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## Title

Interplay between Swelling Kinetics and Nanostructure in Perfluorosulfonic Acid Thin-Films: Role of Hygrothermal Aging

**Permalink** https://escholarship.org/uc/item/4r1005g0

**Journal** ACS Applied Polymer Materials, 1(4)

**ISSN** 2637-6105

### **Authors**

Tesfaye, Meron Kushner, Douglas I Kusoglu, Ahmet

Publication Date 2019-04-12

### DOI

10.1021/acsapm.9b00005

Peer reviewed

1	On the Interplay between Swelling Kinetics and Nanostructure in PFSA Thin-films:
2	Role of Hygrothermal Ageing
3	Meron Tesfaye <sup>†,‡</sup> , Douglas I. Kushner <sup>‡</sup> , Ahmet Kusoglu <sup>‡,*</sup>
4	<sup>†</sup> Chemical and Biomolecular Engineering, University of California Berkeley, Berkeley, CA,
5	94720
6	<sup>‡</sup> Energy Storage & Distributed Resources Division, Lawrence Berkeley National Laboratory,
7	Berkeley, CA, 94720
8	
9	*Author to whom correspondence should be addressed: <u>akusoglu@lbl.gov</u>
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### 11 Abstract

12 Impacts of processing, storage, and operation on thin-film perflurosulfonic-acid (PFSA) 13 ionomer coatings used in electrodes of electrochemical devices remains unestablished. In this work, alteration of structure-function relationship in ionomers is achieved via exposure to elevated 14 15 temperature and humidity (hygrothermal ageing). Findings reflect a strong inverse correlation 16 between ageing-induced ionomer thin-film domain orientation and water-transport kinetics evaluated from swelling. Impact of ageing is shown to be more pronounced on platinum, due to 17 interactions with PFSA, as evidenced by greater increase in nano-domain orientation parallel to 18 19 substrate accompanied by reduced water transport, in contrast to silicon support.

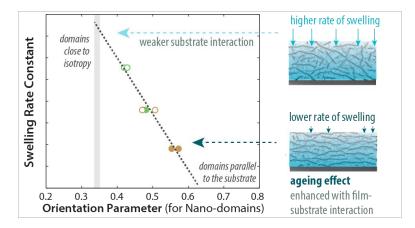
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### 21 Keywords:

ionomer thin-films, hygrothermal ageing, nanostructure, domain orientation, time constant, watertransport kinetics

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# 26 Table of Contents Graphic



28 Perfluorosulfonic-acid (PFSA) ionomers serve as benchmark polymer electrolytes for various clean-water and energy-conversion electrochemical technologies.<sup>1</sup> Integral to 29 its ion- and water-transport capabilities is PFSA's phase-separated structure, which is 30 31 enhanced upon hydration of its -SO<sub>3</sub>H end groups and maintained by the mechanical stability of its inert backbone matrix.<sup>1</sup> As a random copolymer lacking well-defined 32 structure, comprehension of PFSA's intrinsic structure-function relationship has heavily 33 dependent on processing conditions (i.e. pre-treatment, cation contamination, etc.), thermal 34 history (unmanned, annealed, hot-pressed, etc.), and external stimuli (i.e. humidity, solvent, 35 etc.).<sup>1,2</sup> However, engineering high-performance ionomers requires establishment of 36 fundamental relationship between morphology of PFSA and properties that dictate 37 performance.<sup>2</sup> This is even more consequential for PFSA thin-films (nanometers in 38 thickness), as local interfaces and interactions impose additional constraints that limit 39 critical properties, impeding electrode performance in energy-conversion devices.<sup>3,4</sup> 40 Various examples of confinement-driven limitations in thin-film ionomers include 41 decreased water uptake, diffusivity, ion conductivity, accompanied by an overall increase 42 in transport resistance and mechanical stiffness.<sup>1,3-7</sup> While these property changes have 43 been ascribed to substrate-dependent morphological changes, interfacial interactions, and 44 finite-size driven enthalpic and entropic strains, few have resulted in definitive predictive 45 correlations.<sup>5,8</sup> To that effect, this study aims to explicitly correlate substrate-dependent 46 morphology (nano-structural orientation) with water-sorption kinetics (swelling property) 47 by employing hygrothermal ageing. 48

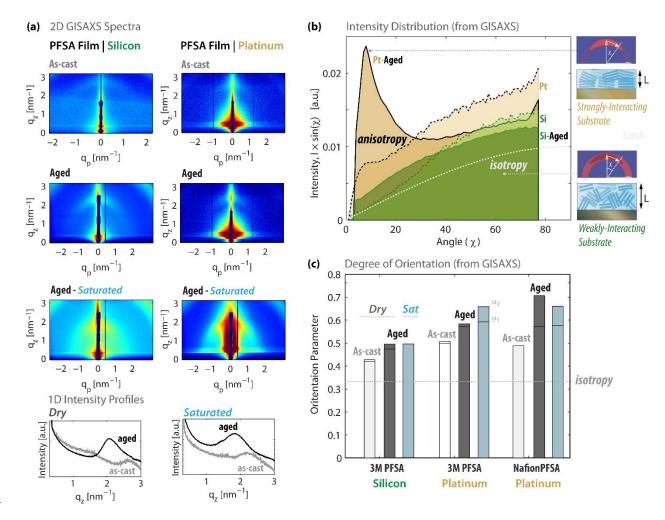
While some environmental conditions intrinsic to device operation (such as
hydration) are necessary to maintain PFSA functionality, they can also permanently alter

structure and functionality over time.<sup>9–11</sup> Notably, prolonged exposure of bulk PFSA to hot 51 and humid conditions (hygrothermal ageing) results in dramatic changes in ionomer 52 properties, altering ionomer's morphology, increasing mechanical properties and reducing 53 water uptake and conductivity.<sup>9-13</sup> Although no such explorations on the impact of ageing 54 in PFSA thin-films exists thus far, the technique of prolonged solvent exposure and 55 different processing conditions are frequently employed in block-copolymer thin-films to 56 access different morphologies.<sup>14–16</sup> In supported ionomer thin-films, similar to ionomers 57 present in electrodes, morphological changes cannot be decoupled from supporting 58 59 substrates. Recent findings have reported ionomer thin-film swelling property depends on the chemical composition of substrates (platinum/oxidized platinum (Pt/PtO<sub>x</sub>), gold (Au), 60 and carbon (C)), which may even be dictated during film formation, highlighting the key 61 role substrate interactions play in controlling film behavior.<sup>3,7,17</sup> Thin-films exhibit phase-62 separated domains under humidification, similar to a membrane, but with anisotropy driven 63 by substrate and confinement effects.<sup>3,18</sup> While thin-films on silicon substrates with native 64 oxide (Si/SiO<sub>x</sub>) exhibits close to isotropic patterns (random domain orientation), films on 65 hydrophobized Si or metallic substrates exhibit anisotropic domain orientation, which is 66 further enhanced for film thicknesses <50 nm.<sup>1,3</sup> This work demonstrates a direct link 67 between ionomer thin-film swelling, which serves as proxy for ion conductivity and water 68 transport, to substrate dependent nano-structural orientation as accessed by hygrothermal 69 70 ageing.

In this study, 30-40nm thin-films of PFSA with long- and medium-side-chain length (Nafion and 3M respectively) were spun-cast on Si/SiO<sub>x</sub> and Pt/PtO<sub>x</sub> substrates to mimic ionomers in electrochemical electrodes. The films were then hygrothermally aged for 2-4 74 days in an environmental chamber held at 85% relative humidity (RH) and a temperature of 70 °C. Structural changes caused by hygrothermal ageing are investigated via in-situ 75 grazing incidence small-angle X-ray scattering (GISAXS). Fig. 1 summarizes the GISAXS 76 patterns of unaged (as-cast) and aged PFSA thin-films on Si/SiOx and Pt/PtOx support. An 77 ionomer peak (at  $q_{\text{peak}} = 2$  to 2.5 nm<sup>-1</sup>) is apparent in all, and more pronounced in aged thin-78 films (Fig. 1a), signifying the phase-separated nanostructure with a correlation length of d 79  $= 2\pi/q_{\text{peak}} = 2.5$  to 3 nm. Upon hydration, the ionomer peak shifts to lower q (large d) due 80 to incorporation of water molecules into hydrophilic nano-domains. Interestingly, changes 81 82 in peak position as well as peak shape upon hydration are less distinct for aged films where the hydrophilic-domain network appears to already be expanded and set as a result of 83 ageing.<sup>3,18</sup> (See SI for further discussion). The high-intensity regions around the specular 84 peak in the GISAXS images (Fig. 1a) highlight the hygrothermal ageing-induced 85 anisotropic structure. To analyze structural anisotropy, the intensity distributions around 86 the ionomer peak position ( $q_{\text{peak}} = 2.5 \pm 0.25 \text{ nm}^{-1}$ ) are taken from the 2D GISAXS spectra 87 and plotted as a function of azimuthal angle,  $I(\chi)$  (Fig. 1b). Films on the Si/SiO<sub>x</sub> substrate 88 exhibit a low degree of anisotropy and closely follow the ideal isotropic distribution  $(I(\chi)$ 89 = constant). Hygrothermally-aged thin-films prepared on  $Pt/PtO_x$  exhibit an intensity 90 distribution, concentrated at  $q_p = 0$ , which indicates more domains are orientated parallel 91 92 to the surface. A domain orientation parameter, O.P., quantifies the anisotropy of such distributions:<sup>3</sup> 93

94 
$$O.P. = \left\langle \cos^2 \chi \right\rangle = \frac{\int_0^{\pi/2} I(\chi) \sin(\chi) \cos^2 \chi d\chi}{\int_0^{\pi/2} I(\chi) \sin(\chi) d\chi} \langle S \rangle_{\text{domain}} = \frac{1}{2} \langle 3\omega_{\text{ori}} - 1 \rangle$$
[1]

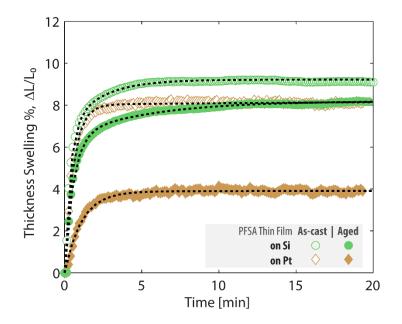
where O.P. is between 0 (domains perpendicular to the substrate) and 1 (domains parallel 95 to the substrate) and is 0.33 for an isotropic structure (random domain orientation). Fig. 1c 96 shows the O.P. for PFSA thin-films increases upon hygrothermal ageing. In addition, there 97 98 is a higher O.P. for aged long side-chain PFSA (Nafion over 3M), revealing the possible role of pendant chain chemistry in these governing interactions.<sup>18</sup> Fig. 1c shows increased 99 nanostructural orientation on Pt/PtOx compared to Si/SiOx, in both as-cast (unaged) and 100 aged state. Aged thin-films on Pt/PtO<sub>x</sub>, also showed greater O.P. distribution probably due 101 to the dynamic interactions between water and Pt/ionomer interface.<sup>3,17,19,20</sup> 102



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Figure 1: (a) 2D GISAXS images of as-cast (unaged) and hygrothermally aged PFSA (3M, 105 825g/mol equivalent weight) thin-films spin-cast on Si/SiOx and Pt/PtOx substrate in dry (33%) 106 and saturated (100% RH) conditions at 26 °C. 1D intensity profiles are shown below for the 3M 107 PFSA films on Si/SiO<sub>x</sub> substrate, based on the 2D images. (b) Intensity distribution as a function 108 of azimuthal angle,  $\chi$ , for the ionomer peak. The solid and dashed lines represent the aged and as-109 cast samples, respectively, and the dotted line shows the distribution calculated for an ideal 110 isotropic case. (c) Orientation parameter calculated from the distributions in (b). The low and high 111 bars correspond to low and high incidence angle. (See SI for details). 112

The effect of hygrothermal ageing on ionomer swelling extent and kinetics was captured by *in-situ* spectroscopic ellipsometry. Fig. 2 compares swelling of aged and ascast PFSA ionomer thin-films on Si/SiO<sub>x</sub> and Pt/PtO<sub>x</sub> substrates at 97% RH. In contrast to hygrothermally-aged films coated on Si/SiO<sub>x</sub>, which show a 12% reduction in swelling, aged films on Pt/PtO<sub>x</sub> exhibit a 50% reduction in extent of swelling normal to the substrate. For comparison, the same water-uptake reduction from reference (unaged) bulk PFSAs (~50 μm thick) would require 100 days of hygrothermal ageing;<sup>21</sup> this difference between
bulk and thin-film aging highlights the enhanced interplay of confinement and substrate on
thin-film PFSAs. The effect of ageing is amplified on strongly-interacting substrates like
Pt/PtOx,<sup>3,17,19,22</sup> which accelerates the ageing process.



### 123

Figure 2: Time-dependent fractional thickness swelling of PFSA (3M) thin-films (nominal thickness,  $L_0 \approx 34$  nm) spin cast on Si/SiO<sub>x</sub> and Pt/PtO<sub>x</sub> substrates during humidification from 0 to 97 % RH. Data are shown for as-cast (unaged) and aged (at 85% RH, 70 °C for 4 days) films. The dashed lines are best-fit to the measured data (symbols) using a two-term exponential expression, discussed later.

129 To quantify this explicitly, ageing-induced changes in thin-film nanostructure (Fig.

130 1) can be correlated with water-sorption dynamics by investigating swelling kinetics (Fig.

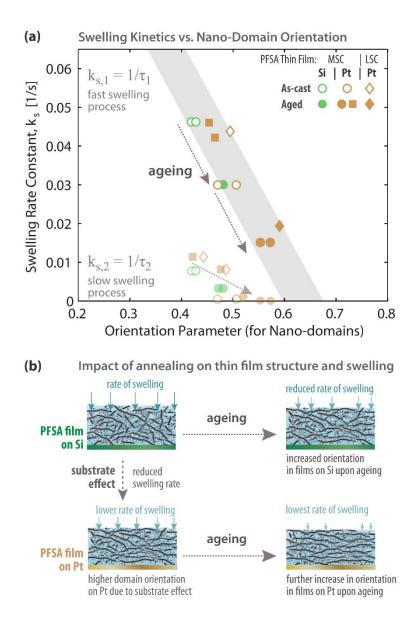
131 2). The time-dependent swelling process is analyzed by fitting the normalized transient

thickness change (Fig. 2) to a two-term exponential expression<sup>1</sup>:

133 
$$\frac{L(t)-L_0}{L_{\infty}-L_0} = 1 - \operatorname{Aexp}\left(-\frac{t}{\tau_1}\right) - (1-A)\exp\left(-\frac{t}{\tau_2}\right)$$
[2]

where  $\tau_1$  and  $\tau_2$  are time constants for the short-term (fast, diffusional transport) and longterm (slow, physical relaxation and interfacial transport) swelling processes,

respectively.<sup>1,23</sup> Fig. 3a shows the inverse time constant of water sorption kinetics,  $k_s = 1/\tau$ , 136 against O.P. Assuming one-dimensional swelling in thin-films, interfacial transport and 137 water diffusion can be estimated from characteristic time constants  $\tau_2$  (=L/k<sub>int</sub>) and  $\tau_1$ 138  $(=L^2/D_{water})$ , respectively.  $D_{water}$  in ionomer thin-films explored here ranges from  $10^{-12}$  cm<sup>2</sup>/s 139 to  $10^{-13}$  cm<sup>2</sup>/s; reduction in D<sub>water</sub> tracks with increasing O.P. and is on the same order of 140 magnitude as those reported by Eastman et. al.<sup>24</sup> Unaged films on Pt/PtO<sub>x</sub> possess strong 141 orientation and ageing increases structural parallel orientation on both supports, 142 presumably impacting through-plane water transport. Considering the primary governing 143 water transport mechanism to be interfacial transport,  $k_{int} = L/\tau_2$  (ranging from 10<sup>-6</sup> to 10<sup>-8</sup>) 144 cm/s) illustrates a different aspect of ageing. Change in O.P. is strongly coupled with 145 diffusional transport, which is accompanied by swelling at early times, while interfacial 146 147 relaxation becomes dominant at longer time-scales ( $\tau_1 > \tau_2$ ) and is dependent on O.P. to a lesser extent. For comparable ageing-induced change in O.P., thin-films on Pt/PtOx show 148 higher time constant relative to Si/SiO<sub>x</sub>, perhaps due to differences in substrate/ionomer 149 interaction compounded by physical ageing (relaxation) that has been shown to be 150 accelerated on metal supports.<sup>20,22,25</sup> However, the key takeaway is ageing-induced 151 morphological rearrangement (O.P.) is inversely correlated to characteristic rate for 152 swelling  $(k_s)$  representing water transport kinetics (or time constant is proportional to O.P., 153 154 i.e.,  $\tau \propto 1/k_s \propto O.P.$ ).



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Figure 3: (a) Swelling rate constants of as-cast (unaged) and aged 3M (■,●) and Nafion (♦) PFSA 156 thin-films during humidification to saturation (from ellipsometry) plotted against the orientation 157 parameter of the same film in dry condition (from GISAXS). All the films were annealed after 158 casting, except the unannealed 3M film on Pt (■), which is shown for comparison. Two rate 159 constants are plotted for each case, representing the fast and slow swelling processes. (From Eq. 160 161 (2)). (b) The schematics illustrate the inverse relationship between increasing domain orientation (parallel to the interfaces) and kinetics of swelling. The shaded region is shown as the guide for-162 the-eye. 163

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166 Ionomer hydration controls transport properties like ion conduction and gas diffusion. Tunable nanostructures that control hydration and transport pathways can be designed via 167 synthesis or processing techniques for well-ordered ionomers like sulfonated block or graft 168 copolymers.<sup>2</sup> A similar attempt is made here via hygrothermal ageing of disordered thin-169 film PFSAs, whose properties are subject to additional surface interaction effects that are 170 heightened with finite size. Previous studies have demonstrated that alignment of 171 hydrophilic moieties at the ionomer surface, governed by the interactions of the ionomer 172 with vapor or liquid, <sup>3,26–28</sup> is related to mass-transport of water.<sup>1,26,28</sup> Findings in this work 173 reflect similar stimulus(hygrothermal ageing)-driven, surface(Pt/PtOx)-controlled, and 174 chemistry (side-chain) influenced preferential orientation resulting in water transport 175 limitations in ionomer thin-films. A number of postulations are made here to explain this 176 177 direct correlation and reduction in ionomer water transport/swelling rate with increase in *O*.*P*.: 178

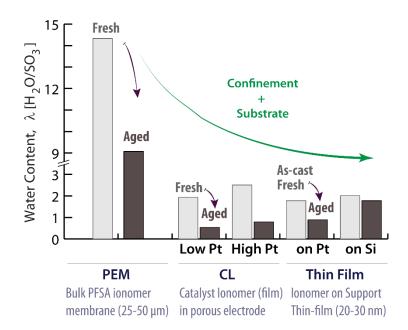
(1) Increasing backbone alignment parallel to the substrate selectively lowers throughplane water transport by lowering hydrophilic surface area and increasing tortuosity,
similar to the effect of structurally-induced orientation on ion transport.<sup>7,8</sup> Ageinginduced chain orientation increases the fraction of -SO<sub>3</sub><sup>-</sup> moieties near the substrate,
intensifies their pairwise interactions, which could further enhance orientation, and
reduces swelling and swelling kinetics perpendicular to the substrate.

(2) Coupled with previous postulate on kinetics, hygrothermal ageing induces order in
 the nanostructure, enabling the ionomer to access different quasi-equilibrium
 states.<sup>29</sup> This is in similar vein to processing induced alignments in di-block
 polymers.<sup>2,16,30</sup> Such a structure could preserve substrate/film interactions that

perpetuate and stabilize the chain configurations that could alter swelling governed by thermodynamic equilibrium. Comparative ageing and morphological studies in other solvents could elucidate the role of solvent and reversibility in achieving alternative structures to decouple the kinetic and thermodynamic effects.

(3) Distinct from the other two possible explanations is sulfonic anhydride formation,
similar to that occurring in bulk PFSA.<sup>9,10,21</sup> Anhydride formation increases
equivalent weight via physical crosslinking of ionomer chains and reduces water
uptake. This explanation would imply that enhanced order and interaction on
Pt/PtO<sub>x</sub> accelerates anhydride formation in ionomer over Si/SiO<sub>x</sub> support.
Spectroscopic studies on aged thin-films could provide insight into presence of other
forms of degradation and anhydride formation.

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Figure 4: Comparison of maximum ionomer water content (at 95-97%RH) in bulk membrane,<sup>10</sup> catalyst layer (CL), and thin-films on a support (PFSA for PEM is Nafion, while for CL and thin-

films 3M PFSA is used. See SI for full description of materials and treatment conditions).

206 These results on thin-film swelling are consistent with our hydration measurements on the unaged and aged catalyst layers (CLs) of varying Pt loading (see SI); higher Pt loading 207 reflects significant reduction in water uptake with hygrothermal ageing compared to low-208 loaded CL (Fig. 4). Thus, findings herein shed light into the ionomer structure-related 209 origins of the performance limitations in fuel-cell electrodes that may occur during 210 operation, and how they might be influenced by the dynamic substrate-ionomer 211 interactions. It must be noted that correlating ionomer's structure-swelling changes to 212 performance is not trivial due to variations in the Pt/ionomer interface such as surface 213 214 roughness (Figure S2 in Supporting Information) and substrate composition along with environmental effects (reducing or oxidizing), which were shown to consequently alter 215 thin-film swelling.<sup>17</sup> 216

This letter reports compelling evidence for the existence of an explicit correlation 217 between the orientation of nano-domains and swelling kinetics in PFSA ionomer thin-films, 218 which is evidenced by ageing-induced changes in the film's nanostructure and swelling. 219 The higher the orientation parallel to film surface, the slower the transport normal to 220 surface. Stronger interactions between ionomer side-chains and Pt/PtOx enhances the 221 222 impact of hygrothermal ageing by preferentially aligning more of the backbone parallel to the substrate. Domain alignment and re-alignment caused by environment (e.g., 223 hygrothermal ageing and saturation) and ionomer/substrate interactions dictates swelling 224 225 behavior and water transport in thin-films. Notably, the temperature used in this study for ageing is much less than the initial annealing temperatures (70 °C vs. 140-160 °C), which 226 227 underscores the role of water as an important stressor driving the film towards structural 228 reorganization that influence material properties. Results presented herein provide new

insights into the impact of environmental conditioning on inducing morphological changes
in ionomer thin-films, a phenomenon of great importance for elucidating and controlling
their structure/function relationship and transport response at dynamic interfaces in energy
conversion devices.

#### 234 Acknowledgements

235 Authors would like to thank William Tong, Peter Dudenas, Julie Fornaciari, and Andrew 236 Crothers for helpful discussions, insights and supplemental work. We thank Rodney Borup 237 of LANL for discussions on ageing procedures and help with CL preparations, and Steve Hamrock and Mike Yandrasits from 3M for providing ionomer dispersions. AK 238 239 acknowledges support from the Fuel Cell Technologies Office, Energy Efficiency and 240 Renewable Energy Office, of the U.S. Department of Energy (DOE), under contract no. DE-AC02-05CH11231. Funding support for MT was provided by University of California 241 242 Chancellors Fellowship. We thank Chenhui Zhu, Polite Stewart and Eric Schaible for their assistance with facilitating the equipment at the Advanced Light Source (ALS) beamline 243 7.3.3, supported by the Office of Science, Office of Basic Energy Sciences, of the U.S. 244 DOE (Contract No. DE-AC02-05CH11231). This work made use of facilities at the Joint 245 Center for Artificial Photosynthesis supported through the Office of Science of the U.S. 246 DOE under Award Number DE-SC0004993<sup>†</sup>. 247

# **Supporting Information**:

Associated Content: Full experimental methods are provided in detail in Supporting Information.

#### 252 **Reference**

253	(1)	Kusoglu, A.; Weber, A. Z. New Insights into Perfluorinated Sulfonic-Acid Ionomers.
254		<i>Chem. Rev.</i> <b>2017</b> , <i>117</i> , 987–1104.

- 255 (2) Park, M. J.; Hong, J.; Kim, S. Y. Role of Nanostructures in Polymer Electrolytes for
- Energy Storage and Delivery. In ACS Symposium Series 1096; Polymers for Energy
- 257 Storage and Delivery: Polyelectrolytes for Batteries and Fuel Cells; 2012; pp 129–146.
- (3) Kusoglu, A. Ionomer Thin Films in PEM Fuel Cells. In *Encyclopedia of Sustainability Science and Technology*; Meyers, R., Ed.; Springer, New York, NY, 2018.
- 260 (4) Kongkanand, A.; Gu, W.; Mathias, M. F. Proton-Exchange Membrane Fuel Cells with
- 261 Low-Pt Content. In *Encyclopedia of Sustainability Science and Technology*; 2018.
- (5) Tesfaye, M.; Kushner, D. I.; McCloskey, B. D.; Weber, A. Z.; Kusoglu, A. Thermal
   Transitions in Perfluorosulfonated Ionomer Thin-Films. *ACS Macro Lett.* 2018, 1237–
- **264** 1242.
- 265 (6) DeCaluwe, S. C.; Baker, A. M.; Bhargava, P.; Fischer, J. E.; Dura, J. A. Structure-
- Property Relationships at Nafion Thin-Film Interfaces: Thickness Effects on Hydration
  and Anisotropic Ion Transport. *Nano Energy* 2018, 46, 91–100.
- 268 (7) Ono, Y.; Nagao, Y. Interfacial Structure and Proton Conductivity of Nafion at the Pt269 Deposited Surface. *Langmuir* 2016, *32*, 352–358.
- 270 (8) Nagao, Y. Proton-Conductivity Enhancement in Polymer Thin Films. *Langmuir* 2017, *33*,
  271 12547–12558.
- 272 (9) Naudy, S.; Collette, F.; Thominette, F.; Gebel, G.; Espuche, E. Influence of Hygrothermal

- Aging on the Gas and Water Transport Properties of Nafions Membranes. *J. Memb. Sci.*274 2014, 451, 293–304.
- (10) Shi, S.; Dursch, T. J.; Blake, C.; Mukundan, R.; Borup, R. L.; Weber, A. Z.; Kusoglu, A.
  Impact of Hygrothermal Aging on Structure/Function Relationship of PerfluorosulfonicAcid Membrane. *J. Polym. Sci. Part B Polym. Phys.* 2016, *54*, 570–581.
- (11) Shi, S.; Chen, G.; Wang, Z.; Chen, X. Mechanical Properties of Nafion 212 Proton
  Exchange Membrane Subjected to Hygrothermal Aging. *J. Power Sources* 2013, *238*,
  318–323.
- (12) Onishi, L. M.; Prausnitz, J. M.; Newman, J. Water Nafion Equilibria. Absence of
  Schroeder's Paradox. 2007, 10166–10173.
- (13) Hensley, J. E.; Way, J. D.; Dec, S. F.; Abney, K. D. The Effects of Thermal Annealing on
  Commercial Nafion Membranes. *J. Memb. Sci.* 2007, 298, 190–201.
- (14) Stewart-Sloan, C. R.; Olsen, B. D. Protonation-Induced Microphase Separation in Thin
  Films of a Polyelectrolyte-Hydrophilic Diblock Copolymer. *ACS Macro Lett.* 2014, *3*,
  410–414.
- 288 (15) Gu, X.; Gunkel, I.; Hexemer, A.; Russell, T. P. Controlling Domain Spacing and Grain
- Size in Cylindrical Block Copolymer Thin Films by Means of Thermal and Solvent Vapor
  Annealing. *Macromolecules* 2016, *49*, 3373–3381.
- (16) Sinturel, C.; Vayer, M.; Morris, M.; Hillmyer, M. A. Solvent Vapor Annealing of Block
  Polymer Thin Films. *Macromolecules* 2013, *46*, 5399–5415.
- 293 (17) Tesfaye, M.; MacDonald, A. N.; Dudenas, P. J.; Kusoglu, A.; Weber, A. Z. Exploring

294		Substrate/Ionomer Interaction under Oxidizing and Reducing Environments. <i>Electrochem</i> .
295		<i>commun.</i> <b>2018</b> , <i>87</i> , 86–90.
296	(18)	Kusoglu, A.; Dursch, T. J.; Weber, A. Z. Nanostructure/Swelling Relationships of Bulk
297		and Thin-Film PFSA Ionomers. Adv. Funct. Mater. 2016, 26, 4961–4975.
298	(19)	Yagi, I.; Inokuma, K.; Kimijima, K.; Notsu, H. Molecular Structure of Buried
299		Perfluorosulfonated Ionomer/Pt Interface Probed by Vibrational Sum Frequency
300		Generation Spectroscopy. J. Phys. Chem. C 2014, 118, 26182–26190.
301	(20)	Kodama, K.; Motobayashi, K.; Shinohara, A.; Hasegawa, N.; Kudo, K.; Jinnouchi, R.;
302		Osawa, M.; Morimoto, Y. Effect of the Side-Chain Structure of Perfluoro-Sulfonic Acid
303		Ionomers on the Oxygen Reduction Reaction on the Surface of Pt. ACS Catal. 2018, 8,
304		694–700.
305	(21)	Collette, F. M.; Lorentz, C.; Gebel, G.; Thominette, F. Hygrothermal Aging of Nafion. J.
306		Memb. Sci. 2009, 330, 21–29.
307	(22)	Selvan, M. E.; He, Q.; Calvo-Munoz, E. M.; Keffer, D. J. Molecular Dynamic Simulations
308		of the Effect on the Hydration of Nafion in the Presence of a Platinum Nanoparticle. J.
309		Phys. Chem. C 2012, 116, 12890–12899.
310	(23)	Kongkanand, A. Interfacial Water Transport Measurements in Nafion Thin Films Using a
311		Quartz-Crystal Microbalance. J. Phys. Chem. C 2011, 115, 11318–11325.
312	(24)	Eastman, S. A.; Kim, S.; Page, K. A.; Rowe, B. W.; Kang, S.; Soles, C. L.; Yager, K. G.
313		Effect of Confinement on Structure, Water Solubility, and Water Transport in Nafion Thin
314		Films. <i>Macromolecules</i> <b>2012</b> , <i>45</i> , 7920–7930.

315	(25)	Kushner, D. I.; Hickner, M. A. Substrate-Dependent Physical Aging of Confined Nafion
316		Thin Films. ACS Macro Lett. 2018, 7, 223–227.
317	(26)	Novitski, D.; Holdcroft, S. Determination of O2 Mass Transport at the Pt   PFSA Ionomer
318		Interface under Reduced Relative Humidity. ACS Appl. Mater. Interfaces 2015, 7, 27314-
319		27323.
320	(27)	Hwang, G. S.; Parkinson, D. Y.; Kusoglu, A.; Macdowell, A. A.; Weber, A. Z.
321		Understanding Water Uptake and Transport in Nafion Using X-ray Microtomography.
322		ACS Macro Lett. 2013, 2, 288–291.
323	(28)	He, Q.; Kusoglu, A.; Lucas, I. T.; Clark, K.; Weber, A. Z.; Kostecki, R. Correlating
324		Humidity-Dependent Ionically Conductive Surface Area with Transport Phenomena in
325		Proton-Exchange Membranes. J. Phys. Chem. B 2011, 115, 11650–11657.
326	(29)	Kim, T. H.; Yi, J. Y.; Jung, C. Y.; Jeong, E.; Yi, S. C. Solvent Effect on the Nafion
327		Agglomerate Morphology in the Catalyst Layer of the Proton Exchange Membrane Fuel
328		Cells. Int. J. Hydrogen Energy 2017, 42, 478–485.
329	(30)	Kim, J.; Kim, B.; Jung, B.; Kang, Y. S.; Ha, H. Y.; Oh, I. H.; Ihn, K. J. Effect of Casting
330		Solvent on Morphology and Physical Properties of Partially Sulfonated Polystyrene-
331		Block-Poly(Ethylene-Ran-Butylene)-Block-Polystyrene Copolymers. Macromol. Rapid
332		<i>Commun.</i> <b>2002</b> , <i>23</i> , 753–756.