the cycloid is observed, projected along $[100]$. The longer period of these oscillations indicates that this corresponds to a k_2 , $[101]$ axis of the cycloid, which is now preferred due to the lower magnetoelastic energy from the smaller epitaxial strain of TSO. Importantly, this variant of the cycloid even persists across the ≈ 20 – 30 nm wide 109° ferroelectric domains and extends over long distances (up to at least 20 μm). This results in the formation of a quasi-long-range, 1D magnonic crystal, as illustrated in (Figure 1e).

This propagation of the k_2 cycloid variant across ferroelectric domains can be understood through the symmetry of the 109° ferroelastic domain walls. Starting with the 71° domains in films on DSO substrates, the in-plane k_1 directions are strongly magnetoelastically preferred. At the ferroelectric domain walls, where polarization P rotates from $[\bar{1}11]$ to $[111]$, the cycloid wave vector directions must be shifted, for example, from $[110]$ to $[\bar{1}10]$, to ensure the perpendicular relation $k_1 \perp P$. This shift leads to a chevron-like pattern being favored as an equilibrium state. In contrast, in films on TSO substrates, the magnetoelastic energy cost is lowered, making the k_2 and k_3 variants closer in energy.

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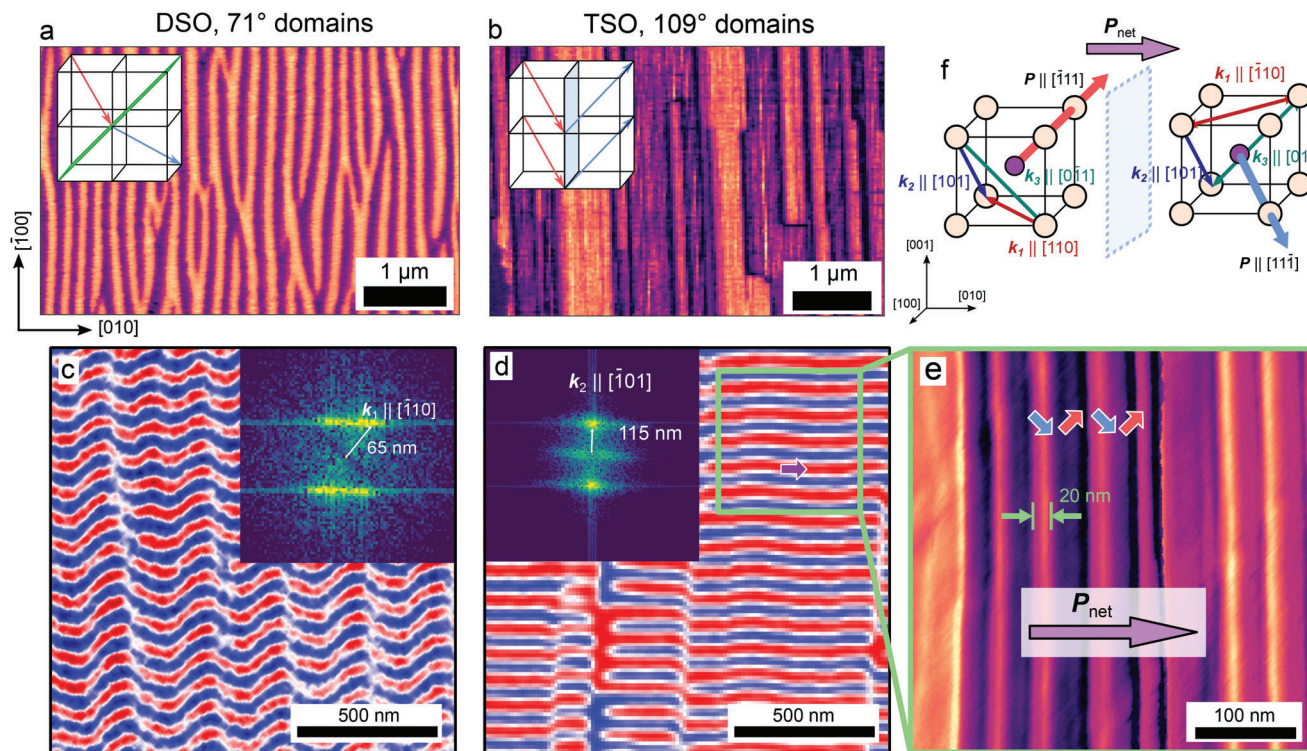


Figure 1. 1D ordered lattice. In the two BFO variants, the ferroelectric domains are arranged into ground states containing a) 109° and b) 71° ferroelastic domain walls. Insets depict the two types of domain walls and their corresponding polarizations in the BFO unit cell. In films deposited on DSO substrates, the cycloid prefers the k_1 variant, which results in a chevron-like pattern across the ferroelectric domains (c). For films on TSO, however, the domain structure allows the selection of the other symmetry-allowed axis, in this case k_2 (d). Corresponding Fourier transforms are shown as insets revealing the period and symmetry of the magnetic structures. e) Zoom of the area shown in green, where the cycloid is unperturbed by the fine, 10–30 nm ferroelastic domain walls. f) This occurs due to the symmetry of k_2 across the 109° domain wall, where the directionality of the cycloid is preserved with the fixed net polarization P_{net} imposed by the ferroelectric domains.

Furthermore, considering the 109° domain walls in the films on TSO substrates, assuming k_2 is now an allowed direction, the propagation axis would not need to alter with the polarization change. This is because [101] is perpendicular to both $[111]$ and $[1\bar{1}\bar{1}]$ (Figure 1f), making k_2 also a symmetry-favored equilibrium in this configuration. Consequently, k_2 emerges as the globally preferred equilibrium state, leading to a single cycloid variant that persists across the 109° ferroelastic domain walls, despite their ≈ 20 –30 nm period^[30].

3. Magnon Transport

To test spin transport with respect to the directions defined by the quasi-1D spin-spiral lattice, we fabricated test structures that are schematically shown in Figure 2a. These structures consist of source and detector wires aligned both parallel and perpendicular to \mathbf{k} . The magnon-induced voltage signal in these structures is compared along multiple directions with respect to \mathbf{k} to show anisotropy, as well as in both the quasi-1D and chevron-like cycloid structures (Figure 1c). First, considering thermally excited magnons in a two-wire scheme, a current in the source wire is used to create a temperature gradient (∇T) in the device, causing thermally excited spin waves to propagate $\mathbf{q} \parallel -\nabla T$, where \mathbf{q} is the magnon propagation direction, via the spin Seebeck effect.^[22] These magnons are then sensed in the detector wire through the

inverse spin Hall effect (ISHE), which converts the spin current to a DC voltage, $\Delta V_{NL, 2\omega}$.^[36,37] To preserve the ferroelectric domains, this measurement was repeated while reversing the directionality of the thermal transport with respect to the in-plane polarization (details in Note S1, Supporting Information). Confirmation that the signal is magnetic in origin comes from testing with different detector materials, e.g., changing the detector wire from Pt to W (with a larger, opposite-signed spin-Hall angle compared to Pt) reverses the sign of $\Delta V_{NL, 2\omega}$. Similarly, moving to a material with a much larger spin-Hall angle, such as SrIrO₃ (SIO),^[38,39] proportionally increases the ISHE voltage. The sensitivity of the output voltage to the spin Hall material indicates the intrinsic origin of the magnon-driven ISHE voltage on the detector and controls for random signal artifacts generated by the measurement geometry.

In films featuring the quasi-1D ordered cycloid lattice, a strong anisotropy is observed with respect to directions $\mathbf{q} \parallel \mathbf{k}$ and $\mathbf{q} \perp \mathbf{k}$. We define θ as the angle between \mathbf{q} and \mathbf{k} . Notably, the ISHE voltage signal, indicative of spin wave conductivity, is approximately three times larger when $\theta = 90^\circ$, aligning with the uniform direction of the texture. In contrast, samples with the alternating chevron-like structure, characterized by a net $\mathbf{k}_{net} = \sum \mathbf{k} = 0$, exhibit minimal spin transport anisotropy (Figure 2c). Additionally, in the chevron-like samples, where locally $\theta = 45^\circ$, $\Delta V_{NL, 2\omega}$ lies between the values at $\theta = 90^\circ$ and 0° seen in the 1D

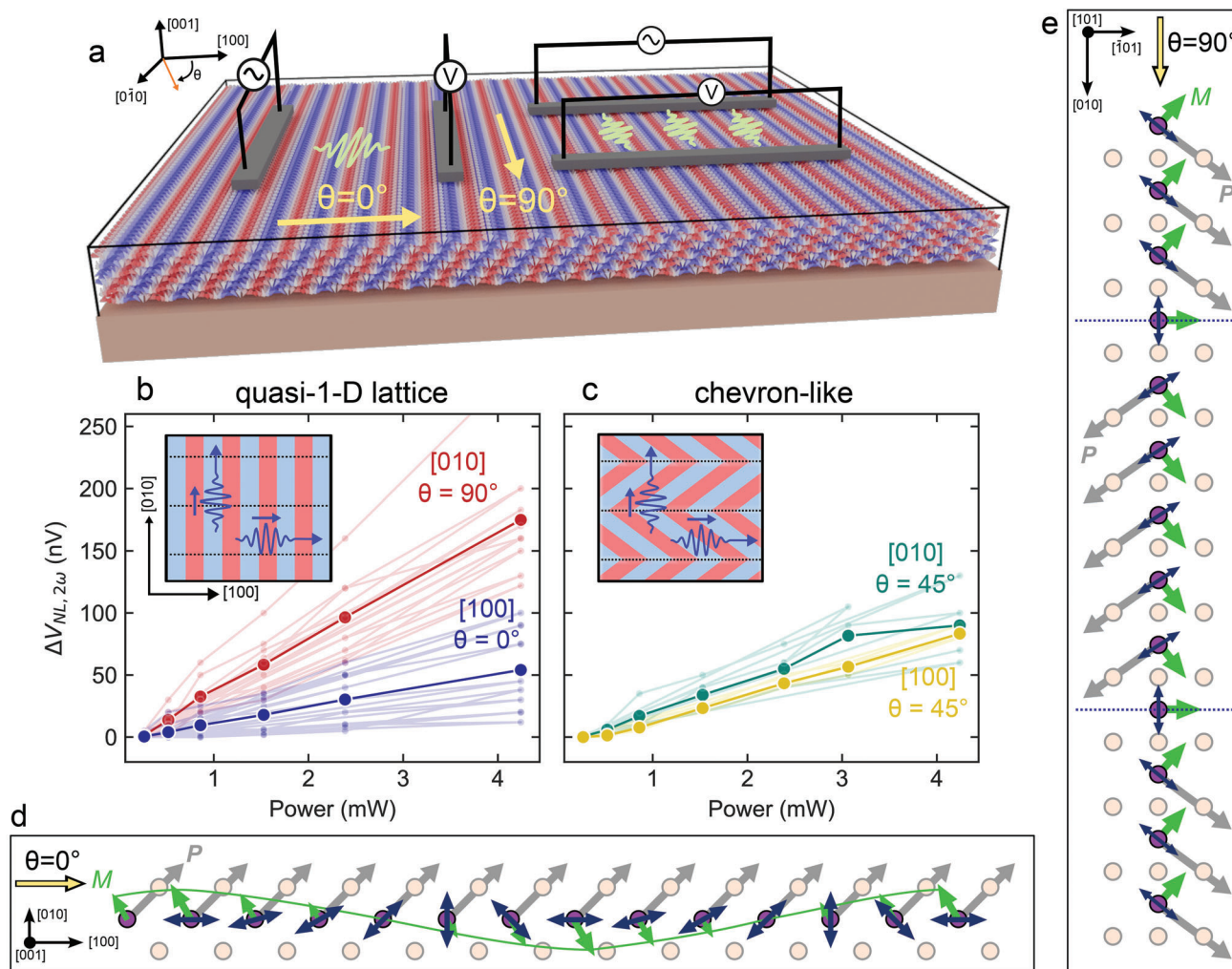


Figure 2. Spin wave anisotropy. a) Schematic of the sample geometry of the test structures \parallel ($\theta = 0^\circ$) and \perp ($\theta = 90^\circ$) to k . These structures give rise to different magnon conductivities due to the directionality of the cycloid. In the quasi-1-D lattice of 109° domains, (b), there is high conductivity along the ferromagnetically coupled direction, $[010]$, and low conductivity along k , $[100]$. Conversely, for the 71° domain scenario, (c), the cycloid, and thus the conductivity, display no distinct anisotropy between the $\theta = 0^\circ$ and 90° directions. This reflects the magnetic cycloid's interaction with both the 109° and 71° ferroelastic domain walls. The data presented in (b,c) is sampled across 40 devices. In the case of 109° cycloid, viewed along the directions parallel to d) $[010]$ ($\theta = 0^\circ$) or e) $[100]$ ($\theta = 90^\circ$), the magnetic moments only deviate slightly across the ferroelastic domain walls, but cycle by 2π in the projected direction of k_2 . Here, the purple (pink) atoms are the Fe (Bi) sites, the gray arrow is P , the dark blue arrow is the antiferromagnetic axis (L), and the green arrow corresponds to M of the spin density wave.^[35] In (d), the antiferromagnetic Fe moments rotate in the $k - P$ plane, along with the magnitude of M traced by the green line. In (e), the 109° ferroelastic domain walls are shown along with the deviation of M across the wall. Unit cells are shown in an approximately 15:1 scale. The orientation of schematics (d) and (e) is viewed along the orthogonal directions shown in (a).

ordered system. This behavior can be initially understood by examining the nanoscopic magnetic spin structure within the cycloid as described schematically in Figure 2d,e. In a line cut $\perp k$, the spins are modulated by the presence of the 109° ferroelastic domain walls but possess a persistent net magnetization component, illustrated in Figure 2e. In contrast, oriented parallel to k , the spins rotate by a full 2π over the period of the cycloid (65 nm, Figure 2d). To further evaluate the physics associated with transport in the quasi-1D structure, we turn to both mesoscale phase-field simulations and model Hamiltonian calculations of spin transport \parallel and $\perp k$.

From phase-field simulations, we observe an anisotropy characterized by reduced spin-wave conductivity along the $q \parallel k$ direc-

tion, compared to $q \perp k$. Simulations are carried out by imposing a cycloidal magnetic ground state comparable to those observed experimentally and by initiating spin waves within a specified frequency range at one edge of the sample (Experimental Section). The efficiency of propagation was quantified by comparing the magnetization precession at the point of origin and a defined distance from the source. In scenarios where $q \perp k$, the initial waveform exhibits negligible damping even at a distance equivalent to 10λ (Figure 3a). When $q \parallel k$, however, the spin wave exhibits significant attenuation over the same distance. This attenuation is evident in both the spectrum of the magnetization precession angle and its integrated energy across the spectrum, which is shown to be influenced by the magnitude of the anisotropy within the

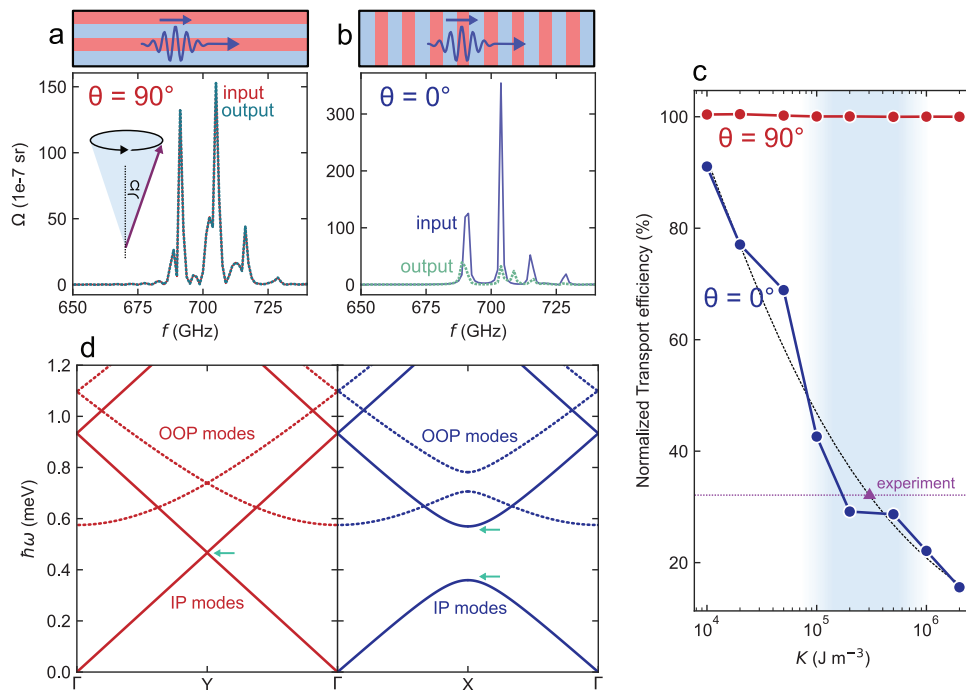


Figure 3. Simulation of magnon conductivity. Simulations of frequency-dependent magnon power when the spin wave is driven along the direction of \mathbf{q} , which is a) perpendicular, and b) parallel to, \mathbf{k} , respectively, in the quasi-1D lattice. The energy carried by the magnons is extracted as $\Omega \times f$, i.e., the product of the conical angle, Ω , of the time-dependent precession of magnetization in Fourier space, and the frequency f . Input and output spectra are measured at the initialization location and far side of the simulated cell. The integral of this spectra is then reported as a relative conductivity, c), simulated as a function of the 1D anisotropy, K , of the cycloid texture. The shaded region represents a range of energies reported in the literature, as well as values matched to the experimental and analytical results presented here. The horizontal dashed line corresponds to the measured efficiency, which is related to the plot by the intercept shown. d) Calculated magnon dispersion, illustrating the emergence of a gap along the $\mathbf{q} \parallel \mathbf{k}$ direction which leads to the reduced spin wave conductivity. X and Y are the high symmetry points corresponding to the [100] ($\parallel \mathbf{k}$) and [010] ($\perp \mathbf{k}$), respectively.

cycloid. As illustrated in Figure 3b, a scenario with an anisotropy of $K = 100 \text{ kJ m}^{-3}$ leads to $\approx 50\%$ damping in spin-wave-energy magnitude over the same length scale. This transport efficiency can then be studied as a function of K to compare with previous calculations and experimental results, defining a reasonable window for the value of K . The precession angles corresponding to other values of K are shown in Figure S6 (Supporting Information).

At the microscopic scale, analytical calculations of the momentum-dependent magnon dispersion in the quasi-1D lattice illustrate the difference in the magnon conduction $\perp \mathbf{k}$ and $\parallel \mathbf{k}$. Analytical solutions for the magnon bands of BFO are based on a continuum Hamiltonian whose density is given by^[40]:

$$H = A(\nabla \mathbf{n})^2 - \alpha \mathbf{P} \cdot [\mathbf{n}(\nabla \cdot \mathbf{n}) + \mathbf{n} \times (\nabla \times \mathbf{n})] - 2\beta M_0 \mathbf{P} \cdot (\mathbf{m} \times \mathbf{n}) - K_u n_c^2 + \lambda m^2 \quad (1)$$

where the first term is the exchange energy, the second and third terms are DMI terms, the fourth term is the anisotropy, and the last term represents the suppression of the net magnetization due to strong antiferromagnetic coupling. The ground-state configuration can then be obtained (Note S3, Supporting Information), showing the well-known anharmonic spin-cycloid structure.^[40–42] Solving the time-dependent Euler–Lagrange equation for the small fluctuations, two spin-wave modes result: 1) An in-plane mode with spin fluctuations within the spin-cycloid plane and 2) an out-of-plane mode whose spin

fluctuations are orthogonal to the spin-cycloid plane. By considering the quasi-2D geometry of the plane-wave-like spin texture of BFO and neglecting the Hamiltonian term $\alpha\beta$ that is known to be much weaker than the rest of the terms in the Hamiltonian,^[43] the band structures of in-plane (IP) magnons and out-of-plane (OOP) magnon modes are obtained analytically (Figure 3d). This clearly shows the formation of a gap in the dispersion along the $\mathbf{q} \parallel \mathbf{k}$ direction, due to the periodicity of the magnetic lattice. This gap indicates a lower density of states along this direction, resulting in an anisotropic, lower magnon conductivity.

In parallel to measurements of thermally excited magnons, non-local transport measurements have been performed using injected spin currents via the spin-Hall effect (SHE) of the source wire^[36] (Figure 4a). In this case, the sign of the output signal will correspond directly to the sign of the input signal (i.e., a positive voltage will result in a positively polarized spin current, which then results in a positive voltage at the detector), rather than only the amplitude of the input (so-called first-harmonic and second-harmonic measurements). While the spin Hall angles of platinum and tungsten are too low to generate an appreciable ISHE signal at the $2 \mu\text{m}$ electrode spacing, the efficient spin-charge transduction of SIO allows detection of the SHE excited magnon signal.^[38,39] In the device geometry where \mathbf{q} is perpendicular to the domain walls, where electric fields allow the ferroelastic domain walls to remain largely stationary^[19,22,35] (Figure S2, Supporting Information), combining the engineered

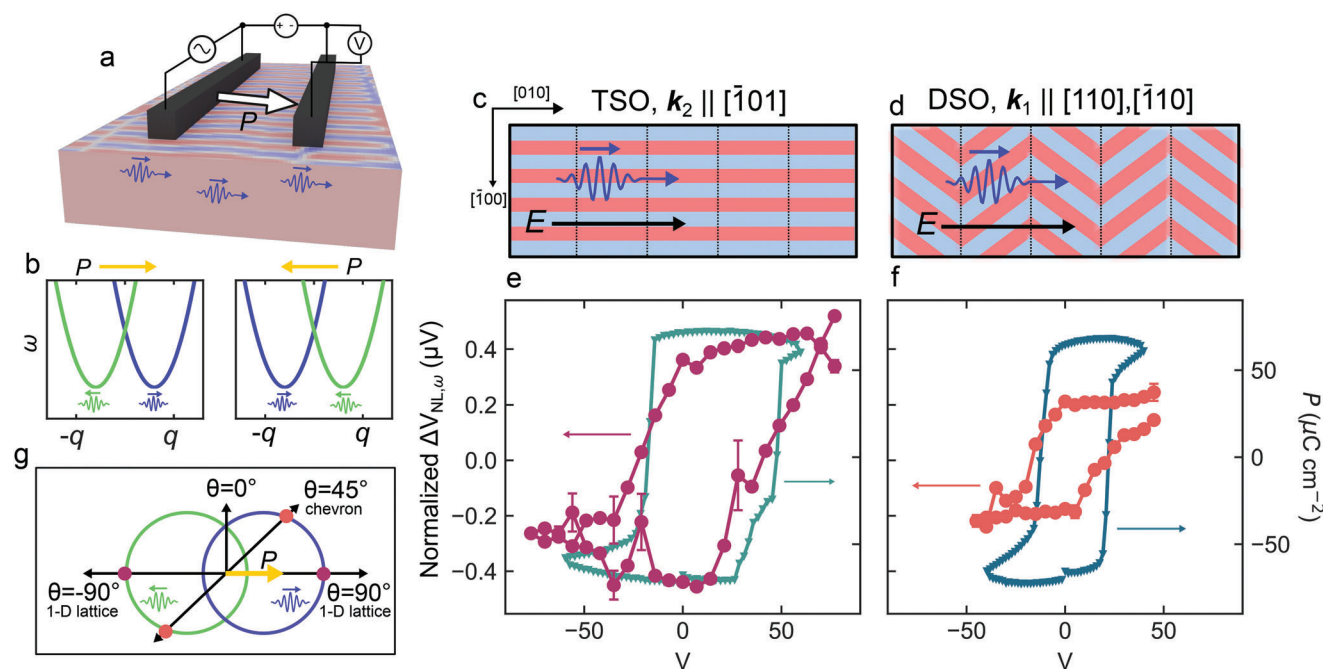


Figure 4. Polarization control of magnon propagation. a) Schematic showing the non-local device geometry with magnons generated by injection of spin from the source wire. b) We hypothesize that the non-reciprocity of the transport is due to a Rashba-like asymmetry of the magnon dispersion emerging from the crystal asymmetry, which is coupled directly to the ferroelectric polarization and DMI. c, d) Illustration showing the sense of the cycloid with respect to the device geometry for films deposited on the two different substrates. In this orientation, the ferroelectric domain structure remains stationary with cycling electric field. e, f) Switchable non-local signal overlaid with the ferroelectric hysteresis for both the single variant and chevron-like cases. g) Combined with the above measurements, we hypothesize that the anisotropy can be explained by a split and anisotropic magnon dispersion, where the magnon conductivity maps to a surface with directions defined by q and k . The distance from the origin is representative of the conductivity, and thus the population.

spin texture with an epitaxial, large SHE oxide can further optimize functionality. This should allow for deterministic switching of the magnetic structure under a continuous rotation of P .^[35]

In test structures using SIO electrodes, we observe a similar anisotropic spin transport, dependent on the underlying magnetic structure. Here, magnons are excited by the SHE and the ferroelectric polarization (noted $P \uparrow$ and $P \downarrow$, where the direction of magnon propagation, $q \uparrow$, is fixed), is switched using an in-plane electric field. Upon switching the in-plane component of the polarization, where the direction of magnon propagation, q , is fixed, the detected first-harmonic non-local voltage, $V_{NL,1\omega}$, switches hysteretically on the scale of $\approx 1 \mu\text{V}$ for a device spacing of $2 \mu\text{m}$. The repeatable difference between the $P \uparrow q \uparrow$ and $P \downarrow q \uparrow$ states is indicative of a non-reciprocity in the magnon dispersion tied to the direction of P . This can be qualitatively rationalized in the form of a Rashba-like, DMI-induced splitting of the spin-dependent magnon bands (sketched in Figure 4b).^[44–48] The result is a $V_{NL,1\omega}$ that is both nonreciprocal and switchable. Since the samples under investigation exhibit no net magnetic moment, the inference is that magnon non-reciprocity in BFO should stem from a chiral component of the magnetic cycloid, which interacts with P .

Under an applied electric field, this hysteresis in $V_{NL,1\omega}$ follows the switching of P , where P and $\Delta V_{NL,1\omega}$ both flip signs at the same coercive voltage (Figure 4d,e).^[49] As the ferroelastic domain

walls remain stationary during this event (Figure S2, Supporting Information), a comparison can be made between samples with quasi-1D and chevron-like configurations (Figure 4c,d). In quasi-1D samples with a single k state, the magnitude of $V_{NL,1\omega} \perp k$ is two-times higher than that of the samples showing a chevron-like magnetic cycloid. The magnitude of this scaling directly parallels the observations in Figure 2, where the chevron-like texture shows a two-times lower conductivity than in the single k samples when $q \perp k$. Taken together, these measurements allow us to sketch the approximate shape of the magnon dispersion projected into the plane of the film (Figure 4g), where $\theta = 90^\circ$ ($P \perp k$) is high conductivity, $\theta = 0^\circ$ ($P \parallel k$) is low, $\theta = 45^\circ$ is intermediate, and $\theta = 90^\circ$ and $\theta = -90^\circ$ carry an opposite sign angular momentum.

These results demonstrate control of the magnetic structure of BFO and a method for effectively decoupling the magnetic structure from the ferroelectric domains. This has significant implications for magnonic devices using BFO, in that fundamental anisotropies of the cycloid structure can be used to optimize transport. The formation of a quasi-1D ordered magnonic lattice comprised of the spin cycloid should be of fundamental interest in terms of phase transitions in a 1D lattice and the possibility of obtaining long-range order in them. In this sense, future studies could be directed towards controlling the ferroelectric/ferroelastic domain structure through the use of vicinally cut substrates^[30] or further defining the magnon dispersion stemming from the complex cycloidal texture of BFO.^[50]

4. Conclusion

Using heteroepitaxy as a building block and imposing a combination of electrostatic and elastic constraints, we can strategically design both the ferroelectric and magnetic domain structure of multiferroic BFO. By doing so, a quasi-1D crystal of the antiferromagnetic spin cycloid can be chosen as the ground state. This anisotropic structure results in preferential spin-wave conduction orthogonal to its axis, due to the opening of a gap in the magnon dispersion created by the periodicity of the magnetic lattice. Leveraging magnetoelectric coupling, this magnonic crystal can be dynamically tuned using an electric field, allowing us to empirically map the magnon dispersion in the film plane with respect to the direction of ferroelectric polarization. This opens the door for the design of magnonic devices utilizing the anisotropy of the spin texture, as well as providing fundamental insights into the conduction of spin waves in complex magnetic systems.

5. Experimental Section

Sample Fabrication: BFO samples (100 nm thick) were deposited on DSO and TSO substrates using both molecular-beam epitaxy (MBE) and pulsed laser deposition (PLD). MBE films were grown by reactive MBE in a VEECO GEN10 system using a mixture of 80% ozone (distilled) and 20% oxygen. Elemental sources of bismuth and iron were used at fluxes of 1.5×10^{14} and 2×10^{13} atoms per cm^2s respectively. All films were grown at a substrate temperature of 675°C and a chamber pressure of 5×10^{-6} Torr. PLD samples were deposited using a 248 nm KrF laser with $\approx 1.5 \text{ J cm}^{-2}$ at 700°C and 90 mTorr O_2 . Pt (15 nm) and W (15 nm) layers were deposited using magnetron sputtering at room temperature. SiO layers were deposited using MBE using a mixture of 80% ozone and 20% O_2 along with elemental sources of strontium and iridium. Due to the high temperatures needed to generate an iridium flux, an electron beam was used to evaporate iridium. The samples were grown at a substrate temperature of 695°C and a background pressure of 1×10^{-6} Torr. Devices were fabricated using standard optical lithography followed by ion milling.

Sample Characterization: X-ray diffraction was performed on a PANalytical Xpert3 diffractometer. Piezoresponse force microscopy was performed on an Asylum MFP3D. NV magnetometry was performed on a Qnami ProteusQ system with a parabolic Quantilever MX+ diamond cantilever.

Phase Field Simulations: MagneX, a substantial GPU-enabled micro-magnetic simulation code package (available on GitHub), was employed to phenomenologically explore the directionality of magnon transport in the cycloid. The time-domain Landau–Lifshitz–Gilbert equation was solved, incorporating exchange coupling, DMI coupling, anisotropy coupling, and demagnetization coupling. A set of abstract ferromagnetic properties was utilized to phenomenologically represent the antiferromagnetic material. The magnetization was pinned by an external applied magnetic field that mimics the cycloid texture. The Gilbert damping constant was set to be 0.0005, the exchange coupling constant was $3.0 \times 10^{-12} \text{ J m}^{-1}$, and the DMI coupling constant was $1.0935 \times 10^{-3} \text{ J m}^{-2}$. The anisotropic hard axis was set to the [111] direction, while the anisotropy constant was swept in the range of 1.0×10^4 and $2.0 \times 10^6 \text{ J m}^{-3}$. The total magnetization was normalized to be one. On one side of the sample, a modified Gaussian pulse in the effective magnetic field was applied in both the x and y direction, mimicking injected spins from the top SHE electrode. The pulse was in the mathematical form of $e^{-(t-t_0)^2/2T_0^2} \cos(2\pi f_0 t)$, where f_0 is 700 GHz. The output port was on the opposite side of the input port in the in-plane direction. The simulated time-varying magnetization components were converted into the frequency domain with the Fourier transform, from which the cone angle of the magnetization was calculated. For more details, see Note S2 (Supporting Information).

Analytical Calculations: See Note S3 (Supporting Information).

Transport Measurements: Transport measurements were performed using three-terminal devices where two were used to inject current and two for output. Two cross-terminals were used to apply electric field for polarization control. All operations were controlled through the in-house developed code. V_{NL} was measured using an SR830 lock-in amplifier and locking to the second (in the case of thermally excited magnons) or first (in the case of spin Hall excited magnons) harmonic of the excitation voltage. An AC signal at 7 Hz was used for both first- and second-harmonic measurements. In the first-harmonic measurement, the magnitude of the injected current was chosen to be small to avoid the thermal drift in the output voltage.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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Conflict of Interest

The authors declare no conflict of interest.

Data Availability Statement

The data that support the findings of this study are available from the corresponding author upon reasonable request.

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antiferromagnet, multiferroics, magnonics

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