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A STUDY OF TRANSPORT PHENOMENA AND INTERFACE STABILITY DURING SOLIDIFICATION OF BINARY SOLUTIONS USING FRONT TRACKING FINITE ELEMENTS

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A STUDY OF TRANSPORT PHENOMENA AND INTERFACE STABILITY DURING SOLIDIFICATION OF BINARY SOLUTIONS USING FRONT TRACKING FINITE ELEMENTS

H.-L. Tsai (Ph.D. Thesis)

May 1984

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### A STUDY OF TRANSPORT AND INTERFACE STABILITY DURING SOLIDIFICATION OF BINARY SOLUTIONS USING FRONT TRACKING FINITE ELEMENTS

Hai-Lung Tsai

Ph.D. Thesis

### Lawrence Berkeley Laboratory University of California Berkeley, California 94720

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May 1985

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by

HAI-LUNG TSAI

#### ABSTRACT

A new numerical method using "front tracking" finite elements was developed to solve the multi-dimensional transient heat and mass transfer equations associated with the solidification of bindary solutions. The energy balance equation at the interface was not treated as a boundary condition, but rather as an independent equation whose solution gave the new position of interface. A special new method was developed by which the interface was tracked in time by two steps: first the magnitude of displacement and the normal direction were independently obtained for each node on the interface, then they were superimposed to determine the new interface position. The numerical method can incorporate realistic thermodynamic conditions on the interface (including the effects of interfacial tension) and can accommodate the non-isothermal as well as the irregular but smooth interface. Α novel meshing system based on a systematic exponential gridding concept was developed to yield accurate temperatures in the solid and liquid phases and concentration distribution in the liquid.

This numerical method was then employed to study the transport phenomena as well as the morphological stability of a planar interface during a solidification process in a saline solution. The transient temperature and concentration distributions in the solid and liquid regions for both planar and curved interfaces were calculated.

To study the stability of a planar interface, several numerical perturbations in space and/or time were performed on either the outer boundary or the interface to simulate various physical effects. The results indicate that for the analyzed conditions, a temperature perturbation on the outer surface of the domain cannot generate the instability of the moving interface, even in a situation in which the solute in front of the interface is constitutionally supercooled. Furthermore, it appears that an increased solute concentration on the interface has a stabilizing effect. These results, obtained through rigorous analysis, contradict the existing interface morphology stability criteria based on concepts of equilibrium thermodynamics prevalent in the technical literature. The new results indicate the importance of the transient dynamic effects in the study of solid-liquid interface stability criteria. These effects have been neglected in previous work.

The numerical study was also used to investigate the effects of concentration perturbations on the solid-liquid interface on the stability of the interface. It was shown that a continuous concentration perturbation can lead to an unstable interface. These results demonstrate the importance of the new numerical method in the study of solidification processes and indicate the need for future studies to promote a fundamental understanding of the physical phenomena associated with the perturbed growth of a solid-liquid interface during solidification.

Bois Rubinst

Chairman, Thesis Committee

## Dedicated, with love, to my parents.

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I wish to thank my wife, Chiu-Liang, for her love, help and understanding during this study.

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### NOMENCLATURE

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[A <sub>i</sub> ]	constant matrices, $i = 1, 2$ , and 3; Eq. (3.60)
A	amplitude of temperature perturbation, Eq. (4.1)
Α'	amplitude of temperature perturbation, Eq. (4.2)
[B]	temperature gradient interpolation matrix, Eq. (3.4)
B	amplitude of concentration perturbation, Eq. (4.3)
[C]	conductance matrix, Eq. (3.10)
C	solute concentration
C <sub>S</sub> , C <sub>1</sub>	solute concentration of the solid phase
C <sub>L</sub> , C <sub>2</sub>	solute concentration of the liquid phase
C <sub>i</sub> , C <sub>∞</sub>	initial solute concentration in the liquid phase
Co	solute concentration on the interface
с	thermal capacity
c <sub>S</sub>	thermal capacity of the solid phase
cL	thermal capacity of the liquid phase
D	mass diffusivity of the liquid phase
f[]	Newton's divided difference, Eq. (3.62)
F(w)	defined in Eq. (1.21)
{g}	defined in Eq. (3.40)
G(ω)	defined in Eq. (1.20)
GC	concentration gradient of liquid phase at the interface
ցլ	temperature gradient of liquid phase at the interface
GS	temperature gradient of solid phase at the interface
h	convection heat transfer coefficient

[J] Jacobian transformation matrix

|J| Jacobian of transformation

[K] defined in Eq. (3.10)

 $[K_i]$  i = c, h, and r; defined in Eq. (3.10)

 $k = C_S / C_L$  partition coefficient

 $k_{S}$ ,  $k_{1}$  thermal conductivity of the solid phase  $k_{L}$ ,  $k_{2}$  thermal conductivity of the liquid phase L latent heat of fusion

m \_\_\_\_\_\_slope of liquidus line

[N] displacement interpolation vector

{dn} displacement vector

n normal direction of the interface

 $\vec{n}_1$  normal direction of the solid boundary

 $\vec{n}_2$  normal direction of the liquid boundary

P defined as P = 1 - k

 $P_n(x)$  interpolation polynomial, Eq. (3.61)

q heat flux, Eq. (3.8)

q<sub>r</sub> radiation heat flux, Eq. (3.8)

 $\{R\}$  defined in Eq. (3.10)

 $\{R_i\}$  i = t, q, h, r; defined in Eq. (3.10)

r-z Cartesian coordinates

 $S(\omega)$  defined in Eq. (1.18)

S(t) moving phase interface

S<sub>i</sub> i = 1, 2, 3, and 4; part of the moving interface, Eq. (3.8) {T} temperature matrix, defined in Eq. (1.18)

T <sub>S</sub> , T <sub>1</sub>	temperature of the solid phase
T <sub>L</sub> , T <sub>2</sub>	temperature of the liquid phase
T <sub>i</sub> , T <sub>∞</sub>	initial temperature of the liquid phase
Tm	melting point of pure liquid in plane interface
Τo	temperature on the interface
t	time
∆t <sub>cr</sub>	critical time step in numerical stability analysis
[U]	matrix of directional cosine of elements on the interface
{ <b>v</b> }	defined in Eq. (3.37)
v	constant interfacial velocity
vn	interfacial normal velocity
W <sub>i</sub> , W <sub>j</sub>	Gauss weights
W(τ)	weighting function
Δ×	mesh size in the x direction
x-z	Cartesian coordinates
ας, α1	thermal diffusivity of the solid phase
α <sub>L</sub> , α <sub>2</sub>	thermal diffusivity of the liquid phase
<sup>a</sup> i	i = 1, 2, 3; coefficients in the boundary condition, Eq. (2.8)
<b>ξ-</b> η	natural or local coordinates
ω	wave frequency, Eq. (1.16)
ωc	defined in Eq. (A.13)
ω	defined in Eq. (A.17)
ωs	defined in Eq. (A.18)
[¢]	interpolation matrix
Φi	interpolation functions

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φ(x,t)	perturbed interface, Eq. (1.16)
[Φ]	defined in Eq. (3.36)
λ <sub>i</sub>	eigenvalues
δ	amplitude of perturbation, Eq. (1.16)
٤s	averaging thermal conductivity of the solid phase, Eq. (A.31)
٤L	averaging thermal conductivity of the liquid phase, Eq. (A.31)
Ω	sum of the solid and liquid domains
Ω1(t)	solid domain
Ω2 <b>(t)</b>	liquid domain
<b>2Ω1(t)</b>	total boundary of the solid domain
∂Ω <b>2(t)</b>	total boundary of the liquid domain
Γ1 <b>(t)</b>	boundary of the solid domain, excluding the interfacial boundary S(t)
$\Gamma_2(t)$	boundary of the liquid domain, excluding the interfacial boundary S(t)
Υ.	mean interfacial curvature
[^]	defined in Eq. (3.39)
σ	Stefan-Boltzmann constant
ε	radiation emissivity
θ	defined in Eq. (3.21)
ρ	density
۶	density of the solid phase
ρ	density of the liquid phase
R	radius of interfacial curvature

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# 4.1 Thermophysical properties of dilute saline solution and ice.

# CHAPTER 1:

1

### INTRODUCTION

### 1.1 SOLIDIFICATION OF SOLUTIONS

### 1.1.1 Applications and Difficulties

Many practical problems in applied science and engineering involve the solidification of solutions, e.g., ice-making, food processing, medicine, crystallography, metallurgy, welding, and many others. Because the properties of the solid phase are not only inherited from its liquid phase, but significantly influenced by the details of the solidification process, a fundamental understanding of the physical phenomena occurring during solidification processes is essential to obtain the desired structure of the solid phase [1-4]. For example, properties of alloys such as strength, toughness, corrosion resistance, etc. are dependent in part on the degree of local segregation resulting from the spacing of dendrite arms in the alloys. By properly controlling the solidification procedures, properties of the alloy can be improved. Of course, the alloy properties can be modified by subsequent heat treatment, but the existing defects cannot in general be completely removed by these additional costly procedures. An important application of current solidification technology may be found in the crystal growth of electronic materials, which requires an understanding of solidification processes [5,6].

From a theoretical point of view, the analysis of solidification of solutions involves the simultaneous solution of transient heat and mass transfer equations in both time-dependent solid and liquid domains. The difficulties of studying solidification stem in part from the nonlinearities associated with the transient position of the interface that separates the two phases with quite different properties. The interface is neither fixed in space nor is its motion known *a priori*; nevertheless, it is part of the solution. Furthermore, the governing heat and mass transfer equations in both phases are nonlinearly coupled at the unknown moving interface.

The most difficult aspect in the study of solidification processes is the establishment of a valid mathematical model that fully describes the real physical phenomena. An incomplete knowledge of, for example, anisotropic interfacial free energy, interfacial structure and kinetics (molecular attachment), etc. and their roles in determining the interface stability leads to difficulties in obtaining the correct governing equations. The phenomena on a molecular level occurring near the interface are intimately related to the fundamental physics of the materials. Although the basic principle of interface instability is partially answered by the supercooling effect, the dendritic shape, dendritic arm spacing, and sidebranching can hardly be predicted. Hence, a complete theoretical simulation of interface time evolution, from a plane transition to instability, to a complete dendrite is still beyond the scope of present technology.

Experimental studies on solidification of solutions are limited to phenomena observations and the empirical correlations among such parameters as the degree of supercooling, dendrite growing velocity,

dendrite arm spacing, etc. [7-17]. These results may have practical importance and value, but do not make significant contributions to the fundamental understanding of solidification processes. It is noted that the comparison between experimental results and those predicted by the over-simplified theories are not adequate. Direct measurements of interfacial free energy and temperatures and solute distributions. especially around the dendrite, are complicated by the moving, microscopic size, and complicated geometry of dendrites.

### 1.1.2 Thermal and Solute Redistributions

\* <u>\*</u>

A simple example will be given to illustrate the physical phenomena occurring during solidification in solutions. Extensions to the more general cases will be discussed thereafter. The example is for the semiinfinite domain shown in Fig. 1.1. The media is a bindary saline solution, at a uniform initial temperature  $T_i$  and concentration  $C_i$ . It is assumed that  $T_i$  is higher than the equilibrium temperature corresponding to C<sub>i</sub>, i.e., the solution is not supercooled. Adiabatic boundary conditions are assumed on both upper and lower sides of the domain. At time t = 0 the cooling process is initiated by suddenly imposing a constant temperature  $T_{\infty}$ , which is less than  $T_i$ , on the left outer surface of the domain. If T\_ is below the freezing point of the solution, the solidification process will start from the left and the solid-liquid interface will move to the right of the domain. This process is called "unidirectional solidification." It is noted that the solution freezing temperature is determined by the amount of solute contained in the solution. In general, the liquid phase and its solidified phase possess



FIGURE 1.1. A unidirectional solidification process and interface time evolution,  $t_3 > t_2 > t_1$ .

quite different thermodynamic, chemical, and physical properties. Problems involving solidification or melting usually are referred to as "phase change," "moving boundary," "free boundary," or "Stefan" problems.

The temperature difference between the outer surface temperature and the interfacial temperature serves as the driving thermal force for advancing the interface. The sensible heat of the solid and the latent heat released by the freezing process are transported through the frozen solid layer by conduction, then rejected into the environment. The sensible heat from the liquid solution will also be transported by conduction and/or convection to the interface and then conducted through the solid region. Hence, the rate of solidification is governed in part by the rate at which the latent heat of fusion generated at the interface can be removed.

According to the phase diagram for solution, the liquid phase and its solid phase will not contain the same amount of solute. Hence, during the solidification of solutions, solute will be partially rejected or incorporated by the solid according to the phase diagram. A partition coefficient takes a value varying from zero (corresponding to the complete rejection of solute by the solid phase) to one (corresponding to the complete incorporation of solute into the solid phase). Usually the solid "phase will contain less solute than the liquid phase, as indicated by the negative slopes of the solidus and liquidus lines in the phase diagram (see Fig. 1.2). The accumulation of rejected solute in the change of phase interface will lower the change of phase temperature. Thus the rate of solidification for solution is determined by the heat transfer



FIGURE 1.2. (a) Part of phase diagram, k < 1. (b) Concentration distribution in a steady-state solidification process.

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process as well as the mass transfer process [18-30]. Due to the low mass diffusion coefficient in the liquid phase, the rejected solute will form a thin solute-rich layer in front of the phase interface. This is called the "concentration boundary layer." This phenomenon of solute segregation will cause the change of phase interface to become unstable according to the various interface stability theories, discussed in Section 1.2.

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Many experiments have found that during the solidification of solutions, the solid-liquid interface is initially planar, but after some time the planar interface will become unstable [31-33,38]. The time evolution of the interface is also shown in Fig. 1.1. It is seen that at the onset of the process a small sinusoidal protuberance appears on the interface. This perturbed interface will then grow outward and become "finger-like" dendrites. This process is called cellular or dendritic growth. Although the name dendrite originates from the Greek word for "tree," we will use it in a more general sense. The shape of the dendrites depends greatly on the properties of the solution and the freezing conditions. In fact, finger-like, tree-like, and many other extremely complex geometries of dendrites have been observed. The dendrite is a microstructure with sizes ranging from a few to approximately 100 µm in diameter. The tips of dendrites are growing faster than the body and eventually neighboring dendrites will merge. Most of the rejected solute will be trapped between dendrites. The dendrite is a more stable form of interface. It is noted that the phenomena of interface instability and dendritic growth are only found when the liquid to be frozen contains solutes or is initially supercooled. For

pure liquids that are not supercooled, the interface is always stable and planar.

A general solidification process of multi-component solutions involves the same basic phenomena as described above. Having many kinds of solutes present with the solution will further complicate the solute redistribution process and may result in the inter-reaction among these solutes. Hence, analyzing a uni-directional solidification process in binary solutions provides us with an optimal opportunity for the study of the fundamental characteristics of solidification processes in solutions and solid-liquid interface morphology [28-30, 32-36].

### 1.2 SOLID-LIQUID INTERFACE STABILITY

#### 1.2.1 Constitutional Supercooling

The interface stability theories will be reviewed in detail here because later work will make extensive reference to this section. Chapter 5 will include commentary on the assumptions from which the theories were derived.

Consider the uni-directional solidification of a binary alloy similar to that in Fig. 1.1, having a partition coefficient k which is assumed constant and less than unity. The relevant part of the phase diagram is shown in Fig. 1.2. The melt is initially at uniform solute concentration  $C_{\infty}$ . The temperature boundary conditions are varied in such a way that the solid-liquid interface advances at a constant velocity V. Steadystate temperature and concentration profiles are assumed in a frame of reference moving with the interface. This signifies that the rate of

solute rejection by the solid during solidification will exactly equal the diffusion rate of solute away from the interface into the liquid. The composition profiles remain constant relative to the interface and are shown in Fig. 1.2. The solute distribution can be determined by solving the diffusion equation in a moving coordinate frame of reference attached to the interface:

$$D \frac{\partial^2 C_L}{\partial z^2} + V \frac{\partial C_L}{\partial z} = 0 , \qquad (1.1)$$

where z is the distance from the interface, D is the solute diffusivity, and V is the constant velocity of the interface. The boundary conditions are:

$$C_{L} = C_{\infty}/k = C_{0} = C_{S}/k \quad \text{at } z = 0$$

$$C_{L} + C_{\infty} \quad \text{at } z + \infty$$

The solution can be obtained easily as:

$$C_{L} = C_{\infty} \left[ 1 + \left( \frac{1-k}{k} \right) \exp \left( -Vz/D \right) \right]. \qquad (1.2)$$

Since

$$C_0 = C_{\infty} \left( 1 + \frac{1-k}{k} \right)$$
 (1.3)

$$G_{C} = C_{\infty} \left( \frac{1-k}{k} \right) \left( -\frac{V}{D} \right) = \left( -\frac{V}{D} \right) (C_{L} - C_{S}) , \qquad (1.4)$$

hence

$$C_{L} = C_{0} + \frac{DG_{C}}{V} \left[ 1 - \exp(-Vz/D) \right],$$
 (1.5)

where  $G_{C}$  is the concentration gradient in the liquid at the interface, Co is the solution concentration at the interface, and k is the partition

coefficient,  $k = C_S/C_L$ .

In addition to the solute distribution derived in Eq. (1.2), it also is necessary to know the thermal diffusion fields of both liquid and solid. The governing equations are similar to that of Eq. (1.1)for temperatures in the liquid and solid states.

$$\alpha_{L} \frac{\partial^{2} T_{L}}{\partial z^{2}} + V \frac{\partial T_{L}}{\partial z} = 0$$
 (1.6)

$$\alpha_{\rm S} \frac{\partial^2 T_{\rm S}}{\partial z^2} + V \frac{\partial T_{\rm S}}{\partial z} = 0 , \qquad (1.7)$$

where  $\alpha_{L}$  and  $\alpha_{S}$  are the thermal diffusivities of the liquid and solid, respectively. The boundary conditions are  $T_{S} = T_{L} = T_{0}$  at the interface z = 0; the temperature gradients in the interface are given by  $G_{L}$ for liquid and  $G_{S}$  for solid at z = 0. With these boundary conditions, the solutions of Eqs. (1.6) and (1.7) are:

$$T_{L} = T_{0} + \frac{\alpha_{L}G_{L}}{V} \left[1 - \exp\left(-\frac{\sqrt{z}}{\alpha_{L}}\right)\right]$$
(1.8)

$$T_{S} = T_{0} + \frac{\alpha_{S}^{G}S}{V} \left[ 1 - \exp(-Vz/\alpha_{S}) \right],$$
 (1.9)

where  $T_0$  is the equilibrium temperature common to solid and liquid at the interface. Notice that Eqs. (1.5), (1.8), and (1.9) have exactly the same form.

Close to the moving interface, the temperature distribution can be expanded in a Maclaurin Series to yield

$$T_{l} = T_{0} + G_{L} z$$
 (1.10)

$$T_{S} = T_{0} + G_{S} z$$
 (1.11)

The energy balance on the interface is given by:

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$$k_{S}G_{S} - k_{L}G_{L} = LV$$
, (1.12)

where L is the latent heat of fusion per unit volume, and  $k_{S}$  and  $k_{L}$  are the thermal conductivities of solid and liquid, respectively.

To study interface stability, the actual temperature distributions and the equilibrium liquid temperature corresponding to the actual concentration are plotted in Fig. 1.3. The equilibrium temperature is directly related to the solute distribution in Eq. (1.5) by means of the phase diagram. From Fig. 1.3, it is seen that there is a region in the liquid near the interface that is supercooled, i.e., the equilibrium phase transformation temperature for the specific concentration at a given location is above the actual temperature. It is noted that in the supercooled region near the interface, the degree of supercooling increases in a direction away from the interface. Thus, if any part of the interface is perturbed and the "tip" of the protuberance enters the increasing supercooled region, it will grow faster and the interface will become unstable.

The above phenomenon is known as "constitutional supercooling" instability. The word constitutional indicates that the supercooling arises from a change in composition. Constitutional supercooling as a cause for interface instability was first proposed by Chalmers and coworkers in 1953 [37], in conjunction with experiments in directional





crystallization of dilute tin alloys. In order for constitutional supercooling instability to occur, the gradient of the equivalent liquidus temperature curve on the change of phase interface (see Fig. 1.3) must be larger than the actual temperature gradient in the liquid on the interface. Thus, if the equilibrium temperature is given in terms of the concentration by

$$T_e = T_m + mC_L$$
, (1.13)

where  $T_m$  is the pure liquid melting point and m is the liquidus slope (assumed to be constant), then constitutional supercooling implies that for an unstable interface,

$$\frac{G_{L}}{G_{e}} = \frac{G_{L}}{mG_{C}} = \frac{G_{L}}{mC_{\infty} \left(\frac{1-k}{k}\right)\left(-\frac{V}{D}\right)} < 1 \qquad (1.14)$$

or

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$$\frac{G_{L}}{V} < \frac{1-k}{k} \left( \frac{mC_{\infty}}{D} \right) , \qquad (1.15)$$

where  $G_{C}$  is the concept of concentration gradient at the interface given in Eq. (1.4).

In summary, the constitutional supercooling was derived under the following assumptions:

- The planar interface is moving at a constant velocity in the z-direction.
- (2) The domain is one-dimensional and semi-infinite in the z positive direction.
- (3) The interface is sharp and infinitesimally thin.
- (4) The interface is in thermodynamic equilibrium.

- (5) The interfacial kinetics are negligible.
- (6) A moving coordinate is used and attached on the interface such that z = 0 marks the plane of interface.
- (7) The temperature and concentration distributions of both solid and liquid are steady-state in the moving coordinate system.
- (8) There is no convection in the liquid, and no change of density in the liquid during freezing.
- (9) The solute diffusion in the solid is negligible.
- (10) All the material properties of solid and liquid are constant.
- (11) The partition coefficient  $k = C_S/C_L$  is constant over the range of interest.

Despite the initial assumptions made to develop the constitutional supercooling criterion, Eq. (1.14), this criterion can also be obtained in the absence of assumptions (1), (2), (6), and (7). Thus, the concept of constitutional supercooling as it was originally proposed is applicable to the multi-dimensional transient solidification processes.

### 1.2.2 Mullins-Sekerka Criterion

The constitutional supercooling criterion was based on a static analysis that showed that constitutional supercooling could be a source of interface instability. The same problem was considered in a dynamic and more rigorous analysis by Mullins and Sekerka (M-S). The same assumptions were made as those in the constitutional supercooling theory. The M-S criterion also considered the diffusion of heat and solute around the perturbed interface, and the interfacial free energy. The linear perturbation analysis was first employed to solve this problem by Mullins and Sekerka in 1964; however, thereafter numerous extensions have been added [38-47]. In the M-S analysis a steady-state system was initially assumed, similar to that described in the previous section. At time t = 0, a small perturbation was imposed on the system, and all equations and boundary conditions were linearized with respect to this perturbation. The time dependence of the amplitude of perturbation was then investigated. In general, if the perturbation is growing in time, the interface is unstable. On the other hand, if the perturbation decays to zero, the interface is stable. It is noted that for any kind of initial disturbances, the final stability criterion of the interface should be the same according to the general theory of stability. The disturbance can be in the temperature, concentration, or interfacial shape.

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Because most functions can be represented by a Fourier series of sinusoidal functions, sinusoidal perturbations of all possible wavelengths can be considered. Only if the rate of growth of the perturbation is negative for all wavelengths is the interface considered stable.

A sinusoidal perturbation of very small amplitude  $\delta$  to the planar interface in the constant moving coordinate can be described by:

$$z = \phi(x,t) = \delta(t) \sin \omega x$$
, (1.16)

where  $\omega = 2\pi/\lambda$  is the wave frequency. Notice that a two-dimensional model is assumed.

Following the work of Mullins and Sekerka (the detailed procedures are outlined in Appendix 1), the interface stability criterion is:

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$$\frac{\delta}{\delta} = \frac{V\left[-2T_{m}\Gamma\omega^{2}\left(\omega_{C}-\frac{V}{D}P\right)-(\xi_{S}+\xi_{L})\left(\omega_{C}-\frac{V}{D}P\right)+2mG_{C}\left(\omega_{C}-\frac{V}{D}\right)\right]}{(\xi_{S}-\xi_{L})\left(\omega_{L}-\frac{V}{D}P\right)+2\omega mG_{C}}$$
(1.17)

Upon examining the physical significance of this interface stability criterion, the time evolution of a perturbation of wavelength  $(2\pi/\omega)$ within the region of applicability of the model used becomes evident. It is apparent that a positive value of  $\delta/\delta$  for any  $\omega$  means the flat interface is unstable, whereas a negative sign for all  $\omega$  indicates a decaying perturbation and a stable interface. It is noted that the sign of Eq. (1.17) depends only on the sign of the numerator, since both terms in the denominator are always positive. The term ( $\boldsymbol{\xi}_{\varsigma} - \boldsymbol{\xi}_{L})$ is proportional to V as shown by Eq. (A.34), and is positive. Notice that  $\omega_{C} > V/D > VP/D$ , which results from the definition of  $\omega_{C}$  [Eq. (A.13)] and from the fact that p = 1-k < 1. Therefore, the value  $\omega_{c}$  - (V/D)P is positive. The second term of the denominator,  $2\omega mG_{c}$ , also is positive because both m and  $G_{\rm C}$  have the same sign. Therefore, we expect the instability or stability of a planar interface to depend only on the sign of the numerator. Dividing the numerator through by a positive value  $2[\omega_{C} - (V/D)P]V$  gives

$$S(\omega) = -T_{m}\Gamma\omega^{2} - \frac{1}{2}(\xi_{S} + \xi_{L}) + \frac{mG_{C}[\omega_{C} - (V/D)]}{\omega_{C} - (V/D)P} . \qquad (1.18)$$

The frequency dependence function  $S(\omega)$  must be negative for stability. The first term in Eq. (1.18) arises from the capillarity, which is always negative and has a stabilizing influence for all wavelengths. Furthermore,

a shorter wavelength (large  $\omega$ ) favors stability. This is exactly the sort of stabilizing effect that would be expected of the surface tension. The second term, representing the temperature gradients, also is stabilizing for positive values. The third term represents the effect of solute accumulation, favors instability, and always is positive. Instability occurs when there is any frequency for which the magnitude of the third term is larger than that of the sum of the first two terms.

It is possible to simplify the M-S criterion and recover the result of the constitutional supercooling theory derived earlier by removing the capillary effects and the dependence of the stability criterion on the wavelength. Under the simplification mentioned above, the M-S criterion for instability can be expressed by:

$$mG_{C} > \frac{1}{2}(\xi_{S} + \xi_{I})$$
 (1.19)

The above criterion is essentially the same as the constitutional supercooling criterion, Eq. (1.14), except that a mean value of  $G_L$  and  $G_S$ , weighted by the thermal conductivities of these two phases, is substituted for  $G_L$  in Eq. (1.14). It is noted that the stability criterion Eq. (1.17) has a greater region of stability than Eq. (1.14) or (1.19).

The M-S stability criterion can be written in a slightly different form by separating out the wavelength-dependent and -independent part of  $S(\omega)$ . Define

$$G(\omega) = \frac{-T_m \Gamma \omega^2}{mG_c} + F(\omega) , \qquad (1.20)$$

where

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$$F(\omega) = \frac{\omega_{\rm C} - (V/D)}{\omega_{\rm C} - (V/D)P}$$
 (1.21)

Then the condition for stability becomes

$$\frac{1}{2}(\xi_{S} + \xi_{L}) > mG_{C}G(\omega) \qquad (1.22)$$

 $G(\omega)$  is composed of two parts. The first one is proportional to  $\omega^2$ . The second,  $F(\omega)$ , is proportional to  $\omega^2$  for small  $\omega$  and tends to unity for large  $\omega$ , since Eq. (1.22) must be valid for all  $\omega$  for the interface to be stable. In other words, the general condition for stability must satisfy

$$\frac{1}{2}(\xi_{c} + \xi_{L}) > mG_{c}G(\omega)_{max} \qquad (1.23)$$

Notice the constitutional supercooling criterion corresponds to  $G(\omega)_{max}$ = 1. In fact,  $G(\omega)_{max}$  has a maximum possible value of unit and will be lower in general. Thus the constitutional supercooling is a necessary but not sufficient condition for instability; the degree of constitutional supercooling must exceed some specific value.

#### 1.3 PURPOSE OF PRESENT STUDY

The major purpose of this work are listed below:

(1) A major difficulty in analyzing solidification processes in solutions is the lack of analytical tools to study the process. The major purpose of this work is to develop a new numerical method to solve the multi-dimensional transient heat and mass transfer equations in the solid and liquid phases during solidification of binary solutions. The finite element computer program developed is capable of handling solidification processes with a non-isothermal phase interface and with geometrically irregular but smooth shapes of the
interface. Currently, there are no other numerical or analytical methods with these capabilities.

(2) The numerical method was used in studying a transient solidification process in a binary solution. Temperature and concentration distributions in both the solid and liquid phases were obtained during the arbitrary transient solidification process. The solute accumulation ahead of a planar interface and around a curved interface were observed. The results of this part of the study, although anticipated, constituted the first rigorous proof of the phenomena.

4.5

(3) The new computer code offers a unique method of studying interface stability phenomena during solidification in solutions. A mathematical model describing a typical solidification process in solutions was established. Then the planar interface stability was studied for several types of perturbation. Surprising and unique results conflicting with previous stability criteria were obtained for the effects of various parameters on interface stability. These results illustrate the importance of the new numerical method and indicate the need for new fundamental studies on the phenomena associated with the interface stability. A discussion of these results is included herein.

Following the introduction in Chapter 1, the mathematical model for typical solidification processes in solution is formulated in Chapter 2. A new general numerical method using front-tracking finite elements that was developed to solve the mathematical model is discussed in Chapter 3. In Chapter 4 this numerical method is employed to study

the planar interface stability problems in solutions. The results are presented and discussed in Chapter 5, followed by conclusions in Chapter 6.

# CHAPTER 2:

# MATHEMATICAL MODEL

# 2.1 A GENERAL SOLIDIFICATION PROBLEM

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Consider a general solidification process in binary solutions at some specific time t as shown in Fig. 2.1. There is a constant domain  $\Omega$  that contains two time-dependent subdomains  $\Omega_1(t)$  and  $\Omega_2(t)$  such that  $\Omega = \Omega_1(t) \cup \Omega_2(t)$ .  $\Omega_1(t)$  and  $\Omega_2(t)$  represent the solid region and the liquid region, respectively. The boundaries of the domains are  $\partial \Omega_1(t) =$  $\Gamma_1(t) \cup S(t)$  of  $\Omega_1(t)$  and  $\partial \Omega_2(t) = \Gamma_1(t) \cup S(t)$  of  $\Omega_2(t)$ , where S(t) is the moving phase interface common to  $\Omega_1(t)$  and  $\Omega_2(t)$ . It is noted that all the domains and their boundaries are time dependent. The outward unit vector normal to boundaries  $\partial \Omega_1(t)$  and  $\partial \Omega_2(t)$  are  $\vec{n}_1$  and  $\vec{n}_2$ , respectively. Any type of boundary conditions, including essential, natural, mixed, and radiation can be applied on parts of the boundaries  $\partial \Omega_1(t)$  and  $\partial \Omega_2(t)$ . These boundary conditions are allowed to be timedependent. We are interested in determining the position of phase interface S(t), the temperature distributions  $T_1(t)$  and  $T_2(t)$ , and the solute concentration distributions  $C_1(t)$  and  $C_2(t)$  at any instant in time.

### 2.2 GOVERNING EQUATIONS

A complete description of solidification processes involves the kinetics of atomic rearrangement near the phase interface, the transport of heat and mass in solid and liquid regions, the convection due to





difference in density of solid and liquid, the natural convection arising from density variation in the liquid, and so forth [48-63]. We will not deal directly with atomic effects or theories of nucleation kinetics. These theories are based on statistical mechanics and concern the fundamental processes of solidification on atomic scale. To make this problem tractable, we propose the following approximations:

(1) The solid-liquid interface is a definite surface in space.

- (2) The effects of interface kinetics are negligible.
- (3) There is thermodynamic equilibrium on the phase interface.
- (4) There is no convection in the liquid.
- (5) The mass diffusion in the solid region is negligible.

The governing equations describing solidification processes under the above assumptions are:

A. Heat Transfer Equations:

$$\nabla \cdot (\mathbf{k}_1 \nabla \mathbf{T}_1) = \rho_1 C_1 \frac{\partial \mathbf{T}_1}{\partial \mathbf{t}} \quad \text{in } \Omega_1(\mathbf{t}) \quad (2.1)$$

$$\nabla \cdot (k_2 \nabla T_2) = \rho_2 C_2 \frac{\partial T_2}{\partial t} \quad \text{in } \Omega_2(t) \qquad (2.2)$$

# B. Mass Transfer Equations:

$$\nabla \cdot (D \nabla C) = \frac{\partial C}{\partial t} \quad \text{in } \Omega_2(t) \qquad (2.3)$$

C. Interface Conditions:

 $(k_1 \nabla T_1 - k_2 \nabla T_2) \cdot \vec{n} = \rho_1 L(\vec{v} \cdot \vec{n}) \quad \text{on } S(t) \quad (2.4)$ 

 $-D \nabla C \cdot \vec{n} = C(1-k)(\vec{v} \cdot \vec{n})$  on S(t) (2.5)

$$T_1 = T_2 = T_m - mC - T_m \Gamma_Y$$
 on S(t). (2.6)

Here L is the latent heat of fusion per unit volume, k is the partition coefficient,  $T_m$  is the melting point of the pure substance with a planar interface, m is the slope of the liquidus line, which may vary as a function of solute concentration,  $\Gamma$  is the Gibbs-Thompson coefficient,  $\gamma$  is the mean interface curvature,  $\vec{v}$  is the local solidification velocity, and  $\vec{n}$  is the unit vector normal to the interface and coinciding with  $\vec{n}_1$  on the interface.

All the material properties in Eqs. (2.1)-(2.6) may vary. Cases with non-isotropic and nonlinear (e.g., temperature and concentration dependent) properties are permitted. These governing equations are written in general vectorial form and are independent of the coordinate systems used.

It can be seen that all the governing equations are coupled at the interface through Eqs. (2.4)-(2.6). Equation (2.4) is obtained from the energy balance at the interface. Equation (2.5) represents the mass balance at the interface. Equation (2.6) indicates that under the thermodynamic equilibrium assumption, the temperature on the interface is determined by the interfacial concentration as well as the interfacial curvature. Hence the interface temperature is not only a function of time, bur varies along the interface. It should also be noted that  $\gamma$ ,  $\vec{n}$ , and  $\vec{v}$  all are functions of time and space.

Before proceeding to the next section, it is worth pausing to compare the assumptions and governing equations of this model with those of interface stability theories discussed previously. The most distinct aspect of this model is the incorporation of transient effects, which makes it close to representing realistic situations. The temperature and concentration

distributions in both solid and liquid are changing with time, but are not fixed with respect to the interface. Every node on the interface may advance at an arbitrary velocity, depending on the applied boundary conditions. The present model provides us with the ability to study more complex situations than the previous quasi-steady models found in the literature [64,65].

# 2.3 INITIAL AND BOUNDARY CONDITIONS

2.3.1 Initial Conditions

Arbitrary initial conditions can be employed in the numerical analysis. However, in this specific study uniform conditions were chosen:

$C = C^{\circ}$	in $\Omega_2$	
$T = T_2^0$	in $\Omega_2$	(2.7)
$T = T_1^0$	in Ω <sub>2</sub>	

# 2.3.2 Boundary Conditions

In addition to the interface conditions [Eqs. (2.4)-(2.6)], which will serve as part of the boundary conditions for domains  $\Omega_1(t)$  and  $\Omega_2(t)$ , the following general boundary conditions may be imposed on any parts of the boundaries:

$$\alpha_1 \nabla T_1 \cdot \vec{n}_1 + \beta_1 T_1 = \gamma_1 \quad \text{on } \Gamma_1(t)$$

$$\alpha_2 \nabla T_2 \cdot \vec{n}_2 + \beta_2 T_2 = \gamma_2 \quad \text{on } \Gamma_2(t) \quad (2.8)$$

$$\alpha_3 \nabla C \cdot \vec{n}_3 + \beta_3 C = \gamma_3 \quad \text{on } \Gamma_3(t) .$$

It should be noted that any type of boundary conditions, including Dirichlet, Neumann, and mixed, can be obtained from the general boundary conditions, Eqs. (2.8), by properly choosing the coefficients  $\alpha_i$ ,  $\beta_i$ , and  $\gamma_i$ , where i = 1, 2, and 3. In particular, essential boundary conditions are derived from Eqs. (2.8) by the so-called penalty method. For example, if one chooses  $\beta_1$  and  $\gamma_1$  to be some very large values compared to  $\alpha_1$ , then the first equation in Eqs. (2.8) is a good approximation to T =  $\gamma_1/\beta_1$  = T<sub>C</sub>, an essential boundary condition. By adopting Eqs. (2.8), the implementation also is simplified. The reader is reminded that these boundary conditions can be functions of time and space.

### 2.4 NONLINEARITY ON THE INTERFACE

Equations (2.1) to (2.6) indicate that diffusion of heat and mass are coupled only at the interface and are considered independent in the rest of the domain. The difficulties in obtaining the solutions stem in part from the fact that Eq. (2.4) is nonlinear. To demonstrate this, consider the simpler situation where  $T_1 = T_2 = T_C$ , a constant value and one-dimensional problem.

Take the total derivatives of  $T_1$  and  $T_2$  on the interface:

$$\left( \frac{\partial T_1}{\partial x} dx + \frac{\partial T_1}{\partial t} dt \right)_{\substack{x=S(t) \\ x=S(t) \\ (2.9) \\ \frac{\partial T_1}{\partial x} \frac{dS(t)}{dt} + \frac{\partial T_1}{\partial t} = \frac{\partial T_2}{\partial x} \frac{dS(t)}{dt} + \frac{\partial T_2}{\partial t} = 0 \text{ at } x = S(t) .$$

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Rearrange to obtain

$$\frac{dS(t)}{dt} = \frac{-\partial T_1/\partial t}{\partial T_1/\partial x} = \frac{-\partial T_2/\partial t}{\partial T_2/\partial x} . \qquad (2.10)$$

Equation (2.4) becomes

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$$k_1 \frac{\partial T_1}{\partial x} - k_2 \frac{\partial T_2}{\partial x} = \rho_1 L \frac{\partial T_1 / \partial t}{\partial T_1 / \partial x} = \rho_1 L \frac{\partial T_2 / \partial t}{\partial T_1 / \partial x}. \quad (2.11)$$

The nonlinearity of Eq. (2.4) is therefore evident.

### 2.5 PREVIOUS NUMERICAL WORK

To date, a large number of analytical and numerical methods have been presented in the technical literature for the solution of problems of heat transfer with phase transformation in a pure substance. Many of these methods are summarized in Refs. 66-69. Most of the analytical methods are restricted to one-dimensional situations [70-72]. Multidimensional situations usually are solved using numerical methods [73-77]. Several solutions using finite elements method also have been reported [78-87].

The numerical techniques using finite elements can be separated into two groups based on the formulation of the problem. In the first group, enthalpy is the dependent variable (see Refs. 77,81-82,86-87). The second group of methods deals with the energy equation written in terms of temperature as the dependent variable (see Refs. 83-85). Solutions using the finite element for the enthalpy formulation include the work by Comini *et al.* [86], Ronel and Baliga [87], and Miller and Miller [81,82].

Bonnerot and Jamet [78] were the first to develop a finite element

that discretizes the domain by means of isoparametric elements corresponding to a six-noded triangular prism in a space defined by the x-y Cartesian coordinates and t, the time variable. The free boundary was approximated by a polygon whose vertices coincided with triangulation nodes. In their method, the elements deformed continuously in time to accommodate the displacement of change of phase interface. The method discussed above is restricted to the solidification processes in pure substances. To study the physical phenomena that occur during solidification processes in solutions and alloys, we have developed a new multidimensional finite element method using "front tracking" finite elements.

The front tracking finite element method uses moving or deforming elements to track continuously in time the position of the change of phase interface. A general front tracking procedure for the study of solidification processes in a pure substance was first established and reported by Rubinsky *et al.* [83,85]. A different front tracking method was also developed by Rubinsky for the study of heat and mass transfer during one-dimensional transient solidification processes in a solution in the presence of forced convection [84].

In this study, a new general multi-dimensional numerical method of solution using front tracking finite elements for the study of heat and mass transfer problems during transient solidification processes in binary solutions was developed. Specific to the front tracking finite element method is the fact that the energy balance on the change of phase interface is not treated as a boundary condition, but rather as an independent equation whose solution gives the position of the interface in time. Because the front tracking method tracks the change of

phase interface continuously in time, the method can deal with irregular interface morphologies and can consider the local thermodynamics on the interface, including capillary effects and nucleation kinetics.

# CHAPTER 3:

# FRONT TRACKING FINITE ELEMENTS

### 3.1 INTRODUCTION

There are no numerical methods that can be used to solve the mathematical model of a typical transient solidification process in binary solutions described by Eqs. (2.1) to (2.8). In this chapter, such a new general numerical method will be developed. Because of the characteristics of this problem, which is specified by an irregular transient change of phase interface, the finite element method, which is able to accommodate irregular geometries, was chosen as the most appropriate method of solution. In general, the finite element method can handle problems with complex geometries, anisotropic materials, and arbitrary boundary conditions. Finite element methods also permit refinement of the domain when necessary.

# 3.2 SPACE DISCRETIZATION

Since the governing equations (2.1) to (2.3) have the same form, we will develop the finite element formulation only for Eq. (2.1). For convenience, Eq. (2.1) is rewritten here and the subscript disregarded:

$$\nabla \cdot (\mathbf{k} \nabla \mathbf{T}) = \rho_{\mathbf{C}} \frac{\partial \mathbf{T}}{\partial \mathbf{t}} .$$
 (3.1)

The finite element formulations will be derived in a general form, so that a general purpose program can be developed (limited to twoor three-dimensional axisymmetry problems). The governing equation is the diffusion equation, but the method can easily be extended to incorporate convection terms if necessary. Following standard procedures [88-96], the solution domain is first divided into M elements of arbitrary shape. In general, these elements can be rectangular, triangular, or mixed, and the number of nodes of each element can be different from each other. We approximate the unknown exact temperature T and its gradients of each element by

$$T(r,z,t) = \sum_{i=1}^{N} \phi_i(r,z) T_i(t)$$

$$\frac{\partial T}{\partial r}(r,z,t) = \sum_{i=1}^{N} \frac{\partial \phi_i}{\partial r}(r,z) T_i(t)$$

$$\frac{\partial T}{\partial z}(r,z,t) = \sum_{i=1}^{N} \frac{\partial \phi_i}{\partial z}(r,z) T_i(t)$$

$$\frac{\partial T}{\partial t}(r,z,t) = \sum_{i=1}^{N} \phi_i(r,z) \frac{dT_i}{dt}(t) ,$$
(3.2)

where N is a finite value representing the number of nodes.  $T_i(t)$  are unknown nodal values to be found,  $\phi_i(r,z)$  are the shape or interpolation functions over element i.

In matrix form,  

$$T(r,z,t) = [\phi(r,z)] \{T(t)\}$$

$$\frac{\partial T}{\partial r} (r,z,t)$$

$$\frac{\partial T}{\partial z} (r,z,t)$$

$$= [\beta(r,z)] \{T(t)\}$$

$$(3.3)$$

$$\frac{\partial T}{\partial t} (r,z,t) = [\phi(r,z)] \{\frac{dT}{dt} (t)\},$$

where  $[\phi(r,z)] = [\phi_1,\phi_2,...,\phi_N]$  is the temperature interpolation matrix.  $\{T(t)\} = [T_1,T_2,...,T_N]^T$ , and  $dT/dt = \frac{dT_1}{dt}, \frac{dT_2}{dt},..., \frac{dT_N}{dt}^T$ :

$$[B(r,z)] = \begin{bmatrix} \frac{\partial \phi_1}{\partial r} & \frac{\partial \phi_2}{\partial r} & \cdots & \frac{\partial \phi_N}{\partial r} \\ \frac{\partial \phi_1}{\partial z} & \frac{\partial \phi_2}{\partial z} & \cdots & \frac{\partial \phi_N}{\partial z} \end{bmatrix}$$
(3.4)

[B] is the temperature gradient interpolation matrix.

By the method of weighted residues, Eq. (3.1) becomes

$$\int_{\Omega_{i}} \left[ \nabla \cdot (k \nabla T) - \rho_{C} \frac{\partial T}{\partial t} \right] \phi_{i} d\Psi = 0 , \qquad (3.5)$$

where  $\Omega_i$  is the volume of element i and  $\phi_i$  are weighting functions. The exact equation has been modified so that it will be satisfied only in a weighted average sense as Eq. (3.5). Integration by parts, followed by the use of Gauss' theorem, yields

$$\int_{\Omega_{i}} (k \nabla T \cdot \nabla \phi_{i} - \rho_{C} \frac{\partial T}{\partial t} \phi_{i}) d\Psi - \int_{\Gamma_{i}} \phi_{i} k \nabla T \cdot \vec{n} dA = 0 , \quad (3.6)$$

where  $\Gamma_i$  is the boundary surface of element i. The integration by parts formula and Gauss' theorem are, respectively,

$$\nabla \cdot (\mathbf{v}\mathbf{k} \ \nabla \ \mathbf{u}) = \mathbf{k} \ \nabla \ \mathbf{u} \cdot \mathbf{v} + \mathbf{v} \nabla \cdot (\mathbf{k} \ \nabla \ \mathbf{u})$$

$$\int \nabla \cdot \vec{\sigma} \ d\mathbf{A} = \int \vec{\sigma} \cdot \vec{n} \ d\mathbf{S}$$

$$\Omega \qquad \qquad \partial \Omega$$
(3.7)

Any type of boundary condition can now be incorporated through the surface integral term in Eq. (3.6). The general boundary conditions are allowed as follows:

$$T = T (r, z, t) \quad \text{on } S_1$$

$$\nabla T \cdot \vec{n} = -q(r, z, t) \quad \text{on } S_2$$

$$\nabla T \cdot \vec{n} = h(t - T_{\infty}) \quad \text{on } S_3$$

$$\nabla T \cdot \vec{n} = \sigma \varepsilon T^4 - \alpha q_r \quad \text{on } S_4$$
(3.8)

where  $\Gamma_i = S_1 \cup S_2 \cup S_3 \cup S_4$ . These boundary conditions represent the specific surface temperature, specified surface heat flux, convective heat transfer, and radiation heat transfer, respectively.

For nonisotropic materials, in general

$$k \nabla T = \begin{bmatrix} k_{11} & k_{12} \\ k_{21} & k_{22} \end{bmatrix} \begin{cases} \frac{\partial T}{\partial r} \\ \frac{\partial T}{\partial z} \end{cases} = [K][B(r,z)]\{T(t)\} \qquad (3.9)$$

Substituting Eqs. (3.8) and (3.9) into Eq. (3.6), we obtain the general form of the governing equation,

$$[C]\left\{\frac{dT}{dt}(t)\right\} + [K]\{T(t)\} = \{R\}, \qquad (3.10)$$

where

$$[C] = \int_{\Omega_{i}} \rho_{C}(\phi)[\phi] d\Psi , \quad \{R\} = \{R_{T}\} + \{R_{q}\} + \{R_{h}\} + \{R_{r}\}$$

$$[K] = [K_{C}] + [K_{h}] + [K_{r}] , \quad \{R_{T}\} = \int_{k} \nabla T\{\phi\} dA$$

$$[K_{C}] = \int_{\Omega_{i}} [B]^{T}[K][B] d\Psi , \quad \{R_{q}\} = \int_{\Omega_{i}} q\{\phi\} dA$$

$$[K_{h}] = \int_{\Omega_{i}} h\{\phi\}[\phi] d\Psi , \quad \{R_{h}\} = \int_{\Omega_{3}} hT_{\infty}\{\phi\} dA$$

$$[K_{r}]\{T\} = \int_{S_{h}} \sigma \varepsilon T^{*}\{\phi\} dA , \quad \{R_{r}\} = \int_{S_{h}} \alpha q_{r}\{\phi\} dA ,$$

and where  $d\Psi$  (= r dr dz) is the volume of an element. It should be noted that the original governing equation, (3.1), has been reduced to a set of ordinary differential equations, (3.10).

The idea and formulations outlined above are quite simple and straightforward. However, it can be imagined that the calculations of any matrix in Eq. (3.10) for a curvilinear element would present great difficulty if performed directly in terms of the r-z coordinates. Furthermore, the character of such calculations (e.g., the limit of integration) would change from element to element in the domain. Thus we introduce an invertible transformation between the original arbitrary element and a master element of simple shape. Figure 3.1 shows this domain transformation, where r-z are the global coordinates and  $\xi$ -n are the local or natural coordinates. They are related through

> $r = r(\xi, n)$   $-1 \le \xi \le 1$  $z = z(\xi, n)$   $-1 \le n \le 1$  (3.11)

by the chain rule of differentiation,

$$\left\{ \begin{array}{c} \frac{\partial}{\partial \xi} \\ \frac{\partial}{\partial \eta} \end{array} \right\} = \begin{bmatrix} \frac{\partial \mathbf{r}}{\partial \xi} & \frac{\partial \mathbf{z}}{\partial \xi} \\ \frac{\partial}{\partial \eta} & \frac{\partial \mathbf{z}}{\partial \eta} \end{bmatrix} \left\{ \begin{array}{c} \frac{\partial}{\partial \mathbf{r}} \\ \frac{\partial}{\partial z} \end{array} \right\} , \qquad (3.12)$$

where the Jacobian transformation matrix [J] is defined by

$$[J] = \begin{bmatrix} \frac{\partial \mathbf{r}}{\partial \xi} & \frac{\partial \mathbf{z}}{\partial \xi} \\ \frac{\partial \mathbf{r}}{\partial \eta} & \frac{\partial \mathbf{z}}{\partial \xi} \end{bmatrix}$$





 $q_3 = \frac{1}{4}(1+3)(1+7), \quad q_4 = \frac{1}{4}(1-3)(1+7)$ 中= 当(1-3)(1-7), 中= 当(1-7)(+3) 中=片(1-3)(1+7), 中=片(1-7)(1-3) \$= (1-3')(1-7')

FIGURE 3.1. Domains mapping from global coordinates r-z to local coordinates ξ-η.

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The function |J| is called the Jacobian of transformation,

$$|\mathbf{J}| = \frac{\partial \mathbf{r}}{\partial \xi} \frac{\partial \mathbf{z}}{\partial \eta} - \frac{\partial \mathbf{r}}{\partial \eta} \frac{\partial \mathbf{z}}{\partial \xi} \qquad (3.14)$$

In order to guarantee that the transformation is unique, that is, no gaps or overlappings among elements, it is necessary to ensure that |J| > 0 for all elements. Then the evaluations of Eq. (3.10) will be performed on the master element.

The nodes in the  $\xi$ -n plane may be mapped into corresponding nodes in the r-z plane by defining

$$r(\xi,n) = \sum_{i=1}^{N} \phi_{i}(\xi,n)r_{i}$$

$$z(\xi,n) = \sum_{i=1}^{N} \phi_{i}(\xi,n)z_{i} , \qquad (3.15)$$

where the  $r_i$  and  $z_i$  are nodal coordinates of an element. It is noted that the so-called isoparametric element is employed by adopting the same interpolation functions that were used previously to interpolate the temperature, i.e.,

$$T(\xi,n) = \sum_{i=1}^{N} \phi_i(\xi,n)T_i$$
, (3.2)

where  $\phi_i$  are no longer functions of r and z, but of  $\xi$  and n. The formula given by Eq. (3.15) is standard [88]. This transformation is also shown in Fig. 3.1. The  $\phi_i$  have the characteristic that, for example, at node 1,  $\phi_1$  =1 and all other  $\phi_i$  are zero, such that Eq. (3.2) is satisfied automatically. Hence, each integration in Eq. (3.10) is evaluated by integrating over the square master element. For example, [C] becomes

$$[C] = \int \int \rho C\{\phi(\xi,n)\} [\phi(\xi,n)] |J(\xi,n)| r d\xi dn . (3.16) -1 -1$$

The integrations are normally carried out by the method of Gauss-Legendre quadrature:

$$[C] = \sum_{i=1}^{NG} \sum_{j=1}^{NG} \omega_{i} \omega_{j} \rho C\{\phi(\xi_{i}, n_{i})\} [\phi(\xi_{i}, n_{j})] \cdot |J(\xi_{i}, n_{j})| r_{i,j} (3.17)$$

where  $\omega_i$  and  $\omega_j$  are Gauss weights,  $\xi_i$  and  $\xi_j$  are the coordinates of Gaussian points, and NG is the number of Gaussian points in each integration direction.

All other matrices in Eq. (3.10) are evaluated in a similar way, except for the temperature gradient interpolation matrix [B]. This matrix is transformed from r-z coordinates to  $\xi$ - $\eta$  coordinates such that

$$[B] = \begin{bmatrix} \frac{\partial \phi_{i}}{\partial r} \\ \frac{\partial \phi_{i}}{\partial z} \end{bmatrix} = [J(\xi, \eta)]^{-1} \begin{cases} \frac{\partial \phi_{i}}{\partial \xi} \\ \frac{\partial \phi_{i}}{\partial \eta} \end{cases} \qquad (3.18)$$

The matrix [K<sub>c</sub>] becomes

$$\begin{bmatrix} K_{c} \end{bmatrix} = \int \int \begin{bmatrix} B(\xi,n) \end{bmatrix}^{T} \begin{bmatrix} K \end{bmatrix} \begin{bmatrix} B(\xi,n) \end{bmatrix} |J(\xi,n)| r d\xi dn , \qquad (3.19)$$
  
-1 -1

which is evaluated by Gauss-Legendre quadrature:

$$[K_{c}] = \sum_{i=1}^{NG} \sum_{j=1}^{NG} \omega_{j} \omega_{j} [B(\xi_{j}, n_{j})]^{T} [K] \cdot [B(\xi_{j}, n_{j})] |J(\xi_{j}, n_{j})| r_{i,j} . (3.20)$$

### 3.3 TIME DISCRETIZATION

The general formulation of the weak form of governing equation (3.1) has been derived as

$$[C(T)]\left\{\frac{dT}{dt}(t)\right\} + [K(T,t)]\{T(t)\} = \{R(T,t)\} . \qquad (3.10)$$

This is a set of nonlinear ordinary differential equations with time t as the independent variable. This equation will be solved by numerical time integration. Two methods of solution are proposed. First, a typical finite different method is used, and then the finite element method in time is employed.

# 3.3.1 General Finite Difference Method

The general  $\theta$  method is introduced such that  $t_{\theta}$  =  $t_n + \theta \Delta t$  , where  $0 \le \theta \le 1$  and

$$\left[C(T)\right]_{\theta}\left\{\frac{dT}{dt}(t)\right\}_{\theta} + \left[K(T,t)\right]_{\theta}\left\{T(t)\right\}_{\theta} = \left\{R(T,t)\right\}_{\theta} . \qquad (3.21)$$

The subscript  $\theta$  indicates the values are evaluated at time  $t_{\theta}$ . We introduce the following approximations of standard finite difference method:

$$\{T\}_{\theta} = (1-\theta)\{T\}_{n} + \theta\{T\}_{n+1}$$

$$\{R\}_{\theta} = (1-\theta)\{R\}_{n} + \theta\{R\}_{n+1}$$

$$\{\frac{dT}{dt}\}_{\theta} = \frac{\{T\}_{n+1} - \{T\}_{n}}{\Delta t}$$

$$(3.22)$$

Similarly for  $[C(T)]_{\theta}$  and  $[K(T,t)]_{\theta}$ . Substitute these into Eq. (3.21)

to obtain the following formulation:

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$$\left[\theta[K] + \frac{[C]_{\theta}}{\Delta t}\right] \{T\}_{n+1} = \left[-(1-\theta)[K] + \frac{[C]_{\theta}}{\Delta t}\right] \{T\}_{n} + \{R\}_{\theta} . (3.23)$$

This equation represents a general family of recurrence relations: for  $\theta = 0$ , the algorithm is a Euler forward method; for  $\theta = \frac{1}{2}$ , a Crank-Nicolson method; for  $\theta = \frac{2}{3}$ , a Galerkin method; and for  $\theta = 1$ , a back-ward method.

3.3.2 Finite Element Method in Time

Equation (3.23) can also be obtained by finite elements in the time domain. First, the time domain is divided into N elements, and for each element the weak form is obtained in the same way as for the spatial domain:

$$t_{m+1}$$
  
 $\int \omega(z) \left[ C[\{\dot{T}\} + [K]\{T\} - \{R\} \right] dz = 0$ , (3.24)  
 $t_{n}$ 

where  $\omega(z)$  is an arbitrary weighting function. The approximation is applied in one-dimensional two-node time elements such that

$$T = (1 - \xi)T_{n} + \xi T_{n+1}$$

$$T = \frac{dT}{dt} = \frac{T_{n+1} - T_{n}}{\Delta t_{n}}$$

$$\xi = \frac{t - t_{0}}{\Delta t_{n}}$$
(3.25)

Substituting in Eq. (3.24) produces

$$\int_{t_{n}}^{t_{n+1}} \omega(\tau) \left\{ \left( \begin{bmatrix} C \end{bmatrix} \frac{\{T\}_{n+1} - \{T\}_{n}}{\Delta t} \right) + \begin{bmatrix} K \end{bmatrix} \begin{bmatrix} (1-\xi)T_{n} + \xi T_{n+1} \end{bmatrix} - \{R\} \right\} d\tau = 0$$
 (3.26)

by the mean value theorem,

$$t_{n+1} \qquad t_{n+1} \qquad \int \xi \omega(\tau) d\tau = \int \omega(\tau) d\tau , \qquad (3.27)$$

$$t_{n} \qquad t_{n}$$

where  $0 \le \theta \le 1$ . Hence,

$$\int_{t_{n}}^{t_{n+1}} \omega(\tau) \left\{ \begin{bmatrix} C \end{bmatrix} \frac{\{T\}_{n+1} - \{T\}_{n}}{\Delta t} + \begin{bmatrix} K \end{bmatrix} \begin{bmatrix} (1-\theta)T_{n} + \theta T_{n+1} \end{bmatrix} - \{R\} \right\} d\tau = 0 .$$
 (3.28)

Since the weighting function  $\omega(\tau)$  is arbitrary,

$$[C]\left\{\frac{T_{n+1}-T_n}{\Delta t}\right\} + [K]\left\{(1-\theta)T_n + \theta T_{n+1}\right\} - \{R\} = 0 \qquad (3.29)$$

is derived.

In general, [C], [K], and  $\{R\}$  are not constants but functions of  $\theta$ .

$$\left[\theta[K]_{\theta} + \frac{[C]_{\theta}}{\Delta t}\right] \{T\}_{n+1} = \left[-(1-\theta)[K] + \frac{[C]_{\theta}}{\Delta t}\right] \{T\}_{n} + \{R\} \quad . \tag{3.30}$$

Thus the same form is obtained in this way as was derived by the finite difference method.

### 3.4 NUMERICAL STABILITY ANALYSIS

Consider the following set of linear differential equations:

$$[C]{T} + [K]{T} = {R}, \qquad (3.10)$$

where the coefficients [C], [K], and {R} are constants. To study their numerical stability, these differential equations are transformed into the modal form, i.e., a set of independent scalar equations. Then the solution of Eq. (3.10) is just the superposition of the solution of each scalar equation. The stability analysis is concentrated on each scalar mode. First, assume the case of free response with R = [0]. The general solution of Eq. (3.10) can be assumed as

$$\{T\} = \{C_{i}\} e^{-\lambda_{i}t}, \quad i = 1,...,n$$
  
$$-\lambda_{i}t \qquad (3.31)$$
  
$$\{T\} = -\lambda_{i} \{C_{i}\} e^{-\lambda_{i}t},$$

where  $\{C_i\}$  is a modal vector of unknown amplitude and  $\lambda_i$  is a modal decay constant. Substituting into Eq. (3.10), we derive

$$\left[-\lambda_{i}[C] + [K]\right] \{C_{i}\}e^{-\lambda_{i}t} = [0],$$
 (3.32)

where

 $\lambda_{i}[C] + [K] = [0]$ 

is the characteristic polynomial for  $\lambda_i$ . Obviously, this is a standard eigenvalue problem; therefore, the following equality has to be satisfied:

 $[K]\{\phi_{i}\} = \lambda_{i}[C]\{\phi_{i}\}, \qquad (3.33)$ 

where  $\{\phi_i\}$  are eigenvectors corresponding to eigenvalues  $\lambda_i$ . These eigenvectors are subject to the orthogonality condition,

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$$\{\phi_{j}\}^{T}[C]\{\phi_{j}\} = \delta_{ij} = \begin{cases} 1 & \text{for } i = j \\ 0 & \text{for } i \neq j \end{cases}, \quad (3.34)$$

so

$$\{\phi_{\mathbf{j}}\}^{\mathsf{T}}[\mathsf{K}]\{\phi_{\mathbf{j}}\} = \lambda_{\mathbf{i}}\{\phi_{\mathbf{j}}\}^{\mathsf{T}}[\mathsf{C}]\{\phi_{\mathbf{i}}\} = \lambda_{\mathbf{i}}\delta_{\mathbf{i}\mathbf{j}} \qquad (3.35)$$

Hence, the solutions are the eigenpairs  $(\lambda_i, \{\phi_i\})$ .

We define

$$[\phi] = [\{\phi_1\} \dots \{\phi_n\}] \quad . \tag{3.36}$$

Then the solution {T} may be expressed as a linear combination of all eigenvectors:

$$\{T\} = [\Phi]\{v\}$$
(3.37)  
$$\{T\} = [\Phi]\{v\} ,$$

where  $\{v\}$  is a vector of generalized modal unknowns. Substituting into Eq. (3.10), we get

$$[C][\phi]{v} + [K][\phi]{v} = {R}$$

$$[\phi]^{T}[C][\phi]{v} + [\phi]^{T}[K][\phi]{v} = [\phi]^{T}{R} .$$
(3.38)

It is noted that

$$\{\phi\}^{T}[C]\{\phi\} = [I]$$
  
 $\{\phi\}^{T}[K]\{\phi\} = [\Lambda]$  (3.39)  
 $[\Lambda] = [\lambda, ..., \lambda]$ 

and

$$\{v\} + [\Lambda]\{v\} = \{g\}$$
 (3.40)

is obtained, where  $\{g\} = \{\phi\}^T \{R\}$ .

Finally the decoupled differential equations are derived for each node:

$$\dot{\mathbf{v}}_{\mathbf{i}} + \lambda_{\mathbf{i}}\mathbf{v}_{\mathbf{i}} = \mathbf{g}_{\mathbf{i}} \qquad (3.41)$$

Discarding the subscript i and applying the  $\theta$  method gives

$$\dot{\mathbf{v}} = (\mathbf{v}_{n+1} - \mathbf{v}_n)/\Delta t$$

$$\mathbf{v} = (1 - \theta)\mathbf{v}_n + \theta\mathbf{v}_{n+1} , \quad 0 \le \theta \le 1 \qquad (3.42)$$

$$\mathbf{v}_{n+1} = \frac{1 - (1 - \theta)\lambda\Delta t}{1 + \Delta t\lambda \theta} \mathbf{v}_n + \frac{1}{1 + \Delta t\lambda \theta} \mathbf{g}_n .$$

The amplification factor is

$$r_{i} = \frac{v_{n+1}}{v_{n}} = \frac{1 - (1 - \theta)\lambda_{i}\Delta t}{1 + \Delta t\lambda_{i}\theta} \qquad (3.43)$$

The requirement of stable solutions is

$$|\mathbf{r}_{i}| < 1$$
, (3.44)

which corresponds to the condition

$$\lambda_{\lambda} \Delta t(-1 + 2\theta) > -2$$
 (3.45)

It is noted that for positive  $\lambda_i \Delta t$ , if  $\theta > \frac{1}{2}$  the algorithm is unconditionally stable. For  $\theta < \frac{1}{2}$ , a conditionally stable method is used and the critical time step is determined by

$$\Delta t_{cr} = \frac{2}{1 - 2\theta} \frac{1}{\lambda_{j}}, \quad 0 \le \theta \le \frac{1}{2}.$$
 (3.46)

Hence, if  $\Delta t \leq \Delta t_{cr}$ , it produces a stable method, and if  $\Delta t > \Delta t_{cr}$ , it results in an unstable method.

The numerical stability for various  $\theta$  are summarized in Fig. 3.2.



FIGURE 3.2. Stability behavior of  $\theta$  method.  $\theta = 0$ , Euler forward;  $\theta = \frac{1}{2}$ , Crank Nicolson,  $\theta = 2/3$ , Galerkin;  $\theta = 1$ , Euler backward.

# 3.5 INTERFACE MOVING SCHEME

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The most important aspect of this work is a new numerical procedure for the solution of the interface condition, Eq. (2.4). The condition will be solved as an independent equation to obtain the new interface positions in time. For convenience, this equation is rewritten here:

 $(K_1 \nabla T_1 - K_2 \nabla T_2) \cdot \vec{n} = \rho_1 L(\vec{v} \cdot \vec{n})$  on S(t). (2.4)

Several methods have been proposed for the solution of this equation [50,51,75,79,85]. Those methods, however, are only applicable to situations in which the interface is isothermal. The isothermal interface occurs when the pure liquid is freezing and the effect of interfacial curvature is neglected. In a general solidification problem such as the one studied here, the temperature can vary along the interface as a function of interfacial concentration and interfacial curvature. This is shown in Eq. (2.6). Therefore, in the present study a new method is developed for the solution of Eq. (2.4).

The simplified example shown in Fig. 3.3 will be used to illustrate this method. Equation (2.4) is integrated along the interface S(t) by the finite element method. First the domain S(t) is divided into N two-node elements. The corresponding four-node isoparametric elements near the interface for both solid and liquid domains are also shown in the same figure. Each four-node element on both sides of the interface has one side that coincides with the interface. The heat flux terms in the left-hand side of Eq. (2.4),  $k_1(\partial T_1/\partial n)$  and  $k_2(\partial T_2/\partial n)$ , will be evaluated on the corresponding four-node isoparametric elements of  $\Omega_1$ and  $\Omega_2$ , respectively.





FIGURE 3.3. Illustration of interface moving scheme.

$$\vec{v} \cdot \vec{n} = v_n = \frac{dn}{dt}$$
, (3.47)

where  $\vec{n}$  indicates the direction normal to the interface S and dn represents the magnitude of the displacement of the interface in that direction.

The normal displacement dn can be expressed within each element along the interface as:

$$dn = \sum_{i=1}^{N} N_i dn_i$$
, (3.48)

or in matrix form as  $dn = [N] \{dn\}$ , where [N] is the displacement interpolation vector and  $\{dn\}$  is the vector of element nodal displacement. Specifically in the simplified two-node element,

$$[N] = [\frac{1}{2}(1+\xi) - \frac{1}{2}(1-\xi)]$$

in the local coordinate system. The finite element Galerkin formulation of Eq. (2.4) is

$$\int (K_1 \nabla T_1 - K_2 \nabla T_2) \cdot \vec{n} [N] dS = \int \frac{\rho_1 L}{dt} \{N\} [N] \{dn\} dS . \qquad (3.49)$$

$$S_i \qquad S_i$$

To evaluate the integration, the global coordinate is changed to the local coordinate by the Jacobian transformation:

$$\int_{-1}^{1} (K_1 \nabla T_1 - K_2 \nabla T_2) \cdot \vec{n}[N] |J| d\xi = \int_{-1}^{1} \frac{\rho_1 L}{dt} \{N\} [N] \{dn\} |J| d\xi .$$
 (3.50)

Since the arc length of interface ds can be expressed as

$$dS = \left\{ \left[ \frac{\partial r(\xi, 1)}{\partial \xi} \right]^2 + \left[ \frac{\partial z(\xi, 1)}{\partial \xi} \right]^2 \right\}^{\frac{5}{2}} d\xi , \qquad (3.51)$$

then

$$J = \left\{ \left[ \frac{\partial r(\xi, 1)}{\partial \xi} \right]^2 + \left[ \frac{\partial z(\xi, 1)}{\partial \xi} \right]^2 \right\}^{\frac{1}{2}} . \quad (3.52)$$

The value of |J| is the two-node element is given by

$$|\mathbf{J}| = \frac{1}{2} [(\mathbf{r}_1 - \mathbf{r}_2)^2 + (z_1 - z_2)^2]^{\frac{1}{2}}, \qquad (3.53)$$

the half-length of an element.

The integration is evaluated by the Gauss-Legendre quadrature:

$$\sum_{i=1}^{NG} \omega_{i} (K_{1} \nabla T_{1} - K_{2} \nabla T_{2}) \cdot \vec{n} [N(\xi_{i})] |J| = \sum_{i=1}^{NG} \omega_{i} \frac{\rho_{1} L}{dt} \{N(\xi_{i})\} [N(\xi_{i})] \{dn\} |J| .$$
(3.54)

Next, the evaluation of the heat flux at each Gauss point will be discussed. The value of  $K_1 \nabla T_1 \cdot \vec{h}$  is calculated on the interface using the corresponding adjacent four-node isoparametric element in domain  $\Omega_1$ . The procedures are the same for computing  $K_2 \nabla T_2 \cdot \vec{n}$ . It is noted that the temperature distributions  $T_1$  and  $T_2$  are known at this stage.

By the chain rule of differentiation,

$$\frac{\partial T_{1}}{\partial n} = \begin{bmatrix} \frac{\partial r}{\partial n} & \frac{\partial z}{\partial n} \end{bmatrix} \begin{cases} \frac{\partial T_{1}}{\partial r} \\ \frac{\partial T_{1}}{\partial z} \end{cases} \text{ on } S(t) , \qquad (3.55)$$

and from Eqs. (3.3) and (3.18),

$$\begin{bmatrix} \frac{\partial T_1}{\partial r} \\ \frac{\partial T_1}{\partial z} \end{bmatrix} = \begin{bmatrix} B_1(r,z) \end{bmatrix} \{T_1(t)\} = \begin{bmatrix} J(\xi,n) \end{bmatrix}^{-1} \begin{cases} \frac{\partial \phi_i}{\partial \xi} \\ \frac{\partial \phi_i}{\partial n} \end{cases} \{T_1(t)\},$$
(3.56)

where  $[B_1]$  is the temperature gradient interpolation matrix. It is noted that the values of interpolation functions in the two-dimensioal four-node isoparametric element, when evaluated on the interface, are the same as those of the one-dimensional two-node element. For the purpose of illustration, the  $[B_1]$  matrix at a Gauss point on the interface is given by

$$\begin{bmatrix} B \end{bmatrix} = \frac{1}{2} \int_{0}^{-1} \begin{bmatrix} 2 & 2 & 0 & 0 \\ 1+\xi_{i} & 1-\xi_{i} & \xi_{i}-1 & -\xi_{i}-1 \end{bmatrix} \text{ on } \xi = \xi_{i}, \eta = 1 \quad (3.57)$$

To simplify notation, the element's directional cosine matrix will be denoted as [U] such that

$$\frac{\partial T_{1}}{\partial n} = \begin{bmatrix} \frac{\partial r}{\partial n} & \frac{\partial z}{\partial n} \end{bmatrix} \begin{cases} \frac{\partial T_{1}}{\partial r} \\ \frac{\partial T_{1}}{\partial z} \end{cases} = \begin{bmatrix} U \end{bmatrix} \begin{bmatrix} B_{1}(\xi, \eta=1) \end{bmatrix} \{T_{1}(t)\}$$
(3.58)

on S(t). The final form of the finite element formulation becomes  $\sum_{i=1}^{NG} \omega_{i} [N(\xi_{i})][U] \quad K_{1}[B_{1}(\xi_{i},n=1)\{T_{1}\} - K_{2}[B_{2}(\xi_{i},n=1)]\{T_{2}\} \quad |J|$   $= \sum_{i=1}^{NG} \omega_{i} \frac{\rho_{1}L}{dt} \{N(\xi_{i})\}[N(\xi_{i})] \{dn\} \mid J| \qquad (3.59)$ 

Or, in matrix form,

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$$[A_1][T_1] - [A_2][T_2] = [A_3][dn] , \qquad (3.60)$$

where  $[A_1]$ ,  $[A_2]$ , and  $[A_3]$  are matrices obtained from matrix multiplications. It is noted that the vector [U] has the same value for each element when the interface is flat. However, in a general curved interface the vector [U] may vary from element to element. In summary, we consider Eq. (3.60) as a one-dimensional problem; the left-hand side  $[A_1][T_1]$  and  $[A_2][T_2]$  are the source terms, while [dn] is unknown. The solution of Eq. (3.60) will yield the magnitude of displacement of each node on the interface.

From thermodynamic considerations it can be shown that each point on the interface moves in a direction locally normal to the interface. The direction in which the interface moves is a function of space along the interface and of time. Assume that at any instant in time, there is a node A, as shown in Fig. 3.3, common to elements 1 and 2, where  $\vec{n}_1$  and  $\vec{n}_2$  are normal directions on the side along the interface of element 1 and element 2, respectively. It is seen that  $\vec{n}_1$  and  $\vec{n}_2$  are different in general. However, we need a unique normal direction of each node on the interface. Newton's divided difference formula has been used to construct the interpolation polynomial:

$$P_{n}(x) = f(x_{0}) + (x-x_{0})f[x_{0},x_{1}] + \dots + (x-x_{0}) \dots (x-x_{n-1})f[x_{0},\dots,x_{n}]$$
(3.61)

where  $(x_0, f(x_0))$ ,  $(x_1, f(x_1))$ , ...,  $(x_n, f(x_n))$  are the coordinates of points to be interpolated. Newton's divided difference,  $f[x_0, x_1, ..., x_n]$ , is given by

$$f[x_0, \dots, x_n] = \frac{f[x_1, \dots, x_n] - f[x_0, \dots, x_{n-1}]}{x_n - x_0} \quad . \quad (3.62)$$

It is well known that an interpolation polynomial of high degree, say n > 8, on the evenly spaced points will result in a larger error when the interpolation point is near both sides of the domain [97-100].

To minimize the interpolation error one has to choose an interpolation point as close as possible to the center of the domain. Hence an interpolation polynomial is not constructed through all the nodes on the interface, but rather, five nodes for a fourth-degree polynomial are constructed to find out the center node normal direction. There are N fourth-degree polynomials. The same polynomial is also used to compute the local radius of curvature on the interface by the formula

$$R = \frac{|P_n'|}{(1+P_n'^2)^{3/2}}, \qquad (3.63)$$

where  $P_n(x)$  is the polynomial obtained before.

The magnitude of the displacement of each node calculated using Eq. (3.60) and the direction normal to the interface evaluated using Eq. (3.62) are combined to determine the new locations of the interface. The magnitude of normal velocity of each node equals the displacement of that node divided by the time step size, dn/dt. The velocity is in the same direction as the displacement.

Interface condition, Eq. (2.5),

$$-D \frac{\partial C}{\partial n} = (1-k)C \frac{\partial n}{\partial t} \quad \text{on } S(t)$$

is one of the boundary conditions used for the solution of the mass transfer equation. Since the nature boundary condition is imposed not on a node but on the whole segment, the value dn/dt on an element is approximated by averaging the value of dn/dt at those two nodes on the interface for the four-node example.

### 3.6 SOLUTION ALGORITHM

The differential equations (2.1) to (2.3), together with interface conditions (2.4) to (2.6) and boundary conditions (2.8) have to be solved simultaneously for given initial conditions (2.7). The solution obtained will include the transient temperature distribution in the solid, the transient temperature and concentration distributions in the liquid, and the transient position of phase interface.

In general, the governing equations (2.1) to (2.3) have to be solved in an iterative way such that the interface conditions (2.4) to (2.6) are satisfied at any time. However, in the present study we employ the "front tracking" method, in which the solutions of governing equations (2.1) to (2.3) are sought individually, and Eq. (2.4) is used to "move" the interface. For each equation from (2.1) to (2.3), the  $\theta$  method of the finite element formulation is used. The characteristic of this numerical method, i.e., implicit, explicit, or mixed, depends on the values of  $\theta$ . The fully implicit method can be obtained by choosing  $\theta$  = 1.0. However, the interface condition (2.4) is solved explicitly. The interface position is tracked continuously in time and will be used for the automatic mesh generation of each domain. The moving scheme will be elaborated in the next section.

Through numerical testing, it has been found that the implicitexplicit method for the solutions of the solidification problem gives good results in terms of accuracy and numerical stability, while significantly reducing the computational time. Hence the governing equations are solved in sequence without iteration and marching in time. Of course, for the problems with nonlinear properties, within each solution of Eqs. (2.1) to (2.3), an iteration scheme has to be used.

In summary, the governing equations will be solved as follows:

- (1) Assume the initial interface location and initial distributions of temperature  $T_1$ ,  $T_2$ , and concentration C.
- (2) Solve Eq. (2.4) to obtain the new interface locations, interface moving velocities, and interface curvatures in the next time step.
- (3) Calculate the new concentration distribution C from Eq. (2.3), together with interface condition (2.5) and boundary condition (2.8). The moving interface velocities are obtained from step (2). The iteration is required for nonlinear properties of Eq. (2.3).
- (4) Calculate the new interface temperature distribution form Eq. (2.6), using the thermodynamic relations between temperature and concentration C on the interface, and the interface curvatures. The concentration C and the curvature associated with each node on the interface are obtained from steps (3) and (2), respectively.
- (5) Calculate the temperature distribution T<sub>1</sub> from Eq. (2.1) with interface condition (2.6) obtained in step (4) and boundary conditions
   (2.8). Iterations are required for nonlinear properties of Eq. (2.1).
- (6) Calculate the temperature distribution  $T_2$  from Eq. (2.2) with interface condition (2.6) obtained in step (4) and boundary condition (2.8). Iterations are required for nonlinear material properties.
- (7) Go to step (2) and march forward in time.

### 3.7 AUTOMATIC MESH GENERATION

Since the phase interface is changing in time, the size and shape of both the solid and liquid domains vary during the solidification process. An automatic generation of nodes and elements in each domain at every time step is necessary to successfully solve this moving boundary problem. The nodes on the interface are tracked at all times, as discussed in the preceding section. Based on these interfacial nodes, an automatic mesh generation scheme is developed. It is noted that the typical information to be obtained from meshing a domain includes the total number of nodes and their global numbering, the total number of elements and their global numberings, the number of nodes and local numbering of each element, the coordinates of each node, etc.

The governing equations (2.1) to (2.3) indicate that the temperatures and concentration distributions are determined only by the diffusion mechanism. The existence of a concentration "boundary layer" near the solid-liquid interface during solidification in solution was discussed in the introduction. The boundary layer thickness is proportional to the magnitude of the diffusion coefficient. The typical relative order of magnitude of the diffusion coefficients in the analyzed problem are: thermal diffusivity of solid,  $\alpha_1 = 0(1)$ , thermal diffusivity in liquid,  $\alpha_2 = 0(10)^{-1}$ , and mass diffusivity of liquid, D =  $0(10)^{-3}$ . For example, in the solidification of saline solutions,  $\alpha_1$ of ice is  $1.26 \times 10^{-6}$  m<sup>2</sup>/sec,  $\alpha_2$  is  $1.33 \times 10^{-7}$  m<sup>2</sup>/sec, and D is  $1.29 \times 10^{-9}$  m<sup>2</sup>/sec. The wide range of values for diffusion coefficients implies that the boundary layer thickness for the temperature distribution
is significantly different from that for concentration. Numerous numerical difficulties are associated with this fact. In fact, strong oscillations in numerical solutions were observed during the initial studies. To overcome this difficulty, three different meshing systems were designed: for the solid temperature, for the liquid temperature, and for the concentration in the liquid. The meshing strategy will be illustrated on the geometrical configuration shown in Fig. 3.4, which will be later used in the stability study.

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The meshing strategy for temperature on domain  $\Omega_1$ , as shown in Fig. 3.4, will be illustrated. The nodes on z = 0 are generated by either predetermining the number of nodes or by determining a reference length between the nodes for a given global dimension, R in the rdirection. The grid size may be uniform, as used in this study, or variable depending on the characteristics of the problems. The number of interfacial nodes is the same as that on the outer boundary. The r coordinate of each interfacial node has the same value as the corresponding node on the outer boundary. Thus every line connecting two nodes, one on the outer boundary and the other on the interface, is parallel to the z-direction. It is noted that this parallel requirement is not necessary, but substantial computer time was saved by using this constraint. This point will be elaborated later in this section.

At every time step the maximum distance of any interfacial node from the outer surface was found. This maximum length was divided by a predetermined reference length. The roundoff integer obtained is the number of segments on each line parallel to the z-direction. Since the





heat flux in Eq. (2.4) must be evaluated on the change of phase interface, the grid size in the z-direction in the vicinity of the interface must be refined for better accuracy. The domains were re-meshed at each time step. The advantage of re-meshing is that smooth solutions are obtained between each time step. The disadvantage is an increase in computer time.

The mesh for temperature  $T_2$  in the domain  $\Omega_2$  in Fig. 3.4 is generated exactly the same way as the meshing procedure described above. The reference length for meshing is, however, smaller, since the thermal diffusivity in the liquid is one order of magnitude smaller than that in solid. It is noted that the number of elements and nodes in  $\Omega_1$  continuously increase in time, but in  $\Omega_2$  they decrease. During the automatic meshing procedures, care was taken to satisfy the compatibility condition between elements in the solid and liquid domains.

The solute concentration boundary layer in the liquid phase is very thin relative to the temperature boundary layer. It is necessary to have good resolution within this layer in order to obtain wiggle-free solutions. Thus from both theoretical and practical points of view, it is inadequate to employ the same fixed mesh size used for the temperature solutions in the solution of mass diffusion equation. It was found that a variable mesh size distribution is more suitable. Experimental results show that the concentration decreases exponentially in a direction normal to the interface [8-21]. Hence a new systematic procedure of using exponential functions to generate the mesh was developed. The resulting mesh size increases exponentially in a direction normal to the

boundary. This new method will be illustrated by an example of meshing using one-dimensional two-node elements. This is shown in Fig. 3.5. Suppose that one desires to distribute N nodes within distance L so that the mesh size  $\Delta x_i$  increases exponentially. The grid size is determined by the formulas

$$A(I) = \exp\left[\frac{-I(Ke)}{N-2}\right]$$
$$C = \sum_{i=0}^{N-2} A(I)$$

 $\Delta x(I) = \frac{L}{C} A(I) ,$ 

where I = 0,1,...,N-2,  $\Delta x(I)$  are the grid size as shown in Fig. 3.5. Ke is the meshing coefficient, which determines the mesh size  $\Delta x_i$ . Small adjustments of Ke can significantly affect the meshing. For example, a meshing coefficient of 5.3 will make the distance between the first two nodes smaller by a factor of about 200 than the distance between the last two nodes,  $\Delta x_{N-2}$ , that is,  $\Delta x_{N-2} = \Delta x_0(200)$ , while a meshing coefficient of 6.4 will result in a factor of about 600. The special case of Ke = 0 corresponds to the equal mesh size. The range to be exponentially gridded can be chosen if desired to be only part of the domain. In situations where the boundary layer thickness changes substantially in time, a time-dependent meshing coefficient could be employed. This concept can be extended to two- and threedimensional domains.

As the interface advances the liquid domain decreases, and one





may change the number of nodes along each z-direction and re-mesh the domain at every time step by a method similar to that used in the temperature domains. The alternate method used in this study is simply to squeeze the coordinate of each node on the line parallel to the r-direction by a ratio proportional to the reduction of the corresponding global length. Because the nodes on the interface move with different velocities, except in the planar interface case, the ratio of contraction for each line parallel to the z coordinate is different.

The unique characteristic of a transient problem with increasing domain will be illustrated by a simple example shown in Fig. 3.6. In this figure, the "old" solid lines represent the meshing system at time  $t = t_1$ , and the "new" dashed lines show the meshing system at time t = $t_1 + \Delta t$ . It is noted that at time  $t = t_1 + \Delta t$ , the temperature, for example, on all nodes in the old meshing system are known. However, in a transient problem we have to know the old temperature on nodes of the new meshing system. Hence the interpolation method has to be employed to find out the old temperatures for these new nodes. As has been indicated before, at every time step all the lines connecting points on the outer boundary and the corresponding points on the interface are parallel to the r-direction. Hence, the old temperature of a new node can be obtained by searching and interpolating only along the line on which it is located. Otherwise, for each node in the new meshing system, it is necessary to identify the old element to which this node belongs, and then use the finite element interpolation functions to calculate its temperature. It is obvious that significant computer time is saved





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using our method.

It is interesting to see that at node A, shown in Fig. 3.6, the previous old temperature is not available. The temperature of this node at the previous time step simply did not exist. This situation occurs when the time step is too large or the interface moving velocity is too fast for the grid size adopted. This is an extra constraint in the moving boundary problem, in addition to the limitation of time step imposed by possible numerical stability considerations.

#### 3.8 COMPUTER PROGRAM

The program is written in a modular form, resulting in flexibility and ease of modification. It is composed of about 65 subroutines. The program was initially developed and tried on the Vax 11/750 UNIX system of the Mechanical Engineering Department of the University of California The results presented in this thesis were obtained from at Berkeley. the CDC-7600 at Lawrence Berkeley Laboratory. The typical computer CPU time is 60 minutes per 1000 time steps without iterations. The program can be used in two- or three-dimensional axisymmetric problems with linear or nonlinear properties. The program was designed to deal with four, eight, and nine-node isoparametric elements. Special care was taken to reduce the memory and computation time. Band stiffness matrix was employed, and standard Gaussian LU decomposition, plus forward and backward substitutions, were used to solve the matrix. A listing of the program appears in Appendix 2.

# CHAPTER 4:

# INTERFACE STABILITY ANALYSIS

## 4.1 PROBLEM DESCRIPTION

Since, as indicated in the Introduction, we plan to study the morphological stability of a planar interface, a typical dendritic domain taken from Fig. 1.1 (designated by broken lines) is illustrated in Fig. 4.1. Both the upper and lower surfaces in this domain are adiabatic from symmetry considerations. The domain is considered in three-dimensional axisymmetry, which is a typical model of a dendrite. An inertial coordinate system rather than a moving coordinate system attached on the phase interface is chosen and illustrated in Fig. 4.1, where R and Z represent the dimension of domain in the r and z coordinates, respectively;  $C_i$  and  $T_i$  are the initial concentration and temperature of the solution. The solidifying medium in this study was chosen for illustration purposes to be a saline solution. The thermophysical properties of saline solutions and ice are listed in Table 4.1. As mentioned previously, many analytic studies on solid-liquid interface stability are based on the assumption that the dimension coinciding with the direction of dendritic growth is semi-infinite. Thus the z-dimension of the domain was taken large enough relative to the r-direction to satisfy this assumption.

To apply the general computer program developed in this study to this specific interface stability problem, it is only necessary to specify some



FIGURE 4.1. Domain of a half-dendrite, three-dimensional axisymmetry case.

	Water	Ice	Units
ρ	999.	999	kg/m <sup>3</sup>
k	5.55×10 <sup>-4</sup>	2.25×10-3	kW/mK
C	1.83	4.22	kJ∕m <sup>3</sup> K
L	353	353	kJ/kg
D	1.18×10 <sup>-9</sup>	-	m <sup>a</sup> /sec
m	-1.86	-1.86	k/M
k	Partition coefficient = 0		

TABLE 4.1. Thermophysical Properties of Dilute Saline Solution and Ice

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parameters at the time of program input data. For example, in the constant properties problem, a parameter is specified such that the subroutine written to solve the nonlinear properties problem will not be called. The boundary conditions are treated in the same way. Since the domain is a three-dimensional axisymmetry, the volume integration in the finite element formulation, dv = r dr dz, is chosen such that r is the average distance of each element from the axis of symmetry. For a two-dimensional domain, r simply equals unity.

# 4.2 NUMERICAL PERTURBATIONS

To study the stability of a planar interface, two types of numerical perturbations have been used: temperature perturbations on the outer boundary and concentration perturbations on the interface. Each kind of perturbation was imposed in the space and/or time domains. In general, in this study the different types of perturbation have been imposed separately. However, if necessary the computer program can handle simultaneously any combination of the various perturbations. In this study only spatial perturbations of these two types have been emphasized, since the main purpose here is to study the stability of a spatially perturbed planar interface. In fact, some perturbations on the time domain were attempted in this study, but their physical significance needs further investigation. Since any arbitrary functions can be represented as the Fourier series of sinusoidal functions, cosine function was used as the perturbation. The procedures of numerical perturbations are illustrated in this section, but the discussion on the physical meanings of the perturbations and the results will be given in Chapter 5.

## 4.2.1 Temperature Perturbation on the Outer Boundary

The schematic illustration of spatial temperature perturbation on the outer boundary is shown in Fig. 4.2. On the outer boundary a constant temperature  $T_{\infty}$  below the freezing temperature of the solution is initially imposed. The temperature difference between the outer boundary  $T_{\infty}$  and the interfacial temperature is the driving force for advancing the interface. It is obvious that under this condition the interface will move in such a way that a planar surface is continuously maintained. Then at time  $t = t_1$  a sinusoidal temperature perturbation described by

$$T = T_0 + A \cos\left(\frac{\pi r_i}{R}\right), \quad t_1 \le t \qquad (4.1)$$

is suddenly imposed, where A is the small amplitude of perturbation, R is the global dimension of outer boundary in the r-direction, and  $r_i$ is the coordinate of node i on the outer boundary. The number of nodes i is preselected. It was expected that a perturbed interface similar to a sinusoidal shape would be gradually generated during the time when the spatial temperature perturbation was applied. In some of the problems, the perturbation was removed at time  $t = t_2$ , and the morphology of the interface was continuously examined. Notce that  $\pi/R$  is equivalent to the frequency, and by changing R any kind of wavelength for the perturbation can be obtained.

A different kind of temperature perturbation in the time domain can be obtained through the boundary condition

$$T = T_{a} + A' \cos\left(\frac{\pi \Delta t}{T}\right) , \quad t_{1} \leq t \leq t_{2} , \quad (4.2)$$





where A' is the small amplitude of perturbation,  $T = t_2 - t_1$  is the half period of a sinusoidal function, and  $\Delta t = t - t_1$ . Here the example only gives a half cycle of cosine wave with an appropriate period of T. This perturbation could be extended in a straightforward manner to any length of time during which arbitrary periods of sinusoidal perturbation is applied. These two perturbations (4.1) and (4.2) can be combined by simple multiplication of the perturbed terms.

## 4.2.2 Concentration Perturbation on the Interface

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A perturbed solid-liquid interface can also be created through a concentration perturbation on the interface. In this case the temperature at the outer boundary is kept at constant value. Similar procedures to those described in the previous paragraph on temperature perturbations are used to initiate a moving planar interface. Then an artificial numerical perturbation of concentration

$$C(r) = C_0(r) + B \cos\left(\frac{\pi r_i}{R}\right), t_1 \le t$$
 (4.3)

is performed on the interface, where  $C_0(r)$  is the original concentration distribution along the interface S(t) and B is the small amplitude of the perturbation. The distribution  $C_0(r)$  is a constant at the initial perturbation; thereafter, it will be a function of r and z, i.e., it will vary along the interface S(t). The concentration perturbation in the time domain on the interface is similar to that in Eq. (4.2).

In this study it is assumed that the amplitude of all kinds of

perturbation is constant. This is not necessarily true, since it also can be a function of time. The results obtained from this study on the effect of various numerical perturbations, discussed in the next chapter, will provide a much deeper understanding of the interface stability phenomena.

# CHAPTER 5:

## **RESULTS AND DISCUSSION**

The computer program developed in this work was used to study the stability of a solid-liquid interface during transient solidification processes.

## 5.1 INTRODUCTION

First the effects on the interface stability of transient temperature fluctuations on the outer surface of the domain were studied. The study was performed for the rectangular geometry shown in Fig. 5.1. The rectangular enclosure contained a liquid solution initially at a constant temperature. The transient solidification process was started by suddenly changing the temperature on one of the narrow walls of the rectangular enclosure to a constant value below the phase transformation temperature. Adiabatic boundary conditions were imposed on the other walls, resulting in a time-dependent propagation of the planar, solid-liquid interface in a direction normal to the constant-temperature wall.

To study the effects of temperature fluctuations on the stability of the moving interface, spatially sinusoidal temperature perturbations were superimposed on the constant temperature boundary for various periods of time and then removed. This was done at different times following the onset of the solidification process. The position and



FIGURE 5.1. Computational domain for studying the interface stability.

velocity of the solid-liquid interface and the multi-dimensional temperature and concentration distributions were continuously calculated using the front tracking finite element method developed in this work. The temperature fluctuation superimposed on the constant temperature boundary condition affected the shape of the planar interface, which became perturbed as well. It was anticipated that on a morphologically stable interface the spatial perturbation would disappear after the removal of the temperature perturbation, while on an unstable interface the spatial perturbation would continue to grow. Numerous computer runs were performed for a medium with the thermophysical properties of a saline solution (Table 4.1). Numerical experiments with different length scales and time scales were attempted. The results of all numerical experiments indicate that a planar solid-liquid interface is stable, during the transient solidification process, to temperature perturbations on the outer boundary, i.e., the spatial perturbation of the change of phase interface disappears after the removal of the tempera-The stability of a planar interface to temperature ture perturbation. fluctuations was observed in all the computer runs, including situations in which the liquid in front of the change of phase interface was thermodynamically supercooled. In such a situation, the constitutional supercooling theory predicts an unstable interface.

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To illustrate the numerical procedure and the observations described above, one set of typical results for certain geometrical and thermal conditions will be initially presented. The length scales used in this analysis are compatible with experimentally determined values for the

dimensions of a perturbed planar interface at the onset of instability (15 µm) during the transient freezing of a saline solution. The length of the narrow wall in the enclosure was, consequently, taken to be 15 µm and the length of the longer wall 2 mm. A large enough ratio between the longer and the narrow wall of the rectangular enclosure was chosen to ensure that a semi-infinite domain could be effectively The initial concentration of the saline solution in the simulated. enclosure was 34.0  $gmol/m^3$  and the initial temperature was -0.121°C. This temperature corresponds to the phase transformation temperature in a 34.0 gmol/m<sup>3</sup> saline solution. To start the transient solidification process a constant temperature of  $-2.0^{\circ}$ C was imposed on one of the narrow walls of the enclosure. Adiabatic boundary conditions were imposed on all the other walls.

# 5.2 PLANAR INTERFACE

First, the transient position of the one-dimensional interface and the transient temperature and concentration profiles were calculated using the front tracking finite element method. Figure 5.2 shows typical temperature and concentration distributions in the solid and liquid regions at various times. The results illustrate several wellknown physical phenomena, which will be discussed in detail since they are of importance in understanding the morphological stability of solid-liquid interfaces during transient solidification processes.

One of these phenomena is the narrow concentration boundary layer adjacent to the change of phase interface in the liquid region.



FIGURE 5.2. Temperature and concentration distributions of the solid and liquid; planar interface.  $R = 0.15 \times 10^{-4}$  m,  $Z = 0.2 \times 10^{-3}$  m,  $C_i = 35.0$  gmol/m<sup>3</sup>,  $T_i = -0.121^{\circ}C$ ,  $T_{\infty} = -2.0^{\circ}C$ .

According to the constitutional phase diagram for saline solutions, ice cannot contain any solute. Consequently, saline is rejected in front of the change of phase interface during solidification. The concentration distribution in front of the change of phase interface is affected by two competitive mechanisms. One is the rejection of solute in front of the moving interface, which is directly related to the interfacial velocity and results in an increase in the solute concentration. In the second method, solute is transported away from the interface by the diffusion mechanism, which is proportional to the concentration gradient and the magnitude of the mass diffusion coefficient.

Figures 5.2, 5.3, and 5.4 show a continuous increase in the solute concentration on the change of phase interface, implying that the solute rejection rate exceeds the rate of solute being diffused away. The relative effect of these two competitive mechanisms is of fundamental importance in undertsanding the stability of a planar interface during a transient solidification process.

The phase transformation temperature is inversely related to solute concentration according to the constitutional phase diagram. Consequently, during the transient solidification process analyzed in this work the temperature on the change of phase interface will continuously decrease in time. This can be observed in the temperature distribution curve in Fig. 5.2.

The velocity of the change of phase interface is proportional to the heat conducted through the solid and removed at the outer boundary. The heat removed is directly related to the temperature gradient in the solid region. This temperature gradient is determined by the temperature

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FIGURE 5.3. Concentration distribution in the liquid, planar interface.  $R = 0.15 \times 10^{-6}$  m,  $Z = 0.2 \times 10^{-3}$  m,  $C_i = 34.0 \text{ gmol/m}^3$ ,  $T_i = -0.121^{\circ}\text{C}$ ,  $T_{\infty} = -2.0^{\circ}\text{C}$ .



FIGURE 5.4. Interfacial concentration, planar interface. R =  $0.15 \times 10^{-4}$  m, Z =  $0.2 \times 10^{-3}$  m, C<sub>1</sub> = 34.0 gmol/m<sup>3</sup>, T<sub>1</sub> = -0.121°C, T<sub>∞</sub> = -2.0°C.

difference between the outer boundary temperature and the interfacial temperature divided by the solid layer thickness. During the transient solidification process the temperature gradient is continuously attenuated in time due to two factors: the continuous increase in distance between the moving interface and the outer boundary and the decrease in change of phase temperature due to solute accumulation on the interface. This phenomenon is illustrated by the results in Figs. 5.5 and 5.6. In Fig. 5.5 the position of the change of phase interface is plotted as a function of time for the freezing of the  $34.0 \text{ gmol/m}^3$ saline solution and for the freezing of pure water. The pure water was at an initial temperature of 0°C and was frozen by imposing a constant temperature of -1.879°C on the outer surface. In Fig. 5.6 the velocity of the change of phase interface is shown as a function of time. The figures show that the solidification process is faster in pure water. As explained above, this is due to the decrease in the change of phase temperature due to solute accumulation on the interface. The observation that solute accumulation on the change of phase interface slows the velocity of the interface during transient solidification processes will be of importance in the forthcoming analysis on the change of phase morphological stability.

Figures 5.7 to 5.10 show results obtained for a solidification process in a rectangular enclosure in which the length of the narrowed wall was taken to be 10  $\mu$ m and the length of the longer wall was taken to be 80  $\mu$ m. The initial concentration of saline solution was taken to be 34.0 gmol/m<sup>3</sup> and the initial temperature was -0.121°C. A constant



FIGURE 5.5. Interfacial position, planar interface. R =  $0.15 \times 10^{-4}$  m, Z =  $0.2 \times 10^{-3}$  m, C<sub>i</sub> = 34.0 gmol/m<sup>3</sup>, T<sub>i</sub> = -0.121°C, T<sub>∞</sub> = -2.0°C.

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# FIGURE 5.6.

Interfacial velocity, planar interface. R =  $0.15 \times 10^{-4}$  m, Z =  $0.2 \times 10^{-3}$  m, C<sub>1</sub> = 34.0 gmol/m<sup>3</sup>, T<sub>1</sub> = -0.121°C, T<sub>∞</sub> = -2.0°C.

temperature of -0.3°C was imposed on one of the narrow walls of the domain. Despite the much lower temperature gradient in the solid region, essentially similar phenomena to those observed in the previous case occurred. Figures 5.7 and 5.8, showing the concentration distribution in the liquid region and the concentration on the interface, indicate a continuous increase in the concentration on the interface. The rate of concentration increase is slower, however, than in the previous example. This can be explained by the lower velocity of the solidification process caused by the lower temperature gradient. The lower velocity can be observed by comparing Figs. 5.9 and 5.10 with Figs. 5.5 and 5.6. Despite this difference, the fundamental behavior remains unchanged, i.e., the solute accumulation on the interface and the distance of the change of phase interface from the outer surface cause a continuous decrease in the change of phase interface velocity.

Figure 5.2 shows that in the liquid region the concentration gradient is much steeper than the temperature gradient. This well-known phenomenon is directly related (according to the various stability criteria) to the solid-liquid phase transformation interface instability.

The "constitutional supercooling" stability theory predicts that an interface will become unstable if the solute in front of the change of phase interface is thermodynamically supercooled. This can be expressed by the relation

$$\frac{m}{\frac{\partial C}{\partial n}} > 1, \qquad (5.1)$$

with the concentration and temperature gradients evaluated on the

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FIGURE 5.7. Concentration distribution in the liquid, planar interface.  $R = 0.1 \times 10^{-4} \text{ m}, Z = 0.8 \times 10^{-4} \text{ m}, C_{i} = 34.0 \text{ gmol/m}^{3}, T_{i} = -0.121^{\circ}\text{C}, T_{\infty} = -0.3^{\circ}\text{C}.$ 



FIGURE 5.8. Interfacial concentration, planar interface.  $R = 0.1 \times 10^{-4} \text{ m}$ ,  $Z = 0.8 \times 10^{-4} \text{ m}$ ,  $C_i = 34.0 \text{ gmol/m}^3$ ,  $T_i = -0.121^{\circ}\text{C}$ ,  $T_{\infty} = -0.3^{\circ}\text{C}$ .





interface in the liquid region in a direction normal to the change of phase interface.

According to the M-S general stability criteria, a surface will become unstable if

$$\left(\frac{\frac{m}{\partial n}}{\frac{\lambda T_{1}}{\lambda n} + K_{2}} \xrightarrow{\partial T_{2}}{\lambda n}\right) > 1, \qquad (5.2)$$

where the concentration and temperature gradients are evaluated on the interface in a direction normal to the change of phase interface. The M-S stability criterion presented here is for the special situation in which capillary effects are neglected. It should be emphasized that the M-S criterion was brought up for completeness only. The criterion is not applicable to the situation discussed here since several of the major assumptions are different in this model. The M-S criterion pertains to a solidification process that is steady in a moving frame of reference and inifinite in domain, whereas the analyzed problem is transient in a finite domain.

## 5.3 TEMPERATURE PERTURBATION ON THE OUTER BOUNDARY

The second step in our analysis was to superimpose on the outer boundary a spatially sinusoidal temperature perturbation. Here we will discuss the results obtained when such a perturbation, with a magnitude of -0.02 cos  $(\pi r_i/R)$  °C, was imposed on  $0.5 \times 10^{-4}$  sec after the onset of the first solidification process. A perturbation with a magnitude of -0.01 cos  $(\pi r_i/R)$  °C was imposed  $0.24 \times 10^{-2}$  sec after the onset of the second case solidification process. At that instant in time the ratio in Eq. (5.1) and (5.2) was on the order of 10<sup>3</sup>, indicating that the liquid adjacent to the change of phase interface was supercooled and the interface unstable, according to considerations of equilibrium thermodynamics. The results were obtained using the front tracking finite element method developed in this work. Since the stability criteria do not include capillary effects, the stabilizing effect of capillarity in this example was not included. However, the computer program utilized can incorporate this effect. The dendrites of the liquid and the solid were also assumed to be the same so as not to introduce convection effects.

Figures 5.11 and 5.12 show the location of the solid-liquid interface relative to that of the central node on the interface during the freezing of a saline solution and of water in the first case. Figure 5.13 shows the solid-liquid interface in the second case. The location of the interface is shown at different times after the perturbation was imposed. Figs. 5.11 and 5.13 present results by continuously imposing a temperature perturbation on the outer boundary.

The results in Fig. 5.12 were obtained by imposing the temperature perturbation  $0.5 \times 10^{-4}$  sec after the onset of the solidification process, followed by removal of temperature perturbation  $0.5 \times 10^{-4}$  sec later. Figures 5.11 and 5.13 indicate that the temperature perturbation resulted in a spatial perturbation on the change of phase interface in both the saline solution and the pure water. It is interesting to notice that the amplitude of the perturbation is much larger in pure water.





FIGURE 5.12. Interfacial position, temperature perturbation.  $T_{\infty} = -2.0 - 0.02 \cos (\pi r_i/R)^{\circ}C$  in  $0.5 \times 10^{-4} < t < 1.0 \times 10^{-3}$  S,  $T_{\infty} = -2.0^{\circ}C$  in  $1.0 \times 10^{-4} < t$  S,  $R = 0.15 \times 10^{-4}$  m,  $Z = 0.2 \times 10^{-3}$  m,  $C_i = 34.0$  gmol/m<sup>3</sup>,  $T_i = -0.121^{\circ}C$ .

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FIGURE 5.13. Interfacial position, temperature perturbation.  $T_{\infty} = -0.3 - 0.01 \cos (\pi r_i/R)^{\circ}C$ , in  $0.24 \times 10^{-2} < t < 0.192 \times 10^{-2} S$ ,  $R = 0.1 \times 10^{-4}$ m,  $Z = 0.8 \times 10^{-6}$  m,  $C_i = 34.0 \text{ gmol/m}^3$ ,  $T_i = -0.121^{\circ}C$ .

The most important results of this analysis is probably that shown in Fig. 5.12. This figure indicates that after the temperature perturbation was removed, the change of phase interface spatial perturbation disappeared and the interface became planar again. The interface became planar more quickly in the saline solution than in pure water. This result indicates that the change of phase interface during a transient solidification process in a saline solution is stable to temperature perturbations on the outer surface. Furthermore, the interface is dynamically stable in situations in which the solution in front of the change of phase interface is supercooled and should be unstable from considerations of equilibrium thermodynamics. Following is an explanation of these results using Figs. 5.14-5.18.

Figures 5.14 and 5.15 show the concentration distribution on the change of phase interface relative to the concentration on the central node on the interface during the solidification processes described in Figs. 5.11-5.13. Figures 5.16-5.18 show the normal velocity of the solid-liquid interface relative to that of the center node on the interface during the two solidification processes in Figs. 5.11-5.13, respectively. The concentration distribution shown in Figs. 5.14 and 5.15 is of major importance in understanding the morphological stability of the change of phase interface during the analyzed transient solidification process.

According to concepts of static thermodynamic equilibrium, the change of phase interface is supposed to be unstable when the solute in front of the change of phase interface is thermodynamically supercooled. From considerations of static thermodynamic equilibrium, if the change





FIGURE 5.15. Interfacial concentration, temperature perturbation.  $T_{\infty} = -0.3 - 0.01 \cos (\pi r_i/R)^{\circ}C$  in  $0.24 \times 10^{-2} < t < 0.192 \times 10^{-2}$  S, R =  $0.1 \times 10^{-4}$  m, Z =  $0.8 \times 10^{-4}$  m, C<sub>i</sub> =  $34.0 \text{ gmol/m}^3$ , T<sub>i</sub> =  $-0.121^{\circ}C$ .

of phase interface is perturbed, the "tip" of the perturbed interface will suddenly find itself in the supercooled region surrounded by a lower solute concentration than the "groove." As a consequence, the tip will grow faster and the planar interface will break. Our rigorous transient numerical analysis shows that during a dynamic transient solidification process, such a phenomenon cannot occur. We would like to emphasize again that this analysis deals with a specific transient solidification process to which the M-S stability criterion cannot be applied.

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Figures 5.14 and 5.15 show that the concentration profile on the interface obtained through a rigorous and exact analysis of the dynamic solidification process is different from that assumed in the static thermodynamic equilibrium stability criterion. The concentration of solute is actually higher at the tip of the perturbed interface, i.e., the point furthest away from the outer surface on which the transient temperature boundary condition was imposed. The concentration is lowest in the groove, the point on the interface closer to the outer surface.

This result, although different from that assumed in the static thermodynamic equilibrium stability criterion, is consistent with the intuitively obvious results obtained previously for the planar solidification process and shown in Figs. 5.2-5.4 and 5.7. These figures indicate that during the transient planar solidification process the solute accumulation in front of the change of phase interface is affected by the solute rejection mechanism due to the transient solidification process and by the solute diffusion mechanism in the liquid. Since



FIGURE 5.16. Interfacial velocity, temperature perturbation.  $T_{\infty} = -2.0 - 0.02 \cos (\pi r_i/R)^{\circ}C$  in  $0.5 \times 10^{-4} < t < 2.0 \times 10^{-4} S$ , R =  $0.15 \times 10^{-4}$  m, Z =  $0.2 \times 10^{-3}$  m,  $C_i = 34.0$  gmol/m<sup>3</sup>,  $T_i = -0.121^{\circ}C$ .





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FIGURE 5.18. Interfacial velocity, temperature perturbation.  $T_{\infty} = -2.0 - 0.02 \cos (\pi r_i/R)^{\circ}C$ in  $0.5 \times 10^{-4} < t < 1.0 \times 10^{-4} S$ ,  $T_{\infty} = -2.0^{\circ}C$  in  $1.0 \times 10^{-4} < t S$ ,  $R = 0.15 \times 10^{-4}$  m,  $Z = 0.2 \times 10^{-3}$  m,  $C_i = 34.0$  gmol/m<sup>3</sup>,  $T_i = -0.121^{\circ}C$ .

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\*\* \*\* during the analyzed transient solidification process the solute rejection rate exceeds the rate of solute being diffused away, i.e., the time scale of the solidification process is much shorter than that of the mass diffusion process, the solute will continuously accumulate on the interface. Therefore, the further the planar change of phase interface is from the outer surface, the higher the solute concentration on the interface.

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The temperature perturbation superimposed on the outer surface temperature resulted in a gradually growing spatial perturbation on the change of phase interface. This is similar to the planar solidification case, since in this problem the time scale of the solidification process and the consequent solute rejection rate are much shorter than that of the mass diffusion process. The solute concentration on the tip of the perturbed interface, which at any instant in time is farther from the outer surface, is higher than that in the groove, which is closer to the outer surface. This results illustrates the importance of incorporating the transient dynamic effects in a study of the morphological stability of the change of phase interfaces.

Figures 5.16-5.18 will be analyzed next in conjunction with Figs. 5.14 and 5.15, and an explanation will be presented for the result in Fig. 5.13. In the previous study on the planar solidification process it was shown that the velocity of the change of phase interface is proportional to the heat conducted through the solid and removed at the outer boundary. The heat transported is determined by the temperature difference between the outer surface and the interfacial temperature

divided by the solid layer thickness. During a planar solidification process, when a temperature perturbation is superimposed on the outer surface temperature, a higher temperature difference will appear between certain points on the change of phase interface and the outer surface. This will increase the solidification rate at these points and consequently will increase the distance between these points on the interface and the outer surface. The increased distance will reduce the temperature gradient and consequently continuously reduce the velocity of the solidification process. The solute accumulation on the change of phase interface discussed with respect to Figs. 5.14 and 5.15 will decrease the change of phase temperature on the moving interface. This temperature will be lower for points on the interface farther away from the outer surface (the tip of the perturbed interface). The increased solute concentration on the interface will also reduce the temperature gradient and consequently will reduce the velocity of the interface.

When the temperature perturbation on the outer surface is removed (i.e., a constant temperature is imposed on the outer surface), the temperature gradient and the consequent heat flux will be lower at points on the change of phase interface furthest away from the outer surface. For the case of pure water, the difference in temperature gradient is affected only by the perturbed interface and the difference in the thickness of the solid layer. For saline solution, the difference is enhanced by the lower temperature on the change of phase interface at points further from the outer surface. Consequently, after removing the temperature perturbation, the velocity of the interface at points closer to the outer surface will increase relative to that at points further from the outer surface until the perturbed interface returns to a stable planar.

The perturbed interface becomes planar faster during the freezing of a solution than during the freezing of pure water. These results are evident in Figs. 5.17 and 5.12. This result indicates that during the analyzed transient solidification process the increased concentration of saline on the interface has a stabilizing effect. The results of this work prove that in the analyzed transient solidification process, a solid-liquid interface surrounded by a thermodynamically supercooled liquid cannot become unstable by means of a transient temperature perturbation on the outer surface. It should be emphasized that this statement is restricted to the transient solidification process analyzed in this work. Since experimental evidence indicates that the solid-liquid interface became unstable during the solidification process, new studies are required to promote the understanding of the physical phenomena associated with the perturbed growth of a solid-liquid interface during transient solidification.

#### 5.4 CONCENTRATION PERTURBATION ON THE INTERFACE

In the previous section it was shown that a temperature perturbation on the outer surface of a solidifying domain cannot induce the instability commonly observed on such an interface during transient solidification. It has been shown that the solute concentration on the interface has a stabilizing effect. Consequently, a study was performed

to determine the effects of perturbing the concentration on the interface on the stability of that interface. The study was performed for the same rectangular enclosure discussed in the previous section. The length of the narrow wall was 10  $\mu$ m and that of the long wall 80  $\mu$ m. The initial concentration of the saline solution was 34.0 gmol/m<sup>3</sup> and the initial temperature -0.121°C. A constant temperature of -0.3°C was imposed on one of the narrow walls of the enclosure and the other walls were adiabatic. The details of the unperturbed solidification process in this system are shown in Figs. 5.7-5.10 and have been discussed in the previous section.

In this part of the study, a concentration perturbation with a magnitude of 2.0 cos  $(\pi r_i/R)$  gmol/m<sup>3</sup> was imposed on the change of phase interface concentration at different instants in time and for various periods of time. This concentration perturbation yielded a continuous decrease in the concentration with a magnitude of 2.0 gmols on the left-hand side of the enclosure relative to the central node in the enclosure, a continuous increase with a magnitude of 2.0 gmols on the right-hand side of the enclosure relative to the central node, and a sinusoidal variation between these two extreme points. Specifically, the concentration on the interface is taken at all times as  $C = C_0(r) + 2.0 \cos (\pi r_i/R) \text{ gmol/m}^3$ , where  $C_0(r)$  is the concentration on the interface is time step. Obviously, since the temperature at points with lower concentration is higher, according to the discussion in the previous section these points will experience a higher temperature gradient and move faster. This is confirmed by the results shown in Fig.

5.16. Figure 5.17 shows the location of the solid-liquid interface relative to that of the central node on the interface during a solidi-fication process in which the concentration perturbation is continu-ously imposed.

The results clearly indicate that the continuous concentration perturbation results in a continuous perturbation of the interface. This behavior has been anticipated as discussed above. It should be emphasized that the results reported in the previous section indicate that a continuous temperature perturbation on the outer surface also yields a continuous perturbation of the interface (see Fig. 5.13). There is, however, a fundamental difference in the behavior of a system perturbed by a concentration perturbation relative to that perturbed by a temperature perturbation. This difference can be observed in Fig. 5.20 and can be explained by Fig. 5.21.

Figure 5.20 shows the velocity of the interface at various instants in time during the perturbation. This figure is especially enlightening when compared with the interface velocity during the temperature perturbation shown in Fig. 5.17. It is seen that for the concentration perturbation the difference in velocity between the tip and the groove of the interface increases in time, while for the concentration perturbation it decreases in time. Thus, while for the temperature perturbation the effects of the temperature perturbation on the interface morphology perturbation will decrease in time, in the concentration perturbation the morphological perturbation on the interface increases in time, eventually leading to the unstable interface observed in experiments.









FIGURE 5.20.

Interfacial velocity, concentration perturbation.  $C = C_0 + 2.0 \cos (\pi r_i/R) \text{ gmol/m}^3, R = 0.1 \times 10^{-4} \text{ m},$   $Z = 0.8 \times 10^{-4} \text{ m}, C_i^{-1} = 34.0 \text{ gmol/m}^3, T_{\infty} = -0.3^{\circ}\text{C},$  $T_i^{-1} = -0.121^{\circ}\text{C}.$ 

An explanation for this phenomenon can be obtained through a comparison of Figs. 5.11 and 5.15. These figures show the interface concentration in time in the case of a temperature perturbation and a concentration perturbation, respectively. It is seen in Fig. 5.15 that following a temperature perturbation, the saline concentration on the tip of the perturbed interface increases in time relative to that in the groove. As explained in the previous section, this has a stabilizing effect since it decreases the temperature gradient on the tip relative to that in the groove and reduces the velocity of the tip relative to that in the groove. Figure 5.22 shows that in the case of a concentration perturbation, a completely different phenomenon occurs. The concentration on the tip of the perturbed interface is lower than that in the groove and consequently a higher temperature gradient is expected on the tip than on the groove. This leads to a higher velocity of the tip relative to the groove, which is essentially what happens during an unstable solidification process.

It should be emphasized, however, that the phenomenon in which the tip velocity continuously increases relative to that in the groove was observed only during a continuous concentration perturbation. In numerical experiments in which concentration perturbations were imposed only for one step and then removed, different behavior was observed. Here the interface became perturbed after the single step concentration perturbation. However, since no mechanism was available to remove the solute from the fastest-growing region of the interface (the tip), the concentration on the tip eventually became larger than that in the groove. This result, shown in Fig. 5.23, led to the disappearance of the morphological



# FIGURE 5.21.

Interfacial concentration, concentration perturbation.  $C = C_0 + 2.0 \cos (\pi r_i/R) \text{ gmol/m}^3$ ,  $R = 0.1 \times 10^{-6} \text{ m}$ ,  $Z = 0.8 \times 10^{-6} \text{ m}$ ,  $C_i = 34.0 \text{ gmol/m}^3$ ,  $T_{\infty} = -0.3^{\circ}\text{C}$ ,  $T_i = -0.121^{\circ}\text{C}$ .



# FIGURE 5.22.

Interfacial concentration, concentration perturbation, one step only.  $C = C_0 + 2.0 \cos (\pi r_i/R) \text{ gmol/m}^3$ ,  $R = 0.1 \times 10^{-6} \text{ m}$ ,  $Z = 0.8 \times 10^{-6} \text{ m}$ ,  $C_i = 34.0 \text{ gmol/m}^3$ ,  $T_{\infty} = -0.3^{\circ}\text{C}$ ,  $T_i = -0.121^{\circ}\text{C}$ .

perturbation on the interface in a similar form to that occurring during transient temperature perturbations.

The results of this part of the study indicate that a continuous concentration perturbation on the interface can lead to an unstable interface during a transient solidification process. This is an extremely important observation since it might indicate the fundamental mechanism responsible for the experimentally observed unstable interfaces during solidification in solution.

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## CHAPTER 6: CONCLUSIONS

A new numerical method using "front tracking" finite elements has been developed to solve the multi-dimensional transient heat and mass diffusion equations associated with the solidification processes in binary solutions. The numerical method can incorporate realistic thermodynamic conditions on the interface (including surface tension effects) and can accommodate non-isothermal interfaces and irregular transient geometries of the interface. At present this is the only method with these capabilities.

In the front tracking method, the thermal and concentration field equations are solved implicitly while the energy balance equation on the interface is treated as an independent equation being solved explicitly to obtain the new interface position in time. Hence the governing equations of solidification of binary solutions are solved in sequence and marching in time without iteration. The front tracking method developed here is unconditionally stable. Essentially there is no constraint on the size of the time step in terms of numerical stability. Specific to this work is the special procedure by which the change of phase interface is tracked in time.

The interface is tracked in time by two steps. First the magnitude of displacement and normal direction are independently obtained for each node on the interface; then they are superimposed to determine the new interface position. This procedure, which is fundamentally different from that in other front tracking methods, was necessary because of the special conditions on the interface in this problem. A new systematic exponential meshing technique was designed for the concentration field. The novel meshing procedure is essential to obtain an accurate result in this problem.

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> This method was used to study the physical phenomena occurring during a transient solidification process in binary solutions. The numerical analysis of a transient solidification process in a saline solution has shown that a thin concentration boundary layer will develop in front of the interface during the process. This is caused by the rejection of solute from the solid, which occurs at a faster rate than the diffusion of solute from the interface in the bulk of the solution. It was shown using equilibrium thermodynamics that this phenomenon causes the liquid in front of the interface to be "constitutionally supercooled." According to existing stability criteria, the planar interface must be morphologically unstable.

The new numerical method was employed in a study of the stability of such an interface for different perturbations. The results of the numerical analysis indicate that a transient temperature fluctuation on the outer surface of the solidifying domain cannot generate the instability of the moving interface even in situations in which the solute in front of the interface is thermodynamically supercooled.

Furthermore, it was shown that the solute concentration in front of the interface has a stabilizing effect. These results contradict

the theoretical predictions of commonly accepted stability criteria, which are based on equilibrium thermodynamics. The discrepancy is caused by the important effect of the transient heat and mass transfer phenomena on the solidification process, a factor that must not be neglected in the stability analysis.

The numerical method was also employed to study the effect of a concentration perturbation on the interface. It was shown that a continuous concentration perturbation can lead to an unstable interface, and it is tentatively proposed that the instability of the interface must be related with the solute convection process. The results of this work demonstrate the importance of this new method, and show that new fundamental studies are needed to promote the understanding of the physical phenomena associated with the perturbed growth of a solid-liquid interface during transient solidification.

#### REFERENCES

1. B. Chalmers, Principles of Solidification, Wiley, New York, 1964.

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- M.C. Flemings, <u>Solidification Processing</u>, McGraw-Hill, New York, 1974.
- K.M. Fisher, "The Effects of Fluid Flow on the Solidification of Industrial Castings and Ingots," <u>Physiocochem. Hydro</u>., <u>2</u>(4), 311-326, 1981.
- C.L. Jones and P. Capper, "Thermal Modelling of Casting of Cd, Hg, Te," <u>J. Crystal Growth</u>, <u>63</u>, 145-153, 1983.
- 5. B.R. Pamplin, Ed., <u>Crystal Growth</u>, 2nd ed., Pergamon, New York, 1981.
- 6. F. Rosenberger, <u>Fundamentals of Crystal Growth I</u>, Springer-Verlag, New York, 1979.
- 7. M.E. Glicksman, R.J. Schaefer, and J.D. Ayers, "Dendritic Growth: A Test of Theory," <u>Metal. Trans. A</u>, <u>7A</u>, 1747-1759.
- 8. J.S. Langer, R.F. Sekerka, and T. Fujioka, "Evidence for a Universal Law of Dendritic Growth Rates," <u>J. Crystal Growth</u>, <u>44</u>, 414-418, 1978.
- 9. H. Muller-Krumbhaar and J.S. Langer, "Siderbranching Instabilities in a Two-Dimensional Model of Dendritic Solidification," Acta Metal., 29, 145-157, 1981.
- S.C. Huang and M.E. Glicksman, "Fundamentals of Dendritic Solidification - I. Steady-State Tip Growth," <u>Acta Metal.</u>, <u>29</u>, 701-715, 1981.
- S.C. Huang and M.E. Glicksman, "Fundamentals of Dendritic Solidification - II. Development of Siderbranch Structure," <u>Acta Metal.</u>, <u>29</u>, 717-734, 1981.
- 12. D.J. Fisher and W. Kurz, "A Theory of Branching Limited Growth of Irregular Eutectics," <u>Acta Metal.</u>, <u>28</u>, 777-794, 1980.
- W. Kurz and D.J. Fisher, "Dendrite Growth at the Limit of Stability: Tip Radius and Spacing," <u>Acta Metal</u>., <u>29</u>, 11-20, 1981.
- M. Solari and H. Bilonia, "Microsegregation in Cellular and Cellular Dendritic Growth," <u>J. Crystal Growth</u>, <u>49</u>, 451-457, 1980.

- 15. M.H. Burden and J.D. Hunt, "Cellular and Dendritic Growth I," J. Crystal Growth, 22, 99-108, 1974.
- 16. M.H. Burden and J.D. Hunt, "Cellular and Dendritic Growth II," J. Crystal Growth, 22, 109-116, 1974.
- S. Witzke, J.P. Riquet, and F. Durand, "Diffusion Field Ahead of a Growing Columnar Front: Discussion of the Columnar-Equiaxed Transition," <u>Acta Metal.</u>, 29, 365374, 1981.
- B.W. Grange, R. Viskanta, and W.H. Stevenson, "Diffusion of Heat and Solute During Freezing of Salt Solutions," <u>Int. J. Heat & Mass Transf.</u>, 19, 373-384, 1976.
- R.L. Levin, "The Freezing of Finite Domain Aqueous Solutions: Solute Redistribution," <u>Int. J. Heat & Mass Transf.</u>, <u>24</u>(9), 1443-1455, 1981.
- J.P. Terwilliger and S.F. Dizio, "Salt Rejection Phenomena in the Freezing of Saline Solution," <u>Chem. Eng. Sci.</u>, <u>25</u>, 1331-1349, 1970.
- 21. C. Korber, M.W. Scheiwe, and K. Wollhoever, "Solute Polarization During Planar Freezing of Aqueous Salt Solutions," <u>Int. J. Heat</u> & Mass Transf., 26(8), 1241-1253, 1983.
- 22. K. Shibrta, J. Sato, and G. Ohira, "The Solute Distributions in Dilute Al-Ti Alloys During Unidirectional Solidification," J. Crystal Growth, 44, 435-445, 1978.
- T.W. Clyne, "Heat Flow in Controlled Directional Solidification of Metals - II. Mathematical Model," <u>J. Crystal Growth</u>, <u>50</u>, 691-700, 1980.
- 24. M.G. O'Callaghan, E.G. Cravalho, and C.E. Huggins, "An Analysis of the Heat and Solute Transport During Solidification of an Aqueous Binary Solution - I. Basal Plane Region," <u>Int. J. Heat</u> & Mass Transf., 25(4), 553-561, 1982.
- 25. M.G. O'Callaghan, E.G. Cravalho, and C.E. Huggins, "An Analysis of the Heat and Solute Transport During Solidification of an Aqueous Binary Solution II. Dendrite Tip Region," <u>Int. J. Heat & Mass Transf.</u>, 25(4), 563-573, 1982.
- 26. B.W. Grange, R. Viskanta, and W.H. Stevenson, "Solute and Thermal Redistribution During Freezing of Salt Solutions," ASME, Cu3.3.
- R. Siegel, "Analysis of Solidification Interface Shape Resulting from Applied Sinusoidal Heating," <u>J. Heat Transf.</u>, <u>104</u>, 13-18, 1982.
- S.R. Coriell and R.F. Sekerka, "Lateral Solute Segregation During Unidirectional Solidification of Binary Alloy with a Curved Solid-Liquid Interface," J. Crystal Growth, 46, 479-482, 1979.

- 29. S.R. Coriell, R.F. Boisvert, R.G. Rehm, and R.F. Sekerka, "Lateral Solute Segregation During Unidirectional Solidification of a Binary Alloy with a Curved Solid-Liquid Interface - II. Large Departures from Planarity," <u>J. Crystal Growth</u>, <u>54</u>, 167-175, 1981.
- T.W. Clyne, "Heat Flow in Controlled Directional Solidification of Metals, I. Experimental Investigation," <u>J. Crystal Growth</u>, <u>50</u>, 684-690, 1980.
- 31. C. Korber and M.W. Scheiwe, "Observations on the Nonplanar Freezing of Aqueous Salt Solutions," <u>J. Crystal Growth</u>, <u>61</u>, 307-316, 1983.
- I. Jin and G.R. Purdy, "Controlled Solidification of a Dilute Binary Alloy - II. Experiment," <u>J. Crystal Growth</u>, <u>23</u>, 37-44, 1974.
- 33. I. Jin and G.R. Purdy, "Controlled Solidification of a Dilute Binary Alloy - I. Theory," J. Crystal Growth, 23, 29-36, 1974.
- R. Trivedi, "Theory of Dendritic Growth During the Directional Solidification of Binary Alloys," <u>J. Crystal Growth</u>, <u>49</u>, 219-232, 1980.
- 35. M.H. Johnston, C.S. Griner, R.A. Parr, and S.J. Robertson, "The Direct Observation of Unidirectional Solidification as a Function of Gravity Level," J. Crystal Growth, 50, 831-838, 1980.
- 36. S.R. Coriell, M.R. Cordes, W.J. Boettinger, and R.F. Sekerka, "Convective and Interfacial Instabilities During Unidirectional Solidification of a Binary Alloy," <u>J. Crystal Growth</u>, <u>49</u>, 13-28, 1980.
- 37. W.A. Tiller, K.A. Jackson, J.W. Rutter, and B. Chalmers, Acta Metal., 1, 428, 1953.
- 38. W.W. Mullins and R.F. Sekerka, "Stability of a Planar Interface During Solidification of a Dilute Binary Alloy," <u>J. Appl. Phys.</u>, <u>35</u>(2), 444-451, 1964.
- 39. W.W. Mullins and R.F. Sekerka, "Morphological Stability of a Particle Growing by Diffusion or Heat Flow," J. Appl. Phys., 34(2), 323-329, 1963.
- R.F. Sekerka, "A Stability Function for Explicit Evaluation of the Mullin-Sekerka Interface Stability Criterion," <u>J. Appl. Phys.</u>, <u>36(1)</u>, 264-268, 1965.
- 41. R. F. Sekerka, "Morphological Stability," <u>J. Crystal Growth</u>, <u>3-4</u>, 71-81, 1968.

- 42. G. Horvay and J.W. Cahn, "Dendritic and Spheroidal Growth," Acta Metal., 9, 695-705, 1961.
- 43. D.P. Woodruff, <u>The Solid-Liquid Interface</u>, Cambridge University Press, 1973.
- 44. R.T. Delves, "<u>Theory of Interface Stability</u>," in <u>Crystal Growth</u>, B.R. Pamplin, Ed., 2nd ed., Pergamon, New York, 1981.
- 45. S.R. Coriell and R.F. Sekerka, "Oscillatory Morphological Instability due to Non-Equilibrium Segregation," <u>J. Crystal Growth</u>, <u>61</u>, 499-508, 1983.
- 46. S.R. Coriell and R.F. Sekerka, "Effect of Convective Flow on Morphological Stability," <u>Physicochem. Hydro.</u>, 2(4), 281-293, 1981.
- 47. J.S. Langer and H. Muller-Krumbhaar, "Theory of Dendritic Growth - I. Element of a Stability Analysis," <u>Acta Metal.</u>, <u>26</u>, 1681-1687, 1978.
- J.S. Langer and H. Muller-Krumbhaar, "Theory of Dendritic Growth -II. Instability in the Limit of Vanishing Surface Tension," <u>Acta</u> <u>Metal., 26</u>, 1689-1695, 1978.
- H. Muller-Krumbhaar and J.S. Langer, "Theory of Dendritic Growth -III. Effects of Surface Tension," Acta Metal., 26, 1697-1708, 1978.
- 50. N. Ramachandran, J.P. Gupta, and Y. Jaluria, "Two-Dimensional Solidification with Natural Convection in the Melt and Convective and Radiative Boundary Conditions," <u>Num. Heat Transf.</u>, <u>4</u>, 469-484, 1981.
- 51. N. Ramachandran, J.P. Gupta, and Y. Jaluria, "Thermal and Fluid Flow Effects During Solidification in a Rectangular Enclosure," Int. J. Heat & Mass Transf., 25(2), 187-194, 1982.
- 52. T.K. Sinha and J.P. Gupta, "Solidification in an Annulus," <u>Int.</u> J. Heat & Mass Transf., <u>25</u>(11), 1771-1773, 1982.
- 53. N. Shamsundar and E.M. Sparrow, "Effect of Density Change on Multi-Dimensional Conduction Phase Change," <u>J. Heat Transf</u>., 551-557, Nov. 1976.
- 54. B. Cantor and A. Vogel, "Dendritic Solidification and Fluid Flow," J. Crystal Growth, 41, 109-123, 1977.
- 55. D.T.J. Hurle, E. Jakeman, and A.A. Wheeler, "Effect of Solutal Convection on the Morphological Stability of a Binary Alloy," J. Crystal Growth, 58, 163-179, 1982.

- 56. M.A. Azouni, "Local Temperature Measurements Over Ice Water Interface and Convective Flows Patterns," <u>J. Crystal Growth</u>, <u>47</u>, 109-114, 1979.
- 57. J.H. Quenisset and R. Naslain, "Effect of Forced Convection on Eutectic Growth," J. Crystal Growth, 54, 465-474, 1981.
- 58. M.E. Glicksman, "Capillary Phenomena During Solidification," J. Crystal Growth, 42, 347-356, 1977.

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- 59. R. Trivedi, H. Franke, and R. Lacmann, "Effects of Interface Kinetics on the Growth Rate of Dendrites," <u>J. Crystal Growth</u>, 47, 389-396, 1979.
- 60. D.T.J. Hurle, "The Effect of Soret Diffusion on the Morphological Stability of a Binary Alloy Crystal," <u>J. Crystal Growth</u>, <u>61</u>, 463-472, 1983.
- 61. B. Cantor and A. Vogel, "Dendritic Solidification and Fluid Flow," J. Crystal Growth, 41, 109-123, 1977.
- 62. C. Gau and R. Viskanta, "Flow Visualization During Solid-Liquid Phase Change Heat Transfer, I. Freezing in a Rectangular Cavity," Int. Comm. Heat Mass Transf., 10, 173-181, 1983.
- 63. C. Gau, R. Viskanta, and C.J. Ho, "Flow Visualization During Solid-Liquid Phase Change Heat Transfer, II. Melting in a Rectangular Cavity," <u>Int. Comm. Heat Mass Transf.</u>, 10, 183-190, 1983.
- 64. R.J. Schaefer, "The Validity of Steady-State Dendrite Growth Models," J. Crystal Growth, 43, 17-20, 1978.
- R. Trivedi, "Comments on the Validity of Steady-State Dendrite Growth Models by R.J. Schaefer," <u>J. Crystal Growth</u>, <u>44</u>, 110-111, 1978.
- 66. D.G. Wilson, A.D. Soloman, and P.T. Boggs, Eds., <u>Moving Boundary</u> <u>Problems</u>, Academic, New York, 1978.
- 67. J.R. Ockenden and W.R. Hodgkins, Eds, <u>Moving Boundary Problems</u> in Heat Flow and Diffusion, Clarendon Press, Oxford, 1977.
- 68. V.J. Lunandini, <u>Heat Transfer in Cold Climates</u>, Van Nostrand Reinhold, New York, 1981.
- 69. M.N. Ozisik, Heat Conduction, Wiley, New York, 1980.
- 70. B. Rubinsky and G. Cravalho, <u>Trans. ASME, J. Heat Transf.</u>, 326-330, May 1979.
- 71. T.R. Goodman, Trans. ASME, 80, 335-342, 1956.
- 72. J. Stefan, Ann. Phy. Chem. Newefalge, 42(2), 269-286, 1881.

- 73. N. Shamsundar and E.M. Sparrow, "Analysis of Multidimensional Conduction Phase Change Via the Enthalpy Model," <u>J. Heat & Mass</u> Transf., 333-340, Aug. 1975.
- 74. N. Shamsundar, "Approximation Calculation of Multidimensional Solidification by Using Conduction Shape Factors," <u>J. Heat Transf.</u>, 104, 8-12, Feb. 1982.
- 75. J.L. Sproston, "Two Dimensional Solidification in Pipes of Rectangular Section" <u>Int. J. Heat & Mass Transf.</u>, <u>24(9)</u>, 1493-1501, 1981.
- 76. V.R. Voller and M. Cross, "Estimating the Solidification/Melting Times of Cylindrically Symmetric Regions," <u>Int. J. Heat & Mass</u> Transf., 24(9), 1457-1462, 1981.
- 77. V. Voller and M. Cross, "Accurate Solutions of Moving Boundary Problems Using the Enthalpy Method," <u>Int. J. Heat & Mass Transf.</u>, 24, 545-556, 1981.
- 78. R. Bonerot and P. Jamet, "Numerical Computation of the Free Boundary for the Two-Dimensional Stefan Problem by Space-Time Finite Elements," <u>J. Comp. Phys.</u>, <u>25</u>, 163-181, 1977.
- 79. H.M. Ettouney and R.A. Brown, "Finite Element Methods for Steady Solidification Problems," J. Comp. Phys., 49, 118-150, 1983.
- 80. D.R. Lynch and K. Q'nell, "Continuously Deforming Finite Elements for the Solution of Parabolic Problems, With and Without Phase Change," Int. J. Num. Meth. in Eng., 17, 81-96, 1981.
- 81. K. Miller and R.M. Miller, "Moving Finite Elements I," <u>SIAM J.</u> Num. Anal., 18(6), 1019-1032, 1981.
- 82. K. Miller, "Moving Finite Elements II," <u>SIAM J. Num. Anal.</u>, <u>18</u>(6), 1033-1057, 1981.
- 83. B. Rubinsky and E.G. Cravalho, "A Finite Elemenet Method for the Solution of One-Dimensional Phase Change Problems," <u>Int. J. Heat</u> & Mass Transf., 24(12), 1987-1989, 1981.
- 84. B. Rubinsky, "Solidification Processes in Saline Solutions," J. Crystal Growth, 62, 513-522, 1983.
- 85. J. Yoo and B. Rubinsky, Num. Heat Transf., 6, 205-222, 1983.
- 86. G. Comini and S. del Guidice, <u>Int. J. Num. Math. Eng</u>., <u>8</u>, 613-624, 1979.
- 87. J. Ronel and B.R. Baliga, ASME Paper No. 79-wa/HT-54.

- 88. E.B. Becker, G.F. Carey, and J.T. Oden, <u>Finite Elements: An</u> Introduction, Vol. 1, Prentice-Hall, Englewood Cliffs, NJ, 1981.
- 89. G.F. Carey and J.T. Oden, <u>Finite Elements: A Second Course</u>, Vol. 2, Prentice-Hall, Englewood Cliffs, NJ, 1983.
- 90. K.J. Bathe, <u>Finite Element Procedures in Engineering Analysis</u>, Prentice-Hall, Englewood Cliffs, NJ, 1982.
- 91. K.H. Huebner and E.M. Thornton, <u>The Finite Element Method for</u> Engineers, 2nd ed., Wiley-Interscience, New York, 1982.
- 92. O.C. Zienkiewicz, <u>The Finite Element Method</u>, 3rd ed., McGraw-Hill, New York, 1977.
- 93. O.C. Zienkiewicz and K. Morgan, <u>Finite Elements and Approximation</u>, Wiley-Interscience, New York, 1982.
- 94. J.E. Akin, <u>Application and Implementation of Finite Element</u> <u>Methods</u>, Academic, New York, 1982.
- 95. T.J.R. Highes, "Finite Element Methods for Convection Dominated Flow," ASME AMO, Vol. 34, 1979.
- 96. A.K. Noor and W.D. Pilkey, Eds., <u>State of the Art Surveys on Finite</u> Element Technology, ASME, New York, 1983.
- 97. P.J. Davis, Introduction and Approximation, Dover, 1975.
- 98. K.E. Atkinson, <u>An Introduction to Numerical Analysis</u>, Wiley, New York, 1978.
- 99. G. Dahlquist and A. Bjorck, <u>Numerical Methods</u>, Prentice-Hall, Englewood Cliffs, NJ, 1974.
- 100. S.D. Conte and C. Boor, <u>Elementary Numerical Analysis</u>, McGraw-Hill, New York, 1980.

#### APPENDIX 1:

### DERIVATION OF MULLINS-SEKERKA CRITERION, EQUATION (1.17)

A sinusoidal perturbation of very small amplitude  $\delta$  applying to the planar interface in the constant moving coordinate can be described by

$$z = \phi(x,t) = \delta(t) \sin \omega x , \qquad (A.1)$$

where  $\omega = 2\pi/\lambda$  is the wave frequency. Notice that a two-dimensional model is used.

The governing equations of solute diffusion in the liquid and heat diffusion of both the solid and liquid in a constant moving frame of reference are:

$$D\left(\frac{\partial^2 C}{\partial z^2} + \frac{\partial^2 C}{\partial x^2}\right) + V \frac{\partial C}{\partial z} = 0 \qquad (A.2)$$

$$\alpha_{S}\left(\frac{\partial^{2}T_{S}}{\partial z^{2}} + \frac{\partial^{2}T_{S}}{\partial x^{2}}\right) + V \frac{\partial T_{S}}{\partial z} = 0 \qquad (A.3)$$

$$\alpha_{L}\left(\frac{\partial^{2}T_{L}}{\partial z^{2}} + \frac{\partial^{2}T_{L}}{\partial x^{2}}\right) + V \frac{\partial T_{L}}{\partial z} = 0 \quad . \tag{A.4}$$

Equation (A.2) will be solved first.

The solution to (A.2) can be written as

$$C(z,x) = C_1(z) + C_1(z) \sin \omega x$$
, (A.5)

where C is the solute perturbation of order  $\delta$  and  $C_L(z)$  is the basic solution of unperturbed situation derived in Eq. (1.5). For convenience,

Eq. (1.5) is rewritten here:

1

$$C_{L}(z) = C_{0} + \frac{DG_{C}}{v} [1 - \exp(-Vz/D)].$$
 (1.5)

It is noted that  $z = \phi(x,t)$  is the coordinate of the perturbed interface, and not z = 0. Substituting Eq. (A.5) into (A.2), C(z) should satisfy

$$D\left(\frac{\partial^2 C_1}{\partial z^2} - \omega^2 C_1\right) + V \frac{\partial C_1}{\partial z} = 0 . \qquad (A.6)$$

At the perturbed interface the assumptions of linear perturbation on temperature and concentration yield

$$T_{\phi} = T_{0} + a\phi(x,t) = T_{0} + a\delta(t) \sin \omega x \qquad (A.7)$$
  

$$C_{\phi} = C_{0} + b\phi(x,t) = T_{0} + b\delta(t) \sin \omega x , \qquad (A.8)$$

where  $T_0$  and  $C_0$  are the temperature and concentration at the unperturbed planar interface, respectively, the same as those obtained in Section 1.2.1. The second terms in Eqs. (A.7) and (A.8) are the first-order corrections corresponding to the infinitesimal perturbation. Subscript  $\phi$  of  $T_{\phi}$  and  $C_{\phi}$  is used to emphasize that these values are derived on the perturbed interface. a and b are constants to be determined. The boundary conditions for Eq. (A.6) are

$$C_1 = \delta b - \frac{\delta DG_C}{V\phi} [1 - \exp(-V\phi/D)] \text{ at } z = \phi \qquad (A.9)$$

 $C_1 \neq 0$  as  $z \neq \infty$ . (A.10)

The solution of Eq. (A.6) under the above two boundary conditions is

$$C_1 = \delta(b - G_C) \exp(-\omega_C z)$$
, (A.11)

where

$$G_{C} = -\frac{V}{D} (C_{L} - C_{S}) \qquad (A.12)$$

$$\omega_{\rm C} = \frac{V}{2D} + \left[ \left( \frac{V}{2D} \right)^2 + \omega^2 \right]^{\frac{1}{2}} \qquad (A.13)$$

The complete solution of Eq. (A.2) is obtained by substituting Eqs. (1.5) and (A.9) into (A.5) to yield

$$C(z,x) = \left\{ C_0 + \frac{DG_C}{V} \left[ 1 - \exp\left(\frac{-VZ}{D}\right) \right] \right\} + \delta(b-G_C) \exp(-\omega_C z) \sin \omega x .$$
(A.14)

Following exactly the same steps above, the solutions of Eqs. (A.3) and (A.4) are

$$T_{L}(z,x) = \left\{ T_{0} + \frac{\alpha_{L}G_{L}}{V} \left[ (1 - \exp(-Vz/\alpha_{L})) \right] \right\} + \left[ \delta(a - G_{L}) \exp(-\omega_{L}z) \sin \omega x \right]$$
(A.15)

$$T_{S}(z,x) = \left\{ T_{0} + \frac{\alpha_{S}G_{S}}{V} \left[ 1 - \exp(-Vz/\alpha_{S}) \right] \right\} + \left[ \delta(a - G_{S}) \exp(-\omega_{S}z) \sin \omega x \right] , \qquad (A.16)$$

where

$$\omega_{L} = \frac{V}{2\alpha_{L}} + \left[ \left( \frac{V}{2\alpha_{L}} \right)^{2} + \omega^{2} \right]^{\frac{1}{2}}$$
(A.17)

$$\omega_{\rm S} = \frac{V}{2\alpha_{\rm S}} - \left[ \left( \frac{V}{2\alpha_{\rm S}} \right)^2 + \omega^2 \right]^{\frac{1}{2}} \qquad (A.18)$$

It is seen that the first part of the solutions in (A.11) and (A.12)

are the unperturbed solutions. If the capillarity is considered on the perturbed interface, then

$$T_{L\phi} = T_{S\phi} = T_{\phi} = T_{m} + mC_{\phi} - T_{m}\Gamma R , \qquad (A.19)$$

where  $\Gamma = \gamma/L$ , the surface free energy divided by the latent heat per unit volume,  $T_m$  is the melting point of pure liquid at the planar situation, m is the slope of liquid line, and  $R = (1/r_1)+(1/r_2)$  is the curvature of the interface.

Since the perturbed interface is a sinusoidal function, r =in the y-direction and R is obtained only from the  $1/r_2$  by the following formula:

$$R = -\frac{\partial^2 \phi}{\partial x^2} \left[ 1 + \left( \frac{\partial \phi}{\partial x} \right)^2 \right]^{-3/2} . \qquad (A.20)$$

Since  $\partial \phi / \partial x \sim \delta$ , which is negligible compared with 1, and  $\phi = \delta \sin \omega x$ , then *R* is obtained as

$$R = \delta \omega^2 \sin \omega x \quad . \tag{A.21}$$

Hence

$$T_{\phi} = T_{m} + mC_{\phi} - T_{m}\Gamma\delta\omega^{2} \sin \omega x \qquad (A.22)$$

Substituting Eqs. (A.7) and (A.8) into (A.18), the relationship between a and b is obtained by

$$a = mb - T_m \Gamma \omega^2 . \qquad (A.23)$$

The energy and solute balance at the perturbed interface requires

$$\left(k_{s}\frac{\partial T_{s}}{\partial z} - k_{L}\frac{\partial T_{L}}{\partial z}\right)_{\phi} = Lv(x) \qquad (A.24)$$

$$\left[D(k-1)\frac{\partial C}{\partial z}\right]_{\phi} = C_{\phi}v(x)$$
 (A.25)

$$\frac{1}{L}\left(k_{S}\frac{\partial T_{S}}{\partial z} - k_{L}\frac{\partial T_{C}}{\partial z}\right)_{\phi} = \frac{1}{C_{\phi}}\left[D(k-1)\frac{\partial C}{\partial z}\right]_{\phi}.$$
 (A.26)

The gradients of concentration in the liquid and of temperatures in the liquid and solid at the interface to the first order of  $\delta$  are:

$$\left(\frac{\partial C}{\partial z}\right)_{\phi}^{\phi} = -\omega_{C} \left[b - G_{C} \left(1 - \frac{V}{\omega_{C}D}\right)\right] \delta \sin \omega x + G_{C}$$

$$\left(\frac{\partial T_{L}}{\partial z}\right)_{\phi}^{\phi} = -\omega_{L} \left[a - G_{L} \left(1 - \frac{V}{\omega_{L}\alpha_{L}}\right)\right] \delta \sin \omega x + G_{L}$$

$$\approx -\omega(a - G_{L})\delta \sin \omega x + G_{L}$$

$$(A.28)$$

$$\left(\frac{\partial T_{S}}{\partial z}\right)_{\phi} = \omega_{S} \left[\alpha - G_{S} \left(1 - \frac{V}{\omega_{S} \alpha_{S}}\right)\right] \delta \sin \omega x + G_{S}$$
$$\approx \omega(\alpha - G_{S}) \delta \sin \omega x + G_{S} , \qquad (A.29)$$

where the approximations  $\alpha_{L}\omega_{L} >> V$  and  $\alpha_{S}\omega_{S} >> V$  have been made. It is noted that in all cases of practical interest,  $V < 3.0 \times 10^{-5}$  m/sec,  $\alpha_{L} > 10^{6}$  m<sup>2</sup>/sec,  $\omega >> V/\alpha_{L} \sim 3.0 \times 10^{-1}$ /m, and  $\lambda = 2\pi/\omega << \alpha_{L}/V \sim 10^{-1}$ cm (typical values for a metallic system). Also, by Eqs. (A.13) and (A.14),  $\omega_{S} \sim \omega_{I} \sim \omega$  can be obtained.

Substituting these gradients into Eq. (A.22), and by Eq. (A.19), the constant b to the first order of  $\delta$  is obtained by

$$b = \frac{2G_{C}T_{m}\Gamma\omega^{2} + \omega G_{C}(\xi_{S} + \xi_{L}) + G_{C}(\omega_{C} - \frac{V}{D})(\xi_{S} - \xi_{L})}{2\omega m G_{C} + (\xi_{S} - \xi_{L})[\omega_{C} - (V/D)P]}, \quad (A.30)$$

where

$$\xi_{L} = \frac{k_{L}}{\overline{k}} G_{L} , \quad \xi_{S} = \frac{k_{S}}{\overline{k}} G_{S}$$
  
 $\overline{k} = \frac{k_{L}}{2}(k_{S} + k_{L}) , \quad P = (1 - k) .$ 
(A.31)

Notice that

$$v(x) = V + \frac{d\delta(t)}{dt} \sin \omega x . \qquad (A.32)$$

Substitute Eq. (A.23) into (A.21) to obtain

$$\mathbf{v}(\mathbf{x}) = \frac{\overline{\mathbf{k}}}{L} \left\{ (\xi_{S} - \xi_{L}) + \omega [2a - (\xi_{S} + \xi_{L})\delta \sin \omega \mathbf{x}] \right\}.$$
(A.33)

Equating the coefficients of Eqs. (A.27) and (A.28) yields

$$V = \frac{\overline{k}}{L} (\xi_{S} - \xi_{L})$$
 (A.34)

$$\delta = \frac{d\delta}{dt} = \frac{2k}{L} \omega [a - \frac{1}{2}(\xi_{S} + \xi_{L})]\delta \qquad (A.35)$$

The value of a can be obtained by substituting b from (A.26) into (A.29). Then (A.30) becomes the central result,

$$\frac{\delta}{\delta} = \frac{V \left[ -2T_{m}\Gamma\omega^{2}\left(\omega_{C} - \frac{V}{D}P\right) - (\xi_{S} + \xi_{L})\left(\omega_{C} - \frac{V}{D}P\right) + 2mG_{C}\left(\omega_{C} - \frac{V}{D}\right) \right]}{(\xi_{S} - \xi_{L})\left(\omega_{L} - \frac{V}{D}P\right) + 2\omega mG_{C}}$$
(A.36)

### APPENDIX 2:

### LISTING OF COMPUTER PROGRAMS
```
program aatsai(input,output,tape5=input,tape6=output)
      level 2. gk.gf.nz
c....Common block for general purpose
      common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/entrl1/ nopt1,nopt2,nopt3
      common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
2
      common/cdoman/ xdim,ydim,prop(4,4)
      common/cmatrx/ gk(1200,25),gf(1200),nz
      common/celem/ ne(1000), node(9,1000)
      common/cwork/ xc(2,1200), nodec(9,1000), uc(1200)
c....Common block for region 1 and 2, both are temp. field
      common/cnode/ x12(2,1200),u12(1200)
c....Common block for region 1, Temp. of solid
      common/econ1/ nnode1,nelem1,npot1,nbc11,nbc12
      common/cbc1/ ndbc11(25),vbc11(25),neb12(20),nsde12(20),vbc12(2,20)
                  ,npt1(10),vpt1(10)
      common/cscal1/ rlen1, rval1
c....Common block for region 2, Temp. of liquid solution
      common/ccon2/ nnode2,nelem2,npot2,nbc21,nbc22
      common/cbc2/ ndbc21(25),vbc21(25),neb22(20),nsde22(20),vbc22(2,20)
                  ,npt2(10),vpt2(10)
      common/cscal2/ rlen2,rval2
c....Common block for region 3, concentration of liquid solution
      common/ccon3/ nnode3,nelem3,npot3,nbc31,nbc32
      common/cnode3/ x3(2,1200),u3(1200)
      common/sbc3/ ndbc31(50),vbc31(50),neb32(40),nsde32(40),vbc32(2,40)
                  ,npt3(10),vpt3(10)
c....Common block for interface, region 4
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                    ,ang1(25),capa(2,25),tin(25),ratio(50)
      common/cmatr4/ gk4(25,3),gf4(25),vnor(25)
      common/celem4/ ne4(25),nodes4(3,25)
C
      call data
      call comn
      call init
      icount=0
100
      icount=icount+1
      time=icount*delt
      call move
      call sov3
      call gett
      call sov1
      call sov2
      if(time.lt.tmax) go to 100
      stop
      end
      subroutine adst4(xim)
С
c....To adjust the interface nodal coordinates
c....To compute the interface normal directions
c....To compute the principal curvatures along the interface
c....To construct the finer interface mesh for computing concentration
c....To use 5-node interpolation polynomial
```

```
common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                     ,angl(25),capa(2,25),tin(25),ratio(50)
      common/cdoman/ xdim,ydim,prop(4,4)
      dimension xim(2,25), xx(5), xy(5)
c....Deal with the end nodes
c....Symmetry on both sides
      const=1.57079632679489
      angl(1) = dasin(1.d0)
      angl(max124) = angl(1)
      ratio(1) = (ydim-xim(2,1))/(ydim-xi3(2,1))
      ratio(nmax3)=(ydim-xim(2,max124))/(ydim-xi3(2,nmax3))
      xi_3(2,1) = xim(2,1)
      xi124(2,1) = xim(2,1)
      xi3(2,nmax3) = xim(2,max124)
      xi124(2, max124) = xim(2, max124)
      capa(1,1) = 0.0
      capa(1, max124) = 0.0
      capa(2,1)=0.0
      capa(2, max124) = 0.0
c....Use 3-node interpolation poly. at the side nodes
      m=2
      do 10 i=1,3
        xx(i) = xim(1,i)
        xy(i) = xim(2,i)
10
      continue
c....Get divided differences
      call aivdi4(xx,3,xy)
c....Get finer mesh for concentration
      call int4(xx,3,xy,xi3(1,2),p)
      ratio(m) = (ydim-p)/(ydim-xi3(2,m))
      xi3(2,m)=p
c....Get common node for temp. and concen.
      call inter4(xx,3,xy,xi124(1,2),p,thita,curv)
      ms=2*m-1
      ratio(ms)=(ydim-p)/(ydim-xi3(2,ms))
      xi3(2,ms)=p
      xi124(2,m)=p
      angl(m)=thita
      capa(1,m) = curv
      if(xi124(1,m).eq.0.0) then
        capa(2,m)=capa(1,m)
        else
          if (thita.gt.const) then
            capa(2,m)=0.0
          else
            capa(2,m)=cos(thita)/xi124(1,m)
          end if
      end if
      do 50 m=3, max124-2
        mi=m-2
        do 40 i=1.5
          xx(i) = xim(1,mi)
          xy(i) = xim(2,mi)
```

```
mi=mi+1
40
        continue
        call divdi4(xx,5,xy)
        mm = 2 * (m - 1)
        call int4(xx,5,xy,xi3(1,mm),p)
        ratio(mm)=(ydim-p)/(ydim-xi3(2,mm))
        xi3(2,mm)=p
        call inter4(xx,5,xy,xi124(1,m),p,thita,curv)
        ms = 2 * m - 1
        ratio(ms)=(ydim-p)/(ydim-xi3(2,ms))
        xi124(2,m)=p
        xi3(2,ms)=p
        angl(m)=thita
        capa(1,m)=curv
        if(thita.gt.const) then
          capa(2,m) = 0.0
        eise
          capa(2,m) = cos(thita)/xi124(1,m)
        end if
50
      continue
      m = max 124 - 2
      do 60 i=1,3
        xx(i) = xim(1,m)
        xy(1) = xim(2.m)
        m=m+1
60
      continue
      mm=nmax3-3
      call divdi4(xx,3,xy)
      do 70 i=mm.mm+2
        call_int4(xx,3,xy,xi3(1,i),p)
        ratio(i) = (ydim-p)/(ydim-xi3(2,i))
        xi3(2.i)=D
70
      continue
      np=max124-1
      call inter4(xx,3,xy,xi124(1,np),p,thita,curv)
      xi124(2.np)=p
      angl(np)=thita
      capa(1,np)=curv
      if(thita.gt.const) then
        capa(2,np)=0.0
      else
        capa(2,np)=cos(thita)/xi124(1,np)
      end if
      return
      end
      subroutine aply1
      level 2, gk.gf.nz
      common/cntrl1/ nopt1,nopt2,nopt3
      common/cntrl2/ time.tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
      common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
      common/cbc1/ ndbc11(25),vbc11(25),neb12(20),nsde12(20),vbc12(2,20)
                   .npt1(10).vpt1(10)
      common/celem/ ne(1000), node(9,1000)
```

```
common/cnode/ x12(2,1200),u12(1200)
      common/cmatrx/ gk(1200,25),gf(1200),nz
      dimension pe(9,9),game(9),xx(2,9),nod(9),uu(9)
C.
c....Apply point loads
      if(npot1.eq.0) go to 20
      do 10 i=1, npot1
        n=npt1(i)
10
        gf(n)=gf(n)+vptl(i)
c....Apply essential boundary conditions
      if(nbc11.eq.0) go to 40
20
        big=1.0e100
      do 30 i=1.nbc11
        nn=ndbc11(i)
        gf(nn)=big*vbc11(i)
        gk(nn,1)=big
30
      continue
c....Apply natural boundary conditions
40
      if(nbc12.eq.0) go to 70
      do 60 itt=1,nbe12
        nel=neb12(itt)
        ns=nsde12(itt)
        nee=ne(nel)
c....Pick out nodal coordinates
     do 50 j=1,nee
        nod(j)=node(j,nel)
        nj=nod(j)
        xx(1,j)=x12(1,nj)
        xx(2,j)=x12(2,nj)
        uu(j)=u12(nj)
50
      continue
      call bcint(vbc12(1,itt),vbc12(2,itt),xx,pe,game,nee,ns,uu)
      call assmb(pe,game,nee,nod)
úΟ
      continue
70
      return
      end
      subroutine aply2
      level 2, gk,gf,nz
      common/cntrl1/ nopt1,nopt2,nopt3
      common/cntrl2/ time.tmax.delt.thet.nprint.niter.tolen.icount.nwrt
      common/ccon2/ nnode2,nelem2,npot2,nbc21,nbc22
      common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
      common/cbc2/ ndbc21(25),vbc21(25),neo22(20),nsde22(20),vbc22(2,20)
                   ,npt2(10),vpt2(10)
      common/celem/ ne(1000),node(9,1000)
      common/cnode/ x12(2,1200),u12(1200)
      common/cmatrx/ gk(1200,25),gf(1200),nz
      common/ccon1/ nnode1, nelem1, npot1, nbc11, nbc12
      dimension pe(9, 9), game(9), xx(2, 9), nod(9), uu(9)
С
c....Apply point loads
      if(npot2.eq.0) go to 20
      do 10 i=1, npot2
        n=npt2(i)
```

```
c.... Apply essential boundary conditions
```

```
if(nbc21.eq.0) go to 40
20
      big=1.0e100
      do 30 i=1.nbc21
        nn=ndbc21(i)
        gf(nn)=big*vbc21(i)
        gk(nn,1) = big
30
      continue
c....Apply natural boundary conditions
      if(nbc22.eq.0) go to 70
40
      do 60 itt=1,nbc22
        nel=neb22(itt)+nelem1
        ns=nsde2∠(itt)
        nee=ne(nel)
c....Pick out nodal coordinates
      do p0 j=1,nee
        nod(j)=node(j,nel)
        nj=noa(j)+nnode1
        xx(1,j)=x12(1,nj)
        xx(z,j) = x12(z,nj)
        uu(j)=u12(nj)
50
      continue
      call bcint(vbc22(1,itt),vbc22(2,itt),xx,pe,game,nee,ns,uu)
      call assmb(pe,game,nee,nod)
bÜ
      continue
70
      return
      end
      subroutine aply3
      level 2, gk,gf,nz
      common/entrl1/ nopt1,nopt2,nopt3
      common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/ccon3/ nnode3,nelem3,npot3,nbc31,nbc32
      common/cint/ xint1(4), wint1(4), xint2(16,2), wint2(16), none, ntwo
      common/cbc3/ ndbc31(50),vbc31(50),neb32(40),nsde32(40),vbc32(2,40)
                   ,npt3(10),vpt3(10)
      common/celem/ ne(1000), node(9,1000)
      common/cnode3/ x3(2,1200),u3(1200)
      common/cmatrx/ gk(1200,25),gf(1200),nz
      dimension pe(9,9), game(9), xx(2,9), nod(9), uu(9)
С
c.... Apply point loads
      if(npot3.eq.0) go to 20
      do 10 i=1, npot3
        n=npt3(i)
10
        gf(n) = gf(n) + vpt3(i)
c....Apply essential boundary conditions
20
      if(nbc31.eq.0) go to 40
      big=1.0e100
      do 30 i=1,nbc31
        nn=ndbc31(i)
        gf(nn)=big*vbc31(i)
        gk(nn,1)=big
30
      continue
```

.

gf(n)=gf(n)+vpt2(i)

```
c....Apply natural boundary conditions
40
      if(nbc32.eq.0) go to 70
      do 60 itt=1,nbc32
        nel=neb32(itt)
        ns=nsde32(itt)
        nee=ne(nel)
c....Pick out nodal coordinates
      do 50 j=1,nee
        nod(j)=node(j,nel)
        nj=nod(j)
        xx(1,j)=x3(1,nj)
        xx(2,j)=x_{3}(2,nj)
        uu(j)=u3(nj)
50
      continue
      call dcint(vbc32(1,itt),vbc32(2,itt),xx,pe,game,nee,ns,uu)
      call assmb(pe,game,nee,nod)
60
      continue
70
      return
      end
      subroutine assmb(ek,ef,nee,nodd)
С
c....To assemble the global matrix from every element contribution
c....Valid only symmetry matrix
С
      level 2, gk,gf,nz
      common/cmatrx/ gk(1200,25),gf(1200),nz
      dimension ek(9,9), ef(9), nodd(9)
      do 10 ii=1,nee
        ig=nodd(ii)
        gf(ig)=gf(ig)+ef(ii)
        do 10 jj=1,nee
          jg=nodd(jj)-ig+1
          if(jg.le.0) go to 10
          gk(ig,jg)=gk(ig,jg)+ek(ii,jj)
10
        continue
      return
      end
      subroutine assmb4(ek,ef,nee,nodd,gk,gf)
С
c....To assemble the global matrix from every element contribution
c....Valid only symmetry matrix
Ĉ
      dimension ek(3,3),ef(3),nodd(9),gk(25,3),gf(25)
      do 10 ii=1,nee
        ig=nodd(ii)
        gf(ig)=gf(ig)+ef(ii)
        do 10 jj=1,nee
          jg=nodd(jj)-ig+1
          if(jg.le.0) go to 10
          gk(ig,jg)=gk(ig,jg)+ek(ii,jj)
10
        continue
      return
      end
      subroutine bcint(p,v,x,pe,game,nee,ns,uu)
```

```
common/cint/ xint1(4), wint1(4), xint2(16,2), wint2(16), none, ntwo
common/cntrl1/ nopt1,nopt2,nopt3
common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
dimension game(9),pe(9,9),x(2,9),xy(2),dxds(2,2)
dimension psi(9), dpsi(9,2)
dimension uu(9), as(9, 9), ab(9), save(9)
do 10 i=1.nee .
 game(i)=0.0
do 10 j=1,nee
   pe(i,j)=0.0
continue
do 70 loop=1, none
if(ns.eq.1) then
xy(2) = -1.0
xy(1) = xint(loop)
else if (ns.eq.2) then
xy(1) = 1.0
xy(2) = xint1(loop)
else if(ns.eq.3) then
xy(2) = 1.0
xy(1) = xint1(loop)
else if(ns.eq.4) then
xy(1) = -1.0
xy(2) = xint1(loop)
eise
write(6,200)
format(2x,#Out of range in bcint.f#,)
end if
call shape(xy,nee,psi,dpsi)
do 20 i=1,2
  do 20 j=1,2
    dxds(i,j)=0.0
     do 20 k=1,nee
       dxds(i,j)=dxds(i,j)+dpsi(k,j)*x(i,k)
continue
if(nopt1.eq.1) then
rx=0.0
do 22 i=1,nee
  rx=rx+x(1,i)*psi(i)
else
rx=1.0
end if
if(ns.eq.1.or.ns.eq.3) then
    valu=dxds(1,1)**2+dxds(2,1)**2
else
   valu=dxds(1,2)**2+dxds(2,2)**2
end if
vjaco=sqrt(valu)
fac=vjaco*wint1(loop)*rx
  do 40 ia=1,nee
    ab(ia)=v*psi(ia)
    do 40 ib=1,nee
       aa(ia,ib)=p*psi(ia)*psi(ib)
  do 45 ic=1.nee
```

\_ ÷

200

20

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save(ic)=0.0do 45 id=1,nee save(ic)=save(ic)+(thet-1.0)\*aa(ic,id)\*uu(id) 45 do 46 im=1,nee game(im) = game(im) + (ab(im) + save(im)) \* fac do 46 in=1,nee pe(im,in)=pe(im,in)+thet\*aa(im,in)\*fac 46 continue 70 continue return end subroutine comn common/cint/ xint1(4), wint1(4), xint2(16,2), wint2(16), none, ntwo call oneint(none) call twoint(ntwo) return end subroutine data common/ontrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt common/entrl1/ nopt1,nopt2,nopt3 common/cdoman/ xdim,ydim,prop(4,4) \_\_common/cint/\_xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo С c....nopt1=1 is axisymmetry, otherwise plane 2-D c...nopt2=1 is transient, otherwise steady state c....nopt3=1 is linear, otherwise nonlinear c...time time for transient calculations c...tmax max. time in calculation time step increment c....delt c....tnet theta method in time integration scheme Euler Backward Method C... tnet=1 thet=2/3 Galerkin Method c . . . . thet=1/2 Midpoint or Crank-Nicolson Method c.... с.... thet=0 Euler Forward Method c...nprint no of steps for print out no of iterations for nonlinear case c....niter c....tolen allowed tolerance in iteration С read(5,\*) nopt1,nopt2,nopt3 if (nopt1.eq.1) then write(6, 100)100 format(1h1,////2x,#THIS IS A 3-D AXISYMMETRY PROBLEM#,/) else write(6, 110)format(/2x, #THIS IS A 2-D PLANE PROBLEM#./) 110 end if if(nopt2.eq.1) then write(6, 120)format(/2x,#THIS IS A TRANSIENT PROBLEM#,/) 120 else write(6.130)130 format(/2x, #THIS IS A STEADY STATE PROBLEM#,/) end if if(nopt3.eq.1) then

```
write(6, 140)
140
        format(/2x, #THIS IS A LINEAR PROBLEM#, /)
      else
        write(6.150)
150
        format(/2x, #THIS IS A NONLINAR PROBLEM#,/)
      end if
      read(5,*) nprint,niter,nwrt
      write(6,160) nprint
160
      format(/2x, #THE NO. OF STEPS FOR PRINT OUT RESULTS IS#, 15/)
      write(6, 170) niter
170
      format(/2x, #THE NO. OF ITERATIONS IS#, 15/)
      read(5,*) delt,tmax,thet,tolen
      write(6,180) celt.tmax.thet
      format(/2x, #THE TIME INCREMENT IS#, e15.5, /2x, #THE MAX. TIME IS#,
180
     .e15.5,/2x,#THE THETA IS#,e15.5,/)
      read(5, *) xdim,ydim
      write(6,190) xdim.ydim
190
      format(/2x, #THE GLOBAL X-DIMENSION IS#, e15.5, /2x, #THE GLOBAL#,
     .# Y-DIMENSION IS#,e15.5,/)
      read(5,*) none,ntwo
      write(6,200) none, ntwo
200
      format(/2x, #THE NO. OF 1-D INTEGRATION PT. IS#, 15,
     ./2x, #THE NO. OF 2-D INTEGRATION PT. IS#, 15, /)
    do 10 i=1,4
      read(5, *) (prop(i,j), j=1, 4)
10
      write(6,210) i,(prop(i,j),j=1,4)
210
      format(/2x, #THE FOLLOWINGS ARE THE DEFAULT COEFFICIENTS IN#,
     .# REGION#,15,//2x,4e15.5,/)
      return
      end
      subroutine divdi4(x.n.d)
c....To get divided differences for interpolation
c....d and x are vectors with entries f(x(i)) and x(i),
c....i=1,...,n respectively. On exit d(i) will contain
c...f[x(1),...,x(i)]
      dimension x(5), d(5)
      do 20 i=1,n-1
        j=n
10
        d(j)=(d(j)-d(j-1))/(x(j)-x(j-i))
        j=j-1
        if(j.ge.(i+1)) go to 10
20
      continue
      return
      end
      subroutine elem(x,n,ek,ef,matt,uu)
      common/entrl1/ nopt1,nopt2,nopt3
      common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
      common/cdoman/ xdim,ydim,prop(4,4)
      dimension ck(9,9),cc(9,9),xy(2),ek(9,9),ef(9)
      dimension x(2,9),psi(9),apsi(9,2)
      dimension dpsix(9),dpsiy(9),dxds(2,2),dsdx(2,2)
      dimension ct(9,9),uu(9),save(9)
```

С

do 10 i=1,n ef(i)=0.0 do 10 j=1,n ek(i,j)=0.010 continue call getmat(x,n,uu,aa,ab,ac,ad,matt) do 70 loop=1.ntwo xy(1) = xint2(loop, 1)xy(2) = xint2(loop, 2) call shape(xy,n,psi,dpsi) do 20 i=1,2 do 20 j=1,2 dxds(i,j)=0.0do 20 k=1.n dxds(i,j)=dxds(i,j)+dpsi(k,j)\*x(i,k) 20 continue if(nopt1.eq.1) then rx = 0.0do 25 ii=1.n 25 rx=rx+x(1,ii)\*psi(ii)else rx=1.0 end if detj=dxds(1,1)\*dxds(2,2)-dxds(1,2)\*dxds(2,1) if(detj.le.0.0) go to 99 dsdx(1,1)=dxds(2,2)/detjdsdx(2,2) = dxds(1,1)/detjdsdx(1,2) = -dxds(1,2)/detjdsdx(2,1) = -dxds(2,1)/detjdo 30 i=1,n dpsix(i)=dpsi(i,1)\*dsdx(1,1)+dpsi(i,2)\*dsdx(2,1) dpsiy(i) = dpsi(i,1) \* dsdx(1,2) + dpsi(i,2) \* dsdx(2,2)30 continue fac=detj\*wint2(loop)\*rx do 40 i=1,n do 40 j=1,n ck(i,j)=aa\*(dpsix(i)\*dpsix(j)+dpsiy(i)\*dpsiy(j)) cc(i,j)=ab/delt\*psi(i)\*psi(j) ct(i,j) = (tnet-1.0) \* ck(i,j) + cc(i,j)40 continue do 45 i=1,n save(i)=0.0 do 45 k=1.n 45 save(i)=save(i)+ct(i,k)\*uu(k) do 50 i=1.n ef(i)=ef(i)+fac\*save(i) do 50 j=1,n ek(i,j)=ek(i,j)+(thet\*ck(i,j)+ce(i,j))\*fac50 continue 70 continue return 99 write(6, 100)100 format(2x,#Bad Jacobian Matrx#,/) stop

## end

1

5

```
subroutine elem4(xx,nee,ek,ef,tmatt,nn,mm,nnode1)
      common/entrl1/ nopt1,nopt2,nopt3
      common/cint/ xint1(4), wint1(4), xint2(16,2), wint2(16), none, ntwo
      common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/cdoman/ xdim,ydim,prop(4,4)
      common/cnode/ x12(2,1200),u12(1200)
      common/celem/ ne(1000), node(9,1000)
      dimension xx(2,9), save(2), ek(3,3), ef(3), dtdx(2)
      dimension xy(2),psi(9),dpsi(9,2),uu(9),xx12(2,9),nod12(9)
      do 1 ii=1.nee
        ef(ii)=0.0
        do 1 jj=1,nee
         ek(ii,jj)=0.0
      if(nopt1.eq.1) then
        rx=0.50*(xx(1,1)+xx(1,2))
      else
        rx=1.0
      end if
      a = (xx(1,1) - xx(1,2)) **2
      b = (xx(2,1) - xx(2,2)) **2
c....Only valid for 4-node element
      detj=0.50*sqrt(a+b)
c....To get two directional cosine
      slop=(xx(2,2)-xx(2,1))/(xx(1,2)-xx(1,1))
      if(slop.le.1.0e-7) then
        theta=asin(1.0)
      else
        tneta=stan(-1.0/slop)
      end if
      dxdn=cos(theta)
      dydn=sin(theta)
      fac=rx*deti
      bx=tmatt/delt
      do 40 in=1, none
        xy(2) = 1.0
        xy(1) = xintl(in)
        call shape(xy,4,psi,dpsi)
        nee1=ne(nn)
        tmatl=prop(1,1)
        do 5 ja=1, neel
          nod12(ja)=node(ja,nn)
          nj=nod12(ja)
          xx12(1,ja)=x12(1,nj)
          xxi2(2,ja)=x12(2,nj)
          uu(ja)=u12(nj)
        continue
        call_flux4(xx12,nee1,tmat1,uu,dpsi,dtdx)
        do 6 mc = 1.2
        save(mc)=dtdx(mc)
        neel=ne(mm)
        tmat1=prop(2,1)
        do 10 ja=1, nee1
          nod12(ja)=node(ja,mm)
```

nj=nod12(ja)+nnode1 xx12(1,ja)=x12(1,nj)xx12(2,ja) = x12(2,nj)uu(ja)=u12(nj)10 continue call flux4(xx12,nee1,tmat1,uu,dpsi,dtdx) do 11 na=1,2 11 save(na)=save(na)-dtdx(na) coef=dxdn\*save(1)+dydn\*save(2) do 20 ii=1,nee c....Node 3 and 4 j=ii+2ef(ii)=ef(ii)+coef\*fac\*psi(j)\*wint1(in) do 20 k=1.nee c... Node 3 and 4 m=k+2ek(ii,k)=ek(ii,k)+bx\*fac\*psi(j)\*psi(m)\*wint1(in) 20 continue 40 continue return end subroutine flux4(xx,nee,tmatt,uu,dpsi,dtdx) common/cint/ xint1(4), wint1(4), xint2(16,2), wint2(16), none, ntwo dimension dpsix(9), dpsiy(9), xx(2,9), uu(9), dtdx(2)dimension dpsi(9,2),dxds(2,2),dsdx(2,2) do 1 if=1,2 1 atax(if)=0.0do 10 ii=1,2 do 10 im=1,2 dxds(ii,im)=0.0 do 10 if=1.nee dxds(ii,im)=dxds(ii,im)+dpsi(if,im)\*xx(ii,if) 10 continue det = dxds(1, 1) \* dxds(2, 2) - dxds(1, 2) \* dxds(2, 1)if(det.le.0.0) go to 1000 dsdx(1,1) = dxds(2,2)/detasdx(2,2) = dxds(1,1)/detdsdx(1,2) = -dxds(1,2)/detdsdx(2,1) = -dxds(2,1)/detdo 20 io=1,nee dpsix(io) = apsi(io,1) \* dsdx(1,1) + dpsi(io,2) \* dsdx(2,1)dpsiy(io)=dpsi(10,1)\*dsdx(1,2)+dpsi(io,2)\*dsdx(2,2) 20 continue do 30 jp=1,nee dtdx(1)=dtdx(1)+dpsix(jp)\*uu(jp)\*tmatt dtdx(2) = dtdx(2) + dpsiy(jp) \* uu(jp) \* tmatt30 continue return 1000 write(6,100) 100 format(/2x, #BAD JACOBIAN MATRIX IN FLUX4.F#,) stop end subroutine form1 level 2, gk,gf,nz

```
common/cntrll/ nopt1,nopt2,nopt3
common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
common/cint/ xint1(4),wint1(4),xint2(10,2),wint2(10),none,ntwo
common/cdoman/ xdim,ydim,prop(4,4)
common/ccon1/ nnode1, nelem1, npot1, nbc11, nbc12
common/celem/ ne(1000), node(9,1000)
common/cnode/ x12(2,1200),u12(1200)
common/cmatrx/ gk(1200,25),gf(1200),nz
dimension xx(2,9), uu(9), nodd(9), ek(9,9), ef(9)
io 10 i=1,nnode1
  gf(i)=0.0
  do 10 j=1,25
    gk(i,j)=0.0
continue
do 40 ielm=1,nelem1
  nec=ne(ielm)
  matt=nz
  do 30 j=1,nee
    noaa(j)=node(j,ielm)
    nj=noda(j)
    xx(1,j)=x12(1,nj)
    xx(2,j)=x12(2,nj)
    uu(j) = u12(nj)
  continue
  call elem(xx, nee, ek, ef, matt, uu)
  call assmb(ek,ef,nee,noda)
continue
return
end
suproutine form2
level 2, gk,gf,nz
common/entrl1/ nopt1,nopt2,nopt3
common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
common/cdoman/ xdim,ydim,prop(4,4)
common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
common/ccon2/ nnode2,nelem2,npot2,nbc21,nbc22
common/celem/ ne(1000),node(9,1000)
common/cnode/ x12(2,1200),u12(1200)
common/cmatrx/ gk(1200,25),gf(1200),nz
dimension xx(2,9),uu(9),nodd(9),ek(9,9),ef(9)
do 10 i=1,nnode2
  gf(i) = 0.0
  do 10 j=1,25
    gk(i,j)=0.0
continue
do 4J ielm=1,nelem2
  iielm=ielm+nelem1
  nee=ne(iielm)
  matt=nz
  do 30 j=1,nee
    nodd(j)=node(j,iielm)
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nj=nodd(j)+nnode1

xx(1,j)=x12(1,nj)xx(2,j)=x12(2,nj)uu(j)=u12(nj)continue call elem(xx,nee,ek,ef,matt,uu) call assmb(ek,ef,nee,nodd) continue return end subroutine form3 level 2, gk,gf,nz common/entrl1/ nopt1,nopt2,nopt3 common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt common/cint/ xint1(4), wint1(4), xint2(16,2), wint2(16), none, ntwo common/cdoman/ xdim,ydim,prop(4,4) common/ccon3/ nnode3, nelem3, npot3, nbc31, nbc32 common/celem/ ne(1000), node(9,1000) common/cnode3/ x3(2,1200),u3(1200) common/ematrx/ gk(1200,25),gf(1200),nz dimension xx(2,9), uu(9), nodd(9), ek(9,9), ef(9)do 10 i=1,nnode3 gf(1)=0.0do 10 j=1,25 gk(i,j)=0.0continue do 40 ielm=1.nelem3 nee=ne(ielm) matt=nz do 30 j=1, nee nodd(j)=node(j,ielm) nj=nodd(j) xx(1,j)=x3(1,nj)xx(2,j)=x3(2,nj)uu(j)=u3(nj)continue call elem(xx, nee, ek, ef, matt, uu) call assmb(ek,ef,nee,nodd) continue return end subroutine form4 common/cinter/xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2), angl(25), capa(2,25), tin(25), ratio(50) common/celem4/ ne4(25),nodes4(3,25) common/celem/ ne(1000), node(9,1000) common/cnode/ x12(2,1200),u12(1200) common/cmatr4/ gk4(25,3),gf4(25),vnor(25) common/entrl1/ nopt1, nopt2, nopt3 common/cntrl2/ time.tmax.delt.thet.nprint,niter,tolen,icount,nwrt common/cint/ xint1(4), wint1(4), xint2(16,2), wint2(16), none, ntwo common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12 common/cdoman/ xdim.ydim.prop(4.4)

```
dimension nodd(9),xx(2,9),ek(3,3),ef(3)
      do 10 i=1, max124
        gf4(i)=0.0
c....Matrix band width is 2 here, one-D two-node symmetry
        do 10 j=1,2
          gk4(i,j)=0.0
10
      continue
c....Get first element No. along the interface in region 1 & 2
      nn=nfirst(1)
      mm=nfirst(2)+nelem1
c....Do the assembly processes
      do c0 ito=1, max124-1
      nee=ne4(ito)
      tmatt=prop(4,3)
      do 40 j=1,nee
        nodd(j)=nodes4(j,ito)
        nj=nodd(j)
        xx(1,j)=xi124(1,nj)
        xx(2,j)=xi124(2,nj)
40
      continue
      call elem4(xx,nee,ek,ef,tmatt,nn,mm,nnode1)
      call assmb4(ek,ef,nee,nodd,gk4,gf4)
      nn=nn+1
      mm = mm + 1
60.
     continue
      return
      end
      subroutine getmat(x,n,u,aa,ab,ac,ad,matt)
c....To compute the coefficients in the differential equations
c.... at any specific point and time, in other words, coefficients
c.... can be function of time, temperature(concentration), and
c.... space. Used in nonlinear differential equations
      common/cntrl1/ nopt1,nopt2,nopt3
      common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
      common/cdoman/ xdim,ydim,prop(4,4)
      dimension x(2,9), u(9)
      aa=prop(matt.1)
      ab=prop(matt,2)
      ac=prop(matt.3)
      ad=prop(matt.4)
     return
      end
      subroutine gett
      common/cnode3/ x3(2.1200), u3(1200)
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                    ,angl(25),capa(2,25),tin(25),ratio(50)
      common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/ccon3/ nnode3, nelem3, npot3, nbc31, nbc32
      dimension cc(2,95)
      cc(1,1)=0.0
      cc(2,1)=0.0
      cc(1,2)=17.0
      cc(2,2) = -0.0620
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cc(1,3)=34.0cc(2,3) = -0.1210cc(1, 4) = 51.0cc(2,4) = -0.1810cc(1,5)=69.0cc(2,5) = -0.240cc(1,6) = 86.0cc(2,6) = -0.2990cc(1,7) = 103.0cc(2,7) = -0.3580 $cc(1, \delta) = 120.0$ cc(2,8) = -0.4170cc(1, 9) = 137.0cc(2, 9) = -0.4750cc(1, 10) = 155.0cc(2,10) = -0.5340cc(1,11)=172.0cc(2,11) = -0.5930cc(1, 12) = 189.0cc(2,12) = -0.6520cc(1, 13) = 207.0cc(2,13) = -0.7110cc(1, 14) = 224.0cc(2, 14) = -0.770cc(1, 15) = 241.0cc(2, 15) = -0.8290cc(1, 16) = 259.0ee(2, 16) = -0.8880cc(1, 17) = 276.0cc(2, 17) = -0.9480cc(1, 18) = 294.0cc(2,18) = -1.0070cc(1, 19) = 311.0cc(2, 19) = -1.0670cc(1,20) = 329.0cc(2,20) = -1.1260cc(1,21) = 346.0cc(2,21) = -1.1860cc(1,22) = 364.0cc(2,22) = -1.2460cc(1,23) = 382.0cc(2,23) = -1.3060cc(1, 24) = 399.0cc(2,24) = -1.3660cc(1,25) = 418.0cc(2,25) = -1.4260cc(1, 26) = 435.0cc(2,26) = -1.4860cc(1,27) = 452.0cc(2,27) = -1.5470cc(1, 28) = 470.0cc(2,28) = -1.6070cc(1,29) = 488.0cc(2,29) = -1.6680

cc(1,30) = 505.0cc(2,30) = -1.7290cc(1,31) = 523.0cc(2,31) = -1.790cc(1, 32) = 541.0cc(2, 32) = -1.8510cc(1,33)=559.0 cc(2,33) = -1.9130cc(1, 34) = 577.0cc(2, 34) = -1.9740cc(1, 35) = 595.0cc(2,35) = -2.0360cc(1,36)=613.0cc(2, 36) = -2.0980cc(1, 37) = 631.0cc(2,37) = -2.1600cc(1, 38) = 649.0cc(2, 38) = -2.2220cc(1,39) = 667.0cc(2,39) = -2.2840cc(1,40)=685.0 cc(2, 40) = -2.3470cc(1, 41) = 703.0cc(2, 41) = -2.4090cc(1, 42) = 721.0cc(2, 42) = -2.4720cc(1, 43) = 739.0cc(2, 43) = -2.5350cc(1, 44) = 757.0cc(2, 44) = -2.5980cc(1, 45) = 775.0cc(2, 45) = -2.6620cc(1, 46) = 794.0cc(2, 46) = -2.7250cc(1, 47) = 812.0cc(2, 47) = -2.7890cc(1.48) = 830.0cc(2, 48) = -2.8530cc(1, 49) = 848.0cc(2, 49) = -2.9170cc(1,50) = 866.0cc(2,50) = -2.9820cc(1,51) = 865.0cc(2,51) = -3.0460cc(1.52) = 921.0cc(2,52) = -3.1760cc(1,53) = 958.0cc(2,53) = -3.3070cc(1,54) = 995.0cc(2,54) = -3.4350cc(1,55)=1032.0 cc(2,55) = -3.5700cc(1,56) = 1069.0cc(2,56) = -3.7030

	cc(1.57) = 1106.0				
	cc(2,57) = -3,8370				
	$cc(2, j_1) = -j_1 c c_1 c_2$				
	$a_0(2, 58) = 2.0720$				
	cc(2, 50) = -5.5(20)				
	22(1,59) = 1101.0				
	CC(2,59) = -4.1070				
	cc(1, 60) = 1218.0				: .
	cc(2, 60) = -4.2440				
	cc(1, o1) = 1256.0				
	cc(2,61) = -4.3780				
	cc(1, 62) = 1294.0				
	cc(2,62) = -4.5160		· .	•	
	cc(1,63)=1331.0				
	cc(2, 63) = -4.6550				
	cc(1, 64) = 1369.0			•	
	cc(2, 64) = -4.7950				
	cc(1,65) = 1407.0			а.	
	cc(2.65) = -4.9370				
	cc(1,66) = 1445.0			•	
	ee(2,66) = -5,0790				
	cc(1,67) = 1484.0				
	cc(2, 67) = -5, 2220				
	co(1, 68) = 1522 0				
	cc(2, 68) = -5, 3670				
	cc(2,00) = -5.50				
	cc(1,0y) = 1000.0				
	cc(2,0) = -5.5 + 20		· · ·		
	cc(2,70) = -5,6590			•	
	cc(1,71) = 1637			ан сайтаан ал ал ан	
	cc(1, 1) = 5.8070			•	
	cc(2,77) = -9.0070				•
	cc(1, 12) = 1010.0				
	cc(2, 72) = -5.9500				
	CC(1, 73) = 1715.0				
	cc(2,73) = -0.1000				
	CC(1, 74) = 1754.0				
	CC(2, 74) = -0.2530				
	CC(1,/5)=1/91.0				
	cc(2,75) = -6.4100				
	cc(2,75) = -6.4100 cc(1,76) = 1832.0				
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640				
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(1,77) = 1930.0				
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(1,77) = 1930.0 cc(2,77) = -6.9540				
• •••• •	cc(2,75) = -6.4100 cc(1,76) = 1632.0 cc(2,76) = -6.5640 cc(1,77) = 1930.0 cc(2,77) = -6.9540 cc(1,78) = 2029.0		· · · · · · · · · · · · · · · · · · ·		
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(1,77) = 1930.0 cc(2,77) = -6.9540 cc(1,78) = 2029.0 cc(2,78) = -7.3530		· · · · · · · · · · · · · · · · · · ·		
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(1,77) = 1930.0 cc(2,77) = -6.9540 cc(1,78) = 2029.0 cc(2,78) = -7.3530 cc(1,79) = 2129.0		• • • • • • • • •	· · · · · · · · · · · · · · · · · · ·	
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(1,77) = 1930.0 cc(2,77) = -6.9540 cc(1,78) = 2029.0 cc(2,78) = -7.3530 cc(1,79) = 2129.0 cc(2,79) = -7.760		· · · · · · · · · · · · · · · · · · ·	· · · · · · · · · · · · · · · · · · ·	
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(1,77) = 1930.0 cc(2,77) = -6.9540 cc(1,78) = 2029.0 cc(2,78) = -7.3530 cc(1,79) = 2129.0 cc(2,79) = -7.760 cc(1,80) = 2229.0		· · · · · · · · · · · · · · · · · · ·	· · · · · · · · · · · · · · · · · · ·	
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -0.5640 cc(1,77) = 1930.0 cc(2,77) = -6.9540 cc(1,78) = 2029.0 cc(2,78) = -7.3530 cc(1,79) = 2129.0 cc(2,79) = -7.760 cc(1,80) = 2229.0 cc(2,80) = -8.1760			-	
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(1,77) = 1930.0 cc(2,77) = -6.9540 cc(1,78) = 2029.0 cc(2,78) = -7.3530 cc(1,79) = 2129.0 cc(2,79) = -7.760 cc(2,80) = -8.1760 cc(1,81) = 2330.0		· · · · · · · · · · · · · · · · · · ·	-	
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(1,77) = 1930.0 cc(2,77) = -6.9540 cc(1,78) = 2029.0 cc(2,78) = -7.3530 cc(1,79) = 2129.0 cc(2,79) = -7.760 cc(2,80) = -8.1760 cc(1,81) = 2330.0 cc(2,81) = -8.0020		· · · · · · · · · · · · · · · · · · ·	-	
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(2,77) = -6.9540 cc(2,77) = -6.9540 cc(1,78) = 2029.0 cc(2,78) = -7.3530 cc(1,79) = 2129.0 cc(2,79) = -7.760 cc(2,80) = -8.1760 cc(1,81) = 2330.0 cc(2,81) = -8.0020 cc(1,82) = 2432.0	* p	· · · · · · · · · · · · · · · · · · ·		
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(2,77) = -6.9540 cc(2,77) = -6.9540 cc(1,78) = 2029.0 cc(2,78) = -7.3530 cc(1,79) = 2129.0 cc(2,79) = -7.760 cc(2,80) = -8.1760 cc(1,81) = 2330.0 cc(2,81) = -8.0020 cc(2,82) = -9.0380				
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(2,77) = -6.9540 cc(1,77) = -6.9540 cc(1,78) = 2029.0 cc(2,78) = -7.3530 cc(1,79) = 2129.0 cc(2,79) = -7.760 cc(2,79) = -7.760 cc(1,80) = 2229.0 cc(2,80) = -8.1760 cc(1,81) = 2330.0 cc(2,81) = -8.0020 cc(2,81) = -8.0020 cc(1,82) = 2432.0 cc(2,82) = -9.0380 cc(1,83) = 2534.0		· · · · · · · · · · · · · · · · · · ·		· · ·
	cc(2,75) = -6.4100 cc(1,76) = 1832.0 cc(2,76) = -6.5640 cc(2,77) = -6.9540 cc(1,77) = -6.9540 cc(1,78) = 2029.0 cc(2,78) = -7.3530 cc(1,79) = 2129.0 cc(2,79) = -7.760 cc(2,79) = -7.760 cc(1,80) = 2229.0 cc(2,80) = -8.1760 cc(1,81) = 2330.0 cc(2,81) = -8.0020 cc(2,82) = -9.0380 cc(1,83) = 2534.0 cc(2,83) = -9.4840				· · ·

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cc(1,84) = 2637.0
      cc(2,84) = -9.940
      cc(1,85) = 2741.0
      cc(2, 85) = -10.4080
      cc(1,86) = 2845.0
      cc(2,86) = -10.8380
      cc(1,87) = 3056.0
      cc(2, \delta7) = -11.8850
      cc(1,88) = 3270.0
      cc(2,68) =-12.935
      cc(1,89) = 3486.0
      cc(2, 69) = -14.0440
      cc(1, 90) = 3706.0
      cc(2,90) = -15.2160
      cc(1,91) = 3923.0
      cc(2,91) = -16.4530
      cc(1, 92) = 4153.0
      cc(2,92) = -17.776
      cc(1,93) = 4382.0
      cc(2,93) = -19.1760
      cc(1, 54) = 4613.0
      cs(2,94) = -20.6670
      cc(1,95) = 4800.0
      cc(2,95) = -21.50
      is=nnode3-nmax3+1
factor 1=0.1e2
      factor2=0.1e3
      iaa=1000
      iaaa=2000
      ibb=2000
      ibbb=20000
      iperiod=100
C . . . . . . . . . .
                       . . . . . . . . . . .
      p_{1}=acos(-1.0)
      if (icount.ge.iaa.and.icount.le.iaaa) then
         k=1
         do 5 i=is,nnode3
           degree=xi3(1,k)*pi/xi3(1,nmax3)
           u3(i)=u3(i)+factor1*cos(degree)
           k = k + 1
5
         continue
      end if
      if(icount.ge.ibb.and.icount.le.ibbb) then
         igap=(icount-ibb)/iperiod
         itest=igap/2*2
         ivalue=igap*iperiod+ibb
         if (itest.eq.igap) then
           degree=(icount-ivalue)*pi/iperiod
           do 7 ii=is,nnode3
             u3(ii)=u3(ii)+factor2*cos(degree)
1
           continue
         else
           degree=(icount-ivalue)*pi/iperiod
```

```
do 9 k=is,nnode3
            u3(k)=u3(k)-factor2*cos(degree)
9
          continue
        end if
      end if
      j=1
      do 50 i=is.nnode3.2
        n1=1
        n2=95
        if(u_3(i).lt.ce(1,1).or.u_3(i).gt.ce(1,95)) go to 1000
10
        nhalf=(n1+n2)/2
        if (nhalf.eq.n1.or.nhalf.eq.n2) go to 20
        if(u_3(i).eq.cc(1,nhalf)) then
          tin(j)=cc(2,nhalf)
          j = j + 1
          go to 50
        else if(u3(i).gt.cc(1,nhalf)) then
          n1=nhalf
          go to 10
        else if(u3(i).lt.cc(1,nhalf)) then
          n2=nhalf
          go to 10
        end if
20
        rato=(cc(2,n1)-cc(2,n2))/(cc(1,n1)-cc(1,n2))
        tin(j)=cc(2,n1)+rato*(u3(i)-cc(1,n1))
        j = j + 1
50
      continue
      return
1000
      write(6.500)
500
      format(2x,#Out of range in gett.f#,/)
      stop
      end
      subroutine init
      common/cnode/ x12(2,1200),u12(1200)
      common/celem/ ne(1000), node(9,1000)
      common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
      common/cbc1/ ndbc11(25),vbc11(25),neb12(20),nsde12(20),vbc12(2,20)
                   ,npt1(10),vpt1(10)
      common/cscal1/ rlen1.rval1
      common/ccon2/ nnode2,nelem2,npot2,nbc21,nbc22
      common/cbc2/ ndbc21(25),vbc21(25),neb22(20),nsde22(20),vbc22(2,20)
                   ,npt2(10),vpt2(10)
      common/cscal2/ rlen2.rval2
      common/cconj/ nnode3.nelem3.npot3.nbc31.nbc32
      common/cnode3/ x3(2,1200),u3(1200)
      common/cbc3/ ndbc31(50),vbc31(50),neb32(40),nsde32(40),vbc32(2,40)
                   .npt3(10),vpt3(10)
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                     ,angl(25),capa(2,25),tin(25),ratio(50)
      common/celem4/ ne4(25),nodes4(3,25)
      common/cmatr4/ gk4(25,3),gf4(25),vnor(25)
      common/cooman/ xdim,ydim,prop(4,4)
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```

call init4

```
call init1
      call init2
      call init3
      return
      end
      subroutine init1
      common/cinter/ xi3(2,50), nmax3, xi124(2,25), max124, nfirst(2)
                     ,angl(25),capa(2,25),tin(25),ratio(50)
      common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
      common/cscal1/ rlen1,rval1
      common/cnode/ x12(2,1200),u12(1200)
      common/celem/ ne(1000), node(9,1000)
      common/cbc1/ ndbc11(25),vbc11(25),neb12(20),nsde12(20),vbc12(2,20)
                   ,npt1(10),vpt1(10)
      common/cdoman/ xdim,ydim,prop(4,4)
      write(6,1000)
      format(//2x, #**** THE FOLLOWING IS THE INITIAL INFORMATION#.
1000
     .# OF REGION 1 ****#./)
      read(5,*) rlen1,rval1
      write(6.100) rlen1.rval1
100
      format(/2x, #THE REFERENCE LENGTH FOR MESHING REGION 1 IS#, e15.5,
     ./2x,#THE REFERENCE VALUE FOR MESHING REGION 1 IS#,e15.5,)
      read(5,*) npot1,nbc12
      nbc11=2*max124
      read(5,*) t0,t1
      do 10 i=1, max124
        ndbcll(i)=i
        u12(i)=t0
        vbc11(i)=t0
        tin(i)=t1
        j=i+max1_24
        ndbc11(j)=j
        vbc11(j)=t1
        u12(j)=t1
10
      continue
      do 20 i=1, max124
        x12(1,i) = xi124(1,i)
        x12(2,i)=0.0
        j=i+max124
        x12(1,j)=x124(1,i)
        x12(2,j) = xi124(2,i)
20
      continue
      nnodel=max124*2
      nel=max124-1
      mrow=1
      nfirst(1)=1
      nelem1=nel*mrow
      do 30 i=1,nelem1
30
        ne(i)=4
      icon=0
      do 7G i=1, mrow
        n1=i*max124
        n2=n1-max124
```

С

```
do 60 j=1,nel
           icon=icon+1
          node(1, icon) = n2 + j
           node(2, icon) = n2 + j + 1
          node(3, icon) = n1 + j + 1
          node(4,icon)=n1+j
60
        continue
70
      continue
      write(6,200) nnode1, nelem1
      write(6,300) npot1, nbc11, nbc12
      if(npot1.eq.0) go to 110
      write(6, 350)
      write(6,400) (i,npt1(i),vpt1(i),i=1,npot1)
110
      if(nbc11.eq.0) go to 120
      write(6, 450)
      write(0,400) (i,ndbc11(i),vbc11(i),i=1,nbc11)
120
      if(nbcl2.eq.0) go to 130
      write(6,550)
      write(6.600) (i,neb12(i),nsde12(i),vbc12(1,i),vbc12(2,i),
     .i=1,nbc12)
130
      write(6.750)
      write(6,800) (i,x12(1,i),x12(2,i),u12(i),i=1,nnode1)
200
      format(/2x,#THE NO OF NODES IS#,i5,
     ./2x.#THE NO OF ELEMENTS IS#, 15, )
300
      format(/2x, #THE NO OF POINT SOURCES IS#, i5,
     ./2x,#THE NO OF ESSENTIAL BOUNDARY COND. IS#, 15,
     ./2x,#THE NO OF NATURAL BOUNDARY COND. 1S#,15)
350
      format(/5x,#NO#,5x,#NODE#,3x,#POINT SOURCE VALUE#,)
400
      format(2x, 15, 3x, 15, e15.5)
450
      format(/5x,#NO#,5x,#NODE#,3x,#ESSEN BOUND VALUE#,)
550
      format(/5x, #NO#, 5x, #ELEMENT#, 5x, #SIDE#, 5x, #P#, 5x, #GAMA#,)
600
      format(2x, 316, 2e15.5)
750
      format(/2x,#NODE NO#,8x,#X#,14x,#Y#,12x,#TEMP#,)
800
      format(2x, 15, 3e15.5)
      return
      end
      subroutine init2
      common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
      common/enode/ x12(2,1200),u12(1200)
      common/celem/ ne(1000), node(9,1000)
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                     ,angl(25),capa(2,25),tin(25),ratio(50)
      common/ccon2/ nnode2,nelem2,npot2,nbc21,nbc22
      common/edoman/ xdim,ydim,prop(4,4)
      common/cscal2/ rlen2,rval2
      common/cbc2/ ndbc21(25),vbc21(25),neb22(20),nsde22(20),vbc22(2,20)
                   .npt2(10).vpt2(10)
c
      write(6.1000)
1000
      format(//2x, #**** THE FULLOWING IS THE INITIAL INFORMATION#,
     .# OF REGION 2 ****#./)
      read(5,*) rlen2,rval2
      write(6,100) rlen2,rval2
100
      format(/2x,#THE REFERENCE LENGTH FOR MESHING REGION 2 IS#, e15.5,
```

./2x,#THE REFERENCE VALUE FOR MESHING REGION 2 IS#,e15.5.) ylen=(ydim-xi124(2,1))/rlen2 num=int(ylen) mp = num + 2nnode2=mp\*max124 ydis=(ydim-xi124(2,1))/float(num) do 50 i=1, max 124do 40 j=1,mp j1=j-1 nmm=j1\*max124+inc=nmm+nnode1 if(j.eq.(mp-1)) go to 25 if(j.eq.mp) go to 30 x12(1,nc)=x124(1,i)x12(2,nc)=ydim-float(j1)\*ydis go to 40 25 x12(1,nc)=xi124(1,i)x12(2,nc)=ydim-(float(j1)-rval2)\*ydis go to 40 30 x12(1,nc)=x1124(1,i)x12(2,nc)=x1124(2,i)40 continue 50 continue nel=max124-1mrow=num+1 nfirst(2) = nel\*num+1 nelem2=nel\*mrow icon=nelem1 do 70 i=1.mrow n1 = i \* max 124n2=n1-max124do 60 j=1,nel icon=icon+1 node(1,icon)=n2+j+1node(2,icon)=n2+j node(3,icon)=n1+j node(4, icon) = n1 + j + 160 continue 70 continue do 80 i=1,nelem2 ii=i+nelem1 30 ne(ii)=4read(5,\*) npot2,nbc22 nbc21 = 2\*max124 read(5,\*) ti,tj k=nnode2-max124+1 do 90 i=1, max 124ndbc21(i)=ivbc21(i)=ti j=max124+i ndbc21(j)=kVbc21(j)=tj k = k + 1continue

```
np=nnode2-max124
      do 95 i=1,np
        j=i+nnode1
95
        u12(j)=ti
      do 96 i=np+1,nnode2
        j=i+nnode1
96
        u12(j)=tj
      write(6,200) nnode2,nelem2
      write(6,300) npot2,nbc21,nbc22
      if(npot2.eq.0) go to 110
      write(6, 350)
      write(0,400) (i,npt2(i),vpt2(i),i=1,npot2)
110
      if(nbc21.eq.0) go to 120
      write(6,450)
      write(6,400) (i,ndbc21(i),vbc21(i),i=1,nbc21)
120
      if(nbc22.eq.0) go to 130
      write(6,550)
      write(6,600) (i,neb22(i),nsde22(i),vbc22(1,i),vbc22(2,i),
     .i=1,nbc22)
130
      ii=nnode1+1
      if=nnode1+nnode2
      j=1
      write(6,750)
      do 150 i=ii.if
        ia=1/11
        ic=11*id
        if(j.eq.ic.or.j.eq.1) then
          write(0,000) j,x12(1,i),x12(2,i),u12(i)
        end if
150
      j=j+1
200
      format(/2x,#THE NO OF NODES 1S#,15,/2x,
     .#THE NO OF ELEMENT IS#,15)
300
      format(/2x,#THE NO OF POINT SOURCES IS#, 15,
     ./2x,#THE NO OF ESSENTIAL BOUNDARY COND. IS#, 15,
     ./2x,#THE NO OF NATURAL BOUNDARY COND. IS#,15)
350
      format(/5x,#NO#,5x,#NODE#,5x,#POINT SOURCE VALUE#,)
400
      format(2x, 15, 3x, 15, e15.5)
450
      format(/5x,#NO#,5x,#NODE#,3x,#ESSEN BOUND VALUE#,)
      format(/5x, #NO#, 5x, #ELEMENT#, 4x, #SIDE#, 5x, #P#, 5x, #GAMA#,)
550
600
      format(2x,316,2e15.5)
750
      format(/x, #NODE NO#, 8x, #X#, 14x, #Y#, 12x, #TEMP#,)
800
      format(2x, 15, 3e15.5)
      return
      end
      subroutine init3
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2),
                      ang1(25), capa(2,25), tin(25), ratio(50)
      common/cdoman/ xdim,ydim,prop(4,4)
      common/ccon3/ nnode3, nelem3, npot3, nbc31, nbc32
      common/cnode3/ x3(2,1200),u3(1200)
      common/cbc3/ ndbc31(50),vbc31(50),neb32(40),nsde32(40),vbc32(2,40)
                   ,npt3(10),vpt3(10)
      common/cmatr4/ gk4(25,3),gf4(25),vnor(25)
      dimension a(55)
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read(5, \*) ny nnode3=ny\*nmax3 npot3=0 nbc31=nmax3 nbc32=nmax3-1 read(5,\*) cexp b=cexp/float(ny-2) c=0.0 do 10 i=1,ny-1 m = i - 1a(i) = exp(-float(m)\*b)c=c+a(i) 10 continue d1 = (ydim - xi3(2, 1))/cdo 30 j=1,nmax3aa = 0.0kk=0do 20 k=j,nnode3,nmax3  $x_3(2,k) = y_dim - d1 * a_a$  $x_3(1,k) = x_{13}(1,j)$ KK = KK + 1if(kk.lt.ny) aa=aa+a(kk) ki=k 20 continue x3(2,ki)=xi3(2,j)30 continue nel=nmax3-1 mrow=ny-1 nelem3=nel\*mrow read(5, \*) ci do 70 i=1, nnodej 70 u3(i)=ci do 00 i=1,nbc31 ndbc31(i)=i 80 voc31(i)=cinn=nel\*(mrow-1)+1do 100 i=1,nbc32 neb32(i)=nn nsde32(i)=3 vbc32(1,i)=0.0vbc32(2,i)=0.0nn=nn+1 100 continue write(6,1000) format(//2x, #\*\*\*\* THE FOLLOWING IS THE INITIAL INFORMATION#, 1000 .# OF REGION 3 \*\*\*\*#,/) write(6,200) nnode3, nelem3 write(6,300) npot3,nbc31,nbc32 if(npot3.eq.0) go to 110 write(6,350) write(6,400) (1,npt3(1),vpt3(1),i=1,npot3) 110 if(nbc31.eq.0) go to 120 write(6,450)

write(6.400) (i.ndbc31(i),vbc31(i),i=1,nbc31) 120 if(nbc32.eq.0) go to 130 write(6,550) write(6,600) (i,neb32(i),nsde32(i),vbc32(1,i),vbc32(2,i), .i=1.nbc32) 130 write(6,750)do 150 i=1.nnode3 ia=i/21ic=21\*idif(i.eg.ic.or.i.eg.1) then write(6, 300) i,x3(1,i),x3(2,i),u3(i) end if continue 150 200 format(/2x,#THE NO OF NODES IS#, 15, ./2x,#THE NO OF ELEMENTS IS#,15) 300 format(/2x, #THE NO OF POINT SOURCES IS#, 15, ./2x,#THE NO OF ESSENTIAL BOUNDARY COND. IS#, 15, ./2x,#THE NO OF NATURAL BOUNDARY COND. IS#,15) 350 format(/5x,#NO#,5x,#NODE#,3x,#POINT SOURCE VALUE#,) 400 format(2x.i5.3x.i5.e15.5) 450 format(/5x,#NO#,5x,#NODE#,5x,#ESSEN\_BOUND\_VALUE#,) 550 format(/4x,#NO#,3x,#ELEMENT#,4x,#SIDE#,9x,#P#,11x, .#GAMA#,) 60U format(2x, i4, 4x, i4, 5x, i4, 2e15.5) 750 format(/2x,#NODE NO#,8x,#X#,14x,#Y#,12x,#CONCEN#,) 300 format(2x,15,3e15.5) return end subroutine init4 common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2) ,angl(25),eapa(2,25),tin(25),ratio(50) common/cmatr4/ gk4(25,3),gf4(25),vnor(25) common/celem4/ ne4(25), nodes4(3, 25) common/cdoman/ xdim,ydim,prop(4,4) С write(6, 1000)1000 format(//2x,#\*\*\*\* THE FOLLOWING IS THE INITIAL INFORMATION#, .# OF INTERFACE, REGION 4 \*\*\*\*\* #./) read(5,\*) max124,nmax3 xunit=xdim/float(max124-1) do 1 i=1,max124-1 xi124(1,i)=(i-1)\*xunit1 continue xi124(1,max124)=xdim read(5, \*) si do 10 i=1, max124xi124(2,i)=si 10 do 20 i=1,nmax3 20 xi3(2,i)=si xi3(1,1) = x1124(1,1)do 30 i=1,max124-1 a=xi124(1,i) b = xi + 124(1, i+1)j=2\*i

```
xi3(1,j)=(a+b)/2.0
        xi3(1,j+1)=b
30
      continue
      pi=asin(1.0)
      do 40 i=1, max 124
40
        angl(i)=pi
      do 50 i=1, max124-1
        ne4(i)=2
        nodes4(1,i)=i
        nodes4(2,i)=i+1
50
      continue
      write(\hat{u}, 200) max124, max124-1
      write(6, 250)
      write(6,300) (i,xi124(1,i),xi124(2,i),i=1,max124)
      write(6.350)
      write(6,400) (i,ne4(i),nodes4(1,i),nodes4(2,i),i=1,max124-1)
200
      format(/2x,#THE NO OF NODES IS#,15,
     ./2x,#THE NO OF ELEMENTS IS#,15,)
250
      format(/ux, #NO#, 10x, #X#, 14x, #Y#,)
300
      format(2x, i0, 2e15.5)
350
      format(/2x, #ELEM NO#, 4x, #NO OF NODES#, 6x, #NODE 1#,
     .6x,#NODE 2#,)
40U
      format(2x, i5, 5x, i3, 5x, i8, 5x, i8)
      return
      end
      subroutine int4(x,n,d,t,p)
c....On entrance, d and x are vectors containing f[x(1), \ldots, x(i)]
c....and x(i), i=1,...,n, respectively. On exit p will contain
c....the value p(t) of the (n-1)-th degree polynomial interpolating
c....to f on x
      dimension x(5).d(5)
      p=d(n)
      i=n-1
10
      p=d(i)+(t-x(i))*p
      i=i-1
      if(i.ge.1) go to 10
      return
      end
      subroutine inter4(x,n,d,t,p,angle,curv)
c....Given vectors d(i)=f[x(1),\ldots,x(i)] and x(i), i=1,\ldots,n
c....Obtain p(t), angle of normal direction and curvature at t
      dimension x(5), d(5)
      p=d(n)
      i=n-1
      p=d(i)+(t-x(i))*p
10
      i = i - 1
      if(i.ge.1) go to 10
c....Compute the coefficients of the first and second derivatives
c....of Newton Divided Difference Formula
      pder1=d(2)
      do 50 i=3.n
        cc=0.0
        do 40 im=1.i-1
          c=1.0
```

do 30 in=1,i-1 if(im.eq.in) go to 30 c=c\*(t-x(in))30 continue cc = cc + c40 continue pder1=pder1+d(i)\*cc 50 continue pder2=2.0\*d(3)if(n.eq.3) go to 1000 c = 0.0do ú0 i=1.3 60 c=c+(t-x(i))pder2=pder2+d(4)\*c\*2.0 if(n.eq.4) go to 1000 c = 0.0do 70 i=1.4 k=i+165 if(k.gt.4) go to 70 c=c+(t-x(i))\*(t-x(k))k = k + 1go to 65 70 continue pder2=pder2+d(5)\*c\*2.0if(n.eq.5) go to 1000 write(0,200) 200 format(/2x,#N CUT OF RANGE IN INTER4.F#,) stop c....Get angle of normal direction 1000 if (pder1.le.0.le-7) then angle=asin(1.0) else slop=-1.0/pder1 angle=atan(slop) end if c....Compute curvature by formula P"/(1+P'\*\*2)\*\*1.5 dumny=(1.0+pder1\*\*2)\*\*1.5 if(abs(pder2).le.0.1e-8) then curv=0.0else curv=abs(pder2)/dumny end if return end subroutine mesh1(nnodec,nelemc) c....To remesh the temperature domain of region 1 in every c.... time step c....1. Given interface nodal coordinates xi124(2,25) c....2. Find Max. y-distance ymax of xi124(2,25) c....3. Obtain the nodal coordinates along each y-dir c....4. Finer mesh is constructed at both end-sides С common/ework/ xc(2,1200),nodec(9,1000),uc(1200)

common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)

```
,angl(25),capa(2,25),tin(25),ratio(50)
      common/cscal1/ rlen1,rval1
      common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
С
c....To find the max. of y coordinate along the interface
      ymax = xi124(2,1)
      do 10 i=2.max124
        if(xi124(2,i).gt.ymax) ymax=xi124(2,i)
10
      continue
c....To find the no. of nodes along each y-dir.
      ylen=ymax/rlen1+0.5
      num=int(ylen)
      if(num.lt.1) stop
      if (num.eq.1) then
        mp=num+2
      else
        mp = num + 3
      end if
      nfirst(1) = (max124-1) *(mp-2)+1
      nnodec=max124*mp
c....For each y-dir obtain the nodal coordinates
c....Special consideration for end side nodes
      do 50 i=1.max124
        yais=xi124(2,i)/num
        do 40 j=1,mp
          j1=j-1
          nmm=j1*max124+i
          if(j.eq.1) go to 20
          if(j.eq.2) go to 25
          if(j.eq.(mp-1).and.j.ne.mp) go to 26
          if (j.eq.mp) go to 30
          xc(1, nmm) = xi124(1, i)
          xc(2,nmm)=(j1-1)*yais
          go to 40
20
          xc(1,nmm) = xi124(1,i)
          xc(2,nmm)=j1*ydis
          go to 40
25
          xc(1.nmm) = xi124(1.i)
          xc(2,nom)=(j1-rval1)*ydis
          go to 40
26
          xc(1,nmm)=xi124(1,i)
          xc(2,nmm)=(j1-rval1-1)*ydis
          go to 40
30
          xc(1,nmm)=xi124(1,1)
          xc(2,nmm) = xi124(2,i)
40
        continue
50
      continue
c....Idetify the element data
c....Only consider 4-node element here
      nel=max124-1
      mrow=num+2
      nelemc=nel*mrow
      icon=0
      do /0 i=1, mrow
```

```
n1=i*max124
         n_{2=n_{1}-m_{2}}
         do ó0 j=1,nel
           icon=icon+1
           nodec(1,icon)=n2+j
           nodec(2, icon) = n2 + j + 1
           nodec(3, icon) = n1 + j + 1
           nodec(4,icon)=n1+j
60
        continue
70
      continue
      return
      end
      subroutine mesh2(nnodec,nelemc)
c....To remesh the temperature domain of region 2 in every.
c.... time step
c....1. Given interface nodal coordinates xi124(2,25)
c....2. Find min. y-distance ymin of xi124(2,25)
c....3. Obtain the nodal coordinates along each y-dir
c....4. Finer mesh is constructed at interface
С
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                      ,ang1(25),capa(2,25),tin(25),ratio(50)
      common/cscal2/ rlen2,rval2
      common/ccon2/ nnode2,nelem2,npot2,nbc21,nbc22
      common/edoman/ xdim,ydim,prop(4,4)
      common/cwork/ xc(2,1200),nodec(9,1000),uc(1200)
C
c....rval1=0.5
c....To find the min. of y coordinates along the interface
      ymin=xi124(2,1)
      do 10 i=2,max124
         if(xi124(2,i).lt.ymin) ymin=xi124(2,i)
10
      continue
c....To find the no. of nodes along each y-dir
      ylen=(ydim-ymin)/rlen2+0.5
      num=int(ylen)
      mp=num+2
      nnodec=max124*mp
c....For each y-dir obtain the nodal coordinates
      do 50 i=1, max124
        ydis=(ydim-xi124(2,i))/num
        do 40 j=1,mp
           j1=j-1
           nmm=j1*max124+i
           if(j.eq.(mp-1)) go to 25
           if(j.eq.mp) go to 30
           xc(1,nmm) = xi124(1,i)
           xc(2,nmm)=ydim-j1*ydis
           go to 40
25
           xc(1,nmm)=xi124(1,i)
           xc(2,nmm)=ydim-(j1-rval2)*ydis
           go to 40
30
           xc(1,nmm) = xi124(1,i)
           xc(2,nmm) = xi124(2,i)
```

```
40
        continue
50
      continue
c....Identify yhe element data
c....Only consider 4-node element here
      nel=max124-1
      nfirst(2)=num*nel+1
      mrow=num+1
      nelemc=nel*mrow
      icon=0
      do 70 i=1, mrow
        n1 = i * max 124
        n2=n1-max124
        do 60 j=1,nel
           icon=icon+1
          nodec(1,icon)=n2+j+1
          nodec(2, icon) = n2 + j
          nodec(3,icon)=n1+j
          nodec(4, icon) = n1 + j + 1
60
        continue
70
      continue
      return
      end
      subroutine mesh3
c....To remesh the concentration domain in every time step
c....Given interface nodal coordinates xi3(2,50)
c....Keep the same meshing system except squeeze each y-dir
c.... proportional to ratio(i) respectively
Ċ
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2),
                       angl(25), capa(2, 25), tin(25), ratio(50)
      common/cdoman/ xdim,ydim,prop(4,4)
      common/ccon3/ nnode3,nelem3,npot3,nbc31,nbc32
      common/unode3/ x3(2,1200),u3(1200)
      common/cwork/ xc(2,1200),nodec(9,1000),uc(1200)
С
c....For each y-direction obtain the nodal coordinates
      do 10 1=1,nmax3
        xc(1,i) = xi3(1,i)
        xc(2,i) = ydim
10
      continue
      do 50 i=1,nmax3
        do 40 j=i+nmax3,nnode3,nmax3
          xc(1,j)=xi3(1,i)
          xc(2,j)=ydim-(ydim-x3(2,j))*ratio(i)
          jj=j
40
       continue
       xc(2,jj)=xi3(2,i)
50
      continue
      return
      end
      subroutine moal(nnodec)
c....To modify the boundary conditions of the temperature region 1
      common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/ubc1/ ndbc11(25),vbc11(25),neb12(20),nsde12(20),vbc12(2,20)
```

```
,npt1(10),vpt1(10)
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                     ,angl(25),capa(2,25),tin(25),ratio(50)
c
      n1 = max 124 + 1
      n2 = max 124 * 2
      k=nnodec-max124+1
      kk=1
      do 10 i=n1,n2
        ndbc11(i) = \kappa
        vbc11(i)=tin(kk)
        k = k + 1
        kk = kk + 1
10
      continue
0...............
      iaa=10000
      iaaa=20000
      tvalue=-0.3
      factor1=-0.01
if (icount.ge.iaa.ang.icount.le.iaaa) then
        pi=acos(-1.0)
        do 30 i=1,max124
          degree=xi124(1,i)*pi/xi124(1,max124)
          vbc11(i)=tvalue+factor1*cos(degree)
30
        continue
      else
        do 40 i=1, max124
          vboll(i)=tvalue
40
        continue
      end if
      return
      end
      subroutine mod2(nnodec)
c....To modify the boundary conditions of the temperature region 2
      common/cbc2/ ndbc21(25),vbc21(25),neb22(20),nsde22(20),vbc22(2,20)
                   ,npt2(10),vpt2(10)
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                     ,ang1(25),capa(2,25),tin(25),ratio(50)
С
      n1 = max 124 + 1
      n2=max124*2
      k=nnodec-max124+1
      kk=1
      do 10 i=n1,n2
        ndbc21(i)=k
        vbc21(i)=tin(kk)
        k = k + 1
        kk = kk + 1
10
      continue
      return
      end
      subroutine mod3
      common/entrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount.nwrt
```

```
common/cbc3/ ndbc31(50),vbc31(50),neb32(40),nsde32(40),vbc32(2,40)
                    ,npt3(10),vpt3(10)
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2),
                       angl(25), capa(2,25), tin(25), ratio(50)
      common/cmatr4/ gk4(25,3),gf4(25),vnor(25)
      dimension tmp(50)
      dc=0.0
      const=1.0-dc
      j=1
      do 10 i=1, max 124-1
        a=-vnor(i)
        b = -vnor(i+1)
        c = (a+b)/2.0
        vbc32(1,j)=(a+c)/2.0*const
        j=j+1
        vbc32(1,j)=(b+c)/2.0*const
        j=j+1
10
      continue
      factor1=0.5e-5
      factor 2 = -0.5 e - 4
      iaa=201
      iaaa=10000
      ibb = 2100
      ibbb = 3000
      iperioa=10
      pi=acos(-1.0)
      if (icount.ge.iaa.and.icount.le.iaaa) then
        do 20 \text{ k}=1, \text{nmax}3
          tmp(k)=xi3(1,k)*pi/xi3(1,nmax3)
20
        continue
        do 30 k=1,nmax3-1
          degree=0.5*(tmp(k)+tmp(k+1))
          vbc32(1,k)=vbc32(1,k)+factor1*cos(degree)
        continue
      end if
      if (icount.ge.ibb.and.icount.le.ibbb) then
        igap=(icount-ibb)/iperiod
        itest=igap/2*2
        ivalue=iperiod*igap+ibb
        if (itest.eq.igap) then
          degree=(icount-ivalue)*pi/iperiod
          do 40^{\circ} k=1,nmax3-1
            vbc32(1,k)=vbc32(1,k)+factor2*cos(degree)
          continue
        else
          degree=(icount-ivalue)*pi/iperiod
          do 50 k=1.nmax3-1
            vbc32(1,k)=vbc32(1,k)-factor2*cos(degree)
          continue
        end if
      end if
```

С

C

30

40

```
return
      end
      subroutine move
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                     ,angl(25),capa(2,25),tin(25),ratio(50)
      common/celem/ ne(1000), node(9,1000)
      common/cnode/ x12(2,1200),u12(1200)
      common/celem4/ ne4(25), nodes4(3,25)
      common/cmatr4/ gk4(25,3),gf4(25),vnor(25)
      common/entrl1/ nopt1,nopt2,nopt3
      common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
      common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
      common/edoman/ xdim,ydim,prop(4,4)
      dimension xim(2,25)
С
      call proc4
      call velo4(xim)
      call adst4(xim)
      call post4
      return
      end
      subroutine new1(nnodec,nelemc)
c....To obtain the nodal temp. in the mesh system of domain 1
c....Update all node and element information
e
      common/cnode/ x12(2,1200),u12(1200)
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                     ,angl(25),capa(2,25),tin(25),ratio(50)
      common/celem/ ne(1000), node(9,1000)
      common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
      common/cwork/ xc(2,1200),nodec(9,1000),uc(1200)
С
      nc=nnodec=(max124*2)+1
      n1=nnode1-max124+1
      do 10 i=1, max124
        if(xc(2,nc).gt.x12(2,n1)) then
          write(6,1000)
1000
          format(2x, #Too large of time step, out of range in new1.f#,)
          stop
        end if
        nc=nc+1
        n1 = n1 + 1
10
      continue
      nf=nnodec-max124
      do 50 i=1.max124
        do 40 j=i,nf,max124
          do 30 k=i.nnode1.max124
            if(x12(2,k).lt.xc(2,j)) go to 30
            if(x_{12}(2,k).eq.xc(2,j)) then
              ue(j) = u12(\kappa)
              go to 40
            end if
            kk = k - max 124
```

slp=(u)2(k)-u)2(kk))/(x)2(2,k)-x)2(2,kk)	
uc(j)=u(z(k)+s(xc(z,j)-x(z(z,k)))	
40 continue	
50 continue	
do 55 i=1.max124	
nf=nf+1	
uc(nf)=tin(i)	
55 continue	
nnode1=nnodec	
$\frac{1}{2} \left( \frac{1}{2} \right) + \frac{1}{2} \left( \frac{1}{2} \right) + \frac{1}$	
$u_{12}(1) = u_{12}(1)$	
x12(j-j)-x2(j-j)	
60 continue	
70 continue	
nelem1=nelemo	
do 90 i=1,nelem1	
ne(i)=4	
do 80 j=1,4	
node(j,i)=nodec(j,i)	
vo continue	• .
return	
end	
subroutine new2(nnodec,nelemc)	
cTo obtain the nodal temp. in domain 2	
cUpdate all node and element information	
C	
common/cnode/ x12(2,1200), u12(1200)	121 - finat (2)
$\frac{1}{2} = \frac{1}{2} $	$i_{0}(50)$
common/celem/ ne(1000).node(9.1000)	
common/ccon2/ nnode2.nelem2.npot2.nbc21.nbc22	
ccmmon/ccon1/ nnode1, nelem1, npot1, nbc11, nbc12	
common/cwork/ xc(2,1200),nodec(9,1000),uc(1200	)
c	
nsum=nnode1+nnode2	
do $\int (1 \pm i) dx i 24$	
leiennodel	
do 30 k=1 nsum max $124$	
if(x)(2,k),g(2,i)) go to 30	
if(x12(2,k).gt.xc(2,j)) go to 30 if(x12(2,k).eq.xc(2,j)) then	
if(x12(2,k).gt.xc(2,j)) go to 30 if(x12(2,k).eq.xc(2,j)) then uc(j)=u12(k)	
<pre>if(x12(2,k).gt.xc(2,j)) go to 30 if(x12(2,k).eq.xc(2,j)) then     uc(j)=u12(k)     go to 40</pre>	
<pre>if(x12(2,k).gt.xc(2,j)) go to 30 if(x12(2,k).eq.xc(2,j)) then     uc(j)=u12(k)     go to 40 end if</pre>	
if $(x12(2,k).gt.xc(2,j))$ go to 30 if $(x12(2,k).eq.xc(2,j))$ then uc(j)=u12(k) go to 40 end if kk=k-max124 cla (u12(2,k).eq.xc(2,j)) then here in the second sec	ς.
if $(x12(2,k).gt.xc(2,j))$ go to 30 if $(x12(2,k).eq.xc(2,j))$ then uc(j)=u12(k) go to 40 end if kk=k-max124 slp=(u12(k)-u12(kk))/(x12(2,k)-x12(2,kk)) uc(j)=u12(k)+alp=(u2(k,j)-x12(2,k))	).
<pre>if(x12(2,k).gt.xc(2,j)) go to 30 if(x12(2,k).eq.xc(2,j)) then     uc(j)=u12(k)     go to 40 end if     kk=k-max124 slp=(u12(k)-u12(kk))/(x12(2,k)-x12(2,kk))     uc(j)=u12(k)+slp*(xc(2,j)-x12(2,k))     go to 40 </pre>	<b>)</b> .
<pre>if(x12(2,k).gt.xc(2,j)) go to 30 if(x12(2,k).eq.xc(2,j)) then uc(j)=u12(k) go to 40 end if kk=k-max124 slp=(u12(k)-u12(kk))/(x12(2,k)-x12(2,kk)) uc(j)=u12(k)+slp*(xc(2,j)-x12(2,k)) go to 40 30 continue</pre>	<b>)</b> .

```
50
      continue
      nnode2=nnodec
      do 70 i=1, nnode2
        k1=i+nnode1
        u12(k1) = uc(i)
        do 60 j=1,2
          x12(j,k1) = xc(j,i)
60
        continue
70
      continue
      nelem2=nelemc
      do 90 i=1,nelem2
        k1=i+nelem1
        ne(k1)=4
        do 80 j=1,4
          node(j,k1)=nodec(j,i)
80
        continue
90
      continue
      return
      end
      subroutine new3
c....To obtain the nodal concentration in the new mesh system
c.... by using old mesh system and old nodal values
c....Update all node and element information
С
      common/cnode3/ x3(2,1200),u3(1200)
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                      ,angl(25),capa(2,25),tin(25),ratio(50)
      common/celem/ ne(1000), node(9,1000)
      common/ccon3/ nnode3,nelem3,npot3,nbc31,nbc32
      common/cwork/ xc(2,1200), nodec(9,1000), uc(1200)
C
      do 10 i=1, nelem3
10
        ne(i)=4
      nel=nmax3-1
      mrow=nelem3/nel
      icon=0
      do 20 i=1, mrow
        n1=i*nmax3
        n2=n1-nmax3
        do 15 j=1.nel
          icon=icon+1
          node(1, icon) = n2+j+1
          node(2, icon) = n2+j
          node(3, icon) = n1 + j
          node(4, icon) = n1 + j + 1
15
        continue
20
      continue
      do 50 i=1,nmax3
         do 40 j=i,nnode3,nmax3
           do 30 k=i,nnode3,nmax3
              if(x_3(2,k).gt.xc(2,j)) go to 30
              if(x_3(2,k).eq.xc(2,j)) then
                uc(j)=u3(k)
                go to 40
```
```
end if
              kk=k-nmax3
              slp=(u3(k)-u3(kk))/(x3(2,k)-x3(2,kk))
              uc(j)=u3(k)+slp*(xc(2,j)-x3(2,k))
              go to 40
30
            continue
40
         continue
50
       continue
      do 70 i=1.nnode3
        u_3(i) = uc(i)
        do 60 j=1,2
          x3(j,i)=xc(j,i)
60
        continue
70
      continue
      return
      end
      subroutine oneint(n)
      common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
      if(n.ea.1) then
        xint1(1)=0.0
        wint1(1) = 2.0
       return
      else if (n.eq.2) then
        xint1(1) = -1.0/sqrt(3.0)
        xint1(2) = -xint1(1)
        wint1(1) = 1.0
        wint1(2) = wint1(1)
      return
      else if (n.eq.3) then -
        xint1(1) = -sqrt(3.0/5.0)
        xint1(2) = 0.0
        xint1(3) = -xint1(1)
        wint1(1) = 5.0/9.0
        wint1(2)=3.0/9.0
        wint1(3) = wint1(1)
       return
      else if(n.eq.4) then
        xint1(1) = -0.361136311594053
        xint1(2) = -0.339981043584856
        xint1(3) = -xint1(2)
        xint1(4) = -xint1(1)
        wint1(1) = 0.347854645137454
        wint1(2) = 0.652145154862546
        wint1(3) = wint1(2)
        wint1(4) = wint1(1)
      return
      else
        write(6.100)
100
        format(2x,#Choose the improper value in oneint.f#)
      stop
      end if
      end
      subroutine post1(npass)
      common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
```

	common/cnode/ x12(2,1200),u12(1200)
	common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
С	
	write(6,100) time,npass
100	format(//2x, #THE RESULTS OF REGION 1 IS PRINTED AS FOLLOWING #.
	.#AT TIME#.e15.5./2x.#THE ITERATION NO IS#.i5.)
	write(6, 200)
200	$format(/2y \ \#NODF\# \ 10y \ \#Y\# \ 1\mu y \ \#Y\# \ 1\mu y \ \#TFMP\# )$
200	do 5 i=1  nnode 1
	if(aba(u1)(i)) = 1 = 0 = 15) = 12(i) = 0 = 0
c	11(abs(u(2(1)), 10, 100 - 15)) u(2(1)=0.0)
2	
10	d0 = 10 = 1, nnode(1)
10	Write(0, 300) = 1, x + 2(1, 1), x + 2(2, 1), u + 2(1)
000	Iormat(10,3015.5)
	return
	end
	subroutine post2(npass)
	common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
	common/cnode/ x12(2,1200),u12(1200)
	common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
	<pre>common/ccon2/ nnode2,nelem2,npot2,nbc21,nbc22</pre>
С	
	write(6,100) time,npass
100	format(//2x,#THE RESULTS OF REGION 2 IS PRINTED AS FOLLOWING #,
	.#AT TIME#,e15.5,/2x,#THE ITERATION NO IS#,i5,)
	write(6,200)
200	format(/2x,#NODE#,10x,#X#,14x,#Y#.14x,#TEMP#.)
	do 5 i=1,nnode2
	kk=i+nnode1
	if(abs(u12(kk))).le.1.0e-15) u12(kk)=0.0
5	continue
-	do 10 i=1.nnode2
	k=i+nnoge1
10	write $(6, 300)$ i x12(1,k) x12(2,k) u12(k)
300	format(i6 3e15 5)
500	return
	and
	subrouting nost 3(nnase)
	common/scon2/ nnodo2 nolsm2 nnot2 nho21 nho22
	$\frac{1}{2} = \frac{1}{2} = \frac{1}$
	common/entrl2/ time traw delt thet angint nites teles icoust sust
•	commony entrizy time, tmax, dert, thet, nprint, niter, toten, icount, nwrt
C	$\frac{1}{100}$
100	Write(0,100) time, npass
100	iormat(//2x, #IHE RESULTS OF REGION 3 IS PRINTED AS FOLLOWING #,
•	.#AT TIME#, e15.5,/2X, #THE ITERATION NO IS#,15,)
	write(6,200)
200	format(/2x,#NODE#,10x,#X#,14x,#Y#,14x,#CONC#,)
	do 5 i=1,nnode3
-	ir(abs(u3(i)).le.1.0e-15) u3(i)=0.0
5	continue
	do 10 i=1,nnode3
10	write( $6,300$ ) i,x3(1,i),x3(2,i),u3(i)
300	format(i6,3e15.5)
	return

**.**.

end subroutine post4 common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2) ,angl(25),capa(2,25),tin(25),ratio(50) if (icount/nprint\*nprint.eq.icount) then write(6,10) time 10 format(/2x,#THE FOLLOWING IS PRINTED OUT AT TIME#,e15.5,) write(6,20)20 format(//5x,#NO#,10x,#X#,14x,#Y#,13x,#ANGL#,10x, #CAPA1#,10x,#CAPA2#.) do 30 i=1, max 12430 write(6,40) i,xi124(1,i),xi124(2,i),angl(i),capa(1,i), capa(2,i)40 format(2x, 15, 5e15.5) write(6,50)50 format(//2x, #NO(CON) #, 7x, #X#, 14x, #Y#, 12x, #RATIO#,) do 60 i=1,nmax3 60 write(6,70) i,xi3(1,i),xi3(2,i),ratio(i) 70 format(16,3e15.5) end if return end subroutine prep1 c....To get the nodal coordinates, element information, nodal c.... values, and boundary conditions С common/entrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12 common/cnode1/ x12(2,1200),u12(1200) ... common/celem/ ne(1000),node(9,1000) common/cbc1/ ndbc11(25),vbc11(25),neb12(20),nsde12(20), vbc12(2,20),npt1(10),vpt1(10) common/cinter/ xi3(2,50), nmax3, xi124(2,25), max124, nfirst(2), angl(25), capa(2, 25), tin(25), ratio(50) common/cwork/ xc(2,1200), nodec(9,1000), uc(1200) C call mesn1(nnodec,nelemc) call new1(nnodec,nelemc) call mod1(nnodec) call set3(nnode1, nelem1, npot1) return end subroutine prep2 c....To get the nodal coordinates, element information, nodal c.... values, and boundary conditions С common/ccon2/ nnode2.nelem2.npot2.nbc21.nbc22 common/cnode1/ x12(2,1200),u12(1200) common/celem/ ne(1000),node(9,1000) common/cbu2/ ndbc21(25),vbc21(25),neb22(20),nsde22(20), vbc22(2,20),npt2(10),vpt2(10) common/cinter/ xi3(2.50).nmax3.xi124(2.25).max124,nfirst(2). angl(25), capa(2, 25), tin(25), ratio(50)

```
common/ccon1/ nnode1, nelem1, npot1, nbc11, nbc12
      common/cscal2/ rlen2,rval2
      common/cdoman/ xdim,ydim,prop(4,4)
      common/cwork/ xc(2,1200), nodec(9,1000), uc(1200)
      mm=nnode1+nnode2
      nn=nelem1+nelem2
      nm=npot1+npot2
C
      call mesh2(nnodec.nelemc)
      call new2(nnodec,nelemc)
      call mod2(nnodec)
      call set3(mm,nn,nm)
      return
      end
      subroutine prep3
c....To get the nodal coordinates, element information, nodal
c.... values, and boundary conditions
С
      common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/ccon3/ nnode3, nelem3, npot3, nbc31, nbc32
      common/cnode3/ x3(2,1200),u3(1200)
      common/celem/ ne(1000).node(9,1000)
      common/cbc3/ ndbc31(50),vbc31(50),neb32(40),nsde32(40),
                  vbc32(2,40),npt3(10),vpt3(10)
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2),
                     angl(25), capa(2,25), tin(25), ratio(50)
      ccmmon/cmatr4/ gk4(25,3),gf4(25),vnor(25)
      common/cdoman/ xdim,ydim,prop(4,4)
      common/ework/ xc(2,1200), nodec(9,1000), uc(1200)
С
      call mesh3
      call new3
      call mod3
      call set3(nnode3, nelem3, npot3)
      return
      end
      subroutine procl
      level 2, gk,gf,nz
      common/cntrl1/ nopt1,nopt2,nopt3
      common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/cdoman/ xdim,ydim,prop(4,4)
      common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
C
      common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
      common/celem/ ne(1000), node(9,1000)
      common/cnode/ x12(2,1200),u12(1200)
      common/cbc1/ ndbc11(25),vbc11(25),neb12(20),nsde12(20),vbc12(2,20)
                  ,npt1(10),vpt1(10)
      common/cscal1/ rlen1,rval1
      common/cmatrx/ gk(1200,25),gf(1200),nz
      call form1
      call aply1
      return
      end
```

```
subroutine proc2
level 2, gk,gf,nz
common/entrl1/ nopt1, nopt2, nopt3
common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none.ntwo
common/cdoman/ xdim,ydim,prop(4,4)
common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
common/ccon2/ nnode2,nelem2,npot2,nbc21,nbc22
common/celem/ ne(1000), node(9,1000)
common/cbc2/ ndbc21(25),vbc21(25),neb22(20),nsde22(20),vbc22(2,20)
            ,npt2(10),vpt2(10)
common/cscal2/ rlen2.rval2
common/ematrx/ gk(1200,25),gf(1200),nz
common/cnode/ x12(2,1200),u12(1200)
call form2
call aply2
return
end
subroutine proc3
level 2, gk,gf,nz
common/entrl1/ nopt1,nopt2,nopt3
common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
common/cdoman/~ xdim,ydim,prop(4,4)
common/ccon3/ nnode3,nelem3,npot3,nbc31,nbc32
common/cnode3/ x3(2,1200),u3(1200)
common/celem/ ne(1000), node(9,1000)
common/cbc3/ ndbc31(50),vbc31(50),ned32(40),nsde32(40),vbc32(2,40)
            ,npt3(10),vpt3(10)
common/ematrx/ gk(1200,25),gf(1200),nz
call form3
call aply3
return
end
subroutine proc4
common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2),
              angl(25), capa(2,25), tin(25), ratio(50)
common/celem4/ ne4(25), nodes4(3,25)
common/celem/ ne(1000), node(9,1000)
common/cnode/ x12(2,1200),u12(1200)
common/cmatr4/ gk4(25,3),gf4(25),vnor(25)
common/cntrl1/ nopt1, nopt2, nopt3
common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
common/cdoman/ xdim,ydim,prop(4,4)
call form4
call solve4
return
end
subroutine rnsb(n,ib)
level 2, gk,gf,nz
```

С

С

common/cmatrx/ gk(1200,25),gf(1200),nz common/cwork/ xc(2,1200),nodec(9,1000),uc(1200) np1=n+1 do 20 i=2,n sum=0.0k1 = minO(ib-1, i-1)do 10 k=1,k1 10 sum=sum+gk(i-k,k+1)/gk(i-k,1)\*gf(i-k)20 gf(i)=gf(i)-sum uc(n)=gf(n)/gk(n,1)do 40 k=2,ni=npl-k j1=i+1 j2=minO(n,i+ib-1)sum=0.0do 30 j=j1,j2 mm=j-j1+2 30 sum=sum+uc(j)\*gk(i,mm) 40 uc(i) = (gf(i) - sum)/gk(i, 1)return end subroutine rhsb4(gk,dn,gf,n,ib) dimension gk(25,3),gf(25),dn(25) np1=n+1do 20 i=2,n sum=0.0 k1=minO(ib-1,i-1)do 10 k=1,k1 10 sum=sum+gk(i-k,k+1)/gk(i-k,1)\*gf(i-k)20 gf(i)=gf(i)-sumdn(n) = gf(n)/gk(n,1)do 40 k=2,n i=npl-k j1=i+1 j2=minO(n,i+ib-1)sum=0.0do 30 j=j1,j2 mm=j-j1+2 30 sum=sum+dn(j)\*gk(i,mm) 40 dn(i) = (gf(i) - sum)/gk(i, 1)return ena subroutine set3(nn,ne,np) if(nn.lt.2.or.nn.gt.1200) go to 1000 if(ne.lt.1.or.ne.gt.1000) go to 1000 if(np.lt.0.or.np.gt.10) go to 1000 return 1000 write(6, 100)100 format(/2x,#Out of dimensional range in set3.f#,) stop end subroutine shape(x,n,psi,dpsi) С С

This subroutine is to calculate the values of the shape С C functions and their gerivatives with respect to the С С С master element coordinate at the spcified point x С С input parameters x,n С output parameters psi dpsi С С С 4-, 8-, and 9-node quadrilateral element are considered Ċ С n....the number of nodes (and shape functions) in the С С element С C x(1), x(2)...coordinates of point in the master element С С coordinate system С С psi....shape functions C С dpsi....derivatives of shape functions С С С С dimension x(2),psi(9),dpsi(9,2) if(n.eq.4) then p=0.250\*(1.0-x(2))q=0.250\*(1.0+x(2))r = 1.0 - x(1)s=1.0+x(1)psi(1)=p\*r psi(2)=p\*s psi(3)=q\*s psi(4) = q\*rdpsi(1,1) = -pdpsi(1,2) = -0.250 \* rdpsi(2,1)=p aps1(2,2)=-0.250\*s dpsi(3,1)=qdpsi(3,2) = -dpsi(2,2)dpsi(4,1) = -qdpsi(4,2)=-dpsi(1,2) return else if (n.eq.3) then psi(1)=0.250\*(1.0-x(1))\*(1.0-x(2))\*(-1.0-x(1)-x(2))psi(2)=0.250\*(1.0+x(1))\*(1.0-x(2))\*(-1.0+x(1)-x(2)) psi(3)=0.250\*(1.0+x(1))\*(1.0+x(2))\*(-1.0+x(1)+x(2)) psi(4) = 0.250\*(1.0-x(1))\*(1.0+x(2))\*(-1.0-x(1)+x(2))psi(5)=0.50\*(1.0-x(1)\*\*2)\*(1.0-x(2)) psi(6)=0.50\*(1.0+x(1))\*(1.0-x(2)\*\*2) psi(7) = 0.50\*(1.0-x(1)\*\*2)\*(1.0+x(2))psi(8) = 0.50\*(1.0-x(1))\*(1.0-x(2)\*\*2)dpsi(1,1)=0.250\*(1.0-x(2))\*(2.0\*x(1)+x(2))dpsi(1,2)=0.250\*(1.0-x(1))\*(x(1)+2.0\*x(2))dpsi(2,1)=0.250\*(1.0-x(2))\*(2.0\*x(1)-x(2))dpsi(2,2)=0.250\*(1.0+x(1))\*(-x(1)+2.0\*x(2))dpsi(3,1)=0.250\*(1.0+x(2))\*(2.0\*x(1)+x(2))dpsi(3,2)=0.250\*(1.0+x(1))\*(x(1)+2.0\*x(2)) dpsi(4,1)=0.250\*(1.0+x(2))\*(2.0\*x(1)-x(2))dpsi(4,2)=0.250\*(1.0-x(1))\*(-x(1)+2.0\*x(2)) dpsi(5,1) = -x(1)\*(1.0-x(2))dpsi(5,2)=-0.50\*(1.0-x(1)\*\*2) dpsi(6,1)=0.50\*(1.0-x(2)\*\*2)

```
dpsi(6,2) = -x(2) * (1.0+x(1))
         dpsi(7,1) = -x(1)*(1.0+x(2))
        dpsi(7,2) = -dpsi(5,2)
         dpsi(0,1) = -dpsi(0,1)
         dpsi(3,2) = -x(2) * (1.0 - x(1))
      return
      else if(n.eq.9) then
         fact1=x(1) * * 2-x(1)
         fact2=x(2)**2-x(2)
         fact3=x(1)**2+x(1)
         fact4=x(2)**2+x(2)
         fact5=1.0-x(2)**2
         fact6=1.0-x(1)**2
         psi(1)=0.250*fact1*fact2
        psi(2) =0.250*fact3*fact2
        psi(3)=0.250*fact3*fact4
         psi(4)=0.250*fact1*fact4
        psi(5)=0.50*fact6*fact2
        psi(6)=0.50*fact3*fact5
         psi(7) = 0.50 * fact6 * fact4
        psi(2)=0.50*fact1*fact5
        psi(9) = fact5*fact6
        dpsi(1,1)=0.250*(2.0*x(1)-1.0)*fact2
        dpsi(1,2) = 0.250 * fact 1 * (2.0 * x(2) - 1.0)
        dpsi(2,1)=0.250*(2.0*x(1)+1.0)*fact4
        dpsi(2,2)=0.250*fact3*(2.0*x(2)+1.0)
        dpsi(3,1)=dpsi(2,1)
        dpsi(3,2) = dpsi(2,2)
        dpsi(4,1)=0.250*(2.0*x(1)-1.0)*fact4
        dpsi(4,2) = 0.250 * fact 1 * (2.0 * x(2) + 1.0)
        dpsi(5,1) = -x(1) * fact2
        dpsi(5,2) = 0.50 * fact \delta * (2.0 * x(2) - 1.0)
        dpsi(6,1)=0.50*(2.0*x(1)+1.0)*fact5
        dpsi(6,2) = -x(2) * fact3
        dpsi(7,1) = -x(1) * fact 4
        apsi(7,2) = 0.50 * fact \circ (2.0 * x(2) + 1.0)
        dpsi(0,1)=0.50*(2.0*x(1)-1.0)*fact5
        dpsi(8,2) = -x(2) * fact1
        dpsi(9,1) = -2.0 \times (1) \times fact5
        dpsi(9,2) = -2.0 * x(2) * fact6
      return
      else
      write(6, 100)
      format(2x, #Choose the wrong no. of shape functions [#,/)
100
      stop
      end if
      end
      subroutine solve(nnode)
      level 2, gk,gf,nz
      common/cmatrx/ gk(1200,25),gf(1200),nz
      common/cwork/ xc(2,1200), nodec(9,1000), uc(1200)
      if(nz.eq.3) ib=23
      if(nz.eq.1.or.nz.eq.2) ib=13
      call trib(nnode, 1b)
```

```
call rhsb(nnode, ib)
return
end
subroutine solve4
common/cmatr4/gk4(25,3),gf4(25),vnor(25)
common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
               ,angl(25),capa(2,25),tin(25),ratio(50)
ib=2
call trib4(gk4,max124,ib)
call rhsb4(gk4,vnor,gf4,max124,ib)
return
ena
subroutine sov1
level 2, gk.gf.nz
common/ontrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
common/cntrl1/ nopt1,nopt2,nopt3
common/cdoman/ xdim,ydim,prop(4,4)
common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
common/cconi/ nnode1, nelem1, npot1, nbc11, nbc12
common/celem/ ne(1000).node(9,1000)
common/cnode/ x12(2,1200),u12(1200)
common/cbc1/ ndbc11(25),vbc11(25),neb12(20),nsde12(20),vbc12(2,20)
            ,npt1(10),vpt1(10)
common/cscal1/ rlen1,rval1
common/ematrx/ gk(1200,25),gf(1200),nz
common/cinter/ xi3(2,50).nmax3,xi124(2,25),max124,nfirst(2),
               angl(25),capa(2,25),tin(25),ratio(50)
common/cwork/ xc(2,1200), nodec(9,1000), uc(1200)
nz=1
npass=0
error=0.0
call prep1
call proc1
call solve(nnode1)
if(nopt3.eq.1) go to 40
npass=npass+1
do 20 i=1,nnode1
 diff=abs(uc(i)-u12(i))
  if(diff.gt.error) error=diff
  u12(i)=uc(i)
continue
if(error.le.tolen) go to 50
if (npass.lt. niter) go to 10
write(6.100)
format(2x,#Dont Converge in sov1.f#,)
stop
do 45 i=1,nnode1
 u12(i) = uc(i)
if(icount/nprint*nprint.eq.icount) call post1(npass)
return
end
subroutine sov2
level 2, gk,gf,nz
common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
```

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common/entrl1/ nopt1,nopt2,nopt3
      common/cdoman/ xdim,ydim,prop(4,4)
      common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
      common/ccon2/ nnode2,nelem2,npot2,nbc21,nbc22
      common/celem/ ne(1000), node(9,1000)
      common/cnode/ x12(2,1200),u12(1200)
      common/ccon1/ nnode1,nelem1,npot1,nbc11,nbc12
      common/cbc2/ ndbc21(25),vbc21(25),neb22(20),nsde22(20),vbc22(2,20)
                   ,npt2(10),vpt2(10)
      common/cscal2/ rlen2,rval2
      common/ematrx/ gk(1200,25),gf(1200),nz
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2),
                     angl(25), capa(2,25), tin(25), ratio(50)
      common/cwork/ xc(2,1200),nodec(9,1000),uc(1200)
      nz=2
      npass=0
      error=0.0
      call prep2
      call proc2
      call solve(nnode2)
      if(nopt3.eq.1) go to 40
      npass=npass+1
      do 20 i=1,nnode2
        k=i+nnode1
        diff=abs(uc(i)-u12(k))
        if (diff.gt.error) error=diff
        u12(k)=uc(i)
      continue
      if(error.le.tolen) go to 50
      if (npass.lt. niter) go to 10
      write(6.100)
100
      format(2x,#Dont Converge in sov2.f#,)
      Stop
      ao 45 i=1, nnode2
        k=i+nnode1
        u12(k) = uc(i)
      if(icount/nprint*nprint.eq.icount) call post2(npass)
      return
      end
      subroutine sov3
      level 2, gk,gf,nz
      common/cntrl2/ time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      common/ontrl1/ nopt1,nopt2,nopt3
      common/edoman/ xdim,ydim,prop(4,4)
      common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo
      common/ccon3/ nnode3, nelem3, npot3, nbc31, nbc32
      common/celem/ ne(1000), node(9,1003)
      common/cnode3/ x3(2,1200),u3(1200)
      common/cbc3/ ndbc31(50),vbc31(50),neb32(40),nsde32(40),vbc32(2,40)
                  ,npt3(10),vpt3(10)
      common/cmatrx/ gk(1200, 25), gf(1200), nz
      common/cmatr4/ gk4(25,3),gf4(25),vnor(25)
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2),
                     angl(25), capa(2,25), tin(25), ratio(50)
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common/cwork/ xc(2,1200), nodec(9,1000), uc(1200) nz=3npass=0 error=0.0 call prep3 10 call proc3 call solve(nnode3) if(nopt3.eq.1) go to 40 npass=npass+1 do 20 i=1,nnode3 diff=abs(uc(i)-u3(i)) if(diff.gt.error) error=diff  $u_3(i) = u_2(i)$ 20 continue if(error.le.tolen) go to 50 if(npass.lt. niter) go to 10 write(6,100) 100 format(2x,#Dont Converge in sov3.f#,) stop 40 do 45 i=1, nnode3 45  $u_3(i) = uc(i)$ 50 if(icount/nprint\*nprint.eq.icount) call post3(npass) return end subroutine trib(n,ib) level 2, gk,gf,nz common/cmatrx/ gk(1200,25),gf(1200),nz ao 20 i=2,n m1=minO(ib-1,n-i+1)do 20 j=1,m1 sum=0.0 k1=minO(i-1,ib-j)do 10 k=1.k1 10 sum=sum+gk(i-k,k+1)\*gk(i-k,j+k)/gk(i-k,1) 20 gk(i,j)=gk(i,j)-sumreturn end subroutine trib4(gk,n,ib) dimension gk(25,3) do 20 i=2,n m1 = minO(ib-1, n-i+1)do 20 j=1,m1 sum=0.0k1=minO(i-1,ib-j)do 10 k=1.k1 10 sum=sum+gk(i-k,k+1)\*gk(i-k,j+k)/gk(i-k,1) 20 gk(i,j)=gk(i,j)-sum return end subroutine twoint(m) С С С This subroutine is to generate the integration points С С and weights for Gaussian Quadrature integration with C

either four-point or nine-point or sixteen-point for С C С c the square element m=4....four-point С С С С m=9....nine-point m=16...sixteen-point С С С С C common/cint/ xint1(4),wint1(4),xint2(16,2),wint2(16),none,ntwo if (m.eq.4) then do 5 i=1,45 wint2(i)=1.0a = -1.0/sqrt(3.0)xint2(1,1) = axint2(1,2) = axint2(2,1) = -axint2(2,2) = axint2(3,1) = axint2(3,2) = -axint2(4,1) = -axint2(4, 2) = -areturn else if (m.eq.9) then wint2(1) = 25.0/81.0wint2(2) = 40.0/81.0wint2(3) = wint2(1) wint2(4) = wint2(2) wint2(5) = 64.0/81.0wint2( $\vec{0}$ ) = wint2( $\vec{2}$ ) wint2(7) = wint2(1)wint2(8) = wint2(2) wint2(9) = wint2(1) aa = -sqrt(3.0/5.0)do 10 i=1,7,310 xint2(i,1)=aado 20 i=2,8,320 xint2(i,1)=0.0do 30 i=3, 9, 330 xint2(i,1) = -aado 40 i=1,340 xint2(i,2)=aado 50 i=4.650 xint2(i,2)=0.0do 60 i=7,960 xint2(i,2) = -aareturn else if (m.eq.16) then a1=0.347854845137454 a2=0.652145154862546d0 wint2(1) = a1\*a1 wint2(2) = a1\*a2wint2(3) = wint2(2) wint2(4) = wint2(1) wint2(5) = wint2(2)

```
wint2(6) = a2*a2
        wint2(7) = wint2(\delta)
        wint2(\delta)=wint2(2)
        wint2(9) = wint2(2)
        wint2(10) = wint2(6)
        wint2(11) = wint 2(0)
        wint2(12) = wint 2(2)
        wint2(13) = wint 2(1)
        wint2(14) = wint2(2)
        wint2(15) = wint 2(2)
        wint2(16) = wint 2(1)
        a=-0.861136311594053
        b = -0.339981043584856
        do 110 i=1, 13, 4
110
           xint2(i,1)=a
        do 120 i=2,14,4
120
           xint2(i,1)=b
        do 130 i=3, 15, 4
130
           xint2(i,1) = -b
        do 140 i=4.16.4
140
           xint2(i,1) = -a
        ao 150 i=1,4
          xint2(i,2)=a
150
        do 160 i=5,ö
           xint2(i,2)=0
160
        do 170 i=9,12
          xint2(i,2) = -b
170
        do 130 i=13,10
          xint2(i,2) = -a
130
        return
        else
        write(ó,100)
100
        format(2x,#Choose the improper value in twoint.f#,)
        stop
        end if
        end
      subroutine velo4(xim)
c.... To obtain the normal nodal velocity along interface
c.... To obtain the nodal coordinates along interface
      dimension xim(2,25)
      common/cinter/ xi3(2,50),nmax3,xi124(2,25),max124,nfirst(2)
                      ,angl(25),capa(2,25),tin(25),ratio(50)
      common/cmatr4/ gk4(25,3),gf4(25),vnor(25)
      common/cntrl2/_time,tmax,delt,thet,nprint,niter,tolen,icount,nwrt
      xim(1,1) = xi124(1,1)
      xim(2,1) = xi124(2,1) + vnor(1)
      xim(1, max124) = xi124(1, max124)
      xim(2, max124) = xi124(2, max124) + vnor(max124)
      do 10 i=2.max124-1
        thita=angl(i)
        xim(1,i)=xi124(1,i)+vnor(i)*cos(thita)
        xim(2,i)=xi124(2,i)+vnor(i)*sin(thita)
10
      continue
      do 20 i=1.max124
```

vnor(i)=vnor(i)/delt 20 continue ibefore=icount-1 if(ibefore/nwrt\*nwrt.eq.ibefore) then tbefore=ibefore\*delt write(6,300) thefore format(//2x, #THE INTERFACIAL NODE LOCATIONS, NORMAL VELOCITIES#, + 300 #, MOVING DIRECTIONS, AND TEMPERATURES #, /2x,#AT TIME #,e15.5,#
write(6,1000) ARE#,) Ψ. 1000 format(//3x,#NO#,13x,#X#,18x,#Y#,18x, #VN#,16x,#ANGL#,15x,#TEMP#,) do 30 i=1, max 124write(0,100) i,xi124(1,i),xi124(2,i),vnor(i),angl(i),tin(i) 30 100 continue format(/i5,5e19.5) end if return end

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