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Restructuring and Activation of Cu(111) under Electrocatalytic Reduction Conditions

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Abstract: The dynamic restructuring of Cu surfaces in electroreduction conditions is of fundamental interest in major electrocatalysistic reactions. We decode the structural dynamics of a Cu-(111) electrode under electrocatalytic reduction conditions by joint first-principles calculations and operando electrochemical scanning tunneling microscopy (ECSTM) experiments. Combining global optimization and grand canonical density functional theory, we unravel the potential- and pH-dependent restructuring of Cu-(111) in acidic electrolyte. At reductive potential, Cu-(111) is covered by a high density of H atoms and, below a threshold potential, Cu adatoms are formed on the surface in a (4×4) superstructure, a restructuring highly unfavorable in vacuum. The strong H adsorption is driving force for the restructuring, itself induced by the electrode potential. On the restructured surface, electrochemical barriers for the hydrogen evolution stion steps are low so that the catalyst surface is not sta but in constant kinetic evolution and ECSTM produc a tim uring average image. <u>R</u>Therefore, restrug electroreduction conditions creates highly act Su ad 6m sites not present on pristine-Cu-(111).

Introduction

Electrocatalysis is one of the pillars of a fu sustainable energy landscape, enabling che storage of renewable energy and allo production of valuable molecules^[1], interaction with metal electrode electrodes V is of fundamental electrocatalysis, importan evolution reaction where hydrogen (HER) plays a central role^[2]. is both a highly desired electrical energy into reaction to transfo the form of molecular chemical energy in water electrolysis and an unwanted for many other electrochemical hydrogen in side reaction example, s^[4] and metal electroplating^[3], processes, for electrosynthesis and electrochemical COa (CO₂RR)^[1b, 5]. For the case of reduction reaction CO₂RR, to date, is proven to be the only transition metal that can catalyze CO_2 to valuable multi-carbon products like ethylene and ethanol^[6]. Experimental evidence suggests that on the one hand, the structure of the Cu catalyst is of key importance in controlling the selectivity between

as products^[7], H₂ and hydro bon SD meanwhile, undergoes dramatic the Surface restructuring durin electrochemical the tion^[7-8] reduction However. these restructuring ev ts, and their impact on the catalytic are far from being perfo e. understood tod

One intriguing example is the Cu–(111) surface in acidic conditions. During a negative potential sweep, a sange of a ucture of the surface has been observed by operando STM at the onset of HER. A superstructure is seen, associated with a (4×4) unit cell with respect to the u–(111) 1×1 lattice^[9]. In addition, a unique redox to the u–0.35 V vs RHE has been found on Cu–(111)

oltammetry cycles, while no such distinct Surrac signatures were found on Cu (100) and Cu (110) surfaces^[10]. Several hypotheses have been proposed to explain these phenomena, such as H intercalation in the Cursurface to form a hydride compound or H-mediated del of this restructuring of Cu—(111) in rest m ectroreduction conditions has been established yet. ery recent work by Sargent et al^[11] showed a remarkable performance of Cu catalyst in strong acid electrolytes toward CO₂ conversion to multi-carbon products, indicating the feasibility and prospects of acidic CO₂RR in the future. However, the lack of knowledge of surface restructuring impedes the rational design of improved catalytic active sites for acidic CO₂RR^[12]. Modeling studies for HER in the literature are performed on conventional non-reconstructed surface models such as the Cu-(111) surface or stepped surfaces^[13], but this approach is not sufficient to describe the realistic surface state under operational conditions. As a result, the impact of surface restructuring on HER electrocatalytic activity, promoting decreasing it, is unknown. Furthermore, the or electrochemical interface is often approximated by a neutral, non-charged interface. However, the dynamic behavior of the adsorbates on the electrocatalyst strongly depends on the electrode potential and solution pH, which cannot be practically handled by such methods.

Here, we explore the potential-dependent restructuring of the Cu-(111) surface induced by H adsorption under electrochemical reduction in acidic conditions by a combination of first-

principles simulations and operando electrochemical scanning tunneling microscopy (ECSTM) experiments. Although the general motivation for the choice of Cu(111) is CO2RR, we will restrict ourselves here to mildly negative potential, before the onset for hydrocarbon product formation, since in situ STM data can be obtained for these experimental conditions. The surface configurations are efficiently sampled by a grand canonical genetic algorithm (GCGA) global optimization method, and grand canonical density functional theory calculations (GCDFT) permitting the explicit potential modelling by surface charging in an implicit electrolyte. Specific surface structures found stable at reducing potential, were corresponding to a (4×4) supercell reconstruction bearing one additional Cu adatom, decorated and stabilized by high coverage of H adsorbates. The simulated STM images match the potentialdependent experimental operando STM images, enabling a detailed interpretation of the underlying surface structures and validating that the theoretically proposed restructured surfaces are the ones occurring on Cu-(111). Jointly, these approaches reveal, upon shifting the potential to a more negative value, the formation first of a (4×4) H adlayer, followed by Cu restructuring by adatom formation below a threshold potential (seen experimentally at -0.32 V vs SHE when pH Higher рΗ conditions 2). are explo computationally and are shown to decrease the H coverage and to impede the reconstruction. Furthermore, the stronger adsorption of H restructured Cu surface is shown to be th force for the Cu adatom formation. M hani studies using a hybrid solvent model Show HER steps are fast on the restructured Cu so that on the one hand, the Cu ada highly active for HER, and on the other hand surface structure is involved in fast kinet transformations between the various intermediates for the reaction. Operando S provides a time-average imaging, of this dyna

Results and Discussion

reactive surface.

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Determining the configurations of the Cu-H surface at given electrochemical conditions is a significant challenge since the hydrogen coverage on the Cu surface varies with the pH and electrode potential, and the surface possesses a largenumber of degrees of freedom for possible restructure Here, we explored the surface ensemble of configurat for Cu-(111) in electrocatalytic conditions with vario hydrogen coverage using GCGA^[14] n effici t global optimization ructures at a fixed method aimed at explor chemical potential rather that fixed composition. hay be hard to Since the pristine, 11) surfac restructure, we se ed the sea with several initial Cu configurations wi varying dens es of Cu adatoms and vacancies details, methods and Cu for see Supplementary F re 1-8).

The GCGA etho employ here uses a hydrogen reservoir, and desi minimize the coveragedependent form on free energy (Ω_f):

Σ

U.

 $\Omega_f = U$

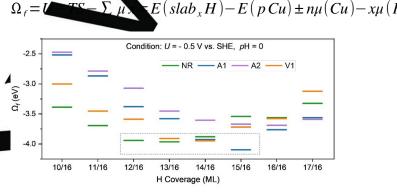


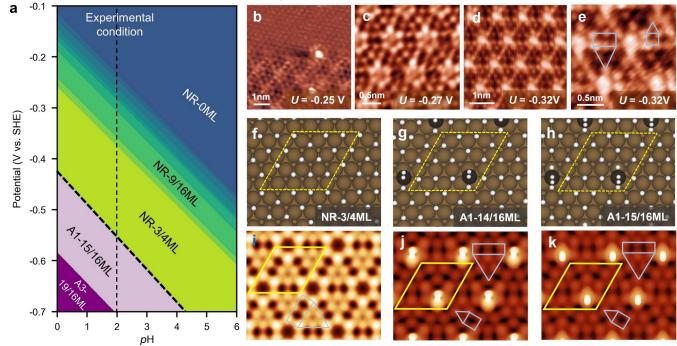
Figure 1. Thermodynamic driving force for Cu (111) restructuring under electro-reduction ditions: Grand Canonical formation free energy ²_f) as a function of H coverage at pH = 0 under a otential of -0.5 V vs SHE. Each energy level corresponds to the global minima(GM) at each coverage for the considered surface structure classes, all using a (4×4) supercell of Cu-(111): NR: non-reconstructed; A1: with one Cu adatom; A2: with a Cu adatom-dimer; V1: with one Cu vacancy in the surface layer. The dotted box frames the grand-canonical (i.e., variable H coverage) low energy structures, including the most stable (4×4) Cu surface with one adatom (A1) and a 15/16 ML H coverage.

where the $E(slab_x H)$ is the electronic energy of the optimized H-covered structure, E(pCu) is the energy of pristine non-reconstructed Cu-(111), n is the number of adatoms or vacancies, denoted as '-' and '+', respectively. $\mu(Cu)$ is the Cu chemical potential derived from Cu bulk. X is the number of hydrogen and $\mu(H)$ is the H chemical potential, which is dependent on pH and electrode potential, U. For simplicity, a series of GCGA has been performed on the different substrates using model conditions of -0.5 V vs SHE and pH of 0, keeping the (4×4) unit cell captured by STM imaging. More detailed potential and pH dependence will be described later. In total, ~10,000 structures were

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sampled in a broad stoichiometric distribution for Cu_nH_x system. After GCGA sampling, we performed accurate calculations on the global minima (GM) and multiple metastable structures (MS) using grand canonical DFT

surface with the formation of one Cu adatom and an H coverage of 15/16 ML in the (4×4) unit cell is globally the most favored configuration at the selected potential of -0.5 V. This is remarkable



(GCDFT) calculations^[15], i.e. explicitly including the charge polarization resulting from the applied potential combined with an implicit model of the water solvand the electrolyte (see methods).

The obtained formation free energies (Ω are shown in Figure 1. For clarity and simplig we only include in Figure 1 the mos stable configuration found for each importan class initial Cu configurations (NR = non-reg A1, A2: surface with 1 or 2 adatoms with 1 Cu vacancy in the surface plane) considered H coverage, while all MSs for class are included in Supplementary Information Configurations with 3 adatoms or with 2 to 4 vacancies were found noncompetitive and are included in Figure 1 (see Supplementary Figu and 8). Ω_f plots show a conve hull as a fur of H coverage for each class although different the structur nodels optimum H coverage for correspond to different the considered While the pristine conditio ns. surface shows an optimal con of 13/16ML at s with vacancie -0.5 V vs SHE, surf or adatoms of H, 14/16, 15/16, and prefer larger amour and A2 structure classes, H coverage, Cu—(111) 16/16 ML, for V1, A1, respectively low energe tically favorable, but restructuring is not 14/16 ML a crossover when coverag cturing by adatom or vacancy appears and res preferred. The restructured formation become

since, as we will discuss later, formation of such defects on Cu–(111) in vacuum is significantly endothermic.

examining these surface structures, we Before will first determine how their stability depends on electro-reduction conditions, evaluate accurately their potential dependence, and make the link with experimental characterization by operando ST . This will allow us to understand how the Cu (1) surface dynamically responds to the ectrode potential and the solution pH, and how he surface configurations under different H coverages can be accessed. The free energy of each most stable surface structure at different H coverages is calculated in the grand canonical ensemble of electrons and adsorbates at various pH and electrode potentials using the surface charging method, which enables us to obtain the potential-dependent electronic free energy of the surface.

In acidic conditions, the first H adsorption and surface restructuring occur at weakly negative potential (Figure 2a). Let us start again at model pH = 0 condition and describe the potential dependence: the proton begins to adsorb on pristine Cu-(111) at -0.12 V vs SHE. As the potential becomes more negative, H coverage increases until -0.26 V where an organized H adlayer

Figure 2. Evolution of the structure of Cu-(111) in electroreduction conditions. (a) Surface Pourbaix diagram assuming global thermodynamic equilibrium. The restructured region is shown in pink. Experimental operando STM images of the Cu-(111) surface during HER in 5 mM H₂SO₄ solution: (b) working electrode potential U = -0.21 V vs SHE (top part), U = -0.25 V vs SHE (bottom part) (tip bias voltage $U_b = 15$ mV; tunnel current $I_t = 10$ nA); (c) U = -0.27 V vs SHE ($U_b = 18$ mV; $I_t = 12$ nA) (d, e) U = -0.32V vs SHE ($U_b = 335$ mV; $I_t = 10$ nA). Images c, d and e have been slightly rotated to align the dense

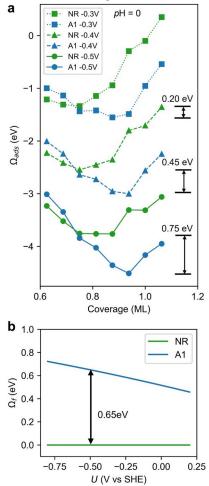
Cu row with that of the calculations. Top view of simulated atomic structures of (f) NR-3/4ML; (g) A1-14/16ML; (h) A1-15/16ML. Color code: terrace Cu: brown, Cu adatom: black, H: white. Simulated STM images for (i) NR-3/4ML, (j) A1-14/16ML, (k) A1-15/16ML, the bias voltages (U_{bias}) in the STM simulations are all 0.05 V.

corresponding to 3/4 ML coverage is formed, combining quasi-square and triangular н adsorption patterns, and remains stable within a large potential range. When the electrode potential reaches -0.42 V vs SHE, the Cu surface is found to restructure to form one Cu adatom in each (4×4) unit cell (structure A1-15). This structure remains stable from -0.42 V to -0.58 V vs SHE, which is a potential window for HER on the Cu-(111) surface in an acidic solution. At a more negative potential, calculations predict further restructuring toward adatom trimers and a coverage of H larger than 1 (19/16 ML).

If the pH is increased, the H chemical potential is decreased, so the H⁺ reduction to provide adsorbed H atoms becomes more difficult and the H coverage decreases at a fixed potential. All domains in Figure 2a shift to more negative potentials to restore the H coverage required for their formation. Therefore, the reconstruction leading to Cu adatom formation will require more negative potential to occur at higher pH (by ~-0.065 V by pH unit).

Let us now specifically consider pH=2, for which t operando STM experiments were performed. One se hexagonal arrangement corresponding to the symmetry of the Cu-(111) surface structure at a potential of <u>0.21</u> V vs SHE (Figure 2b, top part). This does not me h that the surface is free of adsorbates, and it can b covered with highly mobile H not visible with the ST the h 4 cove mobility being expected at low to medium ģe as shown by calculations. At U = -0.25 first organized (4×4) superstructure is found, triangular and square arrangements with a hexag arrangement of bright bumps at the apexes (Figure 2b) bottom part, and 2c). A second superstructure is seen at U = -0.32 V, the main difference being the presence a large bright bump in the center e hexagon (Fi íre 2d and e). The direct interpretation e STM ir iges the is not straightforward, but the comp calculated structures and their simulated Images STM image of the first provides a clear picture. organized structure that ex ntally occurs at the 6 V (Figure. onset potential of -Q rresponds to the simulated image of t calculated ML structure urface is not restructured and (Figure 2f and i). The C the STM contrast cor ponds to a well-defined configuration of the H atom n a (4×4) unit cell (Figure 2i). The H aton t hollow sites on the Cu (111) surface a is associated with a er maximum in the age, where the pattern combines triangular and qua quare motifs, and one quasihexag cell. Hence for that overlayer structure ly assign the contrast in the STM image to the position of H adsorbates. At more negative potential, the Cu surface restructures toward formation of a Cu adatom, giving the A1-15/16 ML structure (Figure 2h). The Cu adatom appears as the main bright feature in the ima while H adatoms provide a secondary pattern which lar to that of the NR-3/4ML image (Figure 2k) adatom is The located in the quasi-hex ngement of H atoms, offering new binding sites atoms with respect to the NR-3/4ML structure. One H a not directly seen in the STM image, is b ing the adate and the surface, and an activated olecular H₂ it is present on the Cu A distance of adatom with a H 804 Å. The observed h from the Cu and H STM contrast the efore results be adatoms. The ex rimental (-0.7 V) and theoretical (ntial wh 0.55 V at ph e such restructuring of 2) forma ccurs (A1 structure) are Cu by adato d the small difference (0.23 V) likely slightly different results from a co bination of approximations in the odelling theore e.g., DFT exchange correlation functional, solver ectrolyte continuum models). A benchmark study on the potential-induced change of configuration of pyridine on Au (111) using the same proach g e an error in the transition potential of 0.3 value Smilar to the difference seen here^[15c].

by a adatom formation is in the HER active region so



that the surface of the catalyst is not static but catalyzes the HER (vide infra). At even more negative potential,

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fast hydrogen evolution makes STM imaging very challenging, and the range of potential where further restructuration forming Cu adatom trimers is predicted from calculations could not be experimentally explored.

Next, we can interrogate the driving force producing the reconstruction toward Cu adatoms. Indeed, the formation of Cu adatoms on the bare Cu-(111) surface is endothermic by about 0.6 -0.65 eV in the considered potential range (Figure 3b). The adsorption free energy of H atoms on the Cu surface, resulting from the electroreduction of protons, depends both on the electrochemical potential and on the coverage (Figure 3a). The H adsorption is first stabilized with increasing H coverage reaching an optimum, which depends on the Cu surface structure. When the electrode potential becomes more negative, the - н adsorption becomes stronger and the H coverage generally increases. However, for the pristine (111) surface the H coverage reaches a stable 3/4 ML value at -0.26 V with a triangle-square pattern (Figure 2f), and does not increase further until the potential of -0.5 V. Indeed, it is difficult to break this pattern and add other H atoms. Adatom formation creates new sites for H adsorption, since the adatom itself presents a quite low coordination number (3). Hence, the defective surface with the Cu adatom can adsorb more H atoms and provides a more favorable adsorption free energy at equilibrium. difference between the optimum adsorption energy for the pristine (NR) surface and the restructured one increases with the ential (Figure 3a). At low potential, this diffe ntial adsorption energy is not large e bugh compensate the cost of Cu adatom format However, for a potential more negative threshold value, the gain in H adsorption surpasses the surface rearrangement cost, the restructuring becomes thermodynamicall favored. We therefore conclude that the catal restructuring is resulting from an optimization Cu surface sites to adsorb H and is hence indu ed by H adsorption, itself induced by the reduc tive potential.

The relatively weak cohesive energy of Cu compared with other metals is an important factor

that affects its restructuring in response to H adsorption^[17]. The chemical potential of proton can be modified by solution pH and electrode potential. More negative potential and lower pH will lead to a stronger H adsorption and render the formation of the Cu adatom easier. Thus, regulating solution pH and applied potential can tune the adsorption strength of adsorbates and then induce the defects formation. Further research can extend this design principle to other metals and adsorbates beyond the Cu-H.

considered proton Until this poj we hay sorption. Ho electrochemical ever, the considered potential (~ -0.3 0.5 V vs Figure 3. Compared stabilizat on by H adsorption Cu-(111) NR and (4×4) unit of non-restructi cell with a Cu ada om A1 Adsorption free energy as the fund of H coverage at different entials and pH = 0. (b) Potentialelectrochemical dependent formation nergy of adatom structure in the absence of

SHE) is a typical operational potential for HER in acidic solution^[9]. The copper surface will undergo a rapid H undergo a rapid H undergo with the solution through multiple Volmer-He rovsky-Tafel elementary steps to achieve the kinetic reaction steady-state. We now explore the reactive evolution of the restructured Cu surface (starting from the truther A1-15, denoting the GM of the A1 surface that contains 15 adsorbed H atoms, Figure 4a). Using a

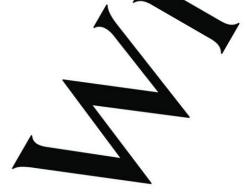
hybrid solvent model, combining explicit water molecules and implicit dielectric continuum, we evaluate the potential-dependent electrochemical reaction barriers of the various HER elementary steps of the menianism of Figure 4a, belonging to three classes:

 $\dot{\iota} + H^{+\dot{\iota} + e^{-\dot{\iota} \rightarrow *H\dot{\iota}}}$ (Volmer)

 $i H + H^{+i + e^{-i \rightarrow * + H_1 i}}$ (Heyrovsky)

 $i H + i H \rightarrow H_2$ (Tafel)

where the * and *H represent the active site and the adsorbed H on the surface, respectively. Since the solution is acidic, solvated H^+ rather than H_2O , is considered as the proton source.



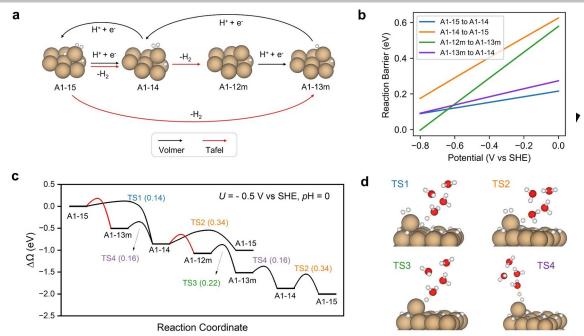


Figure 4. Reactive evolution of the A1-15 structure. (a) Proposed simplified reaction network on A1-15. (b) The reaction barriers for the four proton-coupled electron transfer steps as a function of the potential calculated using the hybrid explict-implicit solvent model. (c) Free energy diagram under - 0.5V vs SHE at pH = 0. (d) Transition state (TS) structures for A1-15 to A1-14 (TS1), A1-14 to A1-15 (TS2), A1-12m to A1-13m (TS3) and A1-13m to A1-14 (TS4).

Starting from A1-15, the desorption of the unit is slightly endothermic with a low (uphill) desorption internal energy of 0.23 eV at -0. SHE, but since the H₂ molecule gains upon desorption, the H₂ desorption re tion exergonic with a (stabilizing) free ener eV. This process was found to sho barrier on the potential energy surf the positive 0.23 eV desorption energy, desorption should be fast the at temperature. When the H₂ molecule undergoes the Tafel step and desorbs from the adatom, H in a bri-site between adatom and diffuses on top of the adate Supplemer om Figure 17) and a metastable A is formed (denoted as A1-13m). The tom adsorbed on top of the adatom wi uickly capture one proton fr the solution and an m electron from the to form an adsorbed H₂ surface 1-14 structure. molecule and the This Volmer step exhibits only a 16 eV barr energy and Qure 4b and d). Once the structure of TS4 on A1-14 structure is form it will either accept a 5 or undergo a Tafel proton and yo to A the H₂ molecule, to step, that is the sorption structure (denoted as form a metasta A1-12m). From the driving force of the negative and the ctivity of the Cu adatom potential 12m will rapidly undergo towards proto three successive three successive Volmer steps to form the A1-13m and A1-14 intermediates and finally return to A1-15 with proton transfer barriers of 0.22 eV, 0.16 eV and 0.34 eV, respectively (Figure 4 b, d).

Finally, Ad-15 is likely to form A1-14 via the Heyrovsky step on the bri-site H as well. When the proton diffuses from the solution and combines with the *bri*-site H, the adsorbed H_2 molecule on the Cu adatom desorbs and the newly formed H_2 molecule from the reacting proton takes its position. Other pathways that may occur on the A1-15 isomer are discussed in the Supplementary Note 1.

Based on the energy profile (Figure 4c), it can be clearly seen that the reaction barriers for the protoncoupled electron transfer steps on the Cu adatom are low, suggesting that the catalyst dynamically rearranges in electroreduction conditions, hopping between different H coverages, linked by Volmer or Tafel reaction elementary steps, and including the evolution of gas phase H₂. The catalyst surface kinetically evolves along several competing pathways and performs the HER reaction. Low barriers also imply a quick rearrangement, especially for TS1 and TS2, with fast A1-15/A1-14/A1-15 interchange. а Other transformations involving metastable isomers A1-13m and A1-12m are also possible. Altogether, compared to the relatively high barriers (0.5-0.6 eV at -0.5 V vs SHE) of the typical predominant Volmer-Heyrovsky steps on Cu-(111) terrace sites from previous work^[13c] and from our calculation (Supplementary Figure 24 and 25), the restructured surface with Cu adatoms appears highly active and is proposed to play a key role for the HER activity. Therefore, the reaction conditions created the restructured Cu surface with its highly active site at the Cu adatom, which can be summarized as the "reaction created the catalyst".

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Δ key consequence of the dynamic rearrangement of the н adsorbates in electroreduction conditions is that the surface imaged by the STM is not static and that the STM contrast results from averaging between several H configurations (A1-15, A1-14, A1-13m and A1-12m) that rapidly exchange on the surface. Indeed, STM imaging, which necessitates scanning the tip over the image area, is a rather slow technique and the acquisition of one frame requires ~tens of seconds. However, our STM simulations show that all these configurations produce rather similar STM images because the STM contrast is dominated by the presence of the Cu adatom and the organization of H atoms on the terrace (Supplementary Figure 18). As a result, the dynamics created by the HER reactions step for the H atoms near the Cu adatom do not affect the image significantly and a well-defined STM image can be recorded.

Conclusion

combination Through theoretical а of calculations and operando ECSTM experiments, we show that the Cu-(111) electrode restructures under electroreduction conditions in acidic electrolyte. Based on global optimization, grand canonical DFT calculations, and STM image simulations compared to the experimental ones, we obtain an atomic scale understanding of the transformation of the Cu-(111) electrocatal **pH** and a function of applied potential, solution related hydrogen coverage. Upon decrea e of th potential, we see first a (4×4) H adlayer with H adsorbates in the unit cell organize structure combining triangular and patterns, followed by a restructuring of the m catalyst itself by creation of a (4×4) array of low coordination Cu adatoms on the surface, seen (a potential of -0.32 V vs SHE at pH = 2 in t experiment. Moreover, the strong H adsorpt at negative potential, associated with the igh coverage, is shown to be the driving for this restructuring. The restructured surface with Cu adatom shows lower barriers for the steps of HER than that of the Cu-(111) surface, and plays a major role in the HER activity. Therefore, electroreduction conditions stabilize adatoms at the Cu surface, creating highly active sites not present on the initial catalyst. The thermodynamic equilibrium approach used here could be completed in the future by kinetic steady state simulations of the

formation of the restructured surface during reaction. This work solves the long-term controversy of the atomic structure transformation of the Cu (111) surface during the electrocatalytic process. Such restructuring originates from the low cohesive energy of Cu and strong H adsorption under reduction potential and at low pH. The results can be generalized to the restructuring of other transition metals since the chemical

potential of H is determined by the electrode potential and solution pH, and the restructuring of soft coinage metals can thus be precisely controlled by these two parameters in electrocatalysis. HER is often an unwanted work, it could be reaction, and, based on the avoided by a well-chosen pH that suppresses the restructuring favorable for the HER activity. HER activity. Finally, the approachican b extended to other electrochemical reaction d other adsorbates such as *CO in electrochemical CO₂ reduction reaction, and ***OH o**r *0 electro-oxidation in reactions. A milar surface roughening by formation of atoms or all islands can be expected in the e cases.

Acknowledgments

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Keywords: Global optimization • Grand canonical density functional theory • Scanning tunnelling microscopy • Surface restructuring • Hydrogen evolution reaction

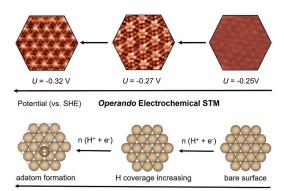
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Entry for the Table of Contents



More negative potential in acidic solution

Potential- and pH- dependent restructuring of the Cu-(111) surface induced by H adsorption under electrochemical reduction in acidic conditions is decoded by a combination of <u>a</u>grand canonical ensemble representation of surface state<u>s</u> and operando electrochemical STM experiments.

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