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# How Pt Influences H<sub>2</sub> Reactions on High Surface-Area Pt/CeO<sub>2</sub> Powder Catalyst Surfaces

Jaeha Lee,<sup>#</sup> Peter Tieu,<sup>#</sup> Jordan Finzel, Wenjie Zang, Xingxu Yan, George Graham, Xiaoqing Pan,\* and Phillip Christopher\*



interrogate the mechanisms of how Pt nanoclusters and single atoms influence H<sub>2</sub> reactions on high-surface area Pt/CeO<sub>2</sub> powder catalysts. TPR showed that Pt promotes H<sub>2</sub> consumption rates on Pt/CeO<sub>2</sub> even when Pt exists on a small fraction of CeO<sub>2</sub> particles, suggesting that H-spillover proceeds far from Pt–CeO<sub>2</sub> interfaces and across CeO<sub>2</sub>–CeO<sub>2</sub> particle interfaces. IR and EELS measurements provided evidence that Pt changes the mechanism of H<sub>2</sub> activation and the rate limiting step for Ce<sup>3+</sup>, oxygen vacancy, and water formation as compared to pure CeO<sub>2</sub>. As a result, higher-saturation surface hydroxyl coverages can be achieved on Pt/CeO<sub>2</sub> compared to pure CeO<sub>2</sub>. Further, Ce<sup>3+</sup> formed by spillover-H from Pt is heterogeneously distributed and localized at and around interparticle CeO<sub>2</sub>–CeO<sub>2</sub> boundaries, while activated H<sub>2</sub> on pure CeO<sub>2</sub> results in homogeneously distributed Ce<sup>3+</sup>. Ce<sup>3+</sup> localization at and around CeO<sub>2</sub>–CeO<sub>2</sub> boundaries for Pt/CeO<sub>2</sub> is accompanied by surface reconstruction that enables faster rates of H<sub>2</sub> consumption. This study reconciles the materials gap between model structures and powder catalysts for H<sub>2</sub> reactions at CeO<sub>2</sub> surfaces.

**KEYWORDS:** platinum, ceria, catalysts, redox reactions, materials gap, spillover

and in situ electron energy loss spectroscopy (EELS) were used to

# INTRODUCTION

CeO<sub>2</sub> has been intensively studied as a catalyst for various heterogeneous catalytic reactions and as a support for platinum-group metals (PGMs).<sup>1</sup> For example, dissociated H<sub>2</sub> on CeO<sub>2</sub> can participate in selective hydrogenation or dehydrogenation reactions.<sup>2–4</sup> In addition, dissociated H<sub>2</sub> can induce the reduction of Ce<sup>4+</sup> to Ce<sup>3+</sup> and the formation of oxygen vacancies (V<sub>o</sub>) through H<sub>2</sub>O formation and desorption, which can facilitate the adsorption and activation of various reactants (e.g., CO<sub>2</sub> and H<sub>2</sub>O).<sup>5–8</sup> Therefore, understanding the H<sub>2</sub> activation mechanism on CeO<sub>2</sub> and the resulting influence on CeO<sub>2</sub> surface chemistry is critical for understanding various catalytic processes.

Surface science analyses have shown that PGMs promote interactions between H<sub>2</sub> and CeO<sub>2</sub> surfaces by dissociating H<sub>2</sub> and spilling over H to CeO<sub>2</sub>.<sup>9–13</sup> However, the translation of this mechanistic picture to high-surface area powder catalysts is complicated by spatial heterogeneities, such as PGMs existing as single atoms (SAs) or nanoclusters (NCs), and the existence of CeO<sub>2</sub>–CeO<sub>2</sub> particle boundaries, which may affect interactions with H<sub>2</sub>. In addition, most reports assert that V<sub>o</sub> (and associated Ce<sup>3+</sup>) species produced via H<sub>2</sub> interactions with PGM/CeO<sub>2</sub> localize around PGMs.<sup>6,14–18</sup> However, the influence of Pt on various elementary steps involved in V<sub>o</sub> formation across the CeO<sub>2</sub> surface has not been resolved, and direct measurements of Ce<sup>3+</sup> spatial distribution in nonmodel PGM-supported CeO<sub>2</sub> samples are lacking. Thus, while it is appreciated that H-spillover from PGMs influences CeO<sub>2</sub> surface chemistry, information is lacking for powder catalysts, particularly the mechanism of PGM-promoted CeO<sub>2</sub> surface reduction, the influence of heterogeneity on the PGM

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structure and  $CeO_2$ -CeO<sub>2</sub> interparticle interactions, and the resulting  $Ce^{3+}$  spatial distribution.

To address these missing mechanistic insights, we prepared a series of high-surface area, powdered Pt/CeO2 materials with varying Pt structures (SAs or NCs), Pt weight loading (0.0005-2 wt %), and dilutions with pure CeO<sub>2</sub> to isolate the role of the Pt structure,  $Pt-CeO_2$  contacts, and  $CeO_2$ -CeO<sub>2</sub> interactions in H<sub>2</sub> reactions with CeO<sub>2</sub> surfaces. In situ spectroscopy and microscopy coupled to measurements of H<sub>2</sub> consumption rates provide sample-averaged and spatially resolved information about the kinetics, mechanisms, and locations of H-spillover and Ce<sup>3+</sup> formation. Experimental evidence suggests that while H<sub>2</sub> dissociates heterolytically on pure CeO<sub>2</sub> surfaces, Pt promotes homolytic H<sub>2</sub> dissociation and subsequent H-spillover onto CeO2 surfaces, resulting in more than a twofold increase in saturation hydroxyl coverage compared to pure CeO<sub>2</sub>. Samples containing predominantly Pt NCs exhibit faster rates of homolytic H<sub>2</sub> dissociation than samples containing predominantly Pt SAs, and further, Hspillover can occur through contact between CeO2 particles even when a significant fraction of CeO<sub>2</sub> particles do not contain Pt. Time-resolved IR spectroscopy of H<sub>2</sub> reactions under isothermal conditions suggests that the conversion of heterolytic H<sub>2</sub> dissociation products into homolytic products is the rate-limiting step for H<sub>2</sub>O evolution on pure CeO<sub>2</sub>, but Ptmediated homolytic H<sub>2</sub> dissociation and H-spillover remove this kinetic limitation and promote the rate of H<sub>2</sub>O evolution from CeO<sub>2</sub> even far from the Pt-CeO<sub>2</sub> interface. In addition, while Ce<sup>3+</sup> formation on pure CeO<sub>2</sub> surfaces occurs homogeneously across the surface, Pt-containing samples exhibit increased hydroxyl coverages on CeO<sub>2</sub> and localization of  $Ce^{3+}$  at and around the boundaries between  $CeO_2$  particles rather than around Pt. The high hydroxyl coverage on CeO<sub>2</sub> caused by H-spillover from Pt induces irreversible changes to interparticle CeO<sub>2</sub> interfaces resulting in the oriented attachment of neighboring CeO<sub>2</sub> particles. The reconstructed CeO<sub>2</sub> surfaces at the boundaries between aligned CeO<sub>2</sub> particles appear to serve as highly reactive sites for H<sub>2</sub> consumption. This study provides mechanistic insights into the influence of spatial heterogeneities in powdered Pt/CeO<sub>2</sub> samples on their H<sub>2</sub> activation and CeO<sub>2</sub> surface reduction reactivity.

#### EXPERIMENTAL SECTION

#### **Materials**

High-purity (99.995%) tetraammineplatinum(II) nitrate (TAPN) purchased from Sigma-Aldrich was used as the Pt precursor. Highsurface area CeO<sub>2</sub> was purchased from US Research Nanomaterials and used as the support in this study. The BET surface area of CeO<sub>2</sub> oxidized at 500 °C with 20% O<sub>2</sub>/He for 2 h was estimated to be 68 m<sup>2</sup>/g from N<sub>2</sub> adsorption/desorption measurements. 28–39% concentration NH<sub>4</sub>OH solution was purchased from Sigma-Aldrich and used in catalyst synthesis.  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> nanoparticles with an average diameter of 5 nm was purchased from US Research Nanomaterials. 10% H<sub>2</sub>/Ar, pure H<sub>2</sub>, Ar, 10% O<sub>2</sub>/He, and 10% CO/He from Airgas were used in this study.

### **Catalyst Synthesis**

Pt was dispersed on CeO<sub>2</sub> using a modified strong electrostatic adsorption method.<sup>19,20</sup> Pt/CeO<sub>2</sub> samples were prepared at a range of Pt weight loadings from 0.0005 to 2 wt %. Briefly, 0.5 g of CeO<sub>2</sub> was dispersed in 50 mL of deionized H<sub>2</sub>O, and the pH of the solution was adjusted to 9.0 using NH<sub>4</sub>OH. The point of zero charge of CeO<sub>2</sub> was measured to be 3.4. Therefore, the CeO<sub>2</sub> surface is deprotonated at a pH of 9.0. Performing syntheses at a pH of 9.0 maximizes Pt dispersion through electrostatic interaction with  $[Pt(NH_3)_4]^{2+}$ 

complexes. Separately, the desired amount of TAPN was dissolved in deionized H<sub>2</sub>O and the solution was brought to a pH of 9.0. The TAPN solution was slowly injected into the CeO<sub>2</sub> solution over 12.5 h, and the final mixed solution was heated to 70 °C until dry. The dried Pt/CeO<sub>2</sub> samples were calcined in a tube furnace at 450 °C for 4 h by flowing dry air after ramping up the temperature at a rate of 10 °C/min. The Pt-free CeO<sub>2</sub> was oxidized at 500 °C for 2 h by flowing dry air (ramp rate, 10 °C/min). Note also that the wetting of CeO<sub>2</sub> with the deionized H<sub>2</sub>O at a pH of 9.0 did not change the surface reducibility of CeO<sub>2</sub>. Pt/CeO<sub>2</sub> was also physically mixed with CeO<sub>2</sub> in different mass ratios (1:19, 1:49, and 1:99) using a vibratory mixer.

## H<sub>2</sub>-Temperature-Programmed Reduction

H<sub>2</sub>-temperature-programmed reduction (H<sub>2</sub>-TPR) curves were obtained with 0.03 g of the sample using an Autochem II (Micromeritics) equipped with a thermal conductivity detector (TCD). Typically, samples were oxidized with 10%  $O_2$ /He at 400  $^{\circ}$ C for 1 h to remove impurities, cooled to 50  $^{\circ}$ C under O<sub>2</sub>, and purged with Ar at 50 °C for 30 min and stabilized under 10% H<sub>2</sub>/Ar at 50 °C for 1 h, and then the temperature was increased in a linear course to collect H<sub>2</sub>-TPR curves. The influence of prereduction on the TPR behavior of pure  $\text{CeO}_2$  was assessed by collecting  $\text{H}_2\text{-}\text{TPR}$ curves after reduction with 10% H<sub>2</sub>/Ar at 450 °C for 1 h followed by re-oxidation with 10% O2/He at 100 °C for 1 h. Similarly, the influence of prereduction on the TPR behavior of Pt/CeO2 was studied by collecting H2-TPR curves after reduction with 10% H2/Ar at 350 °C for 1 h followed by re-oxidation with 10%  $O_2/He$  at 350 °C for 1 h. To confirm that the change in the oxidation state of Pt was not the only cause for the improved H<sub>2</sub> consumption rate following prereduction,  $Pt(2)/CeO_2$  was ex situ reduced with 10% H<sub>2</sub>/Ar at 400 °C for 1 h, diluted with pure CeO<sub>2</sub>, and then re-oxidized in situ with 10%  $O_2/He$  at 350 °C for 1 h. The amount of  $H_2$  consumed during the TPR runs was estimated by integrating the TCD signals.

## Fourier-Transform Infrared Spectroscopy

Fourier-transform infrared (FTIR) spectroscopy experiments were carried out in a diffuse reflectance reaction chamber (Harrick Scientific) equipped with ZnSe windows, mounted inside a Praying Mantis diffuse reflectance adapter (Harrick Scientific), and coupled to a Thermo Scientific Nicolet iS10 FTIR spectrometer with a liquid nitrogen-cooled HgCdTe (MCT) detector. The FTIR and Praying Mantis diffuse reflection accessories were purged with dry N<sub>2</sub> during experiments. In a typical experiment, the reactor was loaded with ~80 mg of 50 nm-sized  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (Sigma-Aldrich), followed by 15 mg of the sample on top of the inert alumina. All spectra are reported in absorbance units.

For the H<sub>2</sub>-FTIR experiments, samples were first oxidized with 10% O<sub>2</sub>/He at a desired temperature for 30 min, purged with Ar at the same temperature for 30 min, and then reduced with 10% H<sub>2</sub>/Ar. Spectra were collected on 1 min time scales. As discussed in the Supporting Information, the IR intensity is sensitive to the "peak to peak" intensity in the interferogram, which is related to the total amount of reflected photons from the sample surface. In addition, the "peak to peak" intensity is sensitive to the temperature or sample color that changes during reduction. Therefore, we maintained the "peak to peak" intensity at 5.0 by focusing/defocusing the lenses. The maximum intensities of the vibrational frequency of surface hydroxyls and the forbidden  ${}^{2}F_{5/2}$ -to- ${}^{2}F_{7/2}$  electronic transition of Ce<sup>3+</sup> were used to compare the rate of formation of these two species. For high-pressure experiments, pure H<sub>2</sub> was used, and the pressure was increased using a back-pressure regulator.

For CO-FTIR experiments, samples were oxidized with  $10\% O_2/$  He at 300 °C for 30 min, cooled to 25 °C under  $O_2$ , and purged with Ar at 50 °C for 30 min before collecting background spectra. 10% CO/Ar was introduced to the sample at 25 °C for 10 min, and the sample was purged by Ar flow for 2 min to remove the gas-phase CO signal in the IR spectra. The flow rate was maintained at 50 mL/min in all conditions.

#### CO Oxidation Reactivity Measurement

The reactivity of the  $Pt(0.01)/CeO_2$  and " $Pt(1)/CeO_2$ :CeO<sub>2</sub> = 1:99" catalysts in CO oxidation was investigated in a tubular packed-bed reactor. 0.015 g of each sample was diluted with 0.3 g of acid-purified SiO<sub>2</sub> gel. Three different pretreatment protocols were used in this study. (i) To study the catalytic reactivity of the oxidized surfaces, samples were oxidized at 400 °C with 10% O2/He for 1 h. After cooling to 50 °C under O<sub>2</sub>, 1% of CO and 10% of O<sub>2</sub> were introduced for reactivity analysis. (ii) To study the catalytic reactivity of the reduced surfaces, samples were reduced with 10%  $H_2/Ar$  at 400  $^\circ C$ for 1 h. After cooling to 50 °C under H<sub>2</sub>, samples were purged with Ar for another hour. 1% CO and 10% O2 were then introduced for reactivity tests. (iii) To study the catalytic reactivity of the re-oxidized surfaces, samples were reduced with 10%  $\rm H_2/Ar$  at 400  $^\circ C$  for 1 h, purged with Ar at 400 °C for 1 h, and then re-oxidized with 10%  $O_2/$ He at 400 °C for 1 h. After cooling to 50 °C under O<sub>2</sub>, 1% CO and 10%  $\mathrm{O}_2$  were introduced. After introduction of  $\mathrm{O}_2$  and CO to all samples, the temperature was increased from 50 to 400 °C at a ramp rate of 2 °C/min to measure the CO oxidation reactivity of Pt/CeO<sub>2</sub> catalysts. In this study, a CO conversion of less than 10% was used to calculated the turnover frequency, and all Pt was assumed to be accessible to the reactants. The Arrhenius equation was used to estimate  $E_a$ . An FTIR spectrometer (Nicolet 6700, Thermo Fisher Scientific) with a 2 m gas cell maintained at 120 °C was used to measure the CO concentration. Spectra were recorded using an MCT detector (16 scans, resolution of 1  $cm^{-1}$ ).

#### **X-ray Powder Diffraction**

X-ray powder diffraction (XRD) patterns were obtained at a voltage of 40 kV and a current of 30 mA with Cu K $\alpha$  radiation ( $\lambda$  = 0.1542 nm) using a Mode 1 Smartlab diffractometer (Rigaku, Japan). A scanning-step size of 0.02° at a speed of 2.5°/min was used. The Scherrer equation was used to estimate grain sizes.

## Surface Area Measurement

 $N_2$  adsorption/desorption isotherms were measured on a Micromeritics 3Flex Porosimeter at liquid  $N_2$  temperature. Before analysis, all samples were degassed at 300  $^\circ C$  for 2 h in vacuum. The Brunauer–Emmett–Teller (BET) method was applied to the  $N_2$  adsorption isotherm to extract surface area estimates.

## X-ray Absorption near Edge Structure Spectroscopy

X-ray absorption spectroscopy (XAS) measurements were completed at beamline 9-3 at the Stanford Synchrotron Radiation Lightsource (SSRL) at SLAC National Laboratory. A Si(220) double-crystal monochromator with a 20% detune was used to scan the Pt L<sub>3</sub> absorption edge (11,564 eV). 30 mg of the Pt/CeO<sub>2</sub> catalyst samples (pressed and sieved, 80-120 mesh) were loaded into a 3 mm quartz capillary tube between quartz wool plugs. The capillary was mounted in an in situ XAS reactor cell.<sup>21</sup> Process gasses were supplied via mass flow controllers (Brooks Instrument). The sample was heated via resistive heaters placed above and below the capillary with PID temperature control and a thermocouple placed in contact with the quartz wool of the sample bed. The sample was placed at a 45° angle relative to the beam path, and fluorescence data were collected with a Canberra 100-element Ge detector orthogonal (90°) to the beam path with a 6 absorption length Zn filter and lead shielding to minimize contributions from the scattering of beamline optics. Kapton tape was used to split the initial monochromated beam, and all spectra were calibrated with respect to a Pt reference foil mounted off-axis with a transmission signal collected using a photodiode. Platinum(II) acetylacetonate [Pt(acac)<sub>2</sub>] and PtO<sub>2</sub> were used as the Pt<sup>+2</sup> and Pt<sup>+4</sup> standards, respectively. Reference compounds were pressed into self-supporting pellets with a diameter of 7 mm, and Xray absorption near edge structure (XANES) scans were collected under ambient conditions.

In situ XANES TPR measurements were used to track the reduction of Pt in Pt/CeO<sub>2</sub> catalysts. Samples were initially oxidized in 20 sccm of 20% O<sub>2</sub>/He at 300 °C (heated at 10 °C/min) for 30 min and cooled to 20 °C. XANES scans of the initial oxidized sample were acquired while purging the sample with He (20 sccm). The

sample was then exposed to 10%  $H_2$ /He (20 sccm), and a TPR was performed up to 400 °C at a ramp rate of 3 °C/min. XANES scans were acquired throughout the TPR.

# **Transmission Electron Microscopy**

High-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) images were acquired using a JEOL JEM300CF equipped with double aberration correction. In the scanning transmission electron microscopy (STEM) mode, a 32 mrad convergence semi-angle was used, and the dark-field annular detector acquires a signal from 53 to 180 mrad. Powder samples were dispersed in ethanol and dropcast onto the Protochips (North Carolina, USA) E-chips. Pre and post in situ imaging was performed on a Protochips Inspection holder. In situ experiments were performed with a Protochips Atmosphere 210 gas cell system with a gas flow rate of 0.1 cm<sup>3</sup>/s at 200 Torr. Pure Ar, 5% H<sub>2</sub>/Ar, and 20% O2/Ar from Airgas were used for the experiments. Prior to imaging, the environment in the holder was pump-purged five times at 200 °C to remove residual atmospheric O and C. For the sequential reduction-oxidation-reduction, the gas cell was exposed to H<sub>2</sub> for 1 h before pump-purging the cell five times with Ar and changing the gas to O<sub>2</sub> for 1 h and finally changing the gas back to H<sub>2</sub> with a pumppurge process. During the pump-purge, the electron beam is turned off to minimize beam exposure.

#### Electron Energy Loss Spectroscopy

EEL spectra were collected using a Gatan K2 direct electron detector with an energy dispersion of 0.250 eV/channel. Spectra were calibrated to the N K edge.  $50 \times 50$  pixel maps with 0.1 s dwell time were used to generate the EELS mapping images.  $10 \times 10$  pixel maps with 0.05 s dwell time were used to measure the Ce oxidation state at the particle boundary in Figure 5c. Principal component analysis (PCA) was used to improve the signal-to-noise ratio before a power law background subtraction is applied. A double arctan background was applied, and the peaks were fitted with a Voigt peak and then the Ce  $M_5$  and  $M_4$  peak heights were measured. The  $M_5/M_4$ ratio was calculated for each pixel to generate the EELS mapping images with a Gaussian blur applied with sigma = [5,5]. Pixels for the Ce EELS map were selected to match the corresponding HAADF-STEM image by applying a threshold for the Ce signal where below said threshold, the  $M_5/M_4$  ratio was set to zero. Custom Python code was used for PCA, double arctan background subtraction, Voigt peak fitting, and  $M_5/M_4$  ratio measurements to estimate the oxidation state of Ce without bias. The code used in this study is available upon request.

# RESULTS AND DISCUSSION

## Effect of Pt Amount and Structure and CeO<sub>2</sub> Particle Boundaries on the Rate and Amount of H<sub>2</sub> Consumption

To understand the influence of the Pt structure, Pt-CeO<sub>2</sub> interfaces, and CeO<sub>2</sub>-CeO<sub>2</sub> particle interfaces on H<sub>2</sub> surface reactions, a series of materials were prepared that traverse each of these variables. Pt was deposited onto CeO<sub>2</sub> nanoparticles (8.5 nm average diameter estimated from XRD, Figure S1a; 68  $m^2/g$ ) via electrostatic adsorption of  $Pt(NH_3)_4(NO_3)_3$  from aqueous solutions, followed by calcination in air at 450 °C for 4 h with Pt weight loadings (wt %) ranging from X = 0.0005 to 2 wt % Pt, denoted as  $Pt(X)/CeO_2$ . The average number of Pt atoms per CeO<sub>2</sub> particle is 1.0 at ~0.01 wt % Pt (Figures 1a and S1b), and Pt is primarily atomically dispersed at  $\leq 0.05$  wt % Pt, as evidenced by in situ CO-FTIR spectra (Figure S2) and CO oxidation kinetic measurements (Figure S3), whereas Pt exists primarily as  $Pt_xO_y$  clusters at a Pt loading of  $\geq 1$  wt % for the calcined samples (see Figures S2 and S3 and Supporting Information Discussion I).<sup>20</sup> Mixtures of Pt(1)/  $CeO_2$  with pure  $CeO_2$  were also prepared at mass ratios up to 1:99  $[Pt(1)/CeO_2:CeO_2]$  (Figure 1b) to introduce inter-



**Figure 1.** Schematics of  $Pt/CeO_2$  with decreasing Pt loading either by decreasing Pt loading (a) or through dilution with pure  $CeO_2$  (b). The number of Pt atoms per  $CeO_2$  particle becomes 1.0 at ~0.01 wt % of Pt. (c) H<sub>2</sub>-TPR curves of  $CeO_2$  and  $Pt/CeO_2$  with different Pt loadings (ramp rate, 10 °C/min; 10% H<sub>2</sub> at 1 atm). (d) H<sub>2</sub>-TPR curves of  $Pt(0.05)/CeO_2$  and "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:19", which have the same nominal Pt loading but different Pt structures (ramp rate, 10 °C/min; 10% H<sub>2</sub> at 1 atm).

particle  $\text{CeO}_2$  contacts and spatially separate Pt from  $\text{CeO}_2$  domains.  $^{22}$ 

H<sub>2</sub>-TPR was used to probe reactions of H<sub>2</sub> on Pt/CeO<sub>2</sub>. H<sub>2</sub>-TPR is often used to assess the reducibility (e.g.,  $V_0$  formation) of oxide surfaces.<sup>23–25</sup> However,  $H_2$ -TPR only directly assesses the rate of  $H_2$  consumption during a temperature increase.  $H_2$ consumption can occur due to surface hydroxyl or hydride formation via H<sub>2</sub> dissociation and does not require simultaneous V<sub>o</sub> formation. Figure 1c shows the H<sub>2</sub> consumption rates on CeO<sub>2</sub> and Pt/CeO<sub>2</sub> with different Pt loadings collected at a temperature ramp rate of 10 °C/min and using a 10% H<sub>2</sub> feed at atmospheric pressure. H<sub>2</sub> is consumed below 500 °C through reactions with CeO<sub>2</sub> (or Pt/ CeO<sub>2</sub>) surfaces, while H<sub>2</sub> consumed above 500  $^{\circ}$ C is due to H<sub>2</sub> reactions with the CeO<sub>2</sub> bulk.<sup>23</sup> TPR measurements were made at <1% H<sub>2</sub> conversion and were devoid of mass transport limitations (Figure S4). Thus, H<sub>2</sub> concentration is effectively constant throughout the catalyst bed, although the Pt/CeO<sub>2</sub> surface structure and composition change during the temperature increase.

Figure 1c shows that the temperature where the peak rate of  $H_2$  consumption occurs due to surface reactions  $(T_m)$  increases as the Pt loading decreases from 2 to 0.0005 wt %, while the consumption of  $H_2$  related to reactions with the CeO<sub>2</sub> bulk is not affected.  $T_m$  is lower for samples containing <1 Pt atom/ CeO<sub>2</sub> particle weight loadings (<0.01 wt %) compared to pure CeO<sub>2</sub>. This demonstrates that  $H_2$  dissociation on Pt sites and diffusion of dissociated H across interparticle CeO<sub>2</sub>–CeO<sub>2</sub> boundaries occur at faster rates than  $H_2$  dissociation on Pt-free CeO<sub>2</sub> surfaces.  $H_2$  consumption by CeO<sub>2</sub> surface reduction

was indistinguishable from H<sub>2</sub> consumed to reduce Pt oxide in Pt/CeO<sub>2</sub> samples when Pt loading  $\leq 1$  wt % (Figure 1c), even for H<sub>2</sub>-TPR collected at a ramp rate of 3 °C/min (Figure S5). This is likely due to the rapid H-spillover from reduced Pt to CeO<sub>2</sub>, linking the reduction of Pt oxide and the H<sub>2</sub> consumption on CeO<sub>2</sub> surfaces.<sup>13,26</sup> However, the H<sub>2</sub> consumption peak centered at 40 °C was observed for Pt(2)/CeO<sub>2</sub> (Figure 1c), and a small feature was also observed at a similar temperature for Pt(1)/CeO<sub>2</sub> (Figures 1c and S6a). The H<sub>2</sub> consumption peak centered at 40 °C for Pt(2)/CeO<sub>2</sub> corresponds to 100  $\mu$ mol/g of H<sub>2</sub> (Figure S6b), which is smaller than the amount expected from the reduction of Pt oxide clusters (205  $\mu$ mol/g of H<sub>2</sub>), assuming that all Pt species reside in the PtO<sub>2</sub> phase. This suggests that only larger Pt<sub>x</sub>O<sub>y</sub> clusters formed at a higher Pt surface density are reduced by H<sub>2</sub> below 50 °C.<sup>27</sup>

It is worth mentioning that  $H_2$  consumption occurs over a wide temperature range for pure CeO<sub>2</sub>, whereas it occurs over a narrower temperature range for Pt/CeO<sub>2</sub> (Figure 1c), with constant temperature ramp rates during the TPR. The consumption of  $H_2$  over a wider temperature range on CeO<sub>2</sub> could be due to the heterogeneous reactivity of CeO<sub>2</sub> surfaces (many different local environments from the exposed basal planes and defect sites) and  $H_2$  dissociation being the rate limiting step. Conversely, the narrower temperature range for  $H_2$  consumption of Pt/CeO<sub>2</sub> samples is likely because  $H_2$  dissociation occurs on Pt sites, followed by H-spillover to the CeO<sub>2</sub> surface. Further mechanistic insights will be discussed.

To deconvolute the influence of total Pt loading and the Pt structure on the rate of  $H_2$  consumption,  $H_2$ -TPR for

Pt(0.05)/CeO<sub>2</sub> and a physical mixture of Pt(1)/CeO<sub>2</sub> and CeO<sub>2</sub> at a mass ratio of 1:19, containing identical macroscopic Pt loadings and different Pt structures (predominantly Pt SAs at 0.05 wt % and predominantly Pt<sub>x</sub>O<sub>y</sub> clusters at 1 wt %), were compared (Figure 1d).  $T_m$  was lower for "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:19" compared to Pt(0.05)/CeO<sub>2</sub>. Hence, Pt NCs are more effective than Pt SAs in promoting H-spillover onto CeO<sub>2</sub> surfaces. Here, the retained promotion in the H<sub>2</sub> consumption rate by the entire sample in the "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:19" physical mixture compared to pure CeO<sub>2</sub> suggests that the local influence of Pt on Ce–O bonds is not responsible for the broad influence of Pt on the rate of H<sub>2</sub> consumption on the CeO<sub>2</sub> surface. This is consistent with the promoted H<sub>2</sub> consumption rate for samples containing <1 Pt atom/CeO<sub>2</sub> particle as compared to pure CeO<sub>2</sub> (Figure 1c).

In situ XANES spectra in Figures S7–S9 show that the reduction of  $Pt_xO_y$  clusters to metallic Pt NCs occurs prior to the consumption of surface-reacting H<sub>2</sub> for "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:19".<sup>28</sup> Here, note that the amount of Pt in the physical mixture is too small ( $\leq 0.05$  wt %) so that the H<sub>2</sub> consumption peak associated with the reduction of Pt oxide, as observed in the in situ XANES spectra, could not be observed in the H<sub>2</sub>-TPR curve. It is interesting that the kinetics of Pt<sub>x</sub>O<sub>y</sub> reduction decouples from that of H-spillover when CeO<sub>2</sub> particles with and without Pt are physically separated. The implications of the existence of interparticle boundaries for H<sub>2</sub> consumption will be discussed in more detail later.

The total amount of  $H_2$  consumption in the surface reaction was also affected by Pt addition beyond the additional expected  $H_2$  consumption for the reduction of Pt oxides (Table 1). It is assumed that all O atoms bound to Pt react

Table 1. Amount of H<sub>2</sub> Consumed Estimated from H<sub>2</sub>-TPR Curves below 500 °C, the Increased Amount of H<sub>2</sub> Consumed after Loading Pt Based on the Amount of H<sub>2</sub> Consumed on Pure CeO<sub>2</sub>, and the Maximum Amount of H<sub>2</sub> Required for Pt<sub>x</sub>O<sub>y</sub> Reduction<sup>*a*</sup>

sample	H <sub>2</sub> consumed below 500°C (μmol/g)	excess $H_2$ consumed below 500°C $(\mu mol/g)^b$
CeO <sub>2</sub>	379	
Pt(0.0005)/CeO <sub>2</sub>	472	93
Pt(0.005)/CeO <sub>2</sub>	480	100
Pt(0.01)/CeO <sub>2</sub>	472	91
Pt(0.02)/CeO <sub>2</sub>	470	86
Pt(0.05)/CeO <sub>2</sub>	467	77
$Pt(1)/CeO_2$	475	0 <sup><i>c</i></sup>
$Pt(2)/CeO_2$	541	0 <sup>c</sup>
$Pt(1)/CeO_2:CeO_2 = 1:99$	423	42

"Here, the Pt-to-O molar ratio is assumed to be 4.0. <sup>b</sup>[Excess H<sub>2</sub> consumed] = [H<sub>2</sub> consumed] – [H<sub>2</sub> consumed on CeO<sub>2</sub>] – [4(from O) × (the amount of Pt)] – [0.5(from Pt-H) × (the amount of Pt)]. <sup>c</sup>[Excess H<sub>2</sub> consumed] is considered 0 if the estimated amount is less than 0.

with H<sub>2</sub> to form H<sub>2</sub>O (Pt-to-O molar ratio was assumed to be 4, the greatest possible amount) and also that H bonds to Pt in a 1:1 molar ratio (e.g., Pt is covered by H). At Pt loadings  $\leq 0.05$  wt %, between 10 and 25% excess H<sub>2</sub> was consumed compared to the expected amount based on H<sub>2</sub> consumption on pure CeO<sub>2</sub> and the maximum amount of H<sub>2</sub> required to reduce and saturate Pt<sub>x</sub>O<sub>y</sub>. This observation is consistent with a recent report where the presence of Pt increased H<sub>2</sub> consumption on CeO<sub>2</sub> (in the form of powder catalysts) by 47%.<sup>29</sup> Note that the excess H<sub>2</sub> consumption was not observed at Pt loading  $\geq 1$  wt % (Table 1). This is because at >1 wt % Pt, the CeO<sub>2</sub> surface becomes significantly populated by Pt, thus decreasing the available CeO<sub>2</sub> surface area. For example, ~0.45 Pt/nm<sup>2</sup> or ~10% coverage of Pt on CeO<sub>2</sub> exists at 1 wt % Pt, which is similar to the magnitude of excess H<sub>2</sub> consumption observed for lower Pt loading samples.

Thus, based on the TPR measurements and analysis, even small amounts of Pt cause significant changes to the rate and amount of  $H_2$  consumption by Pt/CeO<sub>2</sub> samples through surface reactions, even across CeO<sub>2</sub>-CeO<sub>2</sub> particle boundaries, motivating analysis of the sources of these changes.

 $H_2$  can dissociate homolytically or heterolytically on CeO<sub>2</sub> surfaces.<sup>30–34</sup> Homolytic  $H_2$  dissociation reduces two Ce<sup>4+</sup> to Ce<sup>3+</sup> and produces 2 hydroxyls (OH<sub>ads</sub>), (eq 1).<sup>31-35</sup> Heterolytic H<sub>2</sub> dissociation on CeO<sub>2</sub> surfaces does not reduce Ce<sup>4+</sup> and instead produces a hydride  $(H^-)$  on Ce<sup>4+</sup>  $(Ce^{4+}-H^-)$ and an OH<sup>-</sup><sub>ads</sub> (eq 2).<sup>31,32,34</sup> Density functional theory (DFT) calculations predict that heterolytic  $\mathrm{H}_2$  dissociation on  $\mathrm{CeO}_2$ surfaces has a lower activation barrier than homolytic H<sub>2</sub> dissociation (Table S1) but that the homolytic product is thermodynamically preferred.<sup>31,34,36</sup> Therefore, it has been proposed that the heterolytic H<sub>2</sub> dissociation product forms initially on CeO<sub>2</sub> surfaces and then transforms into the homolytic product (producing an additional OH<sub>ads</sub> and 2  $Ce^{3+}$ ) as the species equilibrate (eq 3).<sup>16,17</sup> Neighboring OH<sub>ads</sub> species can react to produce H<sub>2</sub>O<sub>ads</sub> and a lattice oxygen, O<sub>lattice</sub> (eq 4), followed by  $H_2O$  desorption, which leaves behind  $V_o$ (eq 5).<sup>37-39</sup> It has also been proposed that H<sub>2</sub> can homolytically dissociate on neighboring Ce3+ species on nonstoichiometric CeO<sub>2</sub> surfaces (where a considerable V<sub>o</sub> concentration exists) to produce 2  $Ce^{4+}-H^-$  (eq 6), although these species decompose at elevated temperatures (>200  $^{\circ}$ C).<sup>32</sup> The barriers for these elementary steps have been studied extensively via DFT calculations (see Table S1).<sup>30-34,36,38-42</sup> However, the calculated energetics depend on CeO<sub>2</sub> surface structure, OH<sub>ads</sub> coverage, V<sub>o</sub> concentration, and calculation parameters, making it challenging to hypothesize the series of steps and the kinetically relevant steps for OH<sub>ads</sub>, Ce<sup>3+</sup>, and V<sub>o</sub> formation in powdered samples under reaction conditions.

$$H_2 + 2O^{2-} + 2Ce^{4+} \rightarrow 2OH_{ads}^- + 2Ce^{3+}$$
 (1)

$$H_2 + Ce^{4+} + O^{2-} \rightarrow OH_{ads}^- + Ce^{4+} - H^-$$
 (2)

$$OH_{ads}^{-} + Ce^{4+} - H^{-} + Ce^{4+} \rightarrow 2OH_{ads}^{-} + 2Ce^{3+}$$
 (3)

$$2OH_{ads}^{-} \rightarrow H_2O_{ads} + O_{lattice}$$
(4)

$$H_2 O_{ads} \rightarrow H_2 O + V_o \tag{5}$$

$$H_2 + 2Ce^{3+} \to 2Ce^{4+} - H^-$$
 (6)

The addition of PGMs to  $CeO_2$  has been proposed to introduce an additional H<sub>2</sub> reaction pathway on CeO<sub>2</sub> through homolytic H<sub>2</sub> activation on the metal and subsequent H-spillover to CeO<sub>2</sub>, forming 2 OH<sup>-</sup><sub>ads</sub> and 2 Ce<sup>3+</sup> simultaneously (eq 7).<sup>9,11,43,44</sup>

$$Pt + H_2 + 2O^{2-} + 2Ce^{4+} \rightarrow Pt + 2OH_{ads}^{-} + 2Ce^{3+}$$
(7)



**Figure 2.** FTIR spectra for the (a) hydroxyl and (b)  $Ce^{3+}$  region of pure  $CeO_2$  collected under  $H_2$  at 450 °C and for the (d) hydroxyl and (e)  $Ce^{3+}$  region of "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:99" collected under  $H_2$  at 250 °C. Correlation between the maximum absorbance from surface hydroxyl and  $Ce^{3+}$  in the IR spectra of (c) pure CeO<sub>2</sub> from (a,b) and of (f) "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:99" from (d,e). 10% H<sub>2</sub>/Ar is used. The lines connecting the data points are simply to guide the eye and not model fits.

Thus, Pt can facilitate the formation of homolytic  $H_2$  dissociation products on CeO<sub>2</sub> surfaces by bypassing the sequential reaction pathway proposed to occur on pure CeO<sub>2</sub> surfaces—heterolytic  $H_2$  activation followed by transformation into the homolytic products. The relevance of Pt-mediated production of homolytic  $H_2$  dissociation products on CeO<sub>2</sub> to the TPR results requires that H-spillover from Pt to CeO<sub>2</sub> is sufficiently fast across the entire CeO<sub>2</sub> surface to negate the relevance of direct heterolytic  $H_2$  dissociation on CeO<sub>2</sub> surfaces. From eqs 1–7, the rate of formation of OH<sup>-</sup><sub>ads</sub> (d[OH<sup>-</sup><sub>ads</sub>]/dt) and Ce<sup>3+</sup> (d[Ce<sup>3+</sup>]/dt) on CeO<sub>2</sub> surfaces in H<sub>2</sub> can be expressed as eqs 8 and 9, respectively

$$d[OH_{ads}^{-}]/dt = 2 \times r_{homo} + r_{hetero} + r_{hetero \to homo} - 2 \times r_{H_2O}$$
(8)

$$d[Ce^{3^{+}}]/dt = 2 \times r_{homo} + 2 \times r_{hetero \to homo}$$
$$- 2 \times r_{homo \text{ on } Ce^{3^{+}}}$$
(9)

where  $r_{\text{homoo}}$ ,  $r_{\text{hetero} \to \text{homoo}}$ ,  $r_{\text{H}_2\text{O}}$ , and  $r_{\text{homo on Ce}^{3+}}$  represent the rate of homolytic H<sub>2</sub> activation [rate of (1) plus rate of (7)], heterolytic H<sub>2</sub> activation [rate of (2)], transformation of heterolytic products to homolytic products [rate of (3)], H<sub>2</sub>O formation [rate of (4), assuming the rate of (5) is quasiequilibrated], and homolytic H<sub>2</sub> activation on Ce<sup>3+</sup> near V<sub>o</sub> [rate of (6)]. We note that the reverse of (6) has also been proposed to produce  $\rm H_2$  from hydride decomposition, and this will be discussed later.  $^{32,36}$ 

To interrogate the kinetics, mechanisms, and kinetically relevant steps associated with H<sub>2</sub> reactions on CeO<sub>2</sub> and Pt/CeO<sub>2</sub> surfaces, the relative concentrations of surface hydroxyls,  $[OH_{ads}^{-1}]$ , and Ce<sup>3+</sup>,  $[Ce^{3+}]$ , were measured using in situ FTIR while flowing 10% H<sub>2</sub> at constant temperatures (250, 350, or 450 °C) over materials that were pre-oxidized in situ with 10% O<sub>2</sub> for 30 min at 400 °C. In situ FTIR provides the distinct ability, compared to other characterization approaches, to follow both  $[OH_{ads}^{-1}]$  and  $[Ce^{3+}]$  simultaneously with a time resolution of <1 min at atmospheric pressure and above.<sup>45</sup> Details on the FTIR spectra acquisition methodology, which facilitates quantitative comparison of spectra collected from different samples, are provided in Supporting Information Discussion II (Figures S10–S13).

The consumption of H<sub>2</sub> via surface reactions on pure CeO<sub>2</sub> was complete at ~ 450 °C in TPR measurements (Figures 1 and S14). When CeO<sub>2</sub> was exposed to H<sub>2</sub> at a constant temperature of 450 °C, in situ IR evidenced increases in the intensity of bands assigned to  $OH_{ads}^-$  (3000–3500 cm<sup>-1</sup>) and the forbidden  ${}^{2}F_{5/2}$ -to- ${}^{2}F_{7/2}$  electronic transition of Ce<sup>3+</sup> (2130 cm<sup>-1</sup>) (Figure 2a,b), respectively.<sup>46–49</sup> See Supporting Information Discussion III for the assignment of the electronic transition of Ce<sup>3+</sup>. Changes in the intensity of these two bands as a function of time are displayed in Figure 2c, where  $[OH_{ads}^{-1}]$ 



**Figure 3.** (a) Change in the maximum absorbance from surface hydroxyl and  $Ce^{3+}$  estimated from IR spectra of "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:99" collected under H<sub>2</sub> at 250, 350, 400, and 450 °C. Filled and empty symbols represent the absorbance from surface hydroxyl and  $Ce^{3+}$ , respectively. (b) Change in the maximum absorbance from surface hydroxyl estimated from IR spectra collected under H<sub>2</sub> at 350 °C on pure CeO<sub>2</sub> and Pt/CeO<sub>2</sub> with different Pt loadings. 10% H<sub>2</sub>/Ar was used, which equals to 76 Torr of H<sub>2</sub>. The lines connecting the data points are simply to guide the eye and not model fits.

saturates within  $\sim$ 5 min, while [Ce<sup>3+</sup>] increases consistently for more than 60 min.

The time-dependent evolution of  $[OH_{ads}^{-1}]$  and  $[Ce^{3+}]$ provide insights into the mechanism of H<sub>2</sub> reactions on CeO<sub>2</sub> surfaces. First,  $r_{homo \text{ on } Ce^{3+}}$  (6) is assumed to be 0 as  $[Ce^{3+}]$  did not change when the sample was purged with Ar after reaction with H<sub>2</sub> at 450 °C (Figure S15). This suggests that hydrides formed by homolytic activation on Ce<sup>3+</sup> are not stable at 450 °C, and thus, this reaction has no net forward rate under the explored conditions. In addition, it was recently reported that the intensity of the IR band associated with Ce<sup>3+</sup> is linearly proportional to the amount of H<sub>2</sub>O produced during reaction with H<sub>2</sub>.<sup>50</sup> This suggests that the reverse reaction of (6) has a negligible rate as this step does not produce H<sub>2</sub>O.<sup>23,51</sup> Therefore, d[Ce<sup>3+</sup>]/dt in eq 9 can be simplified to eq 10

$$d[Ce^{3+}]/dt = 2 \times r_{homo} + 2 \times r_{hetero \to homo}$$
(10)

The larger initial rate of  $[OH_{ads}^{-}]$  formation compared to  $[Ce^{3+}]$  formation (Figure 2c) suggests that  $r_{hetero}$  is initially larger than the sum of  $r_{homo}$  and  $r_{hetero \rightarrow homo}$ . This is consistent with the hypothesis from theoretical calculations that  $H_2$  initially heterolytically dissociates on  $CeO_2$  and that  $Ce^{3+}$  formation proceeds via transformation of heterolytic  $H_2$  dissociation products. From eq 8, the observation of pseudo-steady-state  $[OH_{ads}^{-}]$  after ~5 min of  $(d[OH_{ads}^{-}]/dt \sim 0)$  leads to eq 11

$$2 \times r_{\rm H_2O} \sim 2 \times r_{\rm homo} + r_{\rm hetero} + r_{\rm hetero \to homo}$$
(11)

 $r_{\text{homo}}$  and  $r_{\text{hetero}}$  are assumed to be 0 once  $d[OH_{ads}^-]/dt \sim 0$ because H<sub>2</sub> consumption on CeO<sub>2</sub> is completed quickly at 450 °C (Figures S16 and S17). Since  $d[Ce^{3+}]/dt > 0$  while  $d[OH_{ads}^-]/dt \sim 0$ , eq 10 suggests that  $r_{\text{hetero} \rightarrow \text{homo}}$  is >0. This then leads to the conclusion that both  $d[Ce^{3+}]/dt$  and  $r_{H_2O}$  are limited by  $r_{\text{hetero} \rightarrow \text{homo}}$  on CeO<sub>2</sub> at 450 °C in H<sub>2</sub>.  $r_{\text{hetero} \rightarrow \text{homo}}$ limiting  $r_{H_2O}$  suggests that H<sub>2</sub> consumption occurs before H<sub>2</sub>O formation on pure CeO<sub>2</sub>, which is consistent with previous reports.<sup>51</sup>

Similar measurements were performed on "Pt(1)/ CeO<sub>2</sub>:CeO<sub>2</sub> = 1:99", where Pt NCs exist on 1 out of every 100 CeO<sub>2</sub> particles and the consumption of surface-reacting  $H_2$ is completed at 250 °C in H<sub>2</sub>-TPR (Figure S14). IR spectra collected at 250 °C under a  $H_2$  flow show that  $[OH_{ads}^-]$  and [Ce<sup>3+</sup>] increased simultaneously (Figure 2d,e), distinct from the behavior of pure CeO<sub>2</sub>. This correlation  $(d[OH_{ads}^-]/dt = \alpha)$ × d[Ce<sup>3+</sup>]/dt;  $\alpha$  is constant) requires that  $r_{\text{homo}}$  be much larger than  $r_{hetero}$  or  $r_{H,O}$  (eqs 8 and 9). This indicates that Pt NCs promote the spillover of homolytic H<sub>2</sub> dissociation products onto the  $CeO_2$  surface (eq 7) but that the rate of  $H_2O$ formation is slow compared to the rate of H-spillover at 250  $^{\circ}$ C. This is consistent with previous reports that H $_2$ consumption occurs on PGM-supported CeO<sub>2</sub> surfaces before H<sub>2</sub>O evolution during H<sub>2</sub>-TPR on high-surface area catalysts.<sup>23</sup> Further, the consistent formation rates of  $[OH_{ads}^-]$  and  $[Ce^{3+}]$ until they reach saturation suggests that homolytically dissociated H<sub>2</sub> on Pt spills over to the entire CeO<sub>2</sub> surface (even to  $CeO_2$  particles not containing Pt) at a faster rate than heterolytic  $H_2$  activation directly on CeO<sub>2</sub>. When the  $H_2$ exposure temperature on " $Pt(1)/CeO_2$ :CeO<sub>2</sub> = 1:99" was increased to >350  $^\circ C$ , both  $[OH^-_{ads}]$  and  $[Ce^{3+}]$  reached maxima within 1-2 min, and subsequently,  $[OH_{ads}^-]$  decreases as a function of time, suggesting that H<sub>2</sub>O formation proceeds (Figure 3a).

Similar experiments were executed for  $Pt(0.01)/CeO_2$ , where Pt is primarily atomically dispersed as single atoms (Figures S2 and S3). Close correlation between d[OH<sub>ads</sub>]/dt and d[Ce<sup>3+</sup>]/dt was again observed (Figure S18). While this may indicate that Pt SAs promote the spillover of homolytic H<sub>2</sub> dissociation products to the CeO<sub>2</sub> surface, there is a debate in the literature on whether Pt SAs can promote homolytic H<sub>2</sub> dissociation.<sup>52–54</sup> We cannot rule out the possibility that Pt NCs present in a low concentration in Pt(0.01)/CeO<sub>2</sub> are responsible for the homolytic H<sub>2</sub> dissociation. Regardless, d[OH<sub>ads</sub>]/dt and d[Ce<sup>3+</sup>]/dt were larger for samples having a larger concentration of Pt NCs, even with identical nominal Pt loadings (Figure S18). This is consistent with Pt NCs activating H<sub>2</sub> at faster rates than Pt SAs, as observed in Figure 1d and reported previously.<sup>55–57</sup>

The in situ FTIR data provide evidence of other changes to  $H_2$  surface chemistry on CeO<sub>2</sub> caused by Pt. First, "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:99" showed ~25% higher saturation of [Ce<sup>3+</sup>]



**Figure 4.** (a) Representative EEL spectra with different  $Ce^{3+}$  concentrations after PCA, power law, and arctan background subtraction, normalized to the Ce M<sub>4</sub> peak height. Processed data are represented by the shaded regions, and the fitted Voigt peaks are represented by the solid lines. HAADF-STEM and EELS mapping images of (b) pure CeO<sub>2</sub> after in situ reduction at 450 °C for 60 min and (c) Pt(2)/CeO<sub>2</sub> with Pt NCs after in situ reduction at 400 °C for 60 min. 10 Torr of H<sub>2</sub> is used. In the EELS maps, a gradual change of the color map from blue to red indicates a gradual increase in Ce<sup>3+</sup> concentration.

compared to that of pure CeO<sub>2</sub> (Figure S19), which is consistent with the increased H<sub>2</sub> consumption observed during TPR (Table 1). The agreement between the IR analysis of relative [Ce<sup>3+</sup>] and the TPR data provides confidence in conclusions drawn from quantitative comparisons of the magnitudes of IR signatures for different samples. Based on this justification, the magnitude of  $[OH_{ads}^-]$  can be compared between samples and H<sub>2</sub> exposure temperatures to understand how Pt influences saturation  $[OH_{ads}^-]$ .

When CeO<sub>2</sub> was exposed to 76 Torr H<sub>2</sub>, the magnitude of the integrated  $OH_{ads}^-$  IR absorbance increased from 0.025 to ~0.09 as exposure temperature increased from 350 to 450 °C (Figure S20). The integrated  $OH_{ads}^-$  absorbance only slightly increased to 0.11 as H<sub>2</sub> pressure was increased to 760 or 5320 Torr (Figure S21). This indicates that the integrated absorbance of ~0.1 is saturation  $OH_{ads}^-$  coverage on CeO<sub>2</sub> due to H<sub>2</sub> exposure. Alternatively, the integrated  $OH_{ads}^$ absorbance during exposure to 76 Torr H<sub>2</sub> at 350 °C was consistently 0.25–0.3 on various Pt/CeO<sub>2</sub> samples (Figure 3b). Increasing the H<sub>2</sub> exposure temperature above 350 °C caused a decrease in the intensity of the  $OH_{ads}^-$  band, associated with H<sub>2</sub>O formation (Figure 3a).

The >2× increase in saturation  $OH_{ads}^-$  coverage on  $Pt/CeO_2$ samples as compared to pure  $CeO_2$  is consistent with the different mechanism of  $H_2$  activation (homolytic vs heterolytic) and the kinetic relevance of converting the heterolytic products to homolytic products for  $H_2O$  formation on pure  $CeO_2$ ; homolytic activation produces  $2OH_{ads}^-$  per  $H_2$ , while heterolytic activation produces  $1OH_{ads}^-$  and  $1 Ce^{4+} - H^-$ . If  $r_{hetero \rightarrow homo}$  were faster than  $r_{H_2O}$  on  $CeO_2$ , then the integrated  $OH_{ads}^-$  absorbance should approach values observed for Pt/  $CeO_2$  (the  $OH_{ads}^-$  concentration should double in converting the heterolytic product to the homolytic product) given that the IR data at temperatures greater than 350 °C demonstrate that  $H_2O$  formation is kinetically feasible.

The in situ FTIR results demonstrate a number of important findings: (1)  $H_2$  activation on pure CeO<sub>2</sub> occurs via heterolytic

 $H_2$  dissociation, followed by equilibration to the homolytic product, before  $H_2O$  formation proceeds; (2) on pure CeO<sub>2</sub>, the rates of Ce<sup>3+</sup> and H<sub>2</sub>O formation are limited by the rate of conversion of the heterolytic H<sub>2</sub> dissociation product to the homolytic product; and (3)  $H_2$  activation on Pt/CeO<sub>2</sub> (even in samples with Pt residing on only 1% of the CeO<sub>2</sub> particles) occurs exclusively via homolytic H<sub>2</sub> dissociation on Pt, followed by rapid and far-reaching H-spillover to CeO<sub>2</sub> at rates much faster than H<sub>2</sub>O formation, allowing for a significant increase in  $OH_{ads}^-$  coverage on  $Pt/CeO_2$  (more than  $2\times$ ) as compared to pure CeO<sub>2</sub>. The results also indicate that Pt indirectly promotes  $H_2O$  (and  $V_0$ ) formation at locations remote from Pt via H-spillover by mitigating the kinetic limitation on pure CeO2 of converting the heterolytic  $H_2$  dissociation products to homolytic  $H_2$  dissociation products. This is also consistent with previous reports that the addition of PGMs on CeO<sub>2</sub> surfaces promotes H<sub>2</sub>O evolution at a lower temperature during H<sub>2</sub>-TPR runs.<sup>23,58,59</sup>

It is worth mentioning that a coincidence of  $[OH_{ads}^-]$  and  $[Ce^{3+}]$  for Pt/CeO<sub>2</sub>, delayed Ce<sup>3+</sup> formation as compared to  $OH_{ads}^-$  on  $CeO_2$ , ~2× lower saturation  $[OH_{ads}^-]$  for  $CeO_2$  than for  $Pt/CeO_2$ , and  $OH_{ads}^-$  on  $CeO_2$  devoid of Pt over a micrometer away from Pt were recently observed in XPS experiments performed at lower H<sub>2</sub> pressures on similar samples (0.8 Torr of H<sub>2</sub> was used in the XPS studies of Ptloaded CeO<sub>2</sub> films with  $\sim$ 10 nm grain sizes, as compared to 75 Torr of  $H_2$  used here in TPR and FTIR experiments). However, no mechanistic discussions associated with the origin of these behaviors apart from the prevalence of H-spillover were provided.<sup>10</sup> The agreement between specific behaviors of  $[OH_{ads}^{-}]$  and  $[Ce^{3+}]$  suggests that the mechanistic details proposed here for the influence of Pt on CeO<sub>2</sub> surface reactions with H<sub>2</sub> are valid over a wide range of H<sub>2</sub> chemical potentials.

# Ce<sup>3+</sup> Distribution on CeO<sub>2</sub> Surfaces

It is evident from Figures 2 and 3 that the addition of Pt to CeO<sub>2</sub> lowered the temperature for H<sub>2</sub> activation and H<sub>2</sub>O formation and increased the saturation  $OH_{ads}^-$  coverage on CeO<sub>2</sub>. Further, these influences spread far from the Pt-CeO<sub>2</sub> interfaces. However, the spatial distribution of OH<sub>ads</sub> and Ce<sup>3+</sup> and the source of increased  $H_2$  consumption and  $Ce^{3+}$ formation on Pt/CeO<sub>2</sub> samples are unclear. To address these questions, the spatial distribution of Ce<sup>3+</sup> was measured using in situ EELS at a high spatial resolution.<sup>60</sup> EELS can distinguish Ce ions with different oxidation states as the M45 edges appear at a lower energy as Ce<sup>4+</sup> is reduced to Ce<sup>3+</sup> and since the relative intensity ratio of the  $M_5$  to  $M_4$  edges  $(I_{M_2}/I_{M_2})$ ) is 1.31 for  $Ce^{3+}$  and 0.91 for  $Ce^{4+}$  (Figure 4a).<sup>15,61,62</sup> In this study, the spatial distribution of Ce4+ and Ce3+ after different thermal treatments was estimated from  $I_{M_s}/I_{M_4}$ . CeO<sub>2</sub> was reduced in situ with H<sub>2</sub> at 450 °C, while Pt/CeO<sub>2</sub> was reduced in situ with  $H_2$  at 400 °C to ensure complete  $H_2$  consumption in <60 min (the time used in between EELS collection on oxidized and reduced samples). The experimental procedures and data analysis associated with the EELS measurements are summarized in Figures S22 and S23.

The spatially resolved EELS data for pure CeO<sub>2</sub> following exposure to 10 Torr of H<sub>2</sub> for 60 min at 450 °C shows that Ce<sup>3+</sup> was homogeneously distributed (Figure 4b, red regions have higher concentrations of  $Ce^{3+}$ ). In contrast, when Pt(2)/CeO<sub>2</sub> (containing Pt NCs) was exposed to 10 Torr of H<sub>2</sub> at 400 °C for 1 h, Ce<sup>3+</sup> was formed heterogeneously across the sample (Figures 4c and S24). By correlation of EELS with TEM images, it is seen that a large fraction of  $Ce^{3+}$  is associated with CeO2-CeO2 interparticle interfaces rather than near Pt NCs, distinct from reports in the literature that hypothesize that reduced Ce species localize at PGMs.<sup>14</sup> Similar to  $Pt(2)/CeO_2$ ,  $Ce^{3+}$  formed heterogeneously throughout the sample when  $Pt(0.01)/CeO_2$  (containing mostly Pt SAs) was exposed to  $H_2$  under similar conditions (Figure S25). Thus, Pt addition changed the spatial distribution of Ce<sup>3+</sup> in powder CeO<sub>2</sub> samples following H<sub>2</sub> exposure.

The spatially averaged Ce<sup>3+</sup> concentrations were quantified for pure CeO<sub>2</sub>,  $Pt(0.01)/CeO_2$ , and  $Pt(2)/CeO_2$  samples from the EELS data following H<sub>2</sub> exposure and agreed with estimates from H<sub>2</sub>-TPR and H<sub>2</sub>-FTIR (Table 2, see Supporting Information Discussion IV for details of calculation). Note that the bulk  $Ce^{3+}$  concentration of 8.1 and 16.3% for pure  $CeO_2$ and  $Pt(0.01)/CeO_2$  in Table 2 represent the Ce<sup>3+</sup> surface coverage of 0.53 and 1.07 monolayers, respectively (assuming that  $CeO_2$  exposes only (111) surface, see Supporting Information Discussion IV for details of calculation). This suggests that surface Ce<sup>4+</sup> remote from Pt (the Pt coverage is very low) were reduced to Ce3+ due to Pt promoting Hspillover, although because of the high Ce<sup>3+</sup> mobility at these temperatures, the formation of  $Ce^{3+}$  at Pt/CeO<sub>2</sub> interfaces followed by diffusion to heterogeneous locations across the sample cannot be ruled out. The EELS data cannot distinguish whether Ce<sup>3+</sup> is associated with OH<sub>ads</sub> or V<sub>o</sub>. However, since both CeO<sub>2</sub> and Pt/CeO<sub>2</sub> samples were reduced with H<sub>2</sub> at elevated temperatures for 60 min, it is likely that some fraction of  $OH_{ads}^-$  condensed to produce  $H_2O$  and  $V_o$ , as deduced from the IR analysis. The EELS maps indicate that a large fraction of Ce<sup>3+</sup> (and likely associated V<sub>o</sub>) localize at and around the boundaries between CeO<sub>2</sub> particles when Pt is added, distinct

Table 2. Comparison of the Estimated Ce <sup>3+</sup> Concentration	ns
from In Situ EEL Spectra vs H <sub>2</sub> -TPR Curves and In Situ H	2-
FTIR Spectra <sup>a</sup>	

sample		$Ce^{3+}$ concentration <sup>b</sup>	
		from EELS $(\%)^c$	from TPR and FTIR (%)
CeO <sub>2</sub>	before reduction	0.3	0 <sup>d</sup>
	after reduction	5.9	8.1 <sup>e</sup>
$Pt(0.01)/CeO_2$	before reduction	$2.3 \pm 0.5$	$0^d$
	after reduction	$11.4 \pm 3.0$	16.3 <sup>f</sup>
$Pt(2)/CeO_2$	before reduction	$2.4 \pm 0.2$	$0^d$
	after reduction	$10.8 \pm 2.9$	18.7 <sup>f</sup>

<sup>a</sup>See Supporting Information Discussion IV for details of calculation. <sup>b</sup>The Ce<sup>3+</sup> concentration is calculated with respect to all Ce (bulk and surface) in the sample. <sup>c</sup>For EELS, CeO<sub>2</sub> was in situ reduced at 450 <sup>o</sup>C for 1 h and Pt/CeO<sub>2</sub> was in situ reduced at 400 <sup>o</sup>C for 1 h in10 Torr H<sub>2</sub>. <sup>d</sup>The Ce<sup>3+</sup> concentration is determined to be 0 because the single-beam intensity from Ce<sup>3+</sup> in the FTIR spectra was 0 before reduction with H<sub>2</sub>. <sup>e</sup>The absorbance from Ce<sup>3+</sup>, estimated from H<sub>2</sub>. FTIR spectra after H<sub>2</sub> reduction at 450 <sup>o</sup>C for 60 min with 75 Torr of H<sub>2</sub>, is used to estimate the Ce<sup>3+</sup> concentration on pure CeO<sub>2</sub>. This is because the Ce<sup>3+</sup> concentration slowly increases over time (Figure 2c). <sup>f</sup>Ce<sup>3+</sup> % on Pt/CeO<sub>2</sub> samples is estimated assuming that all the H<sub>2</sub> reacting with the surface of Pt/CeO<sub>2</sub> (from H<sub>2</sub>-TPR curves) is converted to –OH since Pt promoted the homolytic H<sub>2</sub> dissociation and subsequent H-spillover.

from the common assumption of  $Ce^{3+}$  localization around Pt and from the behavior of pure  $CeO_2$ .

Another interesting observation from the EELS data in Figures 4, S24, and S25 is that for  $Pt(2)/CeO_2$ , the  $Ce^{3+}$  concentration is higher on  $CeO_2$  surfaces near the boundaries between  $CeO_2$  particles as opposed to "free"  $CeO_2$  surfaces. This suggests that both Pt addition and interactions between neighboring  $CeO_2$  particles are required to produce high concentrations of  $Ce^{3+}$  at  $CeO_2$  surfaces following  $H_2$  exposure at high temperatures and that interparticle  $CeO_2$  interactions may be responsible for the excess  $H_2$  consumption (and associated  $Ce^{3+}$  concentration) seen for  $Pt/CeO_2$  samples as compared to pure  $CeO_2$ .

## Irreversible Changes at the Boundaries between CeO<sub>2</sub> Particles by Spillover H

The observation of Ce<sup>3+</sup> localization around CeO<sub>2</sub> interparticle boundaries for Pt/CeO2 samples following H2 exposure suggests the potential for reconstruction at these locations. To analyze potential reconstruction, H<sub>2</sub>-TPR was subsequently executed after an initial reduction-oxidation pretreatment cycle. In Figure 5a, it is observed that the second TPR shows no changes as compared to the first TPR for pure CeO<sub>2</sub>. Alternatively, the  $T_{\rm m}$  for "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:99" occurs at >200 °C lower temperature in the second TPR (Figure 5b). The lower  $T_{\rm m}$  in the second TPR could be due to a milder oxidation of Pt, which could lead to the formation of metallic Pt at a lower temperature and, thus, increased H<sub>2</sub> consumption at a lower temperature. However, when  $Pt(2)/CeO_2$  was exposed to H<sub>2</sub> at 400 °C, diluted with untreated CeO<sub>2</sub> in a 1:99 mass ratio, and oxidized at 350 °C, the consumption of surface-reacting H<sub>2</sub> occurred at a higher temperature than that of the re-oxidized " $Pt(1)/CeO_2$ :  $CeO_2 = 1:99$ " (Figure S26). Moreover, the decrease in  $T_{\rm m}$  in sequential TPR cycles was also observed on  $Pt(0.01)/CeO_2$  that has mostly Pt SAs (Figure S27), although Pt SAs do not significantly sinter under  $H_2$  at 350 °C at these weight loadings (Figure S3).<sup>20</sup> These



**Figure 5.** H<sub>2</sub>-TPR curves of (a) pure CeO<sub>2</sub> and (b) "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:99" (i) after oxidation at 300 °C, (ii) after reduction at (a) 450 °C or (b) 350 °C, and (iii) after (a) reduction at 450 °C followed by re-oxidation at 100 °C or after (b) reduction at 350 °C followed by re-oxidation at 350 °C. (c) Rate of Ce<sup>3+</sup> formation at the boundary of Pt(1)/CeO<sub>2</sub> particles during the 1st and 2nd reductions by using EELS. Sample was re-oxidized at 400 °C after reduction at 400 °C to measure the rate of the 2nd reduction.

results indicate that modified Pt reducibility is not the source of the irreversible change in  $T_{\rm m}$  for Pt/CeO<sub>2</sub> during sequential TPR cycles.

The rate of Ce<sup>3+</sup> formation on "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:99" during sequential isothermal H<sub>2</sub> exposures with re-oxidation in between was also measured via FTIR (Figure S28) and EELS (Figure 5c). In agreement with the TPR data, FTIR data evidenced faster formation rates of OH<sup>-</sup><sub>ads</sub> and Ce<sup>3+</sup> during the second H<sub>2</sub> exposure. EELS also showed that Ce<sup>3+</sup> is formed at a faster rate at the boundaries between CeO<sub>2</sub> particles during the second H<sub>2</sub> exposure (Figure 5c). These results support the hypothesis that spillover H from Pt and the resulting high OH<sup>-</sup><sub>ads</sub> coverage that is unique to Pt/CeO<sub>2</sub> are correlated with the localization of Ce<sup>3+</sup> near the boundaries between CeO<sub>2</sub> particles, which resulted in irreversible changes to the H<sub>2</sub> consumption kinetics.

To suppress the formation of boundaries between CeO<sub>2</sub> particles and examine the influence on the rate of H<sub>2</sub> consumption, CeO<sub>2</sub> and Pt(0.05)/CeO<sub>2</sub> were diluted with  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (diameter, 5 nm) in 1:1 or 1:3 mass ratios (Figure 6a). As an irreducible oxide,  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> can serve as an insulator against H-spillover between CeO<sub>2</sub> particles as the inherent rate of H-spillover on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> is low and the small particle size creates physical barriers between CeO<sub>2</sub> particles.<sup>9</sup> TPR of pure CeO<sub>2</sub> was not affected by dilution with Al<sub>2</sub>O<sub>3</sub> (Figure 6a). On the other hand,  $T_m$  for Pt/CeO<sub>2</sub> shifted to higher temperatures

as the dilution ratio with  $Al_2O_3$  increased (Figure 6b). Also, the amount of  $H_2$  consumed in the TPR decreased to that expected for pure CeO<sub>2</sub> as the dilution ratio increased (Table S2). Although  $Al_2O_3$  may accept some spillover H from Pt,<sup>9</sup> the effect of  $Al_2O_3$  suppressing the formation of the CeO<sub>2</sub>– CeO<sub>2</sub> boundary appears to be more critical in determining the rate and amount of  $H_2$  consumption.

Fast Fourier transforms (FFTs) of STEM images of collections of pure CeO<sub>2</sub> and Pt/CeO<sub>2</sub> particles before and after in situ reactions with H<sub>2</sub> were generated to investigate physical changes at the interfaces between CeO<sub>2</sub> particles (Figure 6c). FFTs of STEM images containing  $CeO_2$  particles with randomly oriented grains exhibit complex diffraction patterns due to the electron beam scattering off variously oriented crystal planes, whereas collections of oriented particles produce simpler diffraction patterns similar to single-crystal materials (Figure S29). FFTs of pure CeO<sub>2</sub> particles exhibited a variety of diffraction patterns before and after in situ reaction with  $H_2$  at 450 °C (Figure 6c). Prior to exposure to H<sub>2</sub>, Pt/CeO<sub>2</sub> particles similarly show a wide variety of diffraction patterns. However, the number of observed diffraction patterns decreased for Pt/CeO<sub>2</sub> after in situ reaction with  $H_2$  at 400 °C (Figure 6c), although no signs of CeO<sub>2</sub> aggregation were observed from XRD patterns and surface area measurements (Figure S30). The change in the diffraction pattern for  $Pt/CeO_2$  following exposure to  $H_2$  at



**Figure 6.** H<sub>2</sub>-TPR curves of (a) CeO<sub>2</sub> or (b) Pt(0.05)/CeO<sub>2</sub> are collected after dilution with  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> in a 1:1 or 1:3 mass ratio. (c) STEM images and FFTs for the corresponding region of pure CeO<sub>2</sub> and Pt/CeO<sub>2</sub> before and after in situ reduction with H<sub>2</sub> at 450 °C for pure CeO<sub>2</sub> and at 400 °C for Pt/CeO<sub>2</sub>.

elevated temperatures suggests re-orientation of the CeO<sub>2</sub> particles to produce aligned grains. Note that the lattice alignment between CeO<sub>2</sub> particles (which is often termed oriented attachment and is noted to have relatively small activation barriers<sup>63,64</sup>) and the removal of some of the surface oxygen are the prerequisites for CeO<sub>2</sub> sintering.<sup>65–67</sup> The underlying principle of particle sintering is to lower the surface energy. Thus, it seems likely that particle orientation in the Pt/CeO<sub>2</sub> samples during H<sub>2</sub> exposure is thermodynamically driven by minimizing surface energy and that the high OH<sup>-</sup><sub>ads</sub> coverage caused by H-spillover from Pt facilitates this process.<sup>65–67</sup> These results also suggest that the interaction between CeO<sub>2</sub>–CeO<sub>2</sub> particles with aligned interfaces may facilitate faster kinetics of H<sub>2</sub> consumption and the increased total H<sub>2</sub> consumption seen in Pt/CeO<sub>2</sub> samples.

## DISCUSSION

This work demonstrates that the addition of Pt to  $CeO_2$  particles changes the reaction mechanism of  $H_2$  with the  $CeO_2$  surfaces in powder catalysts and facilitates broad reconstruction of the system.  $H_2$  is predominantly heterolytically dissociated on pure  $CeO_2$  surfaces (eq 2), and  $H_2O$  evolution

is limited by the transformation of heterolytic H<sub>2</sub> dissociation products into homolytic products (eq 3). On the other hand, H<sub>2</sub> homolytically dissociates on Pt in Pt/CeO<sub>2</sub> at a lower temperature compared to on pure CeO<sub>2</sub> surfaces, followed by rapid H-spillover to the CeO<sub>2</sub> surface, even across CeO<sub>2</sub>- $CeO_2$  interparticle contacts (eq 7). This results in more than twofold higher saturation surface OH<sub>ads</sub> coverage on Pt/CeO<sub>2</sub> compared to pure  $CeO_2$  (Figure 3). The direct production of the homolytic dissociation product enables H<sub>2</sub>O evolution on  $Pt/CeO_2$  at a lower temperature compared to that on pure  $CeO_2$ ,<sup>19,46,47</sup> since  $H_2O$  evolution is no longer limited by the transformation of heterolytic products into homolytic products as is the case for pure  $CeO_2$ . Interestingly, following  $H_2$ exposure to Pt/CeO<sub>2</sub>, the accumulation of  $Ce^{3+}$  and V<sub>o</sub> was observed at the boundary between CeO2 particles instead of at Pt-CeO<sub>2</sub> interfaces. That is, the impact of Pt is global across the whole sample (both in terms of H-spillover and promoted  $V_0$  formation) and not just at the Pt-CeO<sub>2</sub> interface.

Here, note that the H-spillover can occur across  $CeO_2$  particles in physical contact, from  $CeO_2$  particles containing Pt to Pt-free  $CeO_2$  particles (e.g., in "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:99"). This allowed us to separate the roles of the Pt structure,

 $Pt-CeO_2$  contact, and  $CeO_2-CeO_2$  interactions in  $H_2$  reactions with  $CeO_2$  surfaces. The rate of interparticle H-spillover could depend on many different variables related to  $CeO_2$  particles (e.g., the contact area or the extent of lattice alignment of  $CeO_2$  particles). However, detailed insights into interparticle H-spillover are lacking and further investigation is required.

An interesting aspect of our findings is that the inclusion of insulating physical spacers (small  $Al_2O_3$ ) between  $CeO_2$ particles containing Pt increases the  $T_{\rm m}$  and decreases the amount of H<sub>2</sub> consumed in TPR experiments (Figure 6b and Table S2). This indicates that the lattice alignment of  $CeO_2$ particles during the exposure to H<sub>2</sub> at elevated temperatures that is observed for  $Pt/CeO_2$  must occur during the  $H_2$ consumption process, and this alignment promotes the rate of low-temperature H<sub>2</sub> consumption and provides a mechanism for enhanced H<sub>2</sub> consumption. It is noted that the activation barrier for the oriented attachment of CeO<sub>2</sub> particles is reported to be 54 kJ/mol for  $CeO_2$  particles with a diameter of  $\sim 2$  nm and 95 kJ/mol for CeO<sub>2</sub> particles with a diameter of  $\sim$ 60 nm, suggesting that the oriented attachment of the 8 nm CeO<sub>2</sub> particles studied here would have accessible barriers at the  $T_{\rm m}$  observed in our studies.<sup>64–66</sup> It is presumed that the CeO<sub>2</sub> surfaces at the aligned grains of CeO<sub>2</sub> particles can stabilize higher concentrations of surface hydroxyls as well as  $Ce^{3+}$  and are inherently more reactive. It is also possible that the rate of H-spillover at the aligned boundaries is faster than on misaligned grains, thus promoting the rate of  $H_2$ consumption on Pt/CeO<sub>2</sub>.

The alignment of CeO<sub>2</sub> particles after reduction could be related to the accumulation of V<sub>o</sub> at the boundaries. It has been reported that the presence of  $V_{\text{o}}$  at grain boundaries (GBs) of CeO<sub>2</sub> with abrupt structural and chemical inhomogeneities can relax the local structural distortion and stabilize the GB structure.<sup>68,69</sup> GBs are defects in oxide crystals and distinct from boundaries formed between oxide nanoparticles studied here because the nanoparticles do not form contiguous interfaces. However, they share similarities in that both boundaries exhibit abrupt structural and chemical inhomogeneities. Therefore, we hypothesize that the role of V<sub>o</sub> at the boundary between CeO<sub>2</sub> particles is similar to the role of V<sub>o</sub> at GBs. That is,  $V_0$  localized at the boundaries of CeO<sub>2</sub> particles may induce significant surface reconstruction that stabilizes the oxygen nonstoichiometric  $CeO_2$  (e.g., oxygen-deficient) surfaces facing each other. This surface reconstruction may accompany the oriented attachment of CeO<sub>2</sub> particles since  $CeO_2$  particles are mobile, and the structural inhomogeneity is loosened when  $CeO_2$  grains align with each other.

However, it is unclear whether  $V_o$  promotion of the lattice alignment of CeO<sub>2</sub> particles is a direct result of the presence of  $V_o$  at the boundaries between CeO<sub>2</sub> particles or the indirect result of  $V_o$  changing the energetics of H<sub>2</sub>O formation/ desorption steps. The formation of  $V_o$  on CeO<sub>2</sub> surfaces proceeds in 2 steps: (i) recombination of surface hydroxyls (eq 4) followed by (ii) H<sub>2</sub>O desorption (eq 5).  $V_o$  stabilizing the boundaries of nonstoichiometric CeO<sub>2</sub> particles suggests that the surface energy of the product from (i) and (ii) becomes lower. Therefore, it might be possible that step (i) becomes favored at the aligned boundary of CeO<sub>2</sub> particles as a result of a change in reaction energetics. In this case, the high hydroxyl coverage on Pt/CeO<sub>2</sub> attained by the activation of homolytic H-spillover from Pt would be sufficient to promote the alignment or oriented attachment of CeO<sub>2</sub> particles. The accelerated H<sub>2</sub> consumption over "Pt(1)/CeO<sub>2</sub>:CeO<sub>2</sub> = 1:99" after reacting with H<sub>2</sub> at 250 °C, at which temperature H<sub>2</sub>O evolution occurs very slowly, supports this inference (Figure S28), although this requires more rigorous investigation.

While the EELS mapping suggests that Ce<sup>3+</sup> and V<sub>o</sub> accumulate at  $CeO_2$ -CeO<sub>2</sub> boundaries after reacting with H<sub>2</sub> at 400 °C, the question remains as to where  $H_2O$  formation occurs in Pt/CeO<sub>2</sub>. H<sub>2</sub>O formation and V<sub>o</sub> formation could occur directly at the CeO2-CeO2 boundaries, but could also occur at the Pt-CeO<sub>2</sub> interface followed by migration of V<sub>o</sub> to the CeO<sub>2</sub>-CeO<sub>2</sub> boundaries. Although more research is needed to understand which of these two pathways is dominant, it should be emphasized that the H<sub>2</sub> activation step and the V<sub>o</sub> formation step may not necessarily occur in the same location. For example, the formation of H<sub>2</sub>O may occur at the  $CeO_2-CeO_2$  boundaries due to the increased concentration of surface hydroxyls that migrated from Pt through H-spillover. It would be interesting to compare the  $Ce^{3+}$  distribution after reacting with a reducing agent known to be less mobile than H (e.g., CO or hydrocarbons) to better understand how CeO<sub>2</sub>-CeO<sub>2</sub> interparticle boundaries participate in the V<sub>o</sub> formation step.

It is worth emphasizing that this study was conducted on a powder catalyst with a high surface area in which CeO<sub>2</sub> particles form interacting boundaries between each other. This is different from the model structures commonly studied in surface science studies, where the surface extends infinitely and forms no CeO<sub>2</sub>-CeO<sub>2</sub> boundaries or interfaces. It will be important to consider these materials gaps when linking the conclusions of surface science analyses with the findings from this study and other studies on high-surface area CeO<sub>2</sub> materials. For example, a surface science study conducted on a model surface where Pt was dispersed on a  $CeO_2(111)$  film grown on a Cu(111) single crystal reported that H-spillover from Pt to  $\text{CeO}_2$  occurs primarily below 25  $^\circ\text{C}\textsc{,}$  and oxygen spillover from CeO<sub>2</sub> to Pt occurs more preferentially above 150 °C.<sup>11</sup> These conclusions are somewhat different from those reached in the current study. It is possible that the presence of boundaries between CeO<sub>2</sub> particles in the Pt/  $CeO_2$  powder altered the role of Pt in influencing the  $CeO_2$ surface reactions with  $H_2$ . Our observation that  $Ce^{3+}$  is preferentially formed at and around the boundary between  $CeO_2$  particles (Figure 4) and that both the rate and amount of  $H_2$  consumption on Pt/CeO<sub>2</sub> decrease when the formation of boundaries between  $CeO_2$  particles is suppressed (Figure 6b) and Table S2) support this inference. Recently, we also found that the Pt sintering is promoted when the lattice of neighboring CeO<sub>2</sub> particles are aligned.<sup>70</sup> That is, the boundary between CeO<sub>2</sub> particles not only affects how H<sub>2</sub> reacts with Pt/CeO<sub>2</sub> but also plays an important role in determining the thermal stability of dispersed Pt species. Although the boundaries between support particles are often neglected when constructing model structures, a more in-depth consideration of the boundaries between support particles may be important for understanding the properties of powder heterogeneous catalysts.

# CONCLUSIONS

In this contribution, reactions of  $H_2$  on high-surface area Pt/ CeO<sub>2</sub> powder catalysts were examined.  $H_2$  is predominantly heterolytically dissociated on pure CeO<sub>2</sub> surfaces, and  $H_2O$ evolution (and Ce<sup>3+</sup> formation) is limited by the transformation of heterolytic  $H_2$  dissociation products into homolytic products. Pt drives the homolytic dissociation of H<sub>2</sub> and subsequent H-spillover at a much lower temperature compared to heterolytic  $H_2$  dissociation on pure CeO<sub>2</sub> surfaces. The change in the mechanism of H<sub>2</sub> dissociation on the CeO<sub>2</sub> surface caused by Pt results in a >2× larger saturation surface hydroxyl coverage on Pt/CeO<sub>2</sub> compared to pure CeO<sub>2</sub>. As a result of Pt promoting H-spillover onto CeO<sub>2</sub> surfaces, the conversion of heterolytic H<sub>2</sub> dissociation products into homolytic products no longer restricts the rate of H<sub>2</sub>O evolution on the Pt/CeO<sub>2</sub> surface, and thus, it seems that H<sub>2</sub>O evolution is promoted across CeO<sub>2</sub> surfaces, even far away from Pt. In addition, spatially resolved in situ EELS show that Ce<sup>3+</sup> formed by spillover H preferentially localize at and around the boundary between CeO<sub>2</sub> particles rather than localizing around Pt, suggesting that the H<sub>2</sub>O evolution may occur mainly at the CeO<sub>2</sub>-CeO<sub>2</sub> boundary. That is, the impact of Pt on surface reactions on CeO<sub>2</sub> is global and not just localized at the Pt-CeO<sub>2</sub> interface. The V<sub>o</sub> formation (as a result of  $H_2O$  evolution) at the boundary between  $CeO_2$ particles for Pt/CeO<sub>2</sub> is thought to induce surface reconstruction and drive the alignment of neighboring CeO<sub>2</sub> particles to produce boundaries that stabilize the higher concentrations of surface hydroxyls and Ce<sup>3+</sup> and enable the faster rates of H-spillover. This study provides new insights into the PGM-promoted  $CeO_2$  surface reactions with  $H_2$ which are essential for understanding various catalytic processes, by considering the effect of the spatial heterogeneity of powder catalysts.

# ASSOCIATED CONTENT

# **Supporting Information**

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/jacsau.3c00330.

Experimental details, XRD pattern, CO-IR spectra, CO oxidation reactivity data, additional  $H_2$ -TPR curves, additional IR spectra collected under  $H_2$ , additional STEM imaging and EELS mapping, in situ XANES spectra collected under  $H_2$ , summary of DFT studies over  $H_2$  reaction with CeO<sub>2</sub> surfaces, additional discussions on the reactivity of Pt SAs vs NCs, quantitative interpretation of IR, assignment of the IR band at ~2130 cm<sup>-1</sup>, EELS data collection, and Ce<sup>3+</sup> % estimation using EEL spectra,  $H_2$ -TPR curves, and FTIR spectra (PDF)

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## **Author Contributions**

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Notes

The authors declare no competing financial interest.

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