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# Voltage tuning of vibrational mode energies in single-molecule junctions

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**Vibrational modes of molecules are fundamental properties determined by intramolecular bonding, atomic masses, and molecular geometry, and often serve as important channels for dissipation in nanoscale processes. Although single-molecule junctions have been used to manipulate electronic structure and related functional properties of molecules, electrical control of vibrational mode energies has remained elusive. Here we use simultaneous transport and surface-enhanced Raman spectroscopy measurements to demonstrate large, reversible, voltage-driven shifts of vibrational mode energies of C<sub>60</sub> molecules in gold junctions. C<sub>60</sub> mode energies are found to vary approximately quadratically with bias, but in a manner inconsistent with a simple vibrational Stark effect. Our theoretical model instead suggests that the mode shifts are a signature of bias-driven addition of electronic charge to the molecule. These results imply that voltage-controlled tuning of vibrational modes is a general phenomenon at metal–molecule interfaces and is a means of achieving significant shifts in vibrational energies relative to a pure Stark effect.**

plasmonics | nanoscale junctions | molecular electronics

**M**echanical couplings between atoms within molecules, manifested through vibrational spectra, are critically important in many processes at the nanoscale, from energy dissipation to chemical reactions. These couplings originate from the self-consistent electronic structure and ionic positions within the molecule (1). Vibrational spectroscopy examines this bonding, and advanced time-resolved techniques (2–5) can manipulate vibrational populations. Single-molecule junctions (6, 7) also have proven to be valuable tools for examining vibrational physics. Previous work showed that vibrational frequencies may be altered in mechanical break junctions if the chemical linkage to the moving contacts is sufficient to strain bonds in the molecule (8, 9), but also showed vibrations to be unaffected when the linkage to the contacts is less robust (10). Controllably altering vibrational energies in the steady state is difficult, however. Electric fields can redistribute the molecular electron density and shift vibrational modes in the vibrational Stark effect (11), enabling spectroscopic probes of local static electric fields in charge double layers (12, 13) and biosystems (14–16). However, other physics also may be relevant, and studies of electrical tuning of molecular vibrational energies in single- or few-molecule-based solid-state junctions, which often provide clarity that is difficult to obtain from measurements of molecular ensembles, have been lacking.

Surface-enhanced Raman spectroscopy (SERS) (17, 18), in which surface plasmons enhance the Raman scattering rate for molecules, opens up the possibility of performing detailed vibrational studies at the single-molecule level. Plasmonic junctions between extended electrodes (19–25) show correlations of Raman response and conductance implying single- or few-molecule sensitivity, and enable studies of vibrational physics as a function of electrical bias. Spectral diffusion often is observed in single-molecule SERS experiments (18, 25, 26), and there is some preliminary evidence of bias-driven mode shifts in such junctions (23), with the mechanisms of these phenomena remaining unclear.

We report vibrational mode softening in C<sub>60</sub> molecules on the order of tens of wavenumbers, approximately quadratic in the external dc bias,  $V$ , applied across such a junction. We compare these observations with density functional theory (DFT) calculations to determine the underlying mechanism. The calculations suggest that the systematic softening, its magnitude, and its detailed functional dependence on  $V$  are inconsistent with a pure vibrational Stark effect. Instead, changes in molecular charge with bias (27) result in vibrational shifts that closely resemble those observed in the experiments, both in magnitude and sign. This reveals a general physical mechanism, expected to have implications for other systems and measurements.

Fig. 1 shows a typical Raman spectrum from a C<sub>60</sub>-containing junction, prepared by electromigration (28) of a lithographically defined Au constriction on an oxidized Si substrate. This junction is a useful test system, because C<sub>60</sub> is known to adsorb sufficiently strongly to Au to allow the formation of reliable and reproducible junctions (29–31). The fabrication and detailed measurement procedures are discussed in *Materials and Methods*. The incident wavelength for the Raman measurements is 785 nm, and the extended electrode design allows the application of a dc bias,  $V$ , across the junction and the flow of current through it. The sharp mode at 520 cm<sup>-1</sup> is from the Si substrate, and the modes between 1,000 cm<sup>-1</sup> and 1,600 cm<sup>-1</sup> are broadly consistent with expectations from previous C<sub>60</sub> single-molecule SERS experiments and with our own calculations. We note that the close association of the molecule with the surface, necessary for SERS measurements, may result in chemical and symmetry changes that can turn previously Raman-inactive normal modes into active ones (32) and can lift mode degeneracies. Because each of our devices produces a unique Au junction possessing

## Significance

Like guitar strings, molecules have characteristic vibrational frequencies, set by the strength of chemical bonding between the atoms. In an experiment using a special antenna for light, we have found that applying an electrical voltage to a single buckyball molecule systematically lowers its vibrational frequencies, indicating that the bonds are weakened. We can explain this observation in terms of a very simple, general model in which the applied voltage slightly increases the amount of negative charge on the molecule, thus tuning the chemical bond strength. This may be generally useful in understanding and controlling the mechanical properties of molecules.

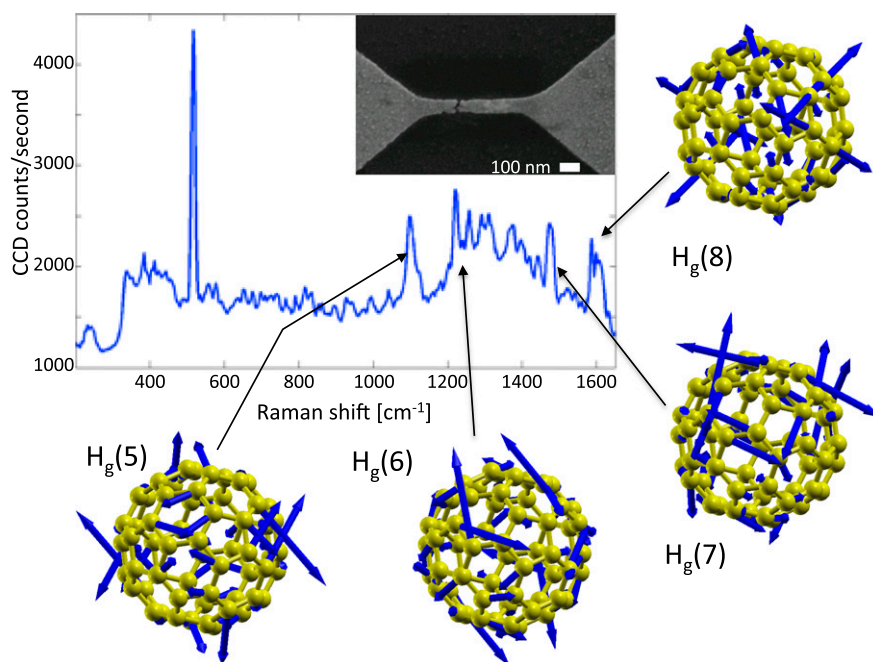
Author contributions: D.N. designed research; Y.L., P.D., and J.B.N. performed research; Y.L., P.D., L.K., J.B.N., and D.N. analyzed data; and Y.L., P.D., L.K., J.B.N., and D.N. wrote the paper.

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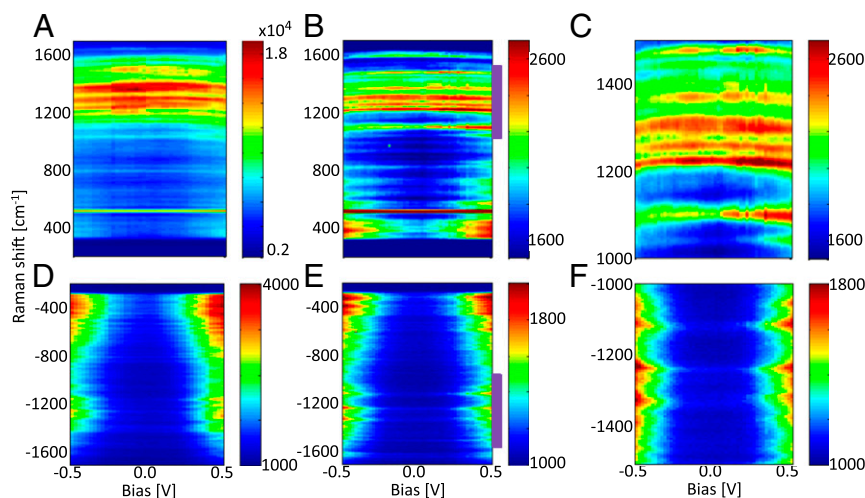
**Fig. 1.** Raman analysis of  $C_{60}$  in an electromigrated junction. (*Main Plot*) Example of a SERS spectrum of  $C_{60}$  in an electromigrated junction. Surrounding diagrams illustrate examples of the complicated displacements associated with Raman-active modes, calculated for an isolated, symmetric, gas-phase molecule. Each such mode is fivefold degenerate in the absence of symmetry breaking. (*Inset*) Scanning electron microscopic image of an electromigrated junction.

a specific molecular association, we track all normal modes of the molecules in the calculations. For the isolated  $C_{60}$  molecule, only the  $A_g$  and  $H_g$  modes are Raman active (they are labeled in Fig. 4D).

As in previous studies (20, 21, 23, 25, 33), correlations as a function of time are observed between spectral intensity fluctuations (“blinking”) and the measured conductance in the tunneling regime. Because the tunneling conductance is dominated by a molecular-scale volume at the point of closest interelectrode separation, these correlations imply few- or single-molecule Raman sensitivity. For this study, the conductances range from 0.1 to a few

$G_0 = 2e^2/h$  and include contributions from both through-molecule and direct metal–metal tunneling or contact in some junctions. These junctions are not in the Coulomb blockade regime.

Fig. 2 shows Stokes and anti-Stokes spectra as a function of applied bias for two representative devices. The main experimental observation is that many of the vibrational modes with energies greater than  $1,000\text{ cm}^{-1}$  shift toward low energies as the applied bias increases. These systematic shifts are observed routinely in  $C_{60}$ -based junctions, having been seen in the 12 of 23 junctions that produced a significant and stable SERS signal. The remaining 11 junctions had blinking (junction configuration



**Fig. 2.** Vibrational modes and their evolution with source-drain bias. (A) Stokes and (D) anti-Stokes spectra of device 1 as a function of  $V$ . (B) Stokes and (E) anti-Stokes spectra of device 2. Color scales indicate counts per integration time. (C and F) Rescaled close-ups of the device 2 data over the wavenumber ranges indicated by the purple bars in B and E, respectively. The vibrational modes curve slightly toward lower energies at larger  $|V|$ , and the anti-Stokes intensities increase at high biases. The latter effect indicates current-driven heating of vibrational degrees of freedom, as reported previously (22, 23). The spectral shifts are more difficult to resolve in the anti-Stokes case because of this evolution of anti-Stokes intensity.

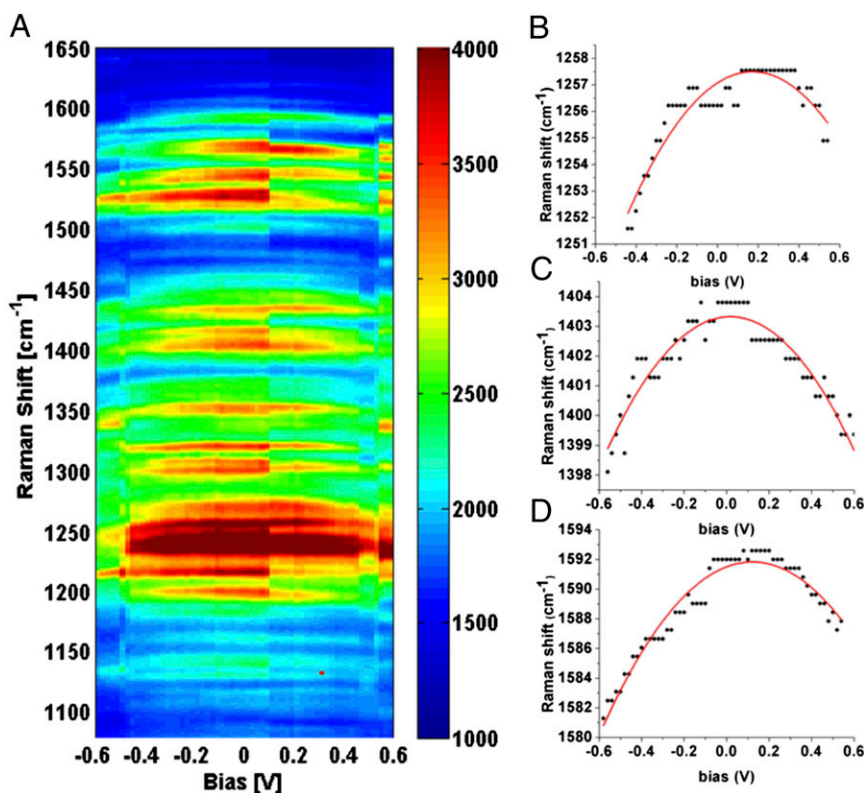
instability) sufficiently strong that it precluded the long measurements required for a clear assessment of bias-driven effects. This yield and variation are consistent with prior experiments in such junctions.

The bias-driven shifts, apparent as a curvature of the spectral features, vary in magnitude, from a few  $\text{cm}^{-1}$  to  $20 \text{ cm}^{-1}$ . Fig. 3 shows data from another device, using a higher-resolution grating in the spectrometer. This particular dataset shows clear discontinuities in the mode intensities at a few bias voltages; these are stochastic blinking, as described above. The bias-driven shifts on the devices are consistent with a quadratic dependence on applied bias,  $\delta\omega \sim V^2$ . Note that electromigration junction experiments do not control the molecule/metal contact geometry precisely at the atomic scale; variability in the contact geometry and molecular environment may give junction-to-junction variations in the precise Raman spectrum. However, the sign, functional form, and magnitude of the bias-driven shifts here are consistent and reproducible.

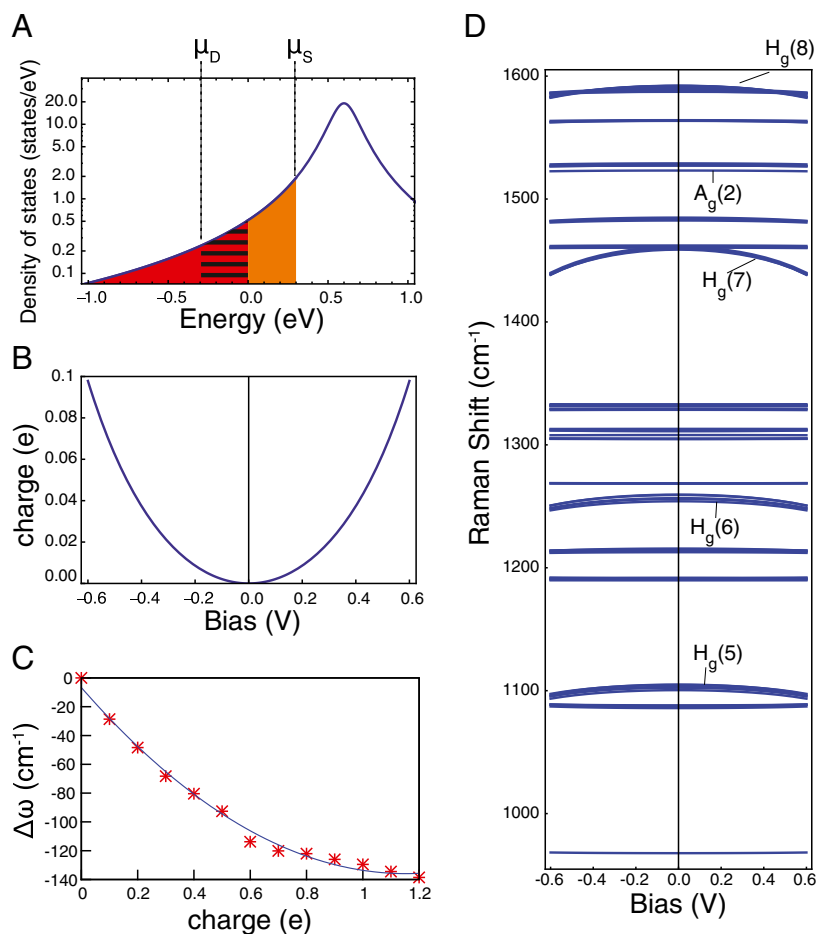
To understand the mechanism at work, we use DFT to compute the vibrational frequencies of  $\text{C}_{60}$  as a function of external field and charge state. In principle, the local field and charge state of a molecule in a junction depend on atomistic features of the molecule–metal contact that are highly complex, are generally unknown, and vary from device to device. Here, we neglect explicit treatment of the electrodes and instead model the  $\text{C}_{60}$  environment, as a function of bias, through changes in fields and steady-state occupation. Initially, we compute the vibrational frequencies of a gas-phase  $\text{C}_{60}$  molecule in the presence of constant electric fields. For several external fields up to  $1.2 \text{ V/nm}$  (approximately twice the range probed by the experiment), the  $\text{C}_{60}$  geometry is relaxed and the vibrational modes are computed within DFT for constant charge state. Both neutral  $\text{C}_{60}$  and the  $\text{C}_{60}^{-1}$

anion are considered. Mirjani et al. (34) recently considered the impact of full reduction or oxidation on vibrational modes of molecules in junctions, observing appreciable shifts relative to the neutral species. However, the lack of resonant transport (*Supporting Information*, Figs. S1–S3) confirms that the applied bias in this experiment is insufficient to fully change the average redox state of the molecule by an entire electron. Therefore, the anion represents the limit of charging possible in the system. For the neutral  $\text{C}_{60}$ , our calculations are in good agreement with previous theory and experiment (*Supporting Information*, Table S1). Calculated field-induced vibrational shifts for the neutral molecule (Fig. S4) (anion, Fig. S5) typically are far less than  $1 \text{ cm}^{-1}$  ( $5 \text{ cm}^{-1}$ ) in magnitude, at maximum field ( $1.2 \text{ V/nm}$ ); moreover, the shifts vary in sign and do not exhibit a generally quadratic functional form with field, in contrast to experiment. This rules out the well-known vibrational Stark effect as the origin of the observed phenomena.

A major clue toward an alternative explanation is that at constant field, differences between specific mode frequencies of the neutral and anion are large, of order  $10\text{--}150 \text{ cm}^{-1}$ , and notably, the affected anion modes are redshifted relative to the neutral molecule (*Supporting Information*). This suggests an explanation in terms of bias-driven changes of the  $\text{C}_{60}$  charge state. To explore this possibility, we recomputed the  $\text{C}_{60}$  vibrational spectrum, adding small fractions of an electron, from 0 to 1, in steps of  $0.1 e$ . In the case of significant hybridization between the molecule and metal contact, partially occupied states in the junction are expected. Furthermore, in an open system, a fractional number of electrons in DFT is defined via an ensemble of integer-electron states and interpreted as a time average of a fluctuating number of particles (35). Therefore, one may infer the effect of partial molecular charging in the junction from calculations of a single partially charged  $\text{C}_{60}$  molecule. A fractional



**Fig. 3.** Bias-driven vibrational energy shifts. (A) Raman response of device 3 as a function of bias ( $x$  axis) and Raman shift ( $y$  axis). The sudden change in the intensity at around  $0.1 \text{ V}$  is the result of blinking. (B–D) Vibrational energy shift as a function of bias for three particular modes:  $1,258 \text{ cm}^{-1}$ ,  $1,404 \text{ cm}^{-1}$ , and  $1,592 \text{ cm}^{-1}$ . The discretized Raman shift data result from pixilation of the detector.



**Fig. 4.** Model of bias-driven changes in molecular charging. (A) At zero bias, the triply degenerate LUMO resonance, centered at  $E_0$  with width  $\Gamma$ , is occupied proportionally by the red shading. As the bias  $V$  is applied, the molecular level gains additional occupation proportional to the area shown by the orange shading and loses occupation proportional to the hatched portion of the Lorentzian. (B) The expression for charging with bias at 80 K (solid) is visually identical to the charging at 0 K (for 300-K charging, see [Supporting Information](#)). The change in partial charge is approximately quadratic in bias. (C) A representative mode's [ $H_g(7)$  at  $1,467\text{ cm}^{-1}$ ] change in vibrational energy with charging, computed via DFT. This dependence, combined with the variation in charge with bias, strongly suggests that bias-driven charging is the origin of the systematic mode softening observed in the experiments. (D) Mode energies as a function of bias from such a calculation.

occupation of the  $C_{60}$  lowest unoccupied molecular orbital (LUMO) upon adsorption into a junction is consistent with the established large electronegativity of  $C_{60}$  (29, 36) and with scanning tunneling microscopy (STM) studies of  $C_{60}$  adsorbed on clean metal surfaces (37). As the molecule is (partially) charged, several  $C_{60}$  vibrational modes shift systematically to lower energies, by tens of centimeters $^{-1}$  (Fig. S6). We compute that these are Raman-active  $H_g$  modes\* that couple strongly to the  $t_{1u}$  LUMO (38–42) and are present throughout the 1,000–1,600- $\text{cm}^{-1}$  measurement range (Fig. 4). This trend is reasonable on general chemistry grounds: Adding an electron to the neutral  $C_{60}$  occupies an antibonding LUMO that is delocalized over the entire molecule, thereby softening many intramolecular bonds. Thus, a redshift of vibrational modes coupled to an antibonding LUMO upon electronic charging would be expected quite generally.

Only a relatively small amount of charging is necessary to result in the mode shifts seen here, and small changes in charge state are very plausible under bias. Fig. 4 shows a simple model for the energy level alignment of the junction. At zero applied

bias, the triply degenerate LUMO resonance will be positioned near the Fermi energy (37, 43), and broadened by its coupling to the source and drain electrodes. Assuming that electrons tunneling from the source to drain and drain to source, respectively; that they are noninteracting and therefore are occupied according to their original source or drain quasi-Fermi levels (44); and that the resonance line shape is Lorentzian with a width  $\Gamma = \Gamma_S + \Gamma_D$ , the change in steady-state occupation,  $\delta\rho$ , of a single triply degenerate level at energy  $E_0$  above the equilibrium Fermi level  $E_F$ , at bias  $V$ , may be expressed as (1, 27)

$$\delta\rho = \int_{-\infty}^{\infty} \frac{1}{2} g(E) \left[ f(E + eV/2) + f(E - eV/2) - 2f(E) \right] dE,$$

where, in this case, the density of states  $g(E) = (d\Gamma/\pi)/(\Gamma^2 + (E - E_0)^2)$  is Lorentzian with degeneracy  $d = 6$  in this case because of spin and orbital degeneracy and  $f(E) = 1/(e^{E/kT} + 1)$  is the Fermi–Dirac distribution function, with the zero-bias Fermi level  $E_F$  taken as the energy reference, i.e.,  $E_F = 0$ . Here a symmetrical voltage drop is assumed, with the source and drain chemical potentials taken to be  $\mu_S = eV/2$  and  $\mu_D = -eV/2$ , respectively. Recently, Kaasbjerg et al. (45) developed a nonequilibrium Green

\* $A_g$  modes also couple to the  $t_{1u}$  LUMO, but because the symmetric  $A_g$  modes do not break the LUMO degeneracy and therefore are not involved in the Jahn–Teller distortion (38–42), they vary less significantly with bias.

function framework for understanding bias-dependent molecular vibrational mode damping and heating in junctions. Including physics similar to what we consider here with our  $C_{60}$ -specific model, this approach also captures vibrational frequency renormalization associated with charging and screening (45), and is consistent with the work presented here in the limit where  $\Gamma$  is small compared with the resonance energy and bias voltage.

Previous STM experiments and DFT calculations of  $C_{60}$  on metal surfaces yielded  $\Gamma \sim 0.1$  eV and  $E_0 \sim 1.0$  eV (43). However, in a junction environment, where  $C_{60}$  is contacted on both sides with rough surfaces,  $E_0$  will be closer to  $E_F$  (37). Depending on the specifics of the Au- $C_{60}$  contact within a particular junction,  $E_0$  may vary somewhat. The value for  $\Gamma$  also will vary to some degree, but numerous experiments have shown significant coupling of  $C_{60}$  to Au. To demonstrate our reasoning, we take  $E_0 = 0.6$  eV above  $E_F$  at zero bias and  $\Gamma = 0.1$  eV. (The effect of finite temperature is shown in Fig. S7, and the effect of other choices for these parameters is explored in Fig. S8, but the general effect of charging is preserved). Together with the above model dependence of  $\delta\rho$  on  $V$  (Fig. 4B) and DFT-computed dependence of the frequency on  $\delta\rho$  (Fig. 4C), we compute the vibrational mode frequency as a function of bias for voltages up to  $\pm 0.6$  V, as shown in Fig. 4D. This simple model explains the measured mode softening trends.

For the choice of model parameters  $E_0$  and  $\Gamma$ , the finite temperature spread of the Fermi-Dirac distribution of the electrons in the source and drain has a negligible effect on the molecular charge at 80 K, but is more important at 300 K (Fig. S7). This suggests that any heating of the electronic distribution at high bias (23) also might play a role in determining the molecular charge and, hence, vibrational energies.

Using nanojunction-based SERS, we observe systematic bias-driven softening of vibrational modes in  $C_{60}$ . Comparisons with DFT calculations show that Stark physics alone cannot be responsible for these effects, and bias-driven alteration of the molecular charge state is the likely explanation. By combining realistic computational models of junctions with measurements of this type, the presence and degree of bias-induced mode softening can turn these junctions into a direct local probe of molecule/metal energetics. Interpreted in light of these observations, the earlier preliminary observations of bias-driven mode softening in a junction based on an oligo(phenylene vinylene) (OPV) molecule (23) suggest that in that particular device, the LUMO must lie close to the electrode Fermi levels. Indeed, recent calculations by Kaasbjerg et al. (45) arrive at similar conclusions regarding the origin of OPV mode shifts under bias observed in ref. 5. Similarly, it is worth considering whether much of the spectral diffusion observed in single-molecule Raman measurements results from small changes in the effective molecular charge density, changes in the occupation of proximal surface states, or the presence/absence of nearby molecular adsorbates. Finally, the observations reported here point out that considerable care should be taken in the interpretation of vibrational Stark effect data in other contexts. Although good agreement between theoretical expectations and observations has been reported (11), when considering

only Stark physics, it is important to note that charging effects may be of similar or greater magnitudes in some circumstances.

## Materials and Methods

The SERS substrates for the measurements are produced by electromigration of lithographically defined metal constrictions fabricated on underlying oxidized Si substrates. The bowtie structures are patterned by electron beam lithography. The constriction of the bowtie structure is 100–120 nm wide and 1  $\mu\text{m}$  long, as shown in Fig. 1, *Inset*. Evaporation of 1 nm Ti and 15 nm Au onto the bowtie structure is followed by lift-off in acetone. The larger, extended contact pads for the bowtie structure are evaporated with 1 nm Ti and 30 nm gold by using a shadow mask. The substrates are cleaned by oxygen plasma for 2 min to remove organic contamination and then spin coated with a solution of  $C_{60}$  at a concentration of 1 mg of  $C_{60}$  per 10 mL toluene. The  $C_{60}$  molecules undergo physisorption directly on the electrode surfaces, with no linker groups or linker chemistry. The chips are wire-bonded to a ceramic carrier and placed within a microscope cryostat. Once cooled to 80 K under a vacuum of  $6 \times 10^{-6}$  millibar, each junction is electromigrated through a computer-controlled procedure to yield interelectrode tunneling conductances near or below  $G_0 = 2e^2/h$ .

Raman spectroscopy is performed using a home-built Raman microscope system, with an incident wavelength of 785 nm. Modes with Stokes or anti-Stokes shifts below 300  $\text{cm}^{-1}$  are cut off by a notch filter. A piezo-actuated lens mount rasters the diffraction-limited beam over the sample surface, allowing the acquisition of spatially mapped Raman response. The mapped Si Raman intensity at 520  $\text{cm}^{-1}$  is used to locate the center of a bowtie structure. Following electromigration, another Raman image determines the location of the nanogap's plasmonic Raman hotspot. Raman spectra at that location then are acquired simultaneously with electronic transport data ( $I$  and  $dI/dV$ ) as a function of  $V$ , using a current preamplifier;  $V$  sourced by a digital-to-analog converter integrated into a lock-in amplifier; and differential conductance measured via lock-in using a 10-mV ac signal added to  $V$  with a summing amplifier. Fig. 1 is an example of a single surface-enhanced Raman spectrum of such a junction, acquired with a 1-s integration time. The sharp mode at 520  $\text{cm}^{-1}$  is from the underlying Si substrate, and the modes between 1,000  $\text{cm}^{-1}$  and 1,600  $\text{cm}^{-1}$  agree reasonably well with expectations from other SERS studies of  $C_{60}$ . Transport data acquisition was synchronized with spectral measurements through triggering. The bias,  $V$ , was swept from  $-0.5$  V to 0.5 V in steps of 0.0125 V or 0.025 V. The acquisition time for each spectrum at every voltage was 1–3 s. A higher-resolution grating is available for detailed studies, although this grating precludes simultaneous measurements of both Stokes and anti-Stokes emission.

All calculations were performed using the Siesta 3.1 code (46), using the generalized gradient exchange-correlation functional of Perdew, Burke, and Ernzerhof (PBE) (47). The C pseudopotential used a core radius cutoff of 1.29 bohr for 2s, 2p, and 2d channels. A triple zeta basis set was used for 2s and 2p functions, with a single zeta 2d polarization function. All calculations were performed with a 1000 Rydberg real space grid and  $30 \times 30 \times 30$ -Å supercells. All structures at all charges and fields were relaxed until forces were less than 0.004 eV/Å. Vibrations were calculated using Siesta's Vibra package (48).

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