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Title

Low energy electron attenuation lengths in core–shell nanoparticles

Permalink

<https://escholarship.org/uc/item/827766rg>

Journal

Physical Chemistry Chemical Physics, 19(20)

ISSN

0956-5000

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Publication Date

2017-05-24

DOI

10.1039/c7cp00663b

Peer reviewed

21 **Introduction:**

22 Electrons scatter both elastically and inelastically as they propagate through a material. The
23 inelastic mean free path (IMFP) is the mean distance an electron with a specific kinetic energy
24 (KE) travels between inelastic scattering events.¹ Understanding the energy dependence of the
25 IMFP in materials is important in several different disciplines. For example, the short distance an
26 electron travels between inelastic collisions makes photoelectron spectroscopy a surface sensitive
27 technique.^{2,3} If the energy dependence of the IMFP is known, then photoemission experiments can
28 be used to obtain depth-resolved chemical information by varying the photoelectron KE.⁴
29 Likewise, the IMFP of electrons is central to understanding photoemission heating of interstellar
30 dust clouds that arises from the absorption of UV radiation. The magnitude of the electron IMFP
31 is predicted to have a direct effect on the extent of warming.⁵ Additionally, inelastic collisions of
32 low energy electrons with DNA produce irreversible, detrimental effects due to bond scission
33 reactions.⁶⁻⁸ The extent of this damage is dependent on both the electron IMFP in the biological
34 medium as well as the cross sections for the interactions.⁹

35 Most measurements of the IMFP have been made at high KE by depositing a thin film of
36 material on a substrate and monitoring photoemission from the substrate as a function of film
37 thickness, or by monitoring intensity of backscattered electrons.¹⁰ Measurements of the IMFP of
38 electrons with KE <100 eV in soft materials (such as liquids) are difficult due to experimental
39 challenges (such as high vapor pressures or difficulties collecting all photoelectrons using a
40 hemispherical analyzer). Studies examining the propagation of low KE electrons in materials
41 generally measure the electron attenuation length (EAL). The EAL is the film thickness that results
42 in a $1/e$ decrease in signal intensity at a given energy compared to a non-coated substrate.¹ It is
43 closely related to the IMFP, but because EAL convolutes elastic and inelastic scattering, it is

44 calculated to be 15-30% shorter (depending on the KE of the electron and material).^{1,11} The first
45 low KE EALs measurements were made by impinging electrons onto a thin film and measuring
46 the resulting transmission current.¹²⁻²⁰ These measurements generally resulted in a single “low
47 energy” EAL that was highly material specific (for example, the low energy EALs for pentacene
48 and perylene were reported to be 7.5 and 80 nm, respectively).²⁰ With the introduction of liquid
49 jet experiments, measurements of the EAL in high vapor pressure liquids (such as water) became
50 possible by either monitoring the angular dependence of photoemission^{21,22} or coupling the O1s
51 photoionization cross-section to the signal intensity from liquid water at different energies.²³
52 Additionally, recent work used the angular distribution of photoemission from free nanoparticles
53 to model low energy electron IMFPs.^{24,25} While still somewhat experiment specific, these energy-
54 resolved measurements have yielded EALs that range from 1-5 nm for 1-25 eV KE photoelectrons
55 in water.

56 Two complementary techniques are typically used to study the properties of low energy
57 electrons in thin films: low energy electron transmission (LEET) and photoelectron transmission.¹²
58 In LEET experiments, thin films are irradiated by electrons and either backscattered electrons or
59 transmission current through the film are detected. In this technique, only the electrons that enter
60 (and not those that exit) the thin film are fully-defined in terms of energy and momentum. Thus,
61 information on an electron’s interactions within the film is lost because the measured value is
62 generally a current that is independent of energy. Conversely, photoelectron transmission
63 experiments input low energy electrons into a film via photoemission from a substrate (such as a
64 platinum electrode). The low energy electrons that escape the film into vacuum are then detected.

65 Recent work has used aerosol photoemission to study surface chemistry^{26,27}, surface
66 segregation^{28,29}, and electronic properties of nanoparticles.^{24,25,30-32} Many of these studies use a

67 velocity map imaging (VMI) spectrometer and vacuum ultraviolet light^{24–26,30,31,33–35} or X-rays^{26,32}
68 from a synchrotron to measure the energy and angular distributions of photoelectrons from free
69 nanoparticles. In this work, we use a VMI spectrometer to measure photoemission from free core-
70 shell nanoparticles and measure the EAL in a liquid hydrocarbon, squalane. By coating squalane
71 onto nanoparticles with a defined photoemission spectrum, we are able to measure the EAL for 1-
72 5 eV photoelectrons. Our measurements, when combined with previous measurements of EAL in
73 covalently-bonded, soft materials, show that the EALs of photoelectrons with >2 eV KE are
74 roughly constant and independent of energy. However, for electrons with KE <2 eV, the behavior
75 of photoelectrons is observed to be material dependent.

76 **Experimental Methods:**

77 Size-selected, core-shell nanoparticles of a specific composition are generated using a
78 method that has been previously described.³⁶ The experimental setup is shown in Figure 1. Briefly,
79 an aqueous solution of the core material (either potassium iodide or sodium chloride) is first
80 atomized and dried over silica gel to a relative humidity (RH) $<15\%$. This yields a stream of
81 polydisperse, solid particles are then size-selected using a scanning mobility particle sizer (SMPS,
82 TSI Inc.). The core diameters used in this experiment are 100, 150 and 200 nm. After size selection,
83 part of the aerosol flow is sampled by a condensation particle counter (CPC) to monitor particle
84 number concentrations. The remaining flow passes through a charcoal denuder to eliminate any
85 unwanted volatile organic contaminants. The core aerosol is then passed through a tube furnace
86 containing a pyrex tube with the coating material (i.e. liquid squalane). The coating forms on the
87 core via heterogeneous nucleation upon exiting the heated section of the oven. The thickness of
88 the coating is controlled by varying the temperature of the oven, and the size of the core-shell
89 particle is measured using a second SMPS. The coating thickness is determined from the difference

90 in diameter between aerosol that passes through the furnace and the uncoated aerosol stream that
91 bypasses the oven. Radial coating widths vary from ~ 1 to 9 nm with roughly ± 0.3 nm uncertainties
92 (Figure S4). This treatment assumes that the nanoparticles are spherical, while in reality they most
93 likely exist as rounded cubes. Previous work studying electron impact ionization of organic coated
94 alkali halides using a similar coating technique found the monolayer coating thickness for a liquid
95 hydrocarbon to be ~ 0.7 nm.³⁷ Because most measurements reported here have coating thicknesses
96 > 0.7 nm, the aerosol particles are assumed to be completely coated. Some systematic errors in the
97 coating thickness could exist due to the non-spherical nature of the particles and the potential for
98 non-uniform coating.

99 Photoemission from free aerosol particles is measured using a VMI spectrometer
100 (described previously).²⁶ Three electrodes in the spectrometer are tuned to achieve velocity
101 mapping conditions,³⁸ where a projection of the nascent velocity distribution of photoelectrons is
102 imaged on a multi-channel plate/phosphor detector with a CMOS camera. Photoelectron images
103 of coated aerosol are collected for 150 s, and sample images at each coating thickness are collected
104 in duplicate. The photoemission images are converted to photoelectron spectra using typical image
105 processing techniques.³⁹ As shown in Figure 2, the images are asymmetric due to the short
106 absorption length of vacuum ultraviolet light. As has been discussed previously, this leads to
107 preferential photoemission from the front of the nanoparticle and shadowing of the back of the
108 particle.^{25,31,33} A discussion of the effect image asymmetry has on the extracted KE spectra is
109 included in the Supplemental Information. Images from the uncoated core are collected after each
110 coating thickness, and the photoelectron spectra from the uncoated core are used to normalize the
111 coated signal for any drifts in instrument or particle generation. Experiments were performed at

112 the Chemical Dynamics Beamline (9.0.2) at the Advanced Light Source, Lawrence Berkeley
113 National Lab.

114 To determine EALs in a coating, squalane is coated onto a KI core and photoemission from
115 the particles is measured as a function of coating thickness. As the coating thickness increases, the
116 intensity of photoemission from the core material decreases due to inelastic collisions. EALs of
117 photoelectrons are determined by tracking the decay of photoemission from the core as a function
118 of coating thickness. KI is chosen as the core material due to its low ionization threshold (6.8 eV)⁴⁰
119 compared to squalane (8.4 eV).⁴¹ Figure 2 shows the KE distribution of photoelectrons from pure,
120 monodispersed KI nanoparticles (diameter, $D_p = 150$ nm, $\sim 10^4$ particles/cm³) and pure,
121 polydisperse squalane nanoparticles ($D_p \sim 220$ nm, $\sim 10^6$ particles/cm³). The KE of photoelectrons
122 from KI extends to higher energies than that of squalane due to the difference in threshold energies.
123 The absolute photoemission intensity from squalane is very small compared to that of KI.
124 Measurable photoemission intensities from pure squalane nanoparticles could only be recorded
125 using a polydisperse aerosol distribution, which has ~ 100 x more particles than the size selected KI
126 flow (Figure 2). Since the amount of squalane coating the KI core is only a small fraction of the
127 squalane in the polydisperse flow, photoemission from squalane is not observable in the
128 experiments to determine EAL. Even so, when determining EALs, we only measured KEs that are
129 large enough to ensure there is no photoemission contribution from squalane. To cover a large
130 range of KEs, photoemission from the core shell nanoparticles is measured at five different photon
131 energies: 8.5, 9, 10, 11 and 12 eV. Photoemission at higher photon energies is not collected due to
132 the interference with water vapor (IE = 12.6 eV).

133 **Results and Analysis:**

134 Figure 3a-e shows photoemission from KI nanoparticles as a function of squalane coating
135 thicknesses at incident photon energies of 8.5, 9, 10, 11, and 12 eV. The initial shape of the
136 uncoated photoemission spectrum (black lines in Figure 3) is determined by the photoionization
137 cross-section of KI at these various energies. At the energies used in this study, the photoelectrons
138 originate from the I 5p state.⁴⁰ As the coating thickness increases, two things are readily apparent
139 from these spectra: i) the intensity of the signal decreases with increasing coating thickness and ii)
140 there appears to be shift of the peak energy with increasing coating thickness. This first observation
141 is due to the inelastic collisions of electrons inside the squalane shell. The latter observation can
142 be attributed to the production of low KE electrons from inelastic collisions.¹² If electrons don't
143 lose all of their KE when they inelastically collide, they can still escape from the particle. In this
144 case, the intensities at lower energies would appear to decay slower because as the shell thickness
145 increases, a portion of their intensities would come from higher KE electrons that have undergone
146 inelastic collisions. To minimize the effects from the cascade to lower energy, a "top most interval"
147 analysis is employed and only the highest energy electrons (those within ~0.5 eV of the largest KE
148 in the initial spectra) are used to determine EALs. Figure 4 shows the normalized photoemission
149 intensity at 1.1 and 4.4 eV KE from the 8.5 and 12 eV spectra, respectively. If the EAL in squalane
150 was independent of energy, these curves would decay via the same constant.

151 By definition, the EAL is the coating thickness that results in a 1/e decrease in signal
152 compared to an uncoated substrate. Thus, the photoemission intensity from the core at a specific
153 energy, $I(E, d)$, is measured as a function of coating thickness, d :

$$154 \quad I(E, d) = I(E, 0) e^{-\frac{y_e(d)}{L_e(E)}}, \quad (1)$$

155 where $I(E, 0)$ is the initial photoemission intensity at energy, E , without any coating, $y_e(d)$ is the
 156 distance the electron must travel through at a given coating thickness, and $L_e(E)$ is the EAL at
 157 energy E . As the coating thickness increases, an increasing amount of the incident light is absorbed
 158 by the coating. Thus, Eq. 1 is modified to include the decrease in photon intensity that reaches the
 159 core due to the increased coating absorption:

$$160 \quad I(E, d) = \left[I(E, 0) e^{-\frac{y_v(d)}{L_v}} \right] e^{-\frac{y_e(d)}{L_e(E)}}. \quad (2)$$

161 In Eq. 2, L_v is the attenuation length of light in the coating and $y_v(d)$ is distance light travels through
 162 at a given coating thickness before encountering the core. At a given photon energy, L_v is equal to
 163 $\lambda/4\pi\kappa$, where λ is the wavelength of the incident light and κ is the imaginary component of the
 164 refractive index of the coating material. Table S1 shows the energy-dependent κ values of squalane
 165 and the associated attenuation length of light, L_v , at the energies we measured.⁴² While
 166 uncertainties in κ values used in this study were not reported,⁴² as will be discussed later, the
 167 calculated absorption lengths are generally larger than the measured EALs, which makes Eq. 2
 168 insensitive to changes in L_v .

169 For the case of a flat surface, the escape length of electrons and penetration length of light
 170 is equal to the coating thickness (i.e. $y_v(d) = y_e(d) = d$). Surface curvature has previously been used
 171 to explain photoemission from nanoparticles³³ and electron impact charging of nanoparticles.³⁷
 172 Because the thickness of the shell is much smaller than the particle radius, surface curvature is
 173 negligible on the scale of electron scattering and coating thickness closely resembles the overlayer
 174 thickness in the EAL definition. Thus, the radial coating thickness (d) is used to describe distance
 175 electrons must escape at each shell thickness. However, as shown in Figure 5, the distance the
 176 photon travels through the coating will only be equal to the coating thickness when the photon

177 enters the core shell nanoparticle normal to its surface. From the geometries shown in Figure 5, it
 178 can be shown that the distance a photon must travel through the coating to reach the core, $y(d, \theta)$,
 179 is:

$$180 \quad y(d, \theta) = (R + d) \cos[\beta(\theta)] - \sqrt{R^2 - (R + d)^2 \sin^2[\beta(\theta)]}, \quad (3)$$

181 where R is the radius of the core and $\beta(\theta)$ is the refracted angle between the incident light and the
 182 particle surface. The refracted angle is determined using Snell's law with the previously measured
 183 refractive indices in squalane.⁴² Previous work by Ziemann *et al.*³⁷ has shown that the probability,
 184 $g(\theta)d\theta$, a photon enters the particle at an angle between θ and $d\theta$ is:

$$185 \quad g(\theta)d(\theta) = 2\pi h dh/\pi R^2 = 2 \sin \theta \cos \theta d\theta, \quad (4)$$

186 where h is defined in Figure 5. The average attenuation of light as it travels through the coating,
 187 $\langle e^{-\frac{y_v(d)}{L_v}} \rangle$, before striking the core is given by:

$$188 \quad \langle e^{-\frac{y_v(d)}{L_v}} \rangle = \int_0^{\pi/2} e^{-\frac{y(d,\theta)}{L_v}} g(\theta) d\theta. \quad (5)$$

189 This integral does not have an exact solution and is solved numerically at each coating thickness.
 190 EALs are determined by using Eq. 5 to describe $\langle e^{-\frac{y_v(d)}{L_v}} \rangle$ and fitting the normalized intensity plots
 191 (e.g. Figure 4) at each KE to the following:

$$192 \quad \frac{I(E,d)}{I_0(E,0)} = \alpha \langle e^{-\frac{y_v(d)}{L_v}} \rangle e^{-\frac{d}{L_e(E)}}, \quad (6)$$

193 where α is a fitting parameter that is constrained to be less than one.¹⁴ As the core particles are
 194 coated by a squalane shell, there is some probability that electrons can be scattered back into the
 195 core. Thus, the intensity from the uncoated aerosol is not used in the fit, and α accounts for the
 196 diminished number of electrons that enter the shell due to interfacial scattering (Figure 4).

197 The EALs calculated from the five different photon energies are given in Table 1. This
198 energy range provides measurements of electron attenuation from electrons with 1.1 to 4.4 eV KE.
199 In general, at KE >2 eV, the EAL is determined to be 3-5 nm and does not appear to be a strong
200 function of energy. At lower KE, the EAL increases, and at KE <2 eV, it is >15 nm.

201 Because the uncertainties in κ are not previously reported,⁴² the sensitivity of Eq. 6 to
202 changes in the attenuation length of light was evaluated by applying a $\pm 20\%$ error to κ and
203 determining the resulting change in computed EALs. At a photon energy of 11 eV, the EAL of
204 electrons with 3.4 ± 0.2 eV KE was calculated to be 3.8 ± 1.0 nm using $\kappa = 0.7$ and $L_v = 12.8$ nm. If
205 κ changes to 0.56 and 0.84 (a -20% and +20% error), the measured EAL at 3.4 eV changes to 3.5
206 and 4.0 nm, respectively. These differences are smaller than the uncertainty in the original
207 measurement and show that the model does not have a strong sensitivity to changes in κ because
208 the EAL is generally significantly shorter than L_v .

209 Furthermore, because the dimensions of the particle are on the order of the wavelength of
210 light, an accurate description of the intensity in the core would require full Mie scattering
211 calculations. To provide a minimum for the reported EALs, we removed the absorption of light by
212 the shell (i.e. constrained $\langle e^{-\frac{\gamma_v(d)}{L_v}} \rangle = 1$ in Eq. 6), and assumed the attenuation of signal with
213 increased coating thicknesses arose only from inelastic scattering of electrons. This treatment had
214 the result of lowering the EALs by roughly 20-30% and represents a potential systematic error
215 from the data analysis.

216 Finally, to confirm the measured EAL are specific to the squalane coating and are not
217 dependent on the experimental approach, the size and material of the core were varied. Table S3
218 shows the EALs in squalane measured with different sized KI cores (100 nm, 150 nm and 200

219 nm). Additionally, Table S2 shows the EALs in squalane using both a KI and NaCl core. While
220 the NaCl photoemission spectra are slightly complicated by squalane photoemission (the
221 ionization thresholds of KI, NaCl and squalane are 6.8, 8.2 and 8.4 eV, respectively),^{40,41} the
222 general agreement in escape lengths suggests the observed EALs are specific to the properties of
223 the squalane shell.

224 **Discussion:**

225 The EALs in squalane measured in this study range from 3.3 nm at 4.4 eV to 15.6 nm at
226 1.1 eV. As shown in Table 1, the EALs are roughly constant (~3-5 nm) when KE >2 eV. However,
227 when KE <3 eV, the attenuation lengths increase to >15 nm. Low energy EALs in solid organic
228 films have previously been measured by monitoring electron transmission currents through films
229 of different thicknesses.^{13-17,19,43} Energy-resolved EALs in n-C₃₆H₇₄ were collected by scanning
230 the incident photon energy on a Pt substrate and changing the film thickness (data shown in Figure
231 6).^{16,17} The measured attenuation lengths of 2-5 eV electrons in n-C₃₆H₇₄ are ~3-5 nm, which are
232 in good agreement with the attenuation lengths reported here for squalane (also 3-5 nm for a similar
233 energy range). However, when the KE is <2 eV, the EALs in n-C₃₆H₇₄ remain constant (~2.5 nm),¹⁶
234 which differs from the measurements for squalane reported here. At low photoelectron energies,
235 Pfluger *et al.* describe phonon excitation associated with the C-H stretching mode as the primary
236 energy-loss scattering source.¹⁹ However, Cartier *et al.* mention that at lower KE, the measured
237 EALs varied with experimental conditions and the length of time a sample was irradiated due to a
238 changing number of trap states.¹⁶ Because the VMI spectrometer constantly probes a new surface,
239 we don't expect a similar "history" effect in our experiments. The first studies looking at
240 transmission of low energy electrons through organic films were not energy-resolved, and thus
241 only an average "low energy" (<3 eV) EAL was measured. The reported attenuation lengths

242 (generally 10-100s of Å) are extremely dependent on film composition.^{13-15,20,43} For example, low
243 energy electrons (<3 eV) had EALs of 7.5 and 80 nm in films composed of pentacene and perylene,
244 respectively.²⁰ Additionally, experiments looking at low energy electron transmission in methane⁴⁴
245 and krypton⁴⁵ films show IMFPs (and thus EALs) that increase at electron KE <2 eV (from 2 nm
246 at 3 eV to 5 nm at 1.7 eV). These studies suggest that at lower KE, there is a corresponding lower
247 density of states and thus fewer possible modes to deposit energy in a given scattering event. Thus,
248 based on the large variability of previously reported EALs, the low KE (<3 eV) EAL appear to be
249 extremely sensitive to the electronic structure of the material.

250 A very recent study also examined the photoemission from core-shell nanoparticles and
251 estimated the EAL of 0.5-1.0 eV photoelectrons using a total electron yield (TEY) measurement.
252 For thin and thick shells of squalane, they report EALs of 8.0 ± 0.5 and 30 ± 3 nm, respectively.⁴⁶
253 The EALs reported here at a similar energy range (15.6 nm for 1.1 ± 0.2 eV electrons) are slightly
254 larger. For a direct comparison with these previous results, TEY of the core is measured as a
255 function of coating thickness at each photon energy. Because TEY combines the photoemission
256 intensity of many different KEs, EALs cannot be directly extracted from this measurement.
257 Instead, the decrease in TEY signal is fit to Eq. 6 to determine an average attenuation coefficient.
258 The average attenuation coefficients are presented in the Supporting Information (Table S3). The
259 work of Amanatidis *et al.* uses a two, 266-nm photon ionization scheme, which results in ~ 9.3 eV
260 radiation. The core material (sodium benzoate) used by Amanatidis *et al.* has a different ionization
261 threshold (~ 7.5 eV)⁴⁶ than the KI core here (6.8 eV).⁴⁰ Thus, the TEY measurements at the specific
262 photon energies are not directly comparable because the KE of the photoelectrons differ. However,
263 the TEY measurements at 8.5 and 9 eV (which result in a KE spectra closest to that of Amanatidis
264 *et al.*) yield average electron attenuation coefficients of 18.8 ± 5.1 nm and 4.9 ± 1.8 nm, respectively.

265 These average attenuation coefficients bound those reported by Amanatidis *et al.* and suggest the
266 TEY measurements are very sensitive to the incident photon energy.

267 As a further means of comparison, there is an increasing body of evidence suggesting the
268 EAL in liquid water approaches a constant value at low KEs.^{21-23,47} The lowest energy EAL
269 reported by Suzuki *et al.*²³ (3.02 ± 0.46 nm at 5 eV) as well as that reported by Buchner *et al.*⁴⁷ (5
270 nm at 4.65 eV) are in reasonable agreement with the measurements reported here. Additionally,
271 recent work by Signorell *et al.* using angle resolved photoelectron spectroscopy of aqueous
272 nanoparticles extracted the IMPF and EAL of low kinetic energy photoelectrons.²⁵ The EAL of
273 electrons with 3 eV KE in water is reported to be 3.9 nm, which is in good agreement with the
274 results reported here. However, at lower KEs (<3 eV), Signorell *et al.* reports a decrease in the
275 EAL, which is attributed to an increase in purely vibrational scattering.²⁵

276 Shown in Figure 6 is a comparison of previously published energy-resolved EAL (and
277 IMFP) measurements for covalently bonded, soft materials and water. Shown in this Figure are
278 EALs (up to 25 eV) for carbon containing species,^{16,44,46,48} liquid water^{22,23,25,47} and solid water.⁴⁹
279 As noted above, there is significant scatter in measurements of the EAL at very low KE (<2 eV)
280 where the EALs are expected to be much more material specific. However, a common feature for
281 all measurements is the consistency of EALs for electrons with >2 eV KE. This is in contrast with
282 the models for electron IMFP and EAL which predict both values to monotonically increase after
283 reaching a minimum at 50-200 eV.^{2,3} Based on the results shown in Figure 6, it appears that (to
284 some extent) low KE electrons in covalently bonded, soft materials have equally short EALs and
285 are as surface sensitive as electrons with larger KE. These results have consequences for future
286 photoemission experiments that utilize electron KE to obtain depth profiles. For example, these
287 data suggest that depth profiling photoemission experiments most likely cannot be performed by

288 moving the KE of photoelectrons to lower energy without explicitly measuring the energy
289 dependence of very low KE photoelectrons.

290 **Conclusion:**

291 In this paper, we have used a VMI spectrometer to probe photoemission from free core-
292 shell nanoparticles. The VMI spectrometer detects all electrons that are emitted from the
293 nanoparticles. By changing the thickness of the shell we have determined the low energy EAL in
294 squalane. At very low kinetic energy (<2 eV), the EAL in squalane is >15 nm. At slightly larger
295 kinetic energies (>2 eV), the EAL drops to 3-5 nm. Together with other energy resolved EALs in
296 covalently bonded materials, these results suggest that the attenuation lengths of electrons with >2
297 eV KE do not change with increasing kinetic energy. This finding suggests that these low energy
298 electrons are equally surface sensitive as higher KE photoelectrons. The energy dependence of
299 EALs for electrons with KE <2 eV appear to be highly material specific.

300 The use of core shell aerosol particles to determine EALs is limited when performed using
301 VUV radiation (due to the absorption of light by the shell material). However, this technique
302 appears to have more promise and applicability at higher energies (such as soft X-rays) where the
303 light penetration lengths are larger (i.e. light absorption by the shell can be neglected) and the
304 spectral features (core-shell excitation) are more defined.

305 **Acknowledgments:**

306 This work and the Advanced Light Source were supported by the Director, Office of
307 Energy Research, Office of Basic Energy Science of the U.S. Department of Energy under
308 Contract No. DE-AC02-05CH11231. M.I.J. thanks the NSF for an NSF Graduate Research

309 Fellowship under DGE-1106400. K.R.W. was supported by the Department of Energy, Office of
310 Science Early Career Research Program.

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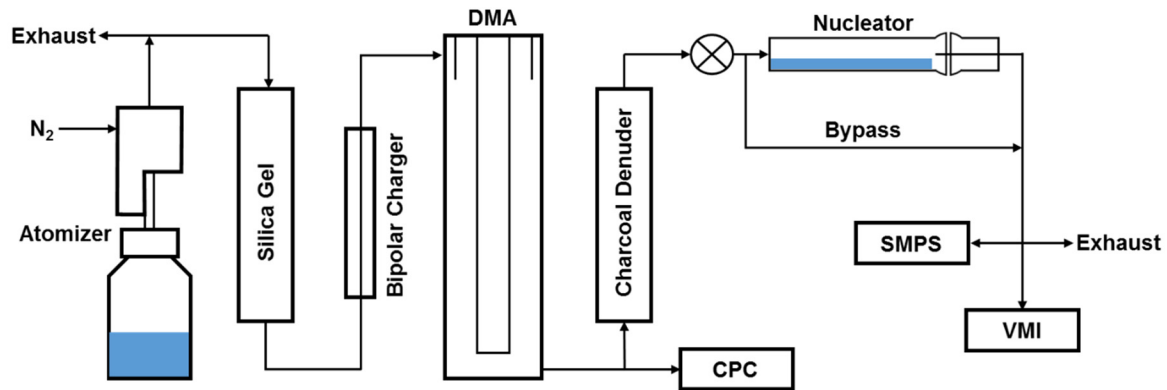
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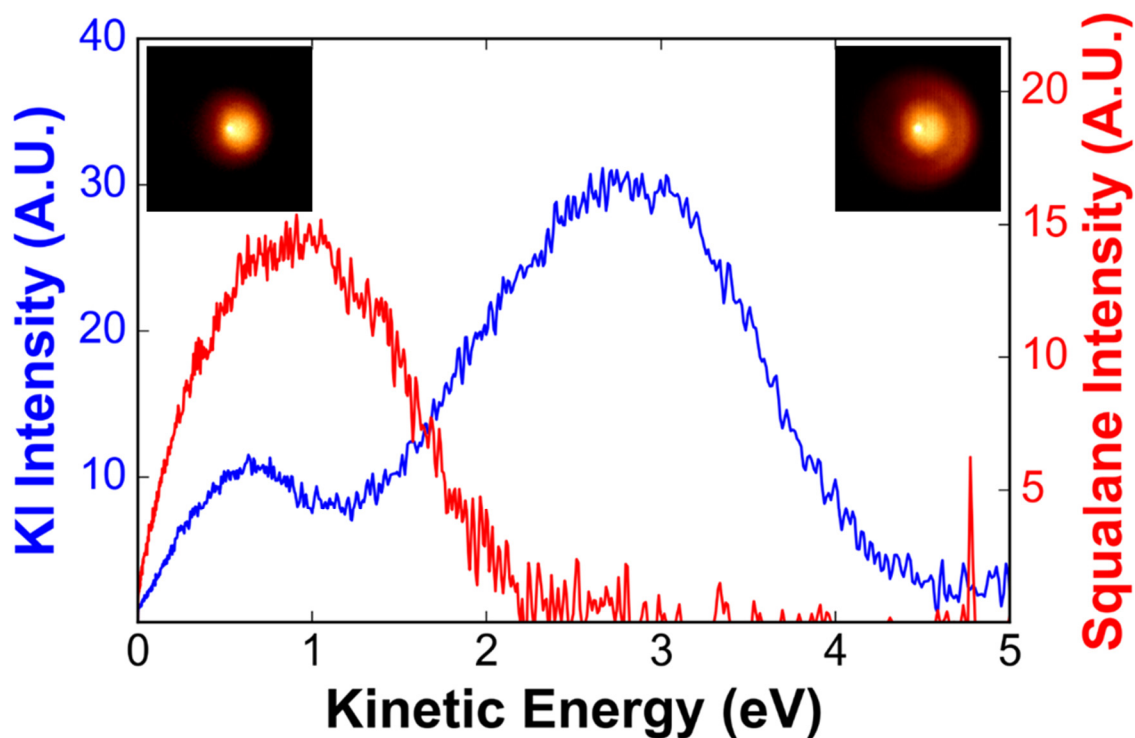
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392 **Figure 1.** Schematic of the coating experimental setup. (DMA = Differential Mobility Analyzer,
393 CPC = Condensation Particle Counter, SMPS = Scanning Mobility Particle Sizer, VMI = Velocity
394 Map Imaging spectrometer).

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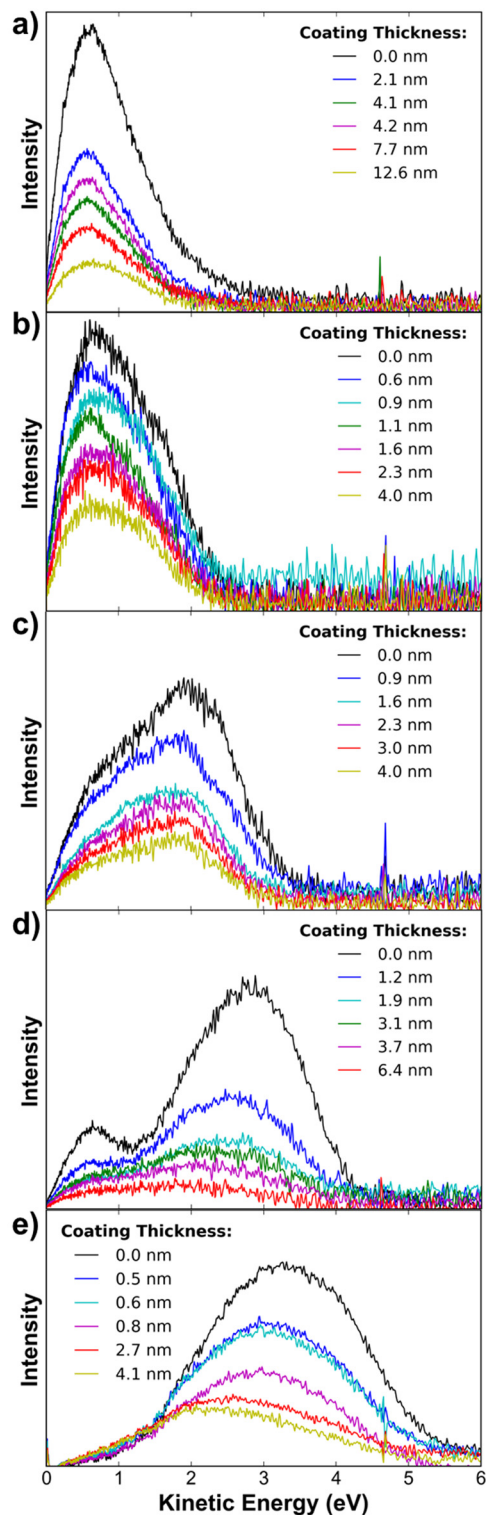


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399 **Figure 2.** Photoemission spectra of monodispersed KI and polydisperse squalane nanoparticles.
400 The incident photon energy is 11 eV. The insets on the left and right show the images that
401 correspond to the squalane and KI photoemission spectra, respectively. Despite similar intensities,
402 the squalane nanoparticle spectrum is obtained using $\sim 100x$ more material than the KI nanoparticle
403 spectrum. Because squalane has a lower ionization cross-section and a higher threshold energy
404 than KI, squalane photoemission does not interfere with the decay of KI photoemission intensity
405 at the higher KEs.

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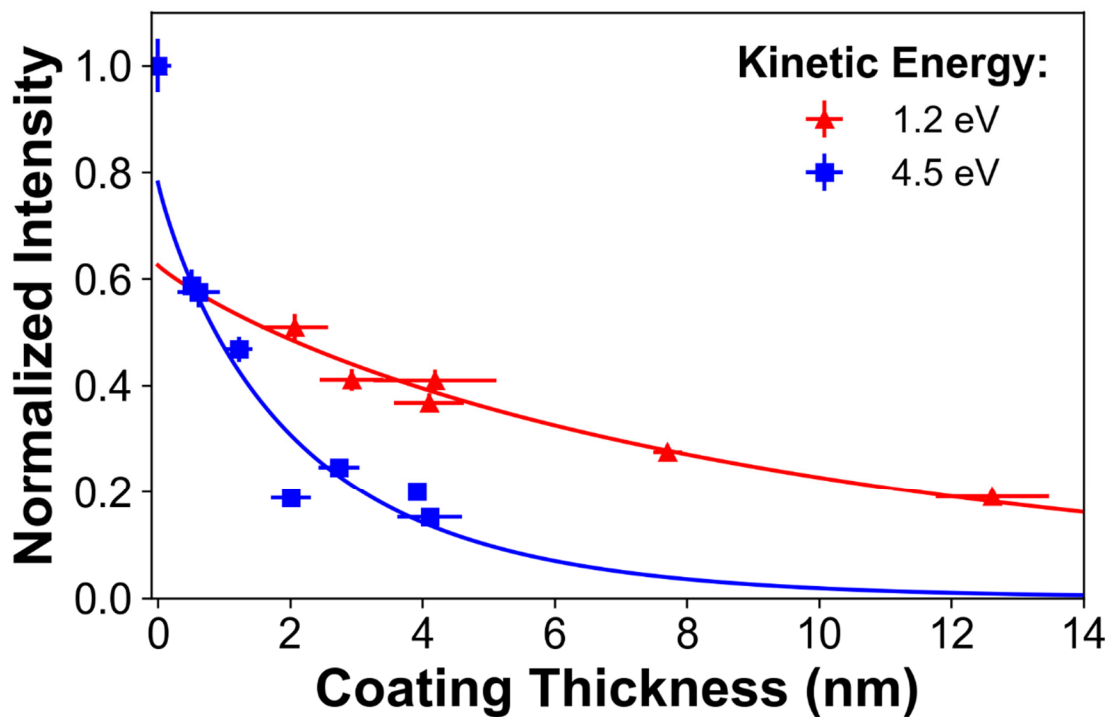
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409 **Figure 3.** Photoemission spectra of KI-squalane core-shell nanoparticles with varying squalane
 410 shell thicknesses. Spectra were collected at incident photon energies of 8.5 eV (a), 9 eV (b), 10
 411 eV (c), 11 eV (d), and 12 eV (e). Note: The side band is missing in the 12 eV spectrum due to
 412 large gas phase background at low KE caused by higher harmonics from beamline 9.0.2.

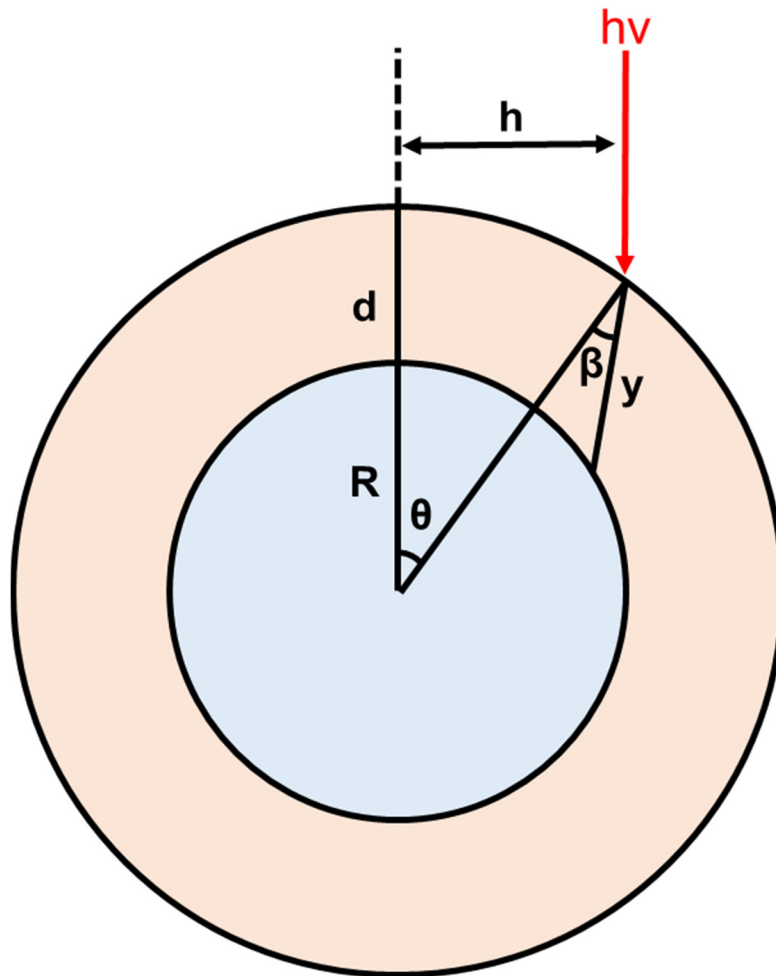
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418 **Figure 4.** Normalized intensity at 1.1 and 4.4 KE from the 8.5 and 12 eV photoemission spectra,
419 respectively. EAL are extracted from the decay of KI photoemission intensity at different KEs
420 using Eq. 6. An energy dependence in the EAL in squalane results in differences in the decay
421 rates at different KEs.

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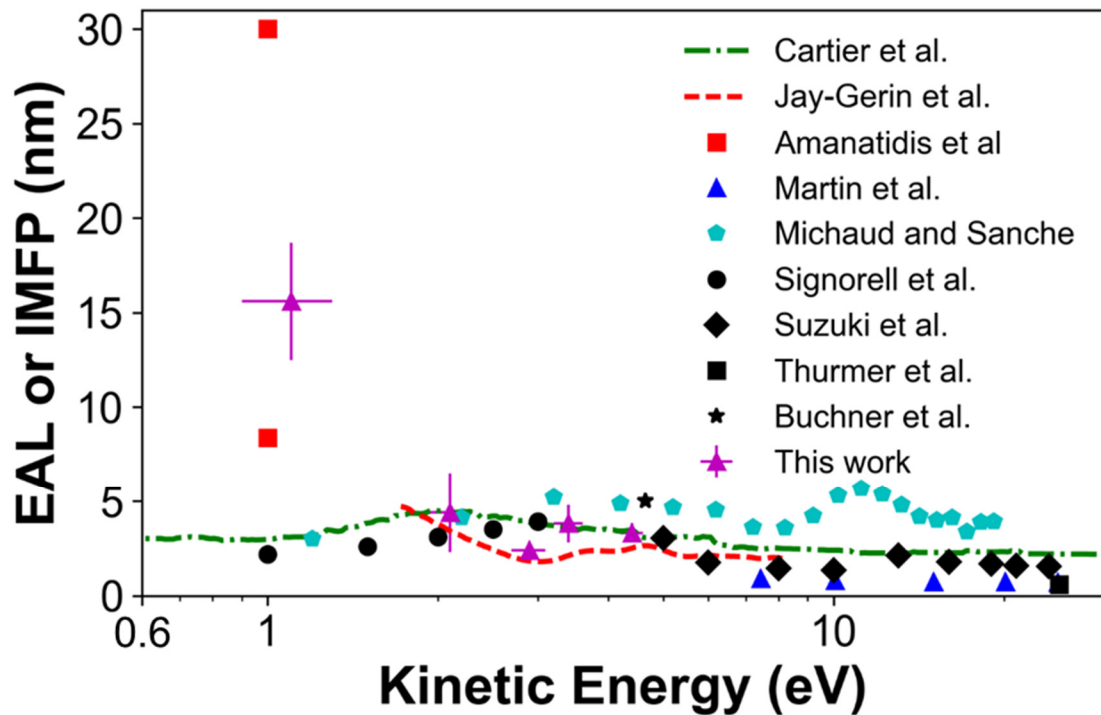
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425 **Figure 5.** Diagram showing the geometry of light entering the core through the squalane layer.

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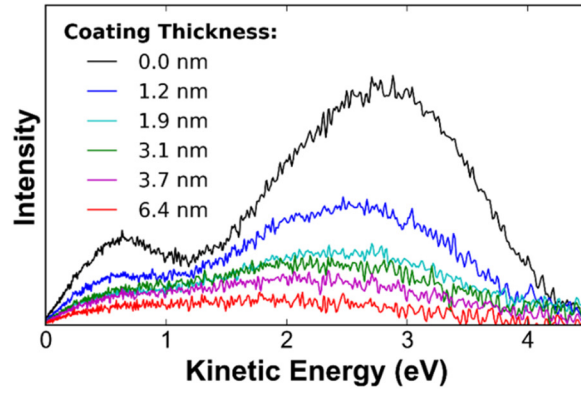
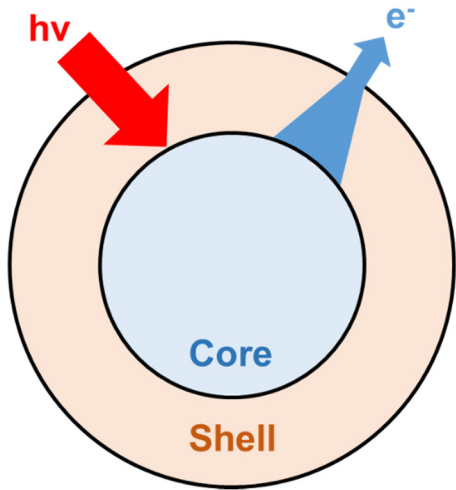
428 **Figure 6.** Collection of energy-resolved, low energy EAL in soft materials. The measurements
 429 from this work are given by the magenta triangles. Error bars represent $\pm 1s$. The green and red
 430 dashed lines represent EALs in paraffin, $n\text{-C}_{36}\text{H}_{74}$ (Cartier et al.^{16,17}) and IMFPs in methane (Jay-
 431 Gerin et al.⁴⁵) films, respectively. The red squares represent the EAL in thick (top) and thin
 432 (bottom) squalane layers (Amanatidis et al.⁴⁶). The blue triangles represent EALs measured in a
 433 free standing carbon film (Martin et al.⁴⁸). The black symbols represent EALs in liquid water from
 434 aerosol particles (circles, Signorell et al.²⁵) and liquid jets (diamonds, Suzuki et al.²³; square,
 435 Thurmer et al.²²; star, Buchner et al.⁴⁷). The pentagons represent EALs in solid water (Michaud
 436 and Sanche⁴⁹). At energies >2 eV, the EAL all remain fairly constant.

437

Kinetic Energy (eV)	EAL (nm)
1.1±0.2	15.6±3.1
2.1±0.2	4.4±2.1
2.9±0.2	2.4±0.4
3.4±0.2	3.8±1.0
4.4±0.2	3.3±0.4

438 **Table 1.** Experimental measurement of the energy dependence of the EAL in squalane. The errors
439 represent uncertainties in the fit to Eq. 6 ($\pm 1s$). Possible systematic errors are discussed in the text.

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442 TOC Graphic