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# In situ X-ray and IR probes relevant to Earth science at the Advanced Light Source at Lawrence Berkeley Laboratory

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## Abstract

Access to synchrotron X-ray facilities has become an important aspect for many disciplines in experimental Earth science. This is especially important for studies that rely on probing samples in situ under natural conditions different from the ones found at the surface of the Earth. The non-ambient condition Earth science program at the Advanced Light Source (ALS), Lawrence Berkeley National Laboratory, offers a variety of tools utilizing the infra-red and hard X-ray spectrum that allow Earth scientists to probe Earth and environmental materials at variable conditions of pressure, stress, temperature, atmospheric composition, and humidity. These facilities are important tools for the user community in that they offer not only considerable capacity (non-ambient condition diffraction) but also complementary (IR spectroscopy, microtomography), and in some cases unique (Laue microdiffraction) instruments. The availability of the ALS' in situ probes to the Earth science community grows especially critical during the ongoing dark time of the Advanced Photon Source in Chicago, which massively reduces available in situ synchrotron user time in North America.

**Keywords** Synchrotron · In situ X-ray diffraction · In situ IR spectroscopy · In situ X-ray tomography

## Introduction

Earth scientists have been at the forefront of utilizing synchrotron radiation for their research from the earliest days of user-facility synchrotron light sources (Calas et al. 1984; Häusermann 1992; Hemley et al. 1998; Sutton et al. 1988; Yagi 1988). This community became pioneering users and drivers of synchrotron development due to their collective need for analysis techniques that could resolve very small sample volumes with high brightness, a wide wavelength spectrum, and a high photon flux. Over time, the number

and complexity of synchrotron techniques that found applications in Earth and environmental sciences has expanded to include X-ray absorption, emission and resonance spectroscopy, as well as sophisticated time-resolved scattering and diffraction techniques (Anzellini et al. 2013; Cnudde and Boone 2013; Fenter et al. 2018; Kuppenko et al. 2012; Lavina et al. 2014; Quartieri 2015). One common thread in these efforts, regardless of the technique applied, was and remains the frequent need to combine the X-ray probe with conditions that are different to those on the Earth's surface. Specifically, many geologic research questions require high pressures and stress/strain (to simulate the planet's interior), extreme temperatures (planet's interior, hydrothermal environments, volcanology, surface of extraterrestrial planetary bodies), variable atmospheric compositions and humidity levels (environmental science, hydrology), and combinations of these. Ancillary equipment that can contain a sample under controlled non-ambient conditions must be transparent to the electromagnetic radiation acting as a probe; light in the infrared (IR) and hard X-ray ranges tends to be the most amenable. It is mainly for this reason that the bulk of

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in situ synchrotron geoscience research relies on techniques that employ IR and/or X-rays with energies above ~5 keV.

The US Department of Energy operates synchrotron user facilities that offer a variety of instruments to the Earth science community at Brookhaven National Lab (National Synchrotron Light Source, NSLS-II), Argonne National Lab (Advanced Photon Source, APS), SLAC National Accelerator Lab (Linac Coherent Light Source, LCLS and Stanford Synchrotron Radiation Lightsource, SSRL) and Lawrence Berkeley Lab (Advanced Light Source, ALS); the Cornell High Energy Synchrotron Source (CHESS) is operated by the National Science Foundation. Recently, Bommannavar et al. (2022), Sutton et al. (2022) and Zhang et al. (2022) nicely summarized capabilities at the APS that are helping to confront many of the current pressing problems in Earth science, and Laasch et al. (2022) documented capabilities at NSLS-II relevant to Earth science research.

The APS of Argonne National Lab and the ALS at Lawrence Berkeley Lab are both scheduled to undergo major accelerator upgrades that will transform the aging electron storage rings towards an ‘ultimate storage ring’, capable of emitting nearly fully coherent X-rays at their respective energies. The APS was first to undergo a 1–2 year long dark time to implement this upgrade, beginning in the spring of 2023. This leaves a temporary but significant gap in domestic synchrotron capabilities in the US. While many of Lawrence Berkeley Lab’s ALS Earth and environmental science-oriented beamlines are distinct from the APS suite, the ALS also hosts some instruments that offer similar capabilities. The purpose of this paper is to introduce (or remind) the Earth science community of the synchrotron IR and hard X-ray capabilities at the ALS in Berkeley. Experiments under non-ambient conditions are conducted on its IR beamlines (1.4, 2.4 and 5.4) and on three of its hard X-ray super-bend beamlines, namely 8.3.2 (X-ray micro-tomography), 12.2.2 (X-ray diffraction under non-ambient conditions) and 12.3.2 (Laue X-ray microdiffraction). Here we present an overview of the in situ non-ambient condition experimental capabilities that these beamlines offer to the Earth science user community, with a special emphasis on aspects not previously published elsewhere.

## Infrared spectroscopy: beamlines 1.4, 2.4 and 5.4

Although the ALS is optimized to produce X-rays, synchrotron bend magnet radiation is broadband and continues to be a bright source into the infrared region. This non-ionizing infrared light can be exploited to probe low-energy electronic excitations and molecular vibrations in a variety of materials using Fourier-transform infrared (FTIR) spectroscopy. The ALS has three IR beamlines, namely Beamlines

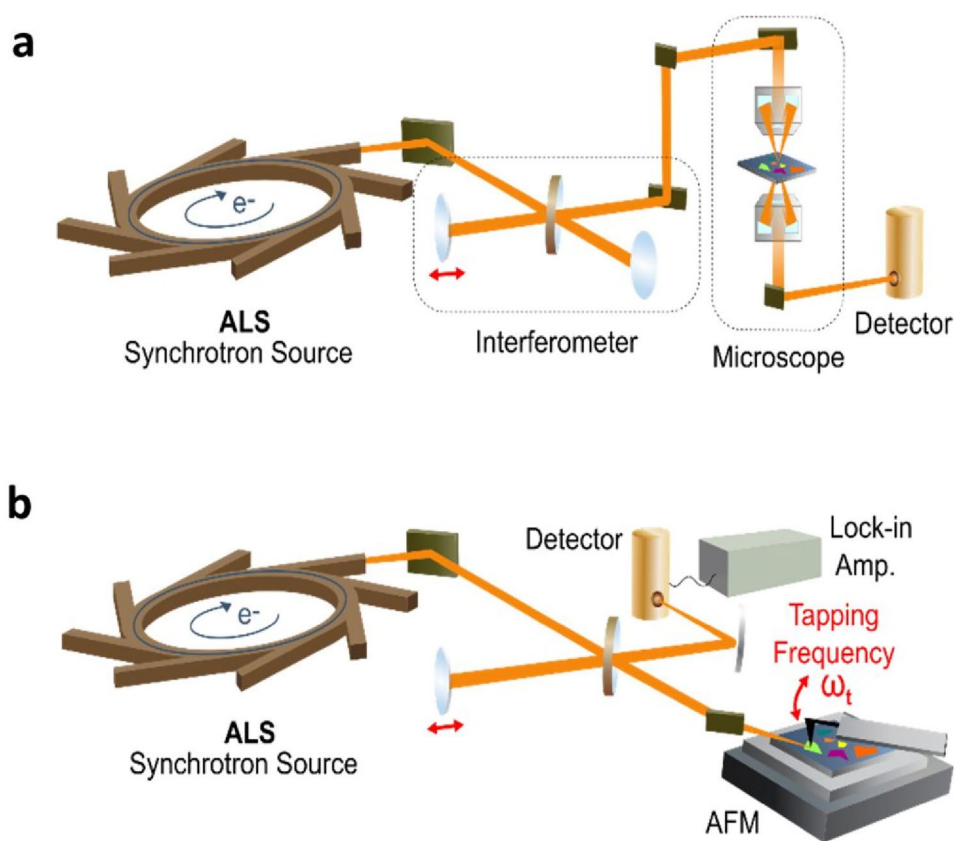
1.4, 2.4, and 5.4, that take advantage of the source characteristics to perform IR micro- and nano-spectroscopy experiments on small and complex heterogeneous materials, with applications ranging from the deep earth to deep space and everywhere in between.

In an IR micro-spectroscopy experiment (Fig. 1a), broadband synchrotron IR light is first passed through a symmetric Michelson interferometer and is then focused onto a sample with reflective Schwarzschild objectives. The diffraction-limited spot size is approximately 2–10  $\mu\text{m}$  for typical mid-infrared wavelengths (Dumas et al. 2020). After interacting with the sample, the reflected or transmitted IR light is detected by a suitable infrared detector (typically an HgCdTe detector) and the resulting interference signal is Fourier transformed to obtain an IR spectrum. Samples are routinely raster scanned to obtain 3D hyperspectral images spanning the IR spectral range. A choice of infrared objectives allows variable working distances up to 8–15 mm. This facilitates the use of a variety of devices that enable sample analysis over a wide range of ambient and non-ambient conditions, including microfluidic cells for studying biomolecules and organisms in hydrated environments (Hazen et al. 2010; Holman et al. 2010; Loutherbak et al. 2016); diamond anvil cells (DACs) and inclusions in naturally occurring diamonds for investigating materials under high pressure (Chen et al. 2020; Nisir et al. 2020; Tschauner et al. 2018, 2021); cryostats for low and high temperature measurements; as well as soils (Bouskill et al. 2016; Keiluweit et al. 2015) and extraterrestrial materials (Ishii et al. 2018; Sandford et al. 2006; Westphal et al. 2014).

Figure 2 illustrates using IR spectroscopy to investigate subsurface biofilms obtained from cold, sulfidic springs at the Sippenauer Moor (Probst et al. 2013). Usually archaea are found in moderate and cold environments as minority members of a much larger microbial community. Archaea in biofilms from the Sippenauer Moor, however, are the dominant species in the symbiotic “string of pearls” community, in which SM1 Euryarchaeon fill the “pearls” and the filamentous bacteria cover the pearl surfaces and form strings between them. Although the two types of microbes were assumed to be syntrophic, the biochemical details were unknown. Here, IR imaging was able to map the locations of the bacteria and archaea and identify the chemical relationships between them, showing unambiguously the sulfate-reducing metabolic activity of the bacteria and lack of such activity in the archaeal cells.

In a synchrotron IR nano-spectroscopy experiment (Fig. 1b), broadband synchrotron IR light is coupled into an asymmetric Michelson interferometer, where half of the light is focused onto an oscillating atomic force microscope (AFM) tip operating in close proximity to the sample surface (Bechtel et al. 2020). The resulting scattered light is collected and recombined with the other half of the

**Fig. 1** **a** Schematic of an IR microscopy experiment in transmission geometry. **b** Schematic of an IR nano-spectroscopy measurement



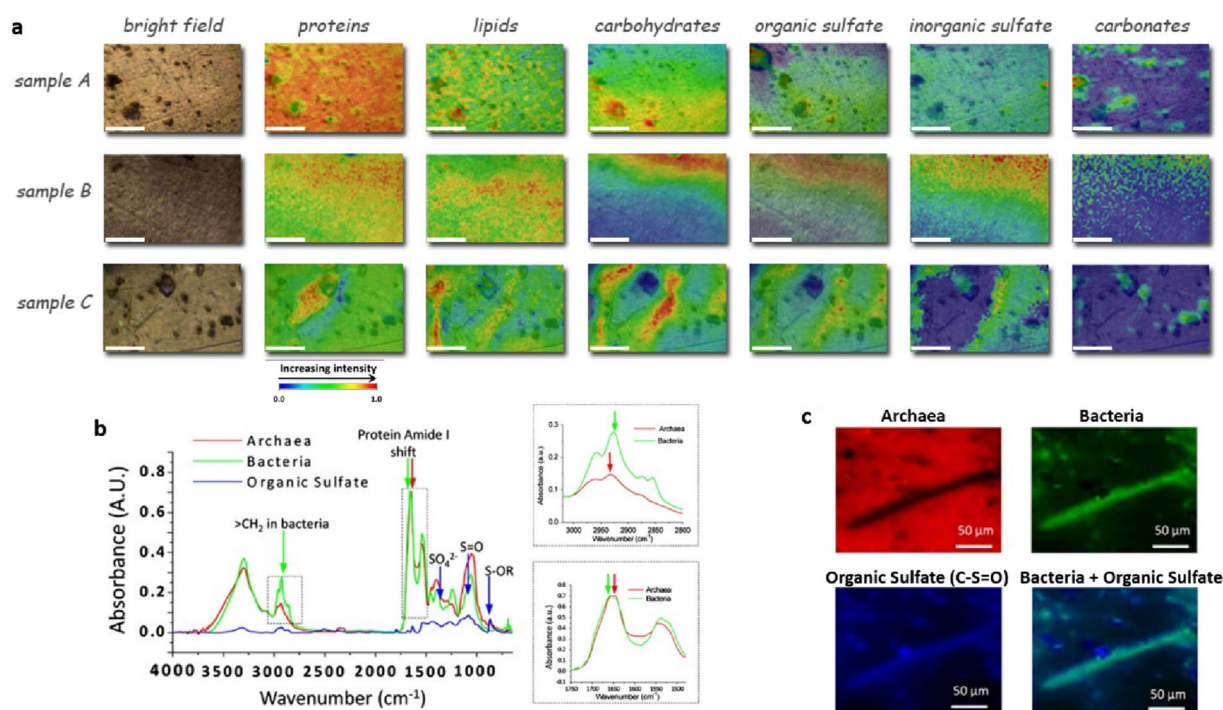
input light, which has been reflected off a moving mirror. The resulting interference between the tip-scattered light and the reference light is measured with an IR detector and demodulated at harmonics of the tip-tapping frequency ( $\omega_t$ ) to remove background scattering before being Fourier-transformed to yield an IR spectrum with information about the real (reflection) and imaginary (absorption) components of the dielectric constant of the sample. In this case, the AFM tip localizes the electric field and provides a tip-limited spatial resolution of <20 nm, enabling chemical imaging at the nanoscale, nearly three orders of magnitude smaller than the diffraction limit. Sample requirements for IR nano-spectroscopy are stricter than for IR micro-spectroscopy measurements, with a major requirement being that the samples are sufficiently flat for AFM measurements. Sample height variations of <500 nm are preferred, but up to 2  $\mu\text{m}$  height variations can be accommodated. Nanoscale heterogeneities can be explored with this technique, as recently demonstrated in naturally occurring shale (Hao et al. 2018); calcium carbonate polymorphs (Bechtel et al. 2014); and meteorite mineralogy (Young et al. 2022). The requirement for the AFM tip to be in close proximity to the sample limits the types of non-ambient conditions that can be accessed using this technique, although variable temperature measurements from room temperature up to 375 K are possible, as well

as limited measurements in liquids with liquid cell designs (Lu et al. 2019; Zhao et al. 2022).

Figure 3 shows synchrotron IR nano-spectroscopy measurements of a small piece of shale (Hao et al. 2018). Shale is the most common sedimentary rock and is composed of mineral particles mixed with variable quantities of organic matter. This mixture forms a nanoporous network that can trap and store fluids, and affects a variety of applications, including hydraulic fracturing, enhanced geothermal heating, carbon sequestration, and water storage. As shown in the figure, the nano-spectroscopy measurements reveal the heterogeneity of the shale and are able to identify different mineral components and organic-rich areas at the nanoscale. These results have been combined with lower spatial resolution IR micro-spectroscopy measurements covering larger fields of view to perform cross-scale molecular analyses that reveal greater detail across nanopore and micrograin scales.

### X-ray micro-tomography: beamline 8.3.2

Beamline 8.3.2 was recommissioned with a new endstation in 2007. It is situated on a superconducting bend magnet and the electron source (of size  $94 \times 17 \mu\text{m}$  FWHM) sees a field of 4.37 T, yielding a critical energy of 11.5 keV. The source-to-sample distance is 20 m; with no focusing optics on the



**Fig. 2** Synchrotron IR micro-spectroscopy images and spectra of bacteria in an archaea-dominated biofilm. **a** Heat map from univariate analysis showing the distribution of key biomolecules and sulfur/carbon biogeochemical cycling products in three different biofilms (A, B, C) from a sulfidic spring. Scale bars = 50  $\mu\text{m}$ . **b** Spectra of three components extracted from multivariate analysis: archaea (red), bacteria (green), and organic sulfate features (blue), with arrows pinpointing the spectral markers used in the analysis. Panel insets are zoomed in to highlight the spectral region of lipids and proteins,

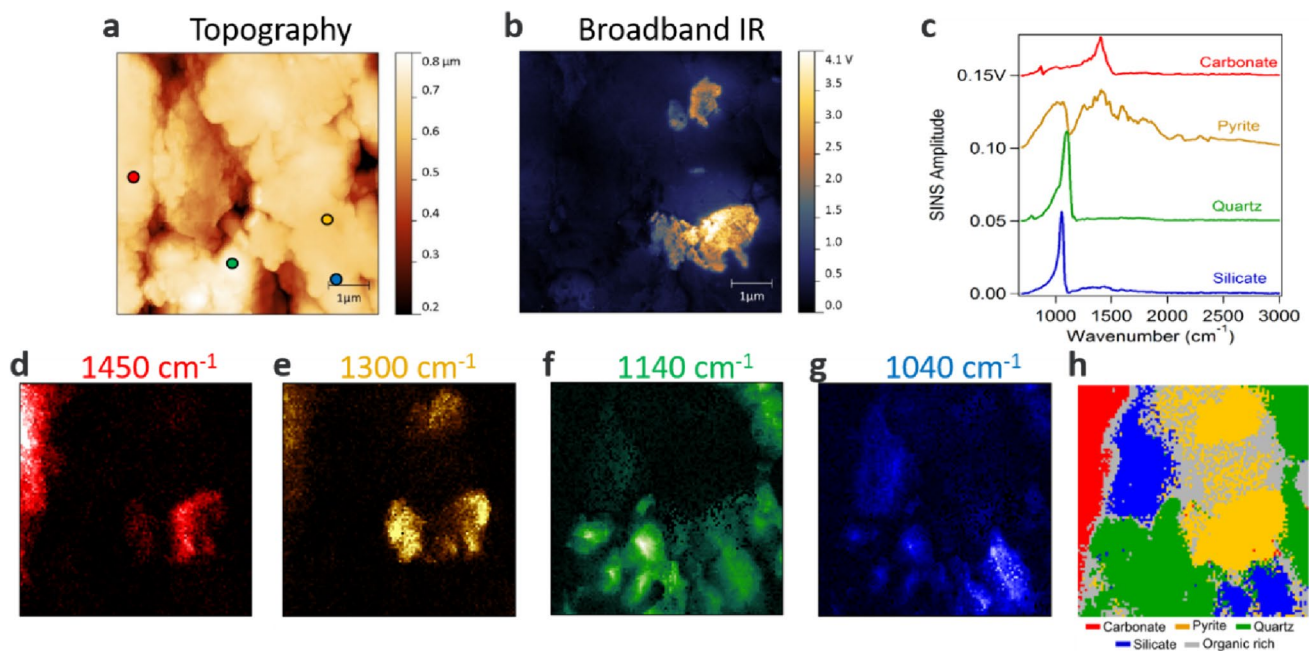
which are an important region for the distinction of bacteria and archaea because of their different membrane and protein composition. **c** Relative concentration images of archaea, bacteria, and organic sulfate (C-S=O) recovered by multivariate analysis. Merging the relative bacterial concentration image (green) with the organic sulfate distribution map (blue) reveals the co-localization of bacteria and organic sulfate. Scale bars = 50  $\mu\text{m}$ . Figure adapted from Probst et al. (2013)

beamline the beam size at the sample is  $\sim 40 \text{ mm} \times 4.6 \text{ mm}$  ( $2 \times 0.23 \text{ mrad}$  @ 20 m). The beamline can currently be operated in either white light mode, in which no optics are used in the beam, or in monochromatic mode, in which a multilayer monochromator (W/B<sub>4</sub>C,  $d = 2 \text{ nm}$ , 1% bandpass) is used over an energy range of approximately 8–40 keV. For many Earth science experiments, white light is used both because it allows higher flux and thus higher time resolution, and because it allows access to X-ray energies higher than the range of the monochromator, which are often necessary to penetrate geologic samples as well as associated sample environments.

The sample is mounted on an air-bearing rotation stage that in turn is mounted over a hole in the optical support table. This arrangement allows the sample mounting system to be translated over a long distance, allowing tall samples up to 60 cm in size to be imaged. Detection is performed using thin scintillators (i.e., 50  $\mu\text{m}$  thick LuAG:Ce), imaging these with optical lenses onto a CMOS detector (currently a PCO.edge). The detection system has multiple resolution and configuration options. An Optique Peter lens system is used with lenses in-line

when imaging in monochromatic mode. Home-built lens systems with lenses out of the X-ray path are employed in white light mode. Lenses span 1 $\times$  to 20 $\times$ , corresponding to pixel sizes of 6.5 to 0.325  $\mu\text{m}$ .

A tomographic scan is carried out by rotating a sample through 180° and collecting hundreds to thousands of images. Exposure times are typically a few milliseconds in white light mode, and a few hundred milliseconds in monochromatic mode. Scan times for a full tomographic scan of one field of view thus range from under a minute to 10 min, depending on the requirements of collection time, spatial resolution, and image quality. An advanced data pipeline has been developed, whereby a data orchestration system automatically moves data as it is collected to the National Energy Research Science Computing Center (NERSC), the high-performance computing system at Berkeley Lab. Users have access to a web-based system based on Jupyter notebooks for data processing, which also runs at NERSC, with templates and processing routines that allow them to run tomographic reconstruction using TomoPy, ASTRA, and SVMBIR. Users can download raw and processed data from NERSC using globus.org or other tools available at NERSC.



**Fig. 3** Synchrotron IR nano-spectroscopy measurements of shale. Simultaneously collected **a** topography and **b** broadband IR amplitude images revealing contrast between the various components of the shale, with highly reflecting components (metals) appearing brighter. **c** IR nano-spectroscopy measurements of the different components in

shale, identified by the dots in panel **a**. **d–g** IR heat maps of a single frequency, representative of the different colored components in **c**. **h** principle component analysis clustering of the different shale components based on the entire spectral measurements. Adapted from Hao et al. (2018)

The sample environments available to users at 8.3.2 include capabilities for heating, cooling, and mechanical testing. There are also sample environments developed by users from the LBNL Earth and Environmental Sciences Area that allow for flowing gas and liquids through samples that are at pressure. Two vessels have been built for use at slightly elevated (20–80 °C) and high-temperature experiments (up to 400 °C). Both vessels are capable of triaxial stress states with confining pressures up to 25 MPa and axial stress up to 40 MPa. The lower temperature vessel is optimized for studying multiphase (Garing et al. 2017a, b; Zuo et al. 2017) and reactive flow (Deng et al. 2016, 2017; Voltolini and Ajo-Franklin 2019) in porous materials, while the higher temperature cell is used for thermally activated processes such as creep (Voltolini 2021) and pyrolysis (Voltolini et al. 2019).

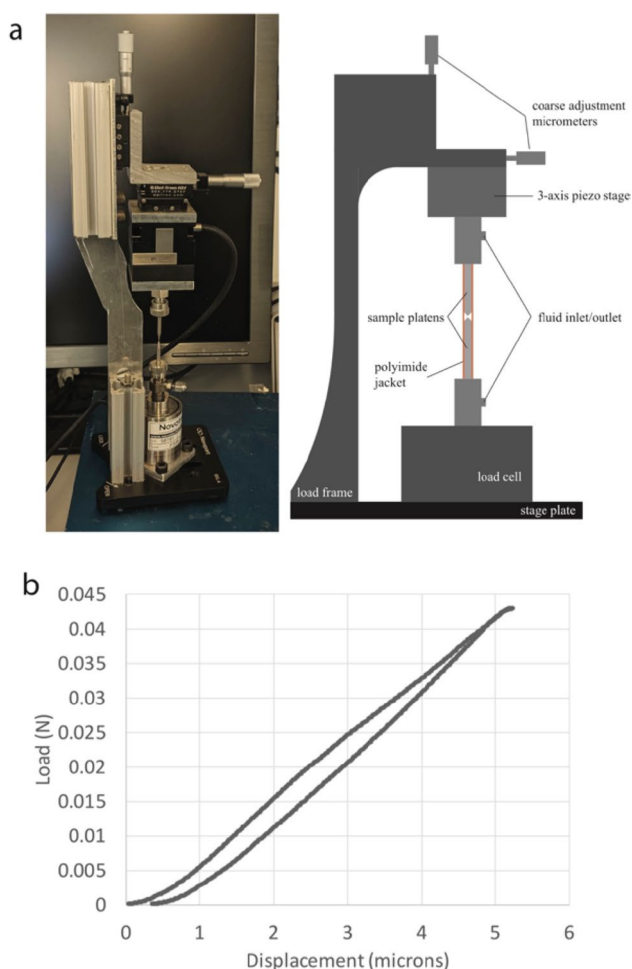
### Nano-tomography: beamline 11.3.1

As a complement to the micro-tomography program, beamline 11.3.1 at the Advanced Light Source is a tender/hard (6–17 keV) X-ray bend magnet beamline that was recently repurposed with a new full-field, nanoscale transmission X-ray microscope. The microscope is designed to image composite and porous materials that have sub-micrometer structure and compositional heterogeneity that governs

materials' performance and geologic behavior. The theoretical and achieved resolutions are 55 and <100 nm, respectively. The microscope is used in tandem with a <25 nm eccentricity rotation stage for high-resolution volume imaging using nanoscale computed tomography. The system also features a novel quadrupole illumination condenser for the illumination of an ~100 μm spot of interest on the sample, followed by a phase-type zone plate magnifying objective of ~52 μm field-of-view and a phase detection ring. The zone plate serves as the system objective and magnifies the sample via projection onto an indirect X-ray detection system, consisting of a polished single crystal CsI (Tl) scintillator and a range of high-quality Plan Fluorite visible light objectives. The objectives project the final visible light image onto a water-cooled CMOS 2048 × 2048-pixel<sup>2</sup> detector. Additional details of the system can be found in Nichols et al. (2022).

### Mechanical-environmental cell

A mechanical loading device has been developed in parallel with the nano-tomography beamline in order to perform in situ mechanical testing with concurrent imaging. The apparatus consists of a single loading column supported by a c-frame (Fig. 4). Samples are fixed between two hollow, needle-shaped loading platens which are sealed to pipe fittings,



**Fig. 4** Nanomechanical loading device. **a** A picture and schematic of the device and **b** example of mechanical data from a constant displacement rate deformation experiment on a single calcite grain

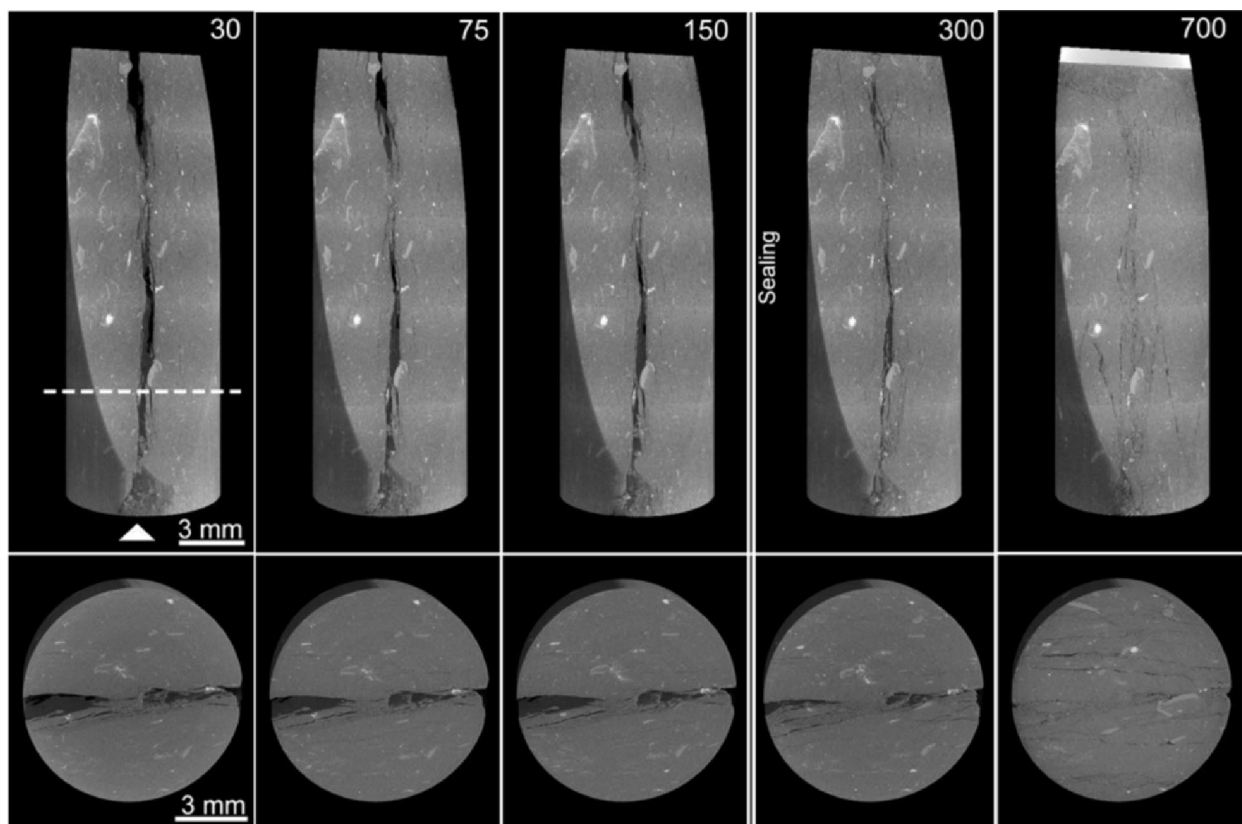
allowing fluid to flow within the annulus between the sample and a polyimide jacket. Force is measured using an integrated donut-style load cell with a linear range of 0–10 N. For a sample with a 40-micron circular cross-section, this amounts to an axial stress range of more than 8 GPa. Three micrometers arranged orthogonally allow for coarse positioning of the platens, while fine control is provided by a three-axis piezo-controlled stage (PI Nanocube). Each axis has a range of 120 microns and the piezoelectric crystal stacks are affixed with strain gauges to allow for closed-loop displacement control with a resolution of 0.2 nm. The stage is driven by a computer-integrated piezo controller (Physik Instrumente E-727) that allows arbitrary loading paths and direct feedback from the integrated load cell. This apparatus is designed for contact-scale mechanics studies in the presence of chemically reactive fluids and is well suited for single grain as well as grain-grain experiments. An example of mechanical data from a constant displacement rate experiment on a single calcite grain is presented in Fig. 4b. This

data can be used to characterize the static elastic properties of granular materials as well as the onset of microplasticity and the role of chemistry on the grain scale.

An example of the kind of scientific questions that can be addressed with micro-tomography (microCT) at Beamline 8.3.2 is given by Voltolini and Ajo-Franklin (2020). In their work, in situ microCT was used to study the sealing of a single fracture in opalinus clay, a proposed nuclear waste repository material (Fig. 5). Opalinus clay (OPA) is a shale with 40–70 wt.% clay minerals (mostly non-swelling clays), 10–30 wt.% carbonates (mainly calcite, subordinate siderite, and ankerite), 10–30 wt.% quartz and accessory feldspars, pyrite, and organic matter. It has well documented self-sealing properties. For the experiment it was soaked in CO<sub>2</sub> saturated water. 4D X-ray microCT sections were collected as a function of confining pressure (up to 700 psi differential pressure). The experiment revealed the influence of different factors in the sealing of the fractured rocks. Hydration-induced swelling, microcracking, and mechanical weakening lead to closing of the asperity contacts. The development of the asperity contacts controls the development of localized choke points, which in turn control the hydraulic properties of the fracture and thus rock sample. Understanding the coupled mechanical and hydraulic properties of fine-grained, heterogeneous materials like shale are important not only for geologic storage, but for energy and water resources as well. In situ observation of the effect of stress on the geometry of a fracture at realistic reservoir conditions is essential to unravel these coupled processes.

### X-ray powder and single-crystal diffraction: beamline 12.2.2

ALS Beamline 12.2.2 was originally designed as a dedicated high-pressure X-ray diffraction beamline focusing on laser-heated diamond anvil cells (LHDAC) and was built and commissioned in 2003–2004. The light for 12.2.2 is produced from the 5.29 T side field of a 6 T superconducting bending magnet yielding a critical energy of 12.7 keV. Its optical design is based on a 2-stage focusing system. The first stage is a vertically collimating mirror, a Kohzu monochromator with both a Si(111) double crystal monochromator ( $E/\Delta E \sim 10,000$ ) and a multilayer double monochromator ( $E/\Delta E \sim 120$ ) followed by a toroidal M2 mirror to yield the primary focus spot at End Station 1 (ES1). Further downstream, the beam can be refocused with demagnification using a set of Kirkpatrick–Baez (K–B) mirrors onto End Station 2 (ES2). Details of these optics are described in Kunz et al. (2005). The beamline offers two end-stations (ES1 and ES2), which can be operated alternately. ES1, positioned at the primary focus point of the beamline, is optimized for high-pressure single-crystal diffraction, while ES2 allows



**Fig. 5** Volume rendering of the OPA sample while increasing the confining pressure (volumes in figure are labeled with differential pressure values, in psi). The inlet is at the bottom (arrow in top left image). On the top row a subvertical section of the sample is shown.

The bottom row displays a slice of sample (at the dashed line in the top left image) looking from the inlet direction. Figure adapted from Voltolini and Ajo-Franklin (2020)

for powder diffraction under various non-ambient conditions. Both ES1 and ES2 have recently installed new developments that are of interest for the Earth- and environmental science community. These are described below:

### 12.2.2: ES1

ES1 hosts a vertically-arranged STOE StadiVari X-ray diffractometer and a Dectris PILATUS3  $\times$  CdTe 300K detector (see Fig. 6) to perform high pressure X-ray single crystal diffraction measurements. The open Eulerian cradle of the diffractometer provides sufficient space to accommodate a set of motorized xyz stages for aligning and active sample position correction. This allows for accurate 4-circle goniometry of samples in holders such as diamond anvil cells weighing up to  $\sim$ 1 kg with a sphere of confusion below 10 microns. Two Aerotech stages move the entire setup transversely and vertically to bring the center of the goniometer onto the primary focus spot of the X-ray beam. A compact and custom-made Ta pinhole (X-ray aperture) with a round cross-section delivers the beam to the sample position with a FWHM spot size of  $65 \mu\text{m}$  (horizontal)  $\times$   $50 \mu\text{m}$  (vertical).

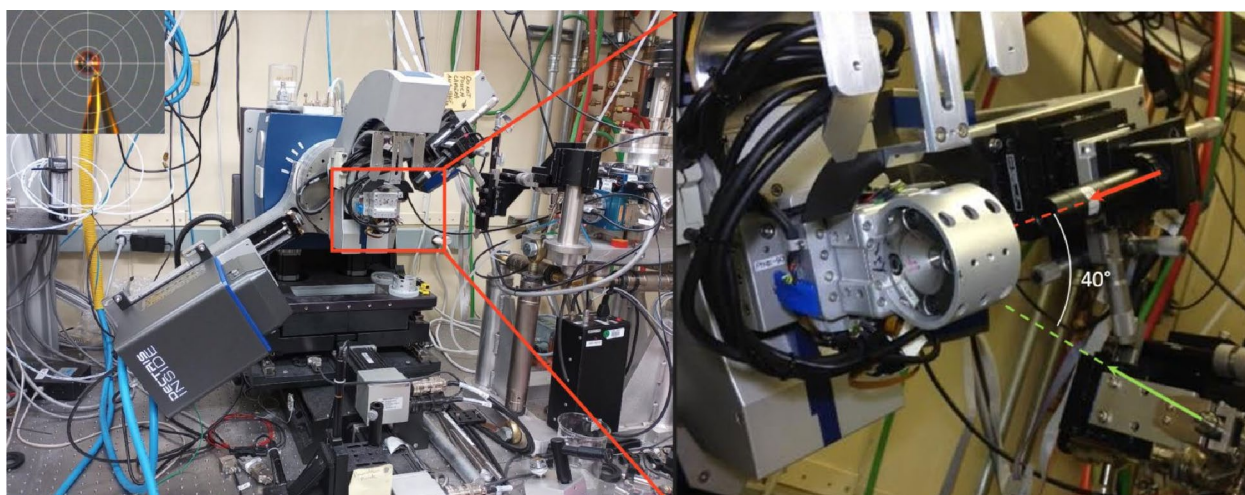
A telescope equipped with a viewing camera at a  $40^\circ$  angle relative to the incoming beam helps with sample alignment and gives a field of view of nearly  $1.2 \times 1.2 \text{ mm}$ .

Shutterless single-crystal X-ray diffraction measurements are performed at energies between 25 keV ( $0.4959 \text{ \AA}$ ) and 33 keV ( $0.3757 \text{ \AA}$ ). A typical exposure time of 0.5–2 s/ $^\circ$  maximizes the intensities of the diffraction peaks without saturation. Beamline characteristics, such as sample distance and precise wavelength, are calibrated with a NIST standard single-crystal ruby sphere (SRM 1990) (Wong-Ng et al. 2001). Data is integrated using XAREA 1.76 (Stoe 2009). Integrated intensities are scaled and corrected for absorption (with the numerical absorption correction) using LANA (Koziskova et al. 2016) and X-RED32 (equivalent to the work of Coppens (1970)) packages embedded in XAREA.

### 12.2.2: ES2

The secondary focus spot of the beamline is created by a set of K–B mirrors that demagnify the primary focus spot in the horizontal and vertical by a factor of 2.19 and 6.14, respectively. ES2 was originally designed for double-sided





**Fig. 6** Left Panel: STOE StadiVari X-ray diffractometer combined with a Dectris PILATUS3 X CdTe 300K detector mounted on 2-theta arm. The insert shows a standard crystal mounted on a MiteGen MicroMesh loop. Right Panel: A DAC, the incoming beam, and the view

laser-heated diamond anvil cell experiments in axial geometry, which remains its primary technique. Details on the LHDAC equipment and its development and characterization can be found in Caldwell et al. (2007), Kunz et al. (2018), Stan et al. (2018), and Yen et al. (2020).

A new and, to our knowledge, currently unique development that has been implemented on beamline 12.2.2 is the ability to perform double-sided laser heating on diamond anvil cells in radial geometry. This adds the possibility of performing texture and viscosity measurements in situ within the LHDAC at conditions of the Earth's interior.

### 12.2.2: ES2—LHDAC in radial geometry

To decipher the dynamic behavior of the deep Earth, in addition to its geochemical and mineralogical composition and geological structure, it is necessary to understand how the solid mantle maintains sufficient plastic flow to sustain convective circulation, which ultimately drives the plate movements at the surface and thus the mass exchange between the inner Earth and its surface. For this, information on the deformation and strength of rocks at the pressure (P) and temperature (T) conditions of the Earth's interior are needed. Diamond anvil cells (DACs) are ideal tools to investigate Earth materials at such conditions.

The DAC technique that allows probing strength and texture of a sample is referred to as radial X-ray diffraction (Bassett 1979; Geng et al. 2018; Kaercher et al. 2016; Kinsland and Bassett 1976; Mao et al. 1996; Merkel 2006; Merkel et al. 2013, 2004; Merkel and Yagi 2005; Miyagi et al. 2010, 2008; Miyagi and Wenk 2016; Raju et al. 2018; Speziale et al. 2019; Wenk et al. 2006; Yue et al. 2019).

While radial Diamond Anvil Cells (rDAC) are today the principal experimental technique for exploring deformation

mechanisms at conditions of the Earth's lower mantle and core, they are acknowledged to have some intrinsic challenges (Wenk et al. 2006). A prominent difficulty for rDACs has been the addition of high temperature. An initial attempt at closing this gap resorted to a one-sided laser heating apparatus (Kunz et al. 2007) at the ALS in 2007. The drawback of this approach (besides the cumbersome alignment procedure) is the potential addition of an axial temperature gradient to the already inherent radial temperature gradient over the probed hot-spot. The problem of large temperature gradients has been addressed in an alternative approach, combining rDACs with resistive heating using a graphite heater that enveloped a mica-boron gasket assembly (Liermann et al. 2009; Miyagi et al. 2013). This method is, however, inherently limited to temperatures <1500 K (corresponding only to upper mantle depths of <500 km). Temperatures in excess of 3000 K, as expected in the basal portions of the lower mantle, can only be reached by means of internal laser heating.

Here, we describe an apparatus installed at ALS beamline 12.2.2 which allows the straightforward combination of a diamond anvil cell in radial geometry with double-sided laser heating.

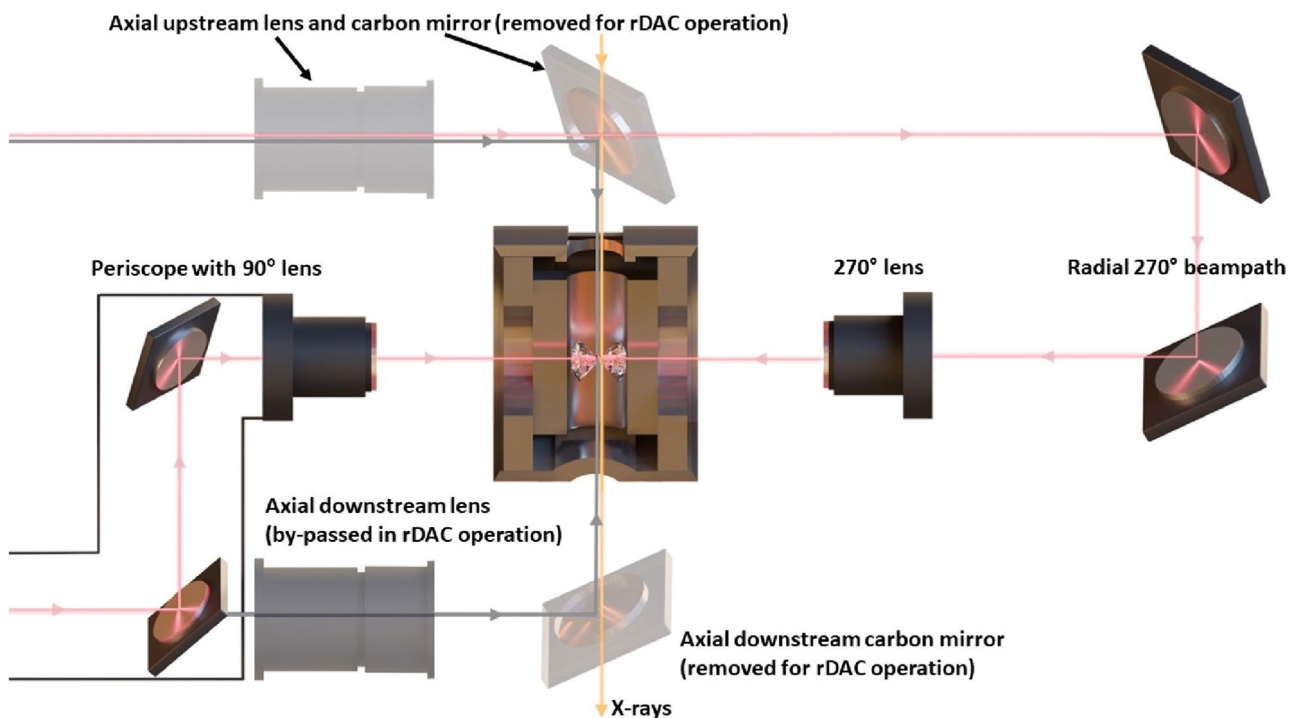
The system is implemented on the laser heating system on ALS beamline 12.2.2 (Kunz et al. 2005, 2018; Yen et al. 2020). In contrast to the previous single-sided system (Kunz et al. 2007), where the rDAC was positioned with the principal stress axis vertical and normal to the incident X-rays, the rDAC in the double-sided system has its compression axis horizontal but perpendicular to the incident beam and thus normal to a vertical rotation axis, which is aligned onto the focused X-ray beam. The heating and pyrometry optics are inserted such that they modify the laser path of the axial system semi-automatically with minimal manual intervention.

As shown in the schematics on Fig. 7, the two sides are referred to as the 90° side (right side, viewing downstream) and 270° side (left side, viewing downstream).

The 90° side is on the right side when looking downstream. It is mounted on a vertically-positioned breadboard that carries the axial upstream and downstream laser and pyrometry equipment, including stages carrying the axial laser heating optics (Kunz et al. 2018). For this, a motorized stage equipped with a pair of adjustable elliptical mirrors (Thorlabs PFE-10-P01 within Thorlabs H45E1 on Thorlabs KM100) is mounted onto the downstream side of the breadboard such that it can be moved to intercept and redirect the laser, imaging, and pyrometry beam paths immediately before the downstream axial objective lens (Fig. 7) onto a separate 80 mm objective lens ( $f/2.85$  apochromatic objective lens by Jenoptik Inc). The 270° equipment is situated on the left side when viewing downstream, i.e., the side which is used for manual access to the sample stage. This makes a permanent automated installation impractical, so we opted for a simple, transferable stage carrying the redirecting mirror optics and objective lens.

In order for the 270° stage to intercept the laser beam, the upstream objective lens and carbon mirror are removed (Fig. 7). The axial carbon mirrors are mounted on magnetic

kinematic bases (Thorlabs KB1X1), which facilitates the transition to radial mode and allows very reproducible positioning without realignment. The base of the stage ( $0.5'' \times 3'' \times 12''$  Al) is bolted onto the optical table with two 1.5'' posts (Thorlabs P14). At the bottom of the optics stack is a Newport MVN80 vertical translation stage, which allows the vertical positioning of the mirror-lens assembly onto the sample. Horizontal adjustment is achieved by a Newport 443 linear stage. A Thorlabs MB6 breadboard is mounted onto a second linear stage (Newport UMR 8.25) that is oriented in line with the objective lens such that it acts as a focusing stage. The breadboard carries two one-inch protected silver coated IR mirrors (Thorlabs PF10-03-P01), each mounted onto a kinematic mirror mount (Newport Performa-i), and an 80 mm objective lens ( $f/2.85$  apochromatic objective lens by Jenoptik Inc), also on a kinematic mount (Thorlabs KM100R). In order to maintain operational flexibility around the sample stage while in the radial laser heating configuration, the objective lens and its mount are held by two small magnetic kinematic bases (Thorlabs KB1X1) that allow quick and reproducible removal and repositioning of the delicate objective lens. To facilitate aligning the 270° optical path orthogonal and centered, the two mounts carrying the IR mirrors are mounted on Newport M423 stages,



**Fig. 7** Schematic of in situ laser heating system for DACs in radial diffraction configuration on ALS beamline 12.2.2. The double-sided axial laser/image/pyrometry beam-path is rotated by 90 degrees into a radial diffraction geometry by means of a motorized periscope mirror assembly (90°-side) and a manually installed mirror and lens set-up (270°-side). The X-ray beam path is depicted in yellow (top

is upstream). The radial laser beam paths are given in pink. The axial laser beam paths are shown in grey with the axial optical elements grayed out for reference. The upstream axial lens and mirror and the downstream mirror are manually removed. The downstream axial lens is by-passed by the periscope assembly. The pyrometry path follows the radial laser path in reverse

allowing independent alignment of the two mirrors onto the incoming laser and focusing lens.

Setting up the system is straightforward. As mentioned above, the 90° side is fully automated and remotely controlled, thus enabling single-sided in situ laser heating within minutes. The manual installation and alignment of the 270° equipment typically takes less than an hour and can be performed independently by experienced users. The equipment not only rotates the laser path by 90 degrees relative to its default axial geometry, but also redirects the optical path for in situ pyrometry. The same pyrometric system as used in the axial geometry can therefore be employed, including quasi-instantaneous temperature mapping of the hot spot (Kunz et al. 2018). The alignment of the rig onto the intercepted IR laser and pyrometry beam path on the one hand and the sample (and thus X-rays) on the other hand is fast and easy thanks to the large number of degrees of freedom allowing independent movement of every optical component of the system. Stable heating up to several thousand K during extended periods of time (up to 30 min) without any temperature induced distortions or translations indicate that the system is mechanically stable despite its high degree of alignment flexibility.

### Application of double-sided rDAC set-up to understand deformations in the lower mantle

The lower mantle consists of perovskite-structured  $\text{MgSiO}_3$  (bridgmanite) and Mg-rich (Mg,Fe)O (ferropericlaase) (e.g. Ringwood (1991)). To understand how the geology of the lower mantle affects tectonic processes such as earthquakes and volcanism we need to understand the deformation behavior of its main constituent minerals on a molecular scale. Plastic deformation of these rocks can lead to texture development and associated seismic anisotropy. If a detailed understanding of the link between deformation and seismic anisotropy is established, observations of seismic anisotropy can be used to understand the dynamics of the deep Earth. Two separate experiments on beamline 12.2.2 investigated the deformation behavior of bridgmanite and

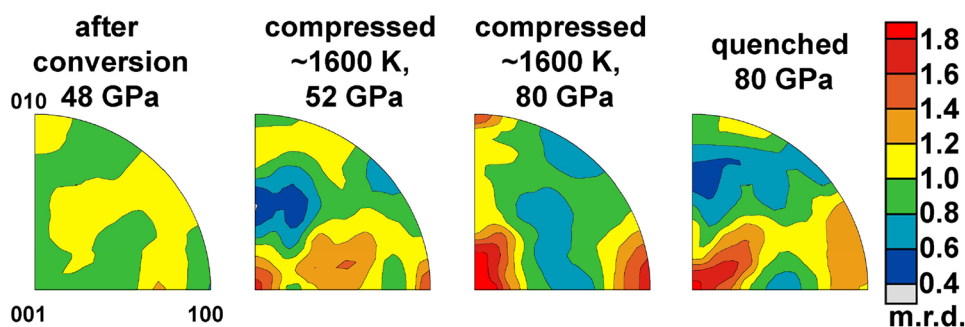
magnesiowüstite (Fe-rich ferropericlaase) using the new double-sided laser heating rDAC capability:

### Deformation of bridgmanite at high pressure and high temperature

Finely-ground  $\text{MgSiO}_3$  enstatite starting material was mixed with Pt into a 30  $\mu\text{m}$  sample chamber within a cBN/Kapton gasket on 300 $\mu\text{m}$ /100 $\mu\text{m}$  beveled diamond anvil culets. The sample was compressed to 45 GPa and then converted to bridgmanite by double sided laser heating. This bridgmanite sample was further compressed while laser heating to obtain its deformation texture in situ at simultaneous high pressures and temperatures. The process was monitored in situ using 25 keV X-rays and a Mar345 image plate detector. Detector and beamline geometry as well as diffraction widths were calibrated using a NIST-SRM674b  $\text{CeO}_2$  standard. Data were reduced and analyzed using a combination of fit2d (Hammersley 2016), MAUD (Lutterotti et al. 1999) and BEARTEX (Wenk et al. 1998). Details on laser heating pyrometry can be found in Kunz et al. (2018), Yen et al. (2020) and Rainey and Kavner (2014).

We find that after conversion from enstatite to bridgmanite at 48 GPa, any compressional texture of enstatite has been erased leaving the newly formed bridgmanite grains with random orientation (Fig. 8). However, during high temperature deformation at ~1600 K and 52 GPa two main maxima form, one at (001) and a second at (100). Further compression to 80 GPa at 1600 K strengthens both maxima. Upon temperature quenching to 300 K the maxima at 100 and 010 decrease in intensity but the (001) maximum remains. A comparison of our results to previous results in the literature (Couper et al. 2020; Miyagi and Wenk 2016) suggests that a change in slip system occurs from (001) at room temperature to (100) at high temperature. The mixed (001), (100) maxima found in our in situ laser heating experiments point to an intrinsic disadvantage of in situ laser heated rDAC experiments: based on the geometry in which the X-rays probe the sample disk with a hot spot in the center along its radial direction, the beam necessarily probes both cold and hot regions of the experiment. Conversely, this serves to

**Fig. 8** Inverse Pole Figure (IPF) of the compression direction for in situ high temperature deformation of bridgmanite. M.r.d is multiples of random distribution with 1 m.r.d being random distribution



show that within a given sample with temperature gradients, different slip systems can co-exist and be characterized using this technique.

### Deformation and phase transitions in $\text{Fe}_{0.92}\text{O}$ at high pressure and high temperature

Wüstite ( $\text{FeO}$ ) is the iron end-member of the  $(\text{Mg,Fe})\text{O}$  solid solution that constitutes the second major phase in the lower mantle after bridgmanite. It forms a continuous solid solution with periclase ( $\text{MgO}$ ). Iron in the lower mantle [often considered to be present between 6 and 7.5 wt%  $\text{FeO}$  (Poirier 2000)] is believed to preferentially partition into Mg-wüstite (Sakai et al. 2009). Low Mg wüstite may be an important component of ultra-low velocity zones at the base of the mantle (Whittaker et al. 2016; Wicks et al. 2010). Periclase ( $\text{MgO}$ ) and wüstite ( $\text{FeO}$ ) both crystallize in the NaCl (B1) structure at ambient conditions. However,  $\text{FeO}$  exhibits a more complex P–T phase diagram than  $\text{MgO}$  (Duffy et al. 1995; Fei 1996; Fischer et al. 2011). While  $\text{MgO}$  stays in the B1 NaCl structure within the P–T range of interest in the Earth's mantle, a series of phase transitions have been reported over the years for  $\text{Fe}_{1-x}\text{O}$  (Fischer et al. (2011) and references therein); most notably a rhombohedral R-3 structure (distortion along the cube diagonal) at above  $\sim 17$  GPa, as well as a B1–B8 transition at high pressure and high temperature. At 10 K and room pressure  $\text{Fe}_{0.99}\text{O}$  was found to have a monoclinic C2/m structure using high resolution neutron diffraction (Fjellvåg et al. 2002). A monoclinic P2<sub>1</sub>/m phase has also been reported at 75 GPa (Kantor et al. 2008).

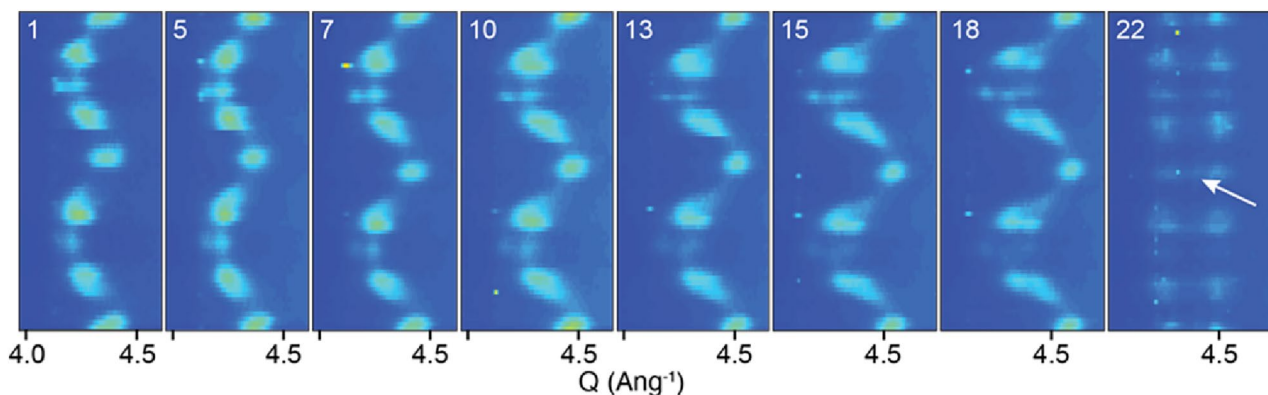
Here we report on reproducible and unambiguous transitions from the high temperature B1 phase to a monoclinic phase in a double-sided laser heated rDAC.

$\text{Fe}_{0.92}\text{O}$  was loaded into a radial diamond BX90 anvil cell (rDAC). Diamond culet diameter was 200  $\mu\text{m}$ . The sample chamber was a 60  $\mu\text{m}$  diameter hole which was laser drilled

from a 40  $\mu\text{m}$  thick and 360  $\mu\text{m}$  in diameter cBN gasket and confined by a Kapton jacket (Miyagi et al. 2013). The monochromatic X-ray beam from ALS beamline 12.2.2 ( $\lambda = 0.4959$  Å) was focused to a size of  $\sim 10$   $\mu\text{m}$  FWHM. Wavelength, detector distance, and detector tilt were calibrated in Dioptas (Prescher and Prakapenka 2015) using a NIST-SRM674b  $\text{CeO}_2$  standard. Diffraction images of  $\text{Fe}_{0.92}\text{O}$  were collected for 60 s on a Mar345 image plate located  $\sim 335$  mm from the sample. Data were analyzed using Dioptas, MAUD (Lutterotti et al. 1999) and BEARTEX (Wenk et al. 1998). Details on laser heating pyrometry can be found in Yen et al. (2020) and Rainey and Kavner (2014).

A monoclinic phase of  $\text{Fe}_{0.92}\text{O}$  was formed at pressures  $> 23$  GPa under deviatoric stress and observed using in situ X-ray diffraction in a radial diamond anvil cell. Based on visual analysis of peak splitting (Fig. 9), an initial phase transition from the rhombohedral phase is suspected to occur between 23 and 27 GPa. Following laser heating, monoclinic peaks are clearly identifiable as distinct from a highly anisotropic rhombohedral structure. This structure can be fit with the C2/m structure as identified by Fjellvåg et al. (2002). The monoclinic structure is consistently reproducible from the high-temperature cubic phase at these pressures, and is stable during compression at least up to 36 GPa. The stability of the monoclinic unit cell is likely dependent on both the degree of non-hydrostaticity within the sample chamber and the high defect density of  $\text{Fe}_{0.92}\text{O}$  as compared to  $\text{Fe}_{1-x}\text{O}$  samples that are closer to stoichiometric. Temperature may also have acted as a catalyst for the phase transformation, and/or the heating sufficiently relaxed elastic strain in the quenched crystal lattice such that highly strained monoclinic peaks present prior to heating were then identifiable.

Using simultaneous X-ray diffraction in radial geometry and double-sided laser heating, texture relationships between the high-temperature B1 structure and the ambient temperature monoclinic structure could be established. As the B1



**Fig. 9** Progression of splitting of the cubic 220 diffraction line during compression at room temperature (image 1–18) and after a single, consecutive stage of laser heating (image 22). Each panel shows an

unrolled diffraction image composed of 72 5° azimuthal sector integrations. A faint diffraction line in image 22 is indicated by a white arrow

structure is unquenchable, this could not have been determined by conventional laser heating of a radial diamond anvil cell in axial geometry.

Texture developments are summarized in Fig. 10 using inverse pole figures (IPF). Inverse pole figures are reported before and after major phase transitions and following periods of compression (Fig. 10). At the beginning of the experiment (15 GPa), maxima in the cubic  $\text{Fe}_{0.92}\text{O}$  IPF are concentrated at 001 (~3–4.0 m.r.d.) and 011 (~2–3 m.r.d.; Fig. 10a). Preferred orientation in the cubic phase becomes more pronounced at 001 (~5–6 m.r.d.) in the IPF during the phase transformation (Fig. 10b). Primary maxima development in the trigonal IPF is at 211 (>7 m.r.d.; Fig. 10b). After a complete transformation to the trigonal phase at 19 GPa and 300 K, IPF intensities are roughly equal at 211 and near 2–21 (~3.0–4.5 m.r.d.; Fig. 10c). Further compression to 23 GPa (Fig. 10d) results in textures similar to those of Kaercher et al. (2012) at 19 GPa. From both graphical and a Rietveld Rwp analysis, we interpret that a phase change occurs above 23 GPa (Fig. 9).

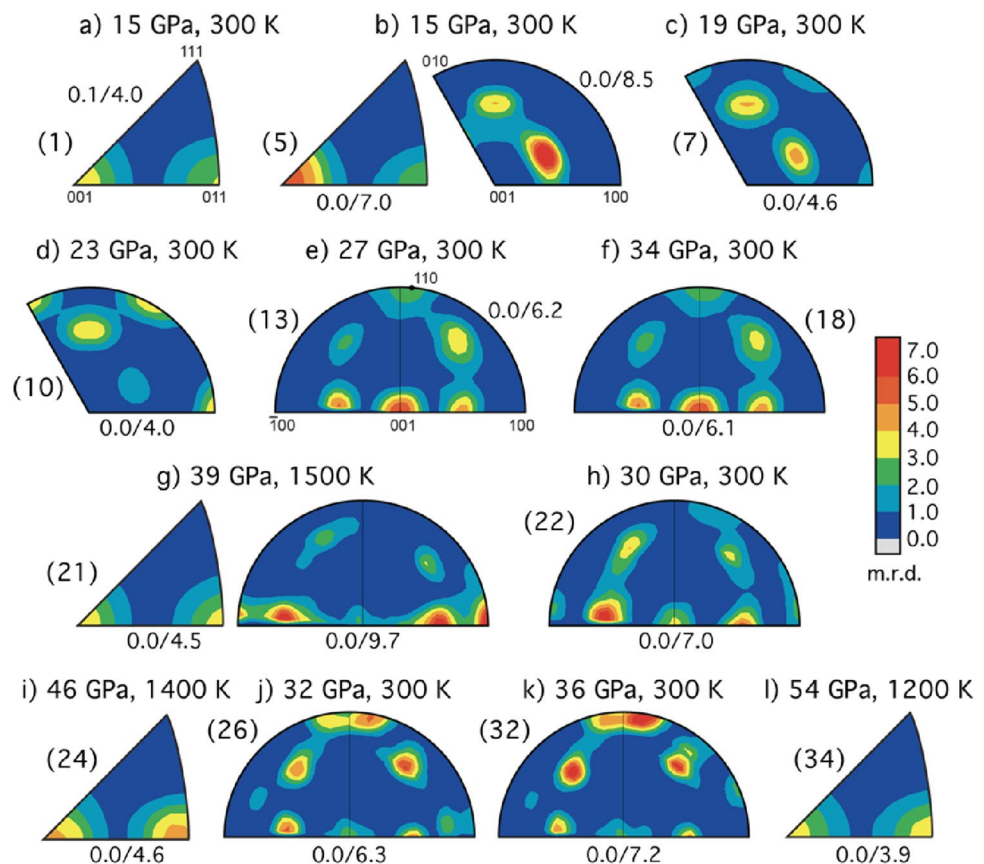
The monoclinic IPF at conditions of ~27 GPa and 300 K shows maxima at 001 (~5–6 m.r.d.), near both 101 and  $\bar{1}01$  (~4–5 m.r.d.), and at 211 (~3–4 m.r.d.; Fig. 10e). Further compression of the monoclinic phase does not result in significant changes in texture (Fig. 10e, f). During the first

stage of heating (39 GPa, 1500 K), only one laser coupled with the sample, and both cubic and monoclinic phases are present. We anticipate that a strong temperature gradient was present due to only one laser having coupled, and suggest that the monoclinic phase is present only in cooler regions of the sample chamber. The cubic IPF shows both 001 and 011 fibers perpendicular to compression after transformation (~3–4 m.r.d, Fig. 10g). A maximum at 001 in the monoclinic IPF present prior to heating weakens during heating (2–3 m.r.d.), and the maxima near 101 and  $\bar{1}01$  (>7 m.r.d.) become elongated between 001 and 100.

Post-heating, the maximum in the monoclinic IPF at 001 is weaker (~3–4 m.r.d.; Fig. 10h) as compared to pre-heating (Fig. 10f). The intensity in the IPF is concentrated near  $\bar{1}01$  and 101 (~6–7 m.r.d.; Fig. 10h). Maxima in the monoclinic IPF at 011 and 211 both have moderate intensities (~3.0–4.0 m.r.d.).

Only the cubic phase is visible in the second round of heating at ~1400 K, where both lasers coupled to the sample. Texture in cubic  $\text{Fe}_{0.92}\text{O}$  (Fig. 10i) is relatively unchanged from that of 39 GPa, 1500 K (Fig. 10g). No monoclinic phase is present. The monoclinic phase returns when the sample is allowed to cool to room temperature (Fig. 10j). The maxima at 001 and near  $\bar{1}01$  and 101 in the monoclinic IPF have weakened compared to the previous

**Fig. 10** Inverse pole figures (IPFs) showing texture development in the direction of compression for  $\text{Fe}_{0.92}\text{O}$  polymorphs over the course of this study. Pressure and temperature at the time of image collection are shown above the IPF. Corresponding image numbers are shown in parenthesis. For monoclinic phases, pressures were determined from the best fit of a trigonal unit cell (Jacobsen et al. 2005). A backslash (/) separates minimum and maximum multiples of random distribution (m.r.d.) values for each IPF



monoclinic phase (~2–3 m.r.d.), and maxima have strengthened both at 211 (4–5 m.r.d.) and near 011 (~5–6 m.r.d.). An additional maximum is also seen near to the top of the monoclinic IPF (~5–6 m.r.d.; Fig. 10j). As the monoclinic phase is compressed, maxima at 211 and 011 and at the top of the IPF (Fig. 10k) continue to develop. When transformed into the cubic phase at 1200 K, the cubic IPF has two maxima of moderate intensities at 001 and 011 (~3–4 m.r.d.; Fig. 10l).

Hence, these results illustrate both the capabilities of the laser-heated radial diffraction system at beamline 12.2.2, and the notable phase and deformational complexity that can be present even within a comparatively chemically simple system at mid-mantle pressures.

### X-ray Laue microdiffraction: beamline 12.3.2

Beamline 12.3.2 was commissioned in 2008, after moving the successful bending magnet Laue X-ray microdiffraction program on beamline 7.3.3 (Tamura et al. 2003) to a 4.37 T superconducting magnet source to increase the flux and energy range (Kunz et al. 2009b). While the original scientific focus of the beamline was the measurement of strains in electronic materials using Laue X-ray microdiffraction (Valek et al. 2002), its program has expanded to include a variety of fields ranging from geosciences to solid state physics and a variety of techniques, including powder microdiffraction and microfluorescence.

The first optic on the path of the synchrotron beam is a horizontally deflecting toroidal mirror, which refocuses the source at the entrance of the experimental hutch, where a pair of roll slits provides a size-adjustable secondary source. The optic chamber inside the hutch contains a pair of JTEC water-cooled Kirkpatrick-Baez mirrors with fixed elliptical shapes that produce a 1  $\mu\text{m}$  X-ray beam on the sample. A 4-bounce Si(111) double channel-cut monochromator can be inserted into the beam to easily switch between polychromatic and monochromatic beams while illuminating the same spot on the sample. The available energy range for the beamline is about 6–24 keV, with a peak intensity around 12 keV. The flexible sample and detector stages allow for both transmission and reflection geometries. 2D diffraction patterns are collected with a DECTRIS Pilatus 1 M, while X-ray microfluorescence signals are collected using a Vortex EM solid-state detector. Additional sample environment chambers are employed to heat samples up to about 1000 K or cool them to about 180 K. A new cooling stage will be made available in the near future, that will be able to cool samples down to liquid N<sub>2</sub> temperatures.

To date, only a few experiments have been conducted under non-ambient conditions at 12.3.2, largely because it was not originally intended for non-ambient condition research. Some of its unique capabilities show promise,

however, particularly in the study of plastic deformation in nanocrystalline materials, a topic that is not well understood. One such study used a diamond anvil cell to perform high-pressure Laue X-ray microdiffraction to investigate the size dependence of nickel nanocrystal grain rotation (from 500 nm down to 3 nm) (Zhou et al. 2017), as grain rotation resulting from grain boundary dislocations is one manifestation of plastic deformation in materials.

6–8  $\mu\text{m}$  tungsten carbide (WC) particles were embedded into a nickel medium of variable particle sizes to act as fiducial markers for the rotation of the nickel nanocrystals. Laue patterns were collected on the WC particles to measure their crystal orientations, and thus their rotations relative to their starting position, at pressures up to 10 GPa. Models such as the Read–Shockley model predict that grain rotation increases with decreasing nanocrystal size. However, the experiment demonstrated that the 70 nm nickel medium induced more rotation than in any other nickel medium, whether larger or smaller in particle size. Subsequent measurements of texture deformation performed at beamline 12.2.2 on the same samples indicated that dislocations in the interior of the grains remain active in grains down to 20 nm. This suite of complementary measurements at two ALS beamlines thus demonstrated that nanocrystal grain rotations are governed by dislocations both in grain interiors and along grain boundaries under external stresses, and the two opposing trends are what caused the observed maximal grain rotation of nanocrystals of a specific size. This study provides a better understanding of plastic deformation in nanocrystalline materials and provides guidance for material properties, functionalities, design, and processing.

In Earth science, Laue X-ray microdiffraction (Laue  $\mu\text{XRD}$ ) has been widely used to examine and map lattice distortion in quartz at the micron scale (Chen et al. 2015; Kunz et al. 2009a). The shape and orientation of the elastic strain tensor can be derived from lattice distortions and explained in terms of residual stress. It was shown that mechanically-induced Dauphine twinning, rhombohedral deformation lamellae, and residual stresses in the Vredefort quartzite are features of shock deformation from the meteorite impact site at Vredefort, South Africa (Chen et al. 2011). For another study, the microstructures of core samples from the San Andreas Fault Observatory at Depth (SAFOD) were shown to comprise complex compositions of quartz, feldspar, clays, and amorphous material, indicating that the primary deformation processes were intense shearing and dissolution–precipitation. The differential stresses are inferred to range from 33 to 132 MPa, with mean values falling between 68 and 168 MPa. The inferred friction coefficient was found to be quite low, assuming hydrostatic pore pressure. The measurements performed support the hypothesis of a weak San Andreas Fault (Rybacki et al. 2011).

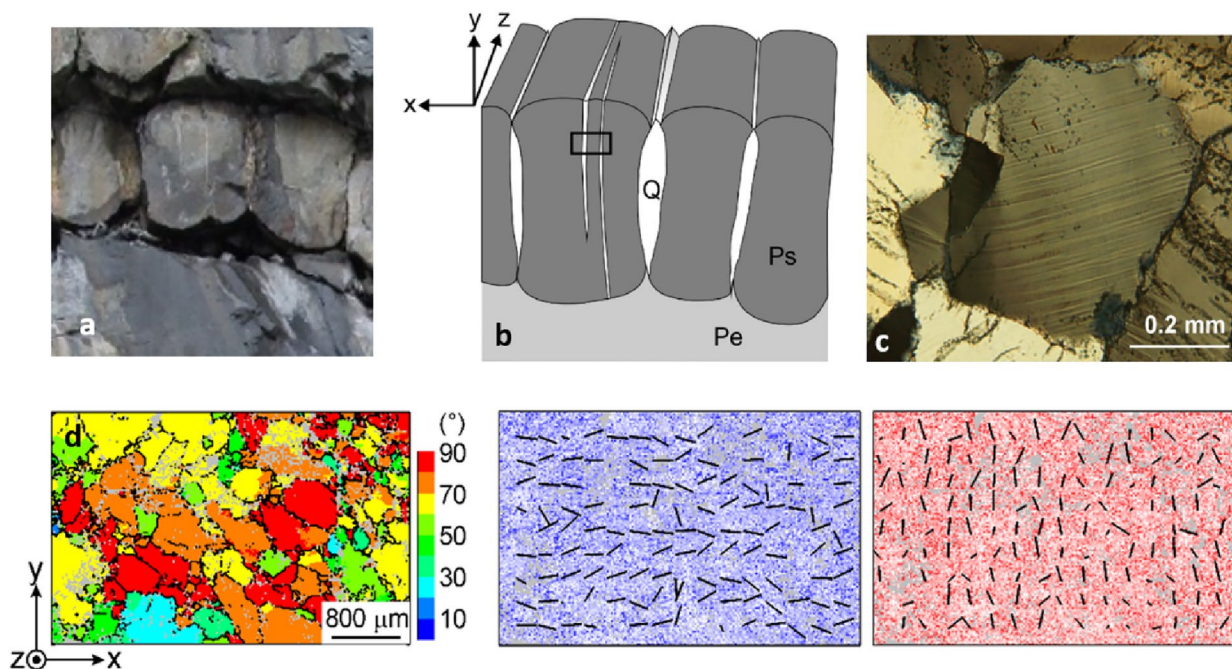
Laue  $\mu$ XRD was also used to investigate vein quartz from Bastogne, Belgium, the original boudinage locality. The measurements showed a residual elastic strain that involves shortening perpendicular to the vein wall (Fig. 11) supporting the hypothesis that the Bastogne boudins formed by layer-parallel shortening rather than layer-parallel extension, as had been previously assumed (Chen et al. 2016).

Additional experiments reveal that metasedimentary quartzites from the Bergell Alps generally shorten perpendicular to the schistosity plane, but with significant asymmetry in relation to foliation and lineation. Laue  $\mu$ XRD is thus a promising technique to detect stress fields during tectonic deformation (Wenk et al. 2020).

Another beamline capability that may be of broader interest is the structure refinement and solution of crystals by Laue  $\mu$ XRD that are either too small or encased in a heterogeneous matrix so that they cannot be studied by conventional crystallographic methods (Dejoie et al. 2013). These methods have been successfully employed in the discovery and naming of eight new minerals since 2019 [including new Pt-group minerals], (Barkov et al. 2019a, b, 2021). In principle, this technique can be applied to crystals synthesized inside a diamond anvil cell or trapped as inclusions in high-pressure charges.

## Off-line capabilities

Many of the experiments performed at the ALS require extensive, delicate sample preparation, particularly high-pressure diamond anvil cell experiments. Users typically prepare their samples at their home institutions, however sample reloading during beamtime is often required. In addition, DAC experiments often require the addition of a gaseous pressure medium into the sample chamber (referred to as gas loading), however many users lack the complex and expensive equipment necessary to gas load a DAC. To address these needs, the ALS has allocated laboratory space near the beamline where 12.2.2 users may prepare and load DACs. The laboratory is well-equipped for DAC sample preparation, hosting three Leica M205C stereomicroscopes as well as an Oxford Lasers Alpha 532 laser milling system. The laser mill primarily serves to machine gaskets for DAC experiments and is capable of drilling sample chambers with  $<5\ \mu\text{m}$  accuracy. The laser mill control software (Cimita v4.3.3) is user-friendly and the instrument is available to beamline users after minimal training. To gas load samples, the laboratory houses a COMPRES/GSECARS gas loading apparatus (Rivers et al.



**Fig. 11** **a** View of psammite(Ps)-pelite(Pe) multilayer sequence in the Mardasson quarry near Bastogne ( $50^{\circ}00'55.01''\text{N}$ ,  $5^{\circ}44'27.63''\text{E}$ ) with the typical “boudin” structures, separated by quartz veins, in the stiffer psammite layers. **b** Schematic line drawing (not to scale) of the Bastogne “boudin” structure (Q: vein quartz, Ps psammite and Pe pelite), indicating the sample location with two quartz veinlets; sample coordinates are indicated (xyz). **c** Detail of one of the scanned

quartz grains with deformation lamellae and fluid inclusions. **d** results from a large scan on a quartzite lamella with grain orientation (Euler orientation angle ( $\phi$ ) and boundaries (left), and magnitudes of the compressional (blue, middle) and extensional principal strain axes (red, right) of the strain ellipsoid and their directions projected on the x–y plane (black lines)). Figure adapted from (Chen et al. 2016)

2008). Though the gas loader is not available for users to operate independently, users may request that beamline staff load their DACs with He, Ne, or Ar in advance of their shift, with other gasses such as CO<sub>2</sub> potentially available on request. While the sample preparation lab is maintained and operated by 12.2.2 staff, it is available upon request and coordination to users of any ALS beamline.

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**Author contributions** M. Kunz wrote 12.2.2 ES-2 section and coordinated manuscript writing; K. Armstrong contributed to 12.2.2 section and off-line capabilities section; H. Barnard contributed to 8.3.2 section; H.A. Bechtel wrote IR section; B. Kalkan wrote 12.2.2 ES-1 section; H. Lisabeth contributed to 8.3.2 section; A.A. MacDowell contributed to 8.3.2 and 12.2.2 sections; D.Y. Parkinson contributed to 8.3.2 section; N. Tamura wrote 12.3.2 section; Q. Williams contributed to 12.2.2 section and introduction; L. Miyagi contributed with experiments and manuscript writing (12.2.2 section); and S. Couper contributed with experiments (12.2.2 section)

**Data availability** Raw data are stored on beamline servers of the various beamlines discussed and can be obtained upon request.

## Declarations

**Conflict of interest** The authors have no competing interests to declare that are relevant to the content of this article.

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