Lawrence Berkeley National Laboratory

Recent Work

Title

THE THERMODYNAMIC PROPERTIES OF SILVER-PALLADIUM ALLOYS

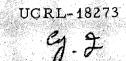
Permalink

https://escholarship.org/uc/item/4dt1b9sv

Authors

Chan, J.P. Hultgren, Ralph.

Publication Date 1968-06-01



1 2 1 1 2 1

- **X**-

2. '.

University of California

, Ernest O. Lawrence Radiation Laboratory

1.16 10

TWO-WEEK LOAN COPY

This is a Library Circulating Copy which may be borrowed for two weeks. For a personal retention copy, call Tech. Info. Division, Ext. 5545

THE THERMODYNAMIC PROPERTIES OF SILVER-PALLADIUM ALLOYS J. P. Chan and Ralph Hultgren

June 1968

-1814 (L.R.-7 (L.R.-,

1.24.28

3 . X. 3 . K. 2 . X

LIBRARY AND Berkeley, California

The second s

DISCLAIMER

This document was prepared as an account of work sponsored by the United States Government. While this document is believed to contain correct information, neither the United States Government nor any agency thereof, nor the Regents of the University of California, nor any of their employees, makes any warranty, express or implied, or assumes any legal responsibility for the accuracy, completeness, or usefulness of any information, apparatus, product, or process disclosed, or represents that its use would not infringe privately owned rights. Reference herein to any specific commercial product, process, or service by its trade name, trademark, manufacturer, or otherwise, does not necessarily constitute or imply its endorsement, recommendation, or favoring by the United States Government or any agency thereof, or the Regents of the University of California. The views and opinions of authors expressed herein do not necessarily state or reflect those of the United States Government or any agency thereof or the Regents of the University of California.

UCRL-18273

Submitted to: Journal of Chemical Thermodynamics

1

્ર્

UNIVERSITY OF CALIFORNIA Lawrence Radiation Laboratory Berkeley, California AEC Contract No. W-7405-eng-48

THE THERMODYNAMIC PROPERTIES OF SILVER-PALLADIUM ALLOYS

J. P. Chan and Ralph Hultgren

June 1968

UCRL-18273

The Thermodynamic Properties of Silver-Palladium Alloys

By

J. P. Chan* and Ralph Hultgren*

*Research Physical Chemist, Materials Science Division I, Sandia Corp., Livermore, Calif. and Professor of Metallurgy, Department of Mineral Technology, College of Engineering and Inorganic Materials Research Division, Lawrence Radiation Laboratory, University of California, Berkeley, California 94720.

<u>Abstract</u>

Heats of formation of silver-palladium solid solutions were measured by liquid tin solution calorimetry. The maximum exothermic heat of formation was found to lie at 40 atomic percent palladium. This was correlated with the magnetic character of the alloys; they are diamagnetic below 40 atomic percent palladium and paramagnetic above. Pauling's theory of metallic chemical bonding provides a satisfactory explanation of this. <u>Introduction</u>. The alloy chemistry of transition metals is poorly understood, both because of lack of thermodynamic data and because of the complexity of electronic structure of these metals. The silver-palladium system was chosen for study because relatively simple bonding concepts lead to a prediction that the exothermic heat of formation should be greater on the silverrich side, reaching a maximum near 40 atomic percent palladium. This was found to be the case.

Transition metals are characterized by having incompletely filled inner d-orbitals; the resulting unpaired electrons make these metals paramagnetic or ferromagnetic. Palladium has approximately 0.6 unpaired electrons per atom. Silver has one more electron than palladium (the atomic number is one greater); this is more than sufficient to fill all the d-orbitals so that silver is diamagnetic, with 0.4 excess electrons per atom in higher energy levels.

Silver and palladium form a complete series of face-centered-cubic solid solutions. Substitution of a silver atom for a palladium atom on the crystal lattice increases the number of electrons by one. On the palladiumrich side, the extra electron enters the d-orbital, reducing the paramagnetic susceptibility. When the silver concentration reaches 60 atomic percent, all the d-orbitals are filled and alloy becomes diamagnetic. Further increases in silver concentration produce a change in bonding electron density with an effect on the heat of formation of the alloy which differs from that on the palladium-rich side. The magnetic susceptibilities^{1, 2, 3} shown in Figure 1 confirm this description.

Experimental. Palladium sponge 99.99% pure and electrolytic silver 99.95% pure were used to make the alloys. Mixtures of filings in zirconia lined graphite crucibles were melted at about 1500°C under helium by induction heating to make ingots weighing about 3 grams each. After cold-working, the ingots were sealed in evacuated quartz tubes and homogenized by heating

-2-

for two to three weeks at temperatures 50-100° below the solidus (1000°C to 1300°C). Homogeneity was tested by X-ray diffraction of mixtures of filings taken from both ends of each ingot, after strain-annealing the filings at 600°C for thirty minutes in evacuated Pyrex glass tubes. Back-reflection lines were sharp and the lattice constants agreed well with those of Coles⁴ for the expected compositions, assuming small losses of silver during melting.

9

192

Heats of formation of the alloys were determined in a liquid-tin solution calorimeter described elsewhere⁵. In this method the heat of solution of the alloy is subtracted from the heats of solution of corresponding amounts of the pure components, to give the heat of formation of the alloy. Experimental data are given in Table I.

The heat contents of the alloys were measured in a diphenyl ether drop calorimeter⁶. The experimental results are given in Table II. A moderate negative deviation from additivity of heat contents was found. From the heat contents the heats of formation, which were measured near room temperature, can be calculated at high temperatures.

The heats of solution of silver and palladium in liquid tin, calculated at 700°K are shown in Table III. The heats of formation of the alloys, calculated at 1200°K are shown in Table IV.

Discussion. The extraordinarily high exothermic heat of solution of palladium in tin is surprising. None of the generalizations on metallic bonding give a hint that palladium would differ much from silver. Palladium and silver have atomic radii (1.375 and 1.445Å) which differ by only 5% and their

-3-

electronegativities are also not very different (2.2 and 1.9). Yet palladium has one of the most exothermic heats of solution known, while silver dissolves endothermically. Silver and palladium, as shown in this paper, behave normally to one another, forming alloys with moderate exothermic heats. Clearly there are large factors in the formation of metallic bonds in alloys which are not explainable on the simple concepts of electronegativity and atomic size.

The heats of formation at 1200°K of silver-palladium alloys are shown in Figure 2. Their dependence on concentration resembles a parabola, such as would be predicted by the well-known quasi-chemical theory in which heats of formation depend on the number of bonds between unlike nearest neighbors, except that they are strongly skewed toward the silver-rich side. The maximum exothermic effect occurs near 40 atomic percent palladium; significantly, this is the composition where magnetic properties drastically change (Figure 1), the alloys are diamagnetic at lower palladium contents, and paramagnetic at higher.

This behavior is not peculiar to silver-palladium alloys. It seems to appear in all binary alloys where one metal is from the Ni-Pd-Pt column of the periodic table and the other member is from the Cu-Ag-Au column. Thus, besides Ag-Pd, Cu-Pd⁷, Au-Pd⁸, Cu-Pt⁹, are exothermic, skewed toward the Cu-rich or Au-rich side, with maximum exothermic effect near 40 atomic percent Pd or Pt. Cu-Ni¹⁰ and Au-Ni¹¹ are endothermic, but are less endothermic on the Cu-rich or Au-rich sides. Ag-Ni¹², Ag-Pt¹², and

-4-

Ŷ

Au-Pt¹² have miscibility gaps; the first is hightly immiscible even in the liquid state, the latter two are more miscible at the Ag-rich and Au-rich sides, respectively. Thus of the nine binary combinations, none disagree with the observation that the exothermic character is greater on the Cu-, Ag-, or Au-rich sides of the binary alloys.

Pauling's theory of metallic bonding agrees nicely with this result. According to Pauling¹³ the transition metals in the middle of the periodic table,' including palladium develop a valence of 6 in the metallic state. Silver, however, is hyperelectronic; the electrons in excess of those necessary to fill the d-orbital are paired with electrons which would otherwise be valence electrons. Thus silver, with 0.4 excess electrons, develops a valence of only 5.6 (or 5.56). Each Pd atom which enters the Ag lattice removes an excess electron into its incompletely filled d-orbital increasing the valence of the Ag; when x_{Pd} reaches 0.40, the valence of the remaining silver atoms reaches 6.0. Thus on the Ag-rich side of the phase diagram each Pd atom increases the valence, contributing to the exothermic character of the alloy; this ceases as the concentration of Pd exceeds 40 atomic percent. The valence is shown graphically in Figure 3, with the extra valence contribution shown in the shaded area.

Myles¹⁴ has recently determined Gibbs energies of Ag-Pd solid solutions from vapor pressure measurements. Combining these values with the enthalpies of formation determined in this paper, the Δ S values shown in Table IV and Figure 2 are obtained.

-5-

<u>Acknowledgments.</u> We are indebted to the U.S. Atomic Energy Commission and to the U.S. Army Office of Ordnance Research for financial assistance. We also express our appreciation to Stanley E. Ross for assistance in the experimental work and to Raymond L. Orr for many helpful suggestions.

TABLE I

Measured	Heats	of Sc	lution	and	Heats	of F	ormation	of the	Alloys
the second se	and the second se	and the second se	the second s	the second s	the second s	the second s		And the Party of t	the second s

· · · · · · · · · · · · · · · · · · ·					
X _{Pd}	T _i , ⁰K	T _f ,°K	^{∆H} sol'n cal/g-atom	$\Delta H_{formation}$ cal/g-atom	
1.000	342.7	700. 8	-23900		
1.000	342.6	700.8	-24210		
0.798	324.6	703.2	-17430	-410	
0.798	343.0	701.5	-17440	-510	
0.612	324.9	701.4	-11240	-990	
0.612	343.1	700.7	-11320	-1030	
0.492	324.8	700, 8	-7470	-1150	
0.492	342.4	700.5	-7580	-1150	
0.394	324.9	699.6	-4380	-1290	
0.394	326.1	706.6	-4400	-1230	
0.200	324.6	698.4	1130	-950	
0.200	342.8	700.2	1010	-930	
0.000	325.6	699.1	6150		
0.000	325.9	705.2	6300		
0.000	342.9	699.2	6120		
0.000	342.6	699 . 2	6090		

ΤА	BI	\mathbf{E}	II
----	----	--------------	----

· · · · ·			
X _{Pd}	T, °K	H _T -H _{298.15} cal/g-atom	$\Delta(\mathrm{H_{T}^{-H}298.15})$ cal/g-atom
0.200	998.3	4560	-40
0.200	1021.5	4670	-90
0.200	1076.4	5040	-120
0.200	1078.7	5050	-130
0.394	989.4	4440	-110
0.394	1095.4	5240	-80
0.394	1095.7	5260	-60
0.492	892.9	3820	-60
0.492	998.6	4530	-100
0.492	1078.2	5210	0
0.798	886.8	3880	10
0.798	998.9	4670	0
0.798	1021.3	4770	-60

Measured Heat Contents for the Alloys

3

TABLE III

Measured Heats of Solution in Tin at 700°K for Pure Silver and Palladium

Substance	∆H 700°K sol'n cal/g-atom
Pd	-26400±160
Ag	3845±55

TABLE IV

-8-

Silver-Palladium Alloys at 1200°K

$$(1-x)Ag(s) + xPd(s) = Ag_{1-x}Pd_{x(s)}$$

$\begin{array}{c ccccccccccccccccccccccccccccccccccc$				
0.2 -1927 -1115 0.677 0.3 -2318 -1320 0.832 0.4 -2429 -1370 0.882 0.5 -2327 -1300 0.856 0.6 -2087 -1110 0.814 0.7 -1751 -850 0.751 0.8 -1338 -590 0.623	X _{Pd}	ΔG	$\Delta \mathrm{H}$	ΔS
0.3 -2318 -1320 0.832 0.4 -2429 -1370 0.882 0.5 -2327 -1300 0.856 0.6 -2087 -1110 0.814 0.7 -1751 -850 0.751 0.8 -1338 -590 0.623	0.1	-1196	-725	0.392
0.4 -2429 -1370 0.882 0.5 -2327 -1300 0.856 0.6 -2087 -1110 0.814 0.7 -1751 -850 0.751 0.8 -1338 -590 0.623	0.2	-1927	-1115	0.677
0.5 -2327 -1300 0.856 0.6 -2087 -1110 0.814 0.7 -1751 -850 0.751 0.8 -1338 -590 0.623	0.3	-2318	-1320	0.832
0.6 -2087 -1110 0.814 0.7 -1751 -850 0.751 0.8 -1338 -590 0.623	0.4	-2429	-1370	0.882
0.7 -1751 -850 0.751 0.8 -1338 -590 0.623	0.5	-2327	-1300	0.856
0.8 -1338 -590 0.623	0.6	-2087	-1110	0.814
	0.7	-1751	-850	0.751
0.9 -825 -310 0.429	0.8	-1338	- 590	0.623
	0.9	-825	-310	0.429

References

- 1. B. Svensson, Ann. Phys. Lpz., 14, 699 (1932).
- 2. J. Wucher, Thesis, Strassbourg (1950).
- 3. F.E. Hoare, J.C. Mathews, and J.C. Walling, Proc. Roy. Soc., <u>216</u>, 502 (1953).
- 4. B.R. Coles, J. Inst. Metals, <u>84</u>, 346 (1956).
- 5. R. L. Orr, A. Goldberg, and R. Hultgren, Rev. Sci. Instr., <u>28</u>, 767 (1957).

- R. Hultgren, P. Newcombe, R. L. Orr, and L. Warner, National Physical Laboratory, Symposium No. 9, 1959, H. M.S.O., London, Paper No. 14.
- 7. J.R. Guadagno, R.L. Orr, and R. Hultgren, 13th Tech. Report, Minerals Research Lab., Berkeley (Sept. 1, 1961).
- 8. J.B. Darby, Jr., Acta Met. <u>14</u>, 265 (1966).
- 9. K.M. Myles and J.B. Darby, Jr., Acta Met., 16, 485 (1968).
- 10. R. Oriani, and W. K. Murphy, Acta Met., 8, 23 (1960).
- 11. R.L. Orr, Acta Met., 8, 489 (1960).
- 12. M. Hansen, "Constitution of Binary Alloys", McGraw Hill, 1958.
- L. Pauling, "Nature of the Chemical Bond", Cornell Univ. Press,
 3rd Ed., 1960.
- 14. K. M. Myles, Acta. Met., <u>13</u>, 109 (1965).

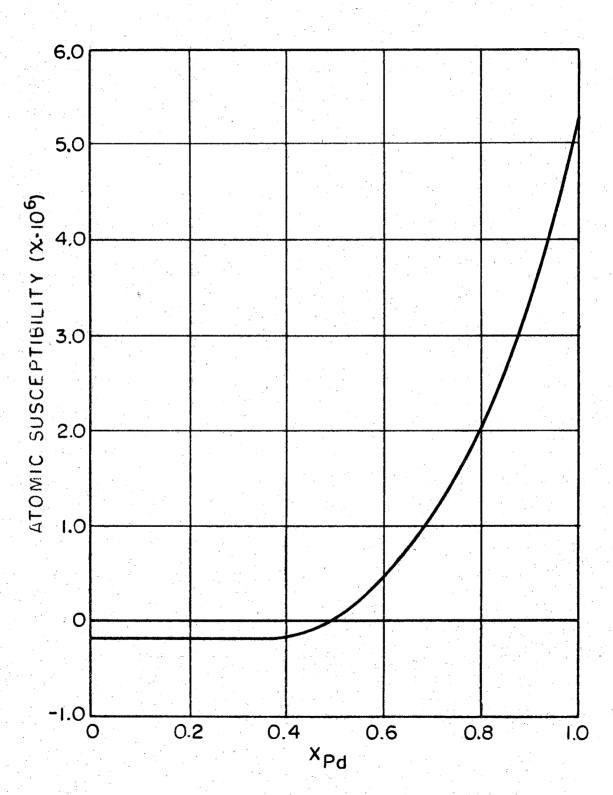
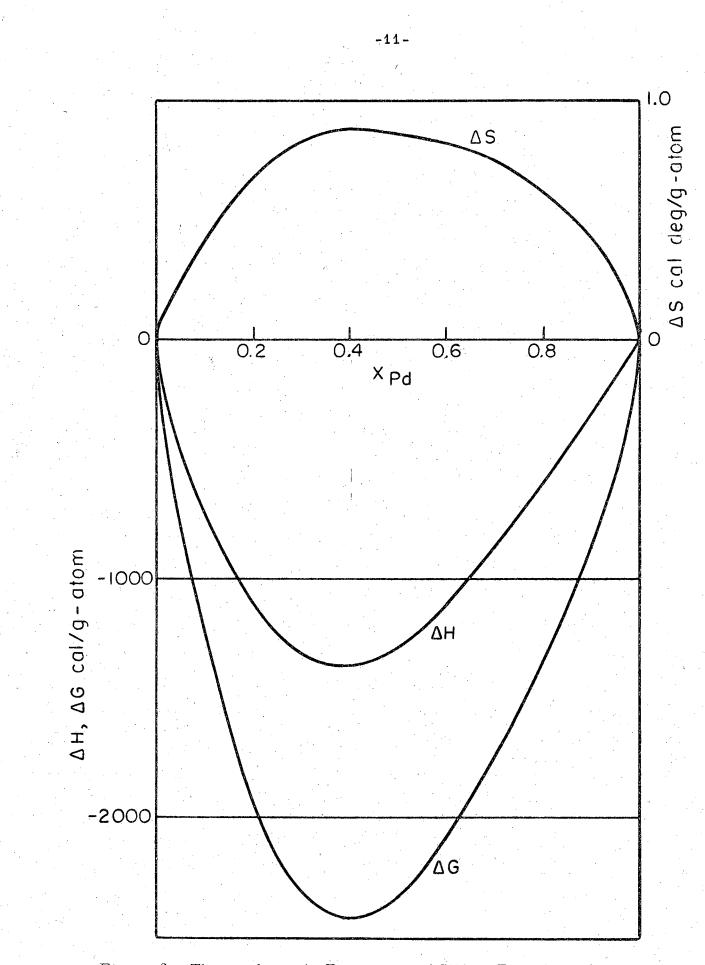


Figure 1. Magnetic Susceptibility versus Composition for Silver-Palladium Alloys at 20°C.

ç

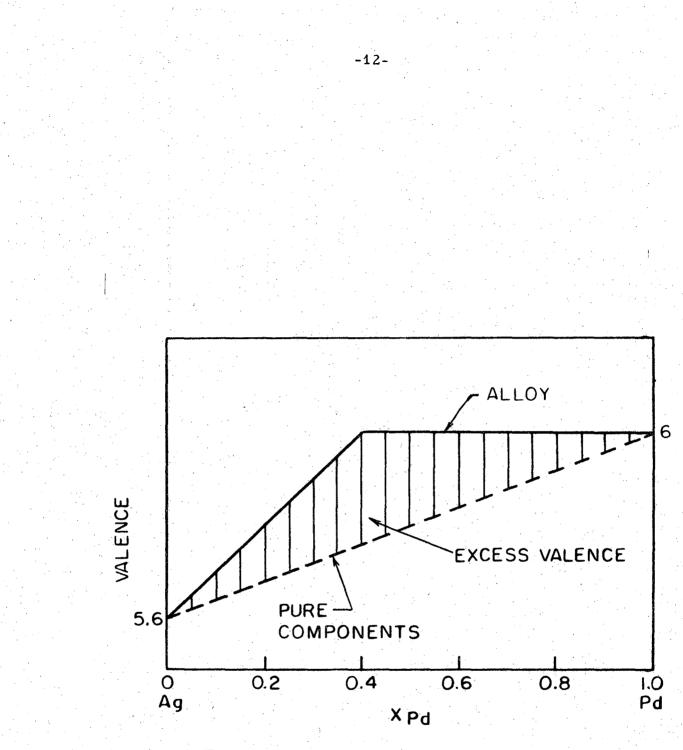
-10-

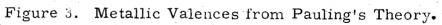


)

2

Figure 2. Thermodynamic Properties of Silver-Palladium Alloys at $1200^{\circ}K_{\bullet}$





£.

This report was prepared as an account of Government sponsored work. Neither the United States, nor the Commission, nor any person acting on behalf of the Commission:

- A. Makes any warranty or representation, expressed or implied, with respect to the accuracy, completeness, or usefulness of the information contained in this report, or that the use of any information, apparatus, method, or process disclosed in this report may not infringe privately owned rights; or
- B. Assumes any liabilities with respect to the use of, or for damages resulting from the use of any information, apparatus, method, or process disclosed in this report.

As used in the above, "person acting on behalf of the Commission" includes any employee or contractor of the Commission, or employee of such contractor, to the extent that such employee or contractor of the Commission, or employee of such contractor prepares, disseminates, or provides access to, any information pursuant to his employment or contract with the Commission, or his employment with such contractor.