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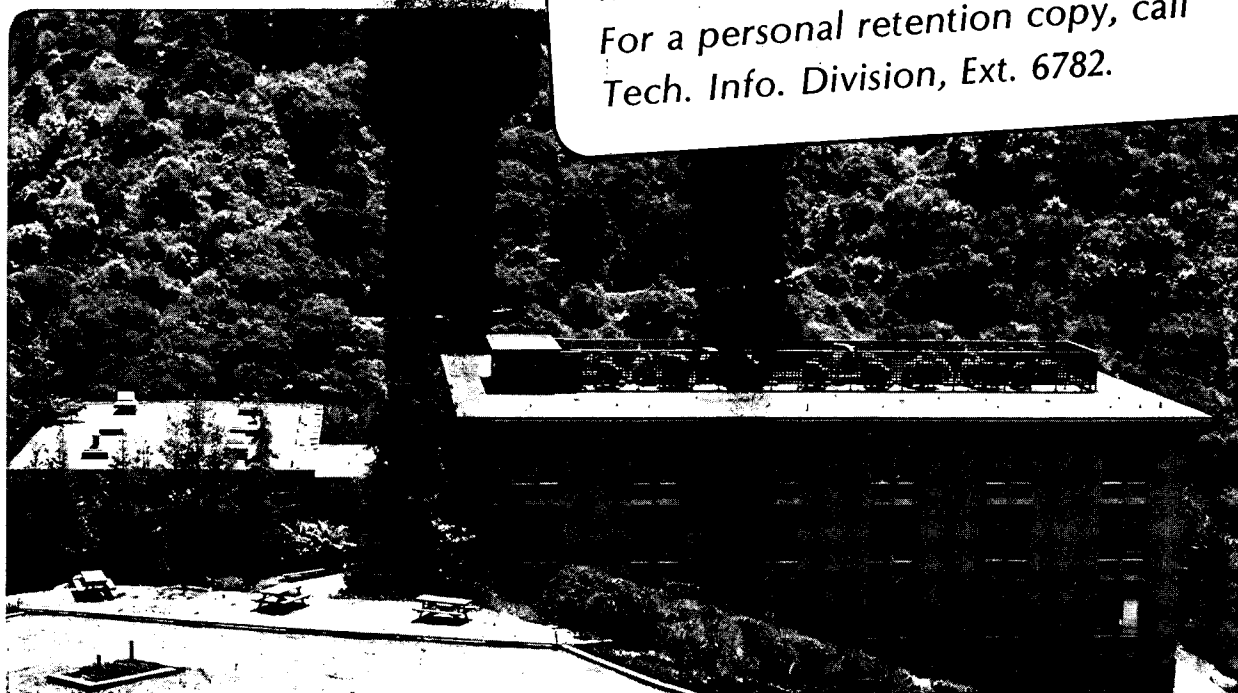
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Oxidation of Singular and Vicinal Surfaces of Silicon: The Structure of Si-SiO₂ Interface

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Abstract

The structure of interfaces between silicon oxides grown in dry oxygen on singular and vicinal (111) surfaces of silicon has been studied using cross-sectional high resolution transmission electron microscopy. Crystalline silicon was found to terminate abruptly on (111) planes, where it transforms to amorphous SiO₂. One interplanar distance high ledges, separated by (111) terraces were found to be present on all surfaces studied. The width of the terraces was surface orientation dependent. It is suggested that such a structure of the interface can be explained by a terrace-ledge-kink model and that high temperature oxidation proceeds by a ledge mechanism similar to that for evaporation from the surface.

Introduction

The Si-SiO₂ interface has been extensively studied during the last few decades due to the application of thermally grown SiO₂ dielectric films in processing of bipolar and especially metal-oxide-silicon (MOS) electronic devices. This research has as its goal a better understanding of the correlation between Si-SiO₂ interface structure and electronic properties, and of the chemistry and mechanism of oxidation. This knowledge will be useful to the development of computer-aided design of integrated circuit fabrication processes.[1,2,3] In this paper high resolution electron microscopy (HREM) studies are reported of the structure of Si-SiO₂ interfaces resulting from oxidation of singular (minimum surface energy) surface of exact (111) orientation and those of vicinal orientations (within 3° of exact (111)). A few HREM studies of Si-SiO₂ interface structures have been published.[4,5,6,7,8] However no systematic structural studies of singular and vicinal (111) oxidized surfaces exist.

Experimental Procedures

The oxide films were grown in dry O₂ on p-type B doped (7-17 Ωcm) silicon singular surfaces having (111) orientation and two vicinal orientations, i.e., 2° off (111) towards [11 $\bar{2}$] and 3° off (111) towards [1 $\bar{1}$ 0]. The orientation of the crystal surfaces were checked by the x-ray back reflection Laue technique.

The wafers were cleaned prior to oxidation in a series of six rinses (1) 5:1:1 H₂O: NH₄OH: H₂O (2) deionized water (D.I.), (3) 5:1:1 H₂O: HCl: H₂O₂, (4) D.I., (5) 50:1 H₂O: HF, (6) D.I. After drying in N₂ the

wafers were transferred into the furnace with an oxygen atmosphere. Oxidation continued for a time sufficient to grow about 100 nm of oxide, and wafers were removed from the furnace within five minutes.

Cross-sectional transmission electron microscopy (TEM) specimens were prepared using the standard technique.^[6] High resolution TEM images were obtained using a JEM 200 CX electron microscope operating at 200 kV and equipped with a high resolution pole piece ($C = 1.2$ mm). The interface was imaged along the $[1\bar{1}0]$ orientation of the exact (111) wafer and the wafer having 2° vicinal orientation. For the 3° vicinal wafer, the interface was imaged along a $[10\bar{1}]$ direction.

These orientations allowed direct imaging of two sets of $\{111\}$ crystal planes and one set of $\{200\}$ Si crystal planes from the Si matrix by phase contrast procedures.^[9]

Experimental Results

Structure of the Oxidized Singular (111) Si Surface

Oxidation at 1100°C in dry O_2 resulted in the structure of the interface shown in Fig. 1. The oxide is amorphous as it can be deduced from the characteristic mottled contrast and the interface between silica and silicon is very abrupt and flat over the entire area observed. Careful observation furthermore reveals the existence of steps only one 111 interplanar distance (.314 nm) high. This can be seen more clearly on the higher magnification micrograph shown in Fig. 2. The width of the terraces between positive and negative steps varies and is dependent upon defocus which indicates that these steps may not extend through the whole TEM specimen thickness. A possible interpretation is shown schematically on Fig. 3. Another interesting feature that was observed is shown in Fig. 2. The last row of crystal image spots is displaced as would be expected if there was a stacking fault parallel to the surface. Computer modeling of the image is in progress to see if alternative explanations exist.^[10]

Structure of an Oxidized 2° Vicinal Si Surface

Vicinal surfaces in contact with vacuum (vapor) are expected to have steps which connect terraces of minimum surface energy, and the intersection of such steps with the terraces are themselves low energy $\langle 110 \rangle$ directions.^[11] Figure 4 shows the 2° vicinal Si-SiO₂ interface which conforms to this model. The interface consists of approximately equally spaced ledges with the width of the terraces 7.0 - 11.0 nm and their height equal to .314 nm. The width of the ledges calculated for the inclination of the surface based on purely geometrical considerations would correspond to 9.0 nm. The terraces are atomically flat, however some positive-negative step pairs are also found to be present on some terraces as is shown in Fig. 5.

Structure Of An Oxidized 3° Vicinal Si Surface

Dry oxidation at 1000°C of a surface 3° off (111) towards $[\bar{1}\bar{1}0]$ resulted in the structure shown in Fig. 6.[7] The HREM image shows that (111) planes are inclined 3° off the interface plane. The Si substrate terminates abruptly at (111) terraces about 6.0 nm wide with the ledges one interplanar distance (.314 nm) high. The structure of this interface was found to be independent of the oxidation temperature and time.

In contrast to the less inclined vicinal surface no extra steps on the terraces were found.

Discussion and Conclusions

The results of this investigation demonstrate that oxidation occurs layer by layer very uniformly over large areas of the Si surface. The roughness of the surface corresponds to the height of the individual steps (.314 nm). The oxide growth corresponds to removal of the Si atoms from the substrate surface. The observed structure of the interface suggests that this process occurs at the ledges. In the case of singular (111) surfaces formation of the ledges might be envisioned as occurring by two dimensional nucleation corresponding to the formation of an oxide island in the next layer of silicon atoms. In the case of vicinal surfaces, structural ledges are already present at the interface providing sites for oxidation. However for too low a density of such ledges two dimensional nucleation still takes place resulting in terraces with additional positive and negative ledges. A similar process is observed for evaporation or dissolution of atoms from a surface into vapor or solution.[12,13] Although the Si surface in this case is in contact with solid silica the interface structure appears to behave very much as it would in contact with a liquid. This is perhaps not surprising because viscous flow of silica occurs above 960°C[14] and oxidations in these experiments were performed at 1100 and 1000°C.

In conclusion, high resolution electron microscopy studies of cross-sectional specimens of oxidized singular and vicinal (111) surfaces have demonstrated that:

- (a) The SiO₂ is amorphous right up to the interface.
- (b) The Si substrate terminates abruptly on atomically flat (111) terraces at the Si-SiO₂ interface.
- (c) Steps one interplanar distance ($d_{\{111\}} = .314$ nm) high are observed on both singular and vicinal oxidized surfaces.
- (d) The observed structure suggests a terrace-ledge-kink model for the interface and that high temperature oxidation proceeds by a ledge mechanism similar to evaporation from the surface.

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Figure Captions

Figure 1. Low magnification HREM image of cross-section of oxide film grown on exact (111) Si wafer (dry O₂, 1100°C).

Figure 2. Higher magnification HREM image of a cross-section of the oxide film grown on exact (111) Si surface with clearly resolved steps one interplanar distance high.

Figure 3. Schematic of possible <110> ledge arrangement in the Si-SiO₂ interface (exact (111) Si wafer). Broken lines in the top figure are possible ledge contours in the interface plane.

Figure 4. Low magnification HREM image of a cross-section of the oxide film grown on Si surface 2° off (111) towards [11 $\bar{2}$].

Figure 5. Higher magnification image of Fig 4. Structural (one sign) and additional positive and negative ledges on atomically smooth terraces are clearly resolved.

Figure 6. High magnification image of a cross-section of the oxide film grown on Si surface 3° off (111) towards [1 $\bar{1}$ 0] (dry O₂, 1000°C). Structural steps are clearly resolved. (Courtesy of San Francisco Press, Inc., from reference [7]).

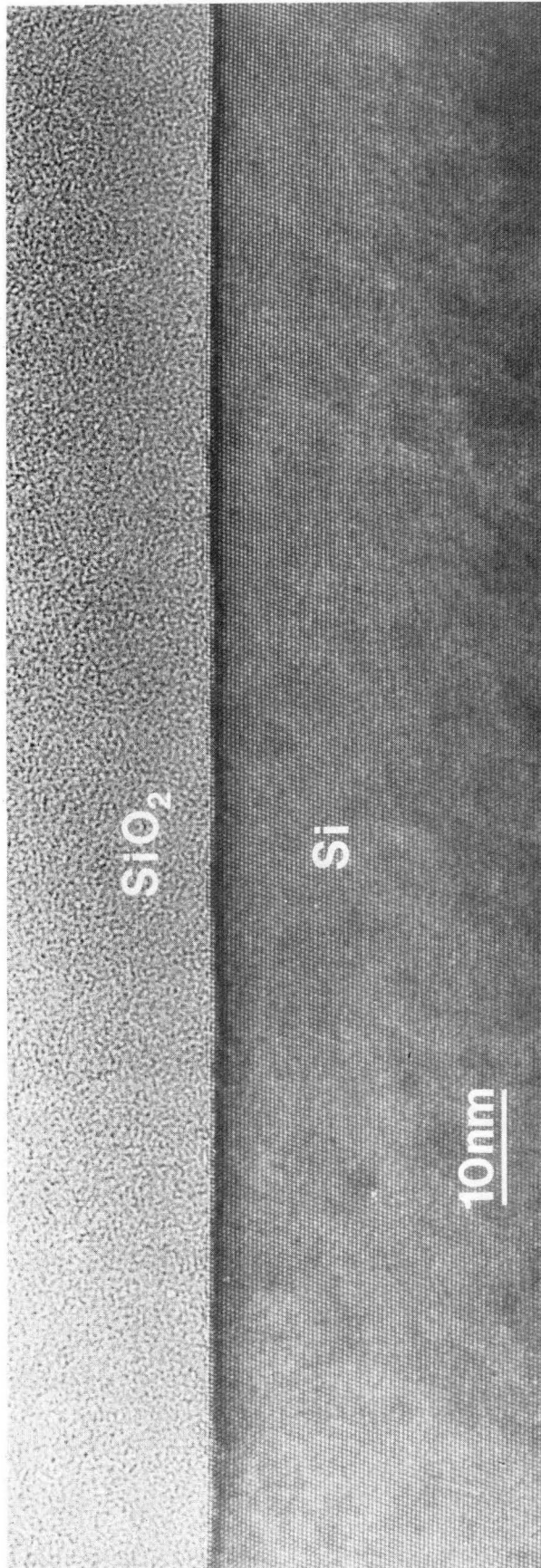
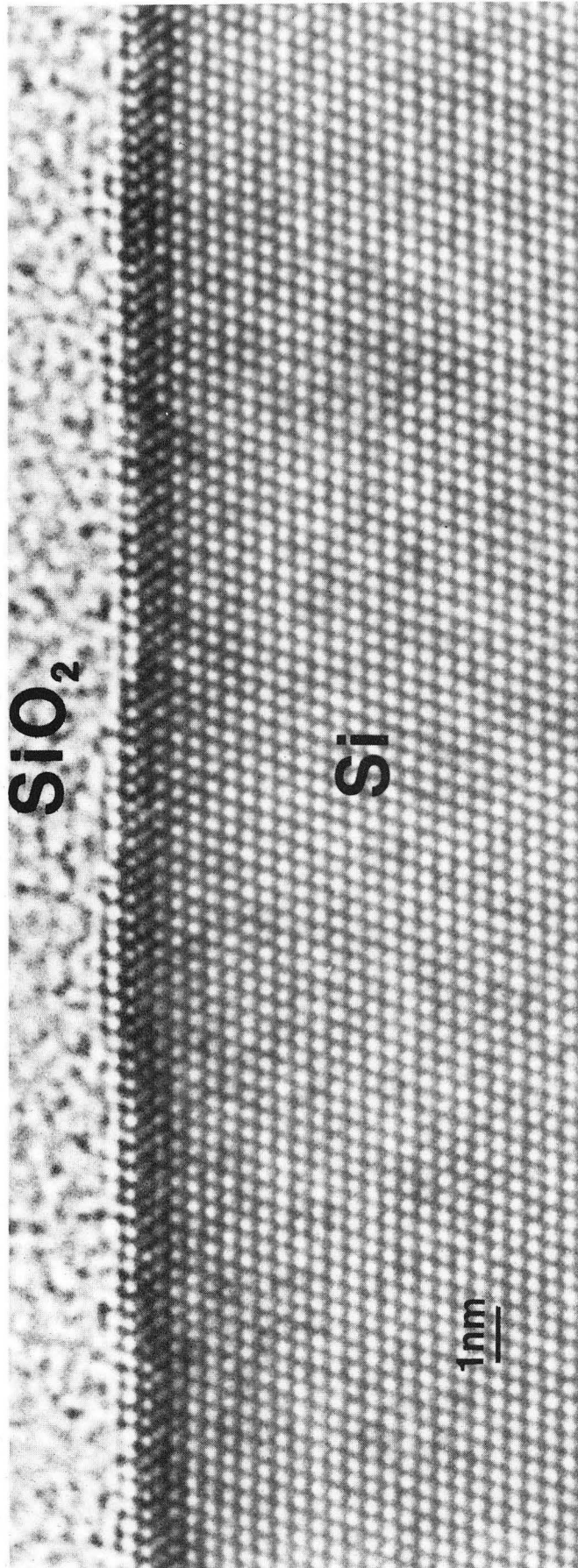


Figure 1.



SiO₂

Si

1nm

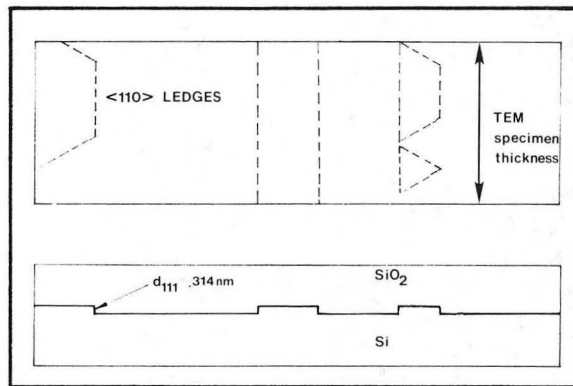


Figure 3

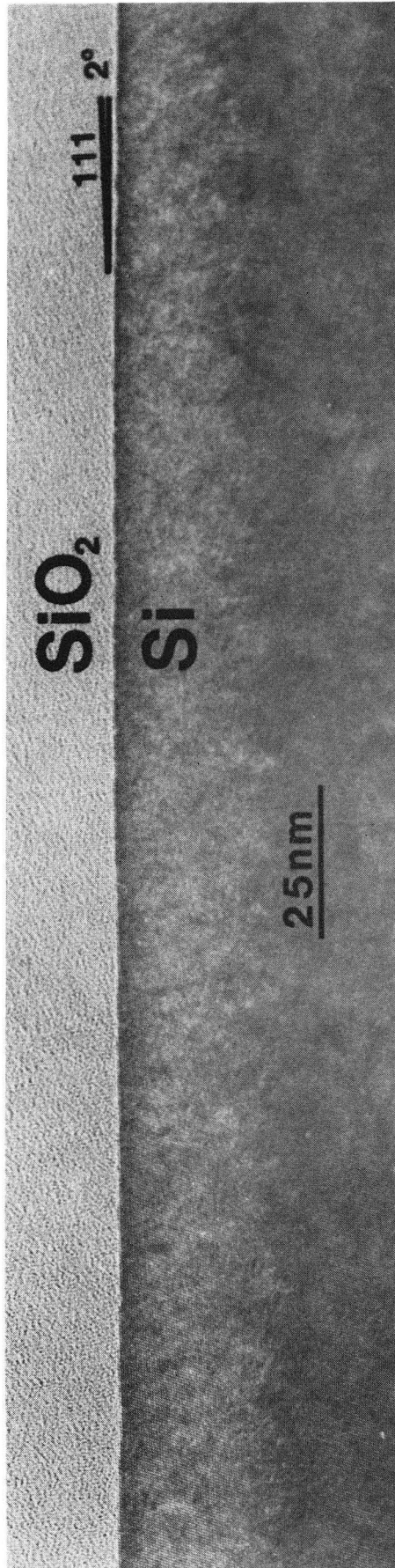


Figure 4.

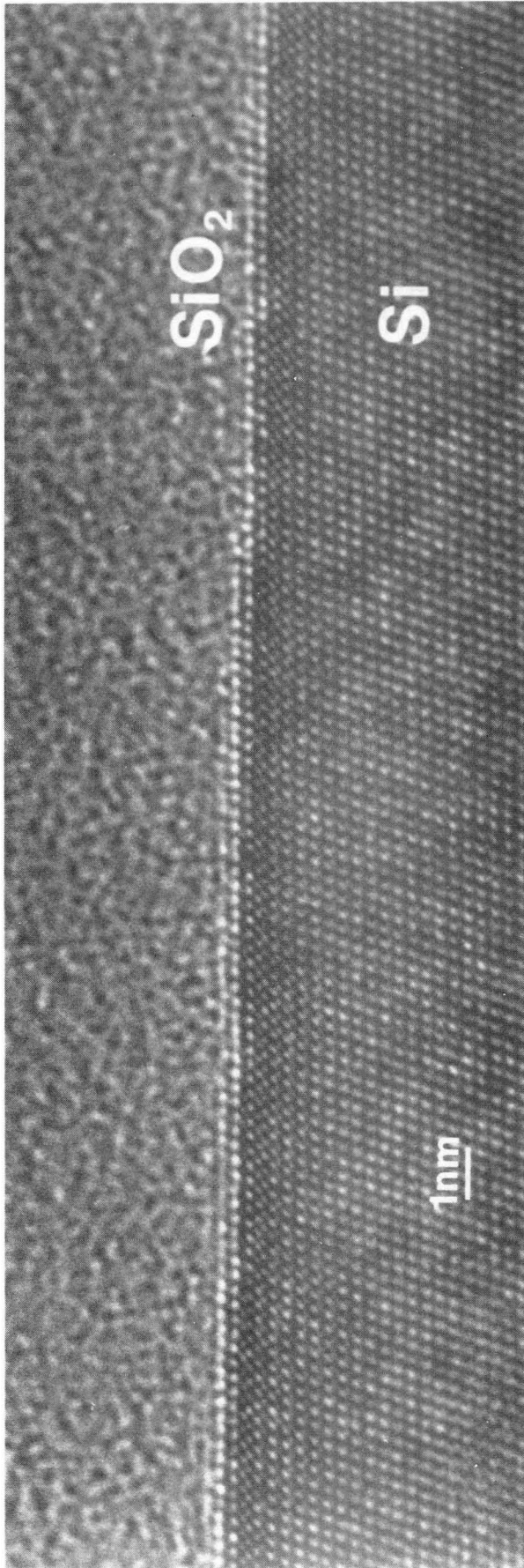


Figure 5.

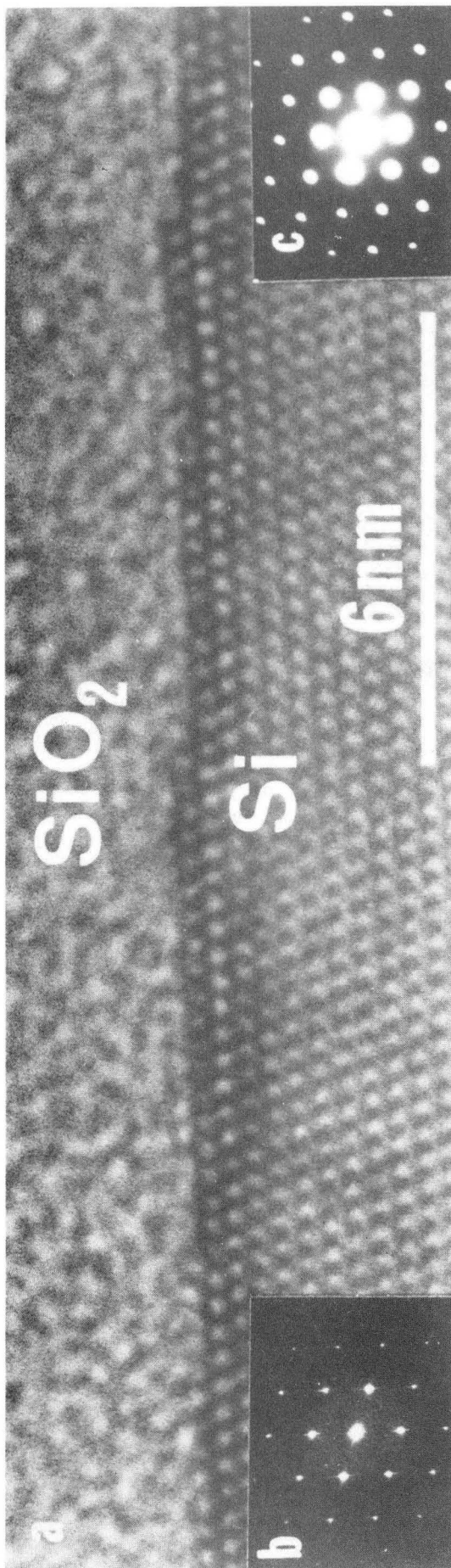


Figure 6.

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