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January 1979

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WETTING AND ADHERENCE OF Na-BORATE GLASS ON GOLD

Stephen T. Tso and Joseph A. Pask

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INTRODUCTION

Previous work in this laboratory on the wetting behavior of sodium borate 'solder' glass (nominally 94% B_2O_3 and 6% $\text{Na}_2\text{O}^{\star}$) on platinum has been reported.¹ It was found that the Pt surface easily adsorbed carbonaceous contaminants causing a reduction of its surface energy which affected the sessile drop experiments. It was also found that the glass readily adsorbed water vapor resulting in a reaction between the glass and Pt which. caused spreading of the glass and development of adherence. The question arose as to whether this glass would behave similarly on gold. This report presents the results of the study.

EXPERIMENTAL PROCEDURE

Gold plate of 99.95% purity and 0.02 in. thickness was cut into 0.5 in. x 0.5 in. squares. The squares were polished through a set of dry polishing papers and given a final polish on a lap wheel with 0.3 micron α -alumina powder. They were cleaned with an ultrasonic cleaner, degreased in acetone, rinsed with alcohol and distilled water, and stored in a desiccator before use.

Based on thesis submitted by Stephen T. Tso for Master of Science in Materials Science, University of California at Berkeley, December 1976. * Obtained from Dow Chemical Co., Walnut Creek, CA. [†]Obtained from Lawrence Livermore Laboratory.

The previously mentioned glass (composition given in Ref. 1) was used in this study. Since it usually contained a great deal of dissolved water, it was vacuum degassed (2 x 10^{-6} torr) for about one day at 900°C in an alumina crucible until the glass ceased to bubble. Only the central portion of the degassed glass was stored in a desiccator until ready to use. From the original glass, about 3% sodium oxide boiled off while \sim 0.3% alumina diffused into the glass from the crucible during the process of vacuum degassing. Use of glass without degassing always resulted in bubbling during the sessile drop experiments.

Two set-ups were used for the sessile drop experiments. The first set-up contained a tantalum tube resistance heating element, which was described previously.² The total pressure in the chamber was about 2 $x 10^{-6}$ torr. The second set-up consisted of a one-inch diameter fused silica tube, a portion of which was placed in the interior of a clam shell Kanthal resistance wire furnace.¹ This arrangement permitted experiments with atmosphere control without its exposure to the heating element.

The nitrogen gas used had a purity of 99.999%. The oxygen gas used was 99.6% pure; its major impurities were Argon (4000 ppm max.), nitrogen (500 ppm max.), and carbon dioxide (1.5 ppm). Water vapor was introduced into the furnace tube by aspiration. During the course of the experiments a liquid nitrogen cold trap was activated as required. Contact angles in both cases were measured at 900°C with an estimated accuracy of $\pm 2^{\circ}$ by sighting with a telescope. Sufficient time (~6h) was taken to assure that steady state contact angles had been achieved.

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RESULTS AND DISCUSSION

Sessile drop experiments of the sodium borate glass on platinum indicated that without a cold trap carbon adsorption occurred on the platinum due to carbonaceous vapor backstreaming from the oil diffusion pump. $\begin{smallmatrix}1&&1\end{smallmatrix}$ In order to determine whether the wetting behavior of the test glass on gold would indicate possible carbon contamination of the gold surface, specimens were treated as follows: (1) as received, (2) heattreated in air at 900°C for 8 hrs, and (3) heat-treated in vacuum $(10^{-8}$ atm) at 900°C immersed in carbon powder for 24 hrs. The observed contact angles in vacuum (10⁻⁸ atm) at 900°C with no cold trap decreased gradually but became constant at 46° after 3 hrs for *all* 3 metal specimens, and remained unchanged for an additional 10 hrs. This behavior was reproducible in the pressure range of 10^{-8} atm to 10^{-5} atm indicating that the gold surfaces had reached a constant structure irrespective of any initial differences that may have existed. In order to eliminate the possibility that *all* of the gold surfaces might have had carbon contamination due to carbonaceous vapor backstreaming from the oil diffusion pump, an experiment was repeated with an activated liquid nitrogen cold trap using a gold specimen that was heat treated in air. A contact angle of 46° was obtained indicating that the gold surface was inert to carbon adsorption from carbonaceous vapor.

At a pressure of 500µm Hg water vapor, bubbles were produced at the gold/glass interface at 900°C. The glass drop vaporized completely in less than 2 hrs. The evaporation rate in oxygen was 2.8 x 10^{-7} g/sec-cm² and in water vapor, 1.8 x 10^{-4} g/sec-cm². The increased weight loss and bubbling suggests that the water vapor was absorbed by the glass and also

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reacted with the glass. Additional evidence for this reaction was an unsteady drop with a contact angle showing a range of 39 to 60°. The lack of development of adherence suggested that the bubbling was due to a boiling phenomenon and not a reaction with the substrate with evolution of a gaseous reaction product.

At an ambient pressure of 1 atm nitrogen, the wetting behavior was found to be the same as that in vacuum. Steady state contact angles of 46° were again obtained. Using Young's equation and a value of 80 ergs/ cm^2 for the glass³, $(\gamma_{\text{sv}}-\gamma_{\text{sl}})$ is 56 ergs/cm², or the surface energy of the substrate is reduced by the glass by this amount. Using the published value of 1370 ergs/cm² for gold⁴, γ_{s1} becomes 1314 ergs/cm².

At ambient pressures greater than 15µm Hg oxygen the contact angle was 6°; at lower pressures, 46°. The relation between the steady state contact angle and the partial pressure of oxygen is shown in Fig. 1. It was also observed that in the oxygen atmospheres the glass drops tended to migrate until they reached the edge of the gold substrate. This phenomenon was not observed in vacuum, nitrogen or water vapor. As the glass drop drifted, the whole drop did not move as a unit but a film of glass was left behind the drop like a tail, which was so thin that it was difficult to detect. When the gold pieces were bent, the films fractured indicating the presence of chemical adherence. In the absence of chemical bonding the glass pops off cleanly on bending the metal piece. The fact that in oxygen a film of glass formed behind the moving glass drop that was contiguous with the drop indicates that its composition was different from the bulk glass since the glass would otherwise retain its drop shape. It is postulated that this film is approaching

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saturation with gold oxide at the interface because of the appearance of good adherence. These quantities are too small and too restricted to be determined by present analytical tools. Deterioration of adherence occurred with prolonged heating and was attributed to diffusion of gold oxide into the bulk glass resulting in loss of saturation at the interface. The edge of the glass drop or film, however, still retained its adherence due to some diffusion of oxygen along the interface from the gold/glass/ . air periphery line.

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The decrease of contact angle to 6° in oxygen atmospheres indicates that $({\bf \gamma}_{_{\bf S}{\bf v}}-{\bf \gamma}_{_{\bf S}1})$ is 80 ergs/cm 2 since oxygen has little effect on the sur-5 face energy of liquid glass. Oxygen, however, chemisorbs on gold surto some diffusion of oxygen along the interface from
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The decrease of contact angle to 6° in oxygen atmos:
 $(Y_{\text{SV}}-Y_{\text{S1}})$ is 80 ergs/cm² since oxygen has little efi
energy of liquid glass.⁵ Oxygen, faces⁶ and using a value of 1210 ergs/cm² for γ_{cv} for gold in oxygen,⁴ $Y_{\rm c1}$ becomes 1130 ergs/cm². The decrease of $Y_{\rm c1}$ by 184 ergs/cm² on exposure to oxygen occurs with the development of. chemical bonding. This deduction is consistent with previous findings^{7,8} and with the theory that adherence occurs when the glass and metal at the interface are saturated with the substrate metal oxide.⁹

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FIGURES

Fig. 1. Contact angle vs. oxygen pressure at 900°C for Na-borate glass on gold.

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