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Journal

Chemistry of Materials, 28(11)

ISSN

0897-4756 1520-5002

Authors

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Publication Date 2016-06-14

DOI

10.1021/acs.chemmater.6b00633

Peer reviewed

(TTF)Pb₂I₅: a Radical Cation-Stabilized Hybrid Lead Iodide with Synergistic Optoelectronic Signatures

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ABSTRACT: Hybrid organic–inorganic materials offer the opportunity to combine functionalities of two distinct chemistries. Tetrathiafulvalene (TTF) is the precursor to some of the first organic metals, and perovskite lead iodides have recently proved their efficacy as photovoltaic materials. Here we describe a hybrid material that incorporates both, comprising TTF radical cations stabilizing an anionic layered lead–iodide network. Preparation, property analysis, and DFT calculations are reported, verifying that (TTF)Pb₂I₅ is a narrow gap semiconductor with optical properties synergistically influenced by the TTF⁺⁺ radical cation. This radical cation contributes to states in the electronic gap formed by orbitals in the inorganic framework.

Hybrid organic-inorganic materials can potentially combine the properties of highly tunable functional organics with those of complementary inorganic networks in a single material. Distinct control over what is desired from each component becomes possible in such materials, allowing the functions of the organic and inorganic components to work synergistically. Tetrathiafulvalene (TTF), the organic component in the new hybrid material reported here, is a precursor to some of the early organic metals and was a clear choice when choosing a functional organic. It was first reported by Wudl *et al.* in the early 1970's,^{1,2} and has been studied for its impressive electron donating abilities,^{3,4} including derivatives displaying superconductivity.⁵

Hybrid compounds containing TTF have also shown novel electronic and magnetic properties. Devic *et al.*^{6,7} reported a ethylenedithio-1,2-diiodotetrathiafulvalene (EDT-TTF-I₂⁺) hybrid with metallic character, and showed its metal-insulator transition to be preserved within a hybrid compound with PbI₂. Kondo *et al.*⁸ similarly reported TTF⁺ hybrids containing Sn and Cl/Br, all which display semiconducting behavior due to interactions through the extended TTF networks. Magnetic compounds with TTF derivatives have also been made with transition metal halides, and found to be composed of two kinds of donor columns which magnetically couple with the anions through close contact between metal halide interactions.⁹ Metal-organic-frameworks (MOFs), a related class of hybrid materials, generally possess poor electronic functionality, but attempts to prepare electrically conductive MOFs have recently been made. Examples include incorporating a conjugated molecule within the framework directly,¹⁰ as a guest that electronically couples metal atoms in the framework using (TCNQ),¹¹ or post-synthetic modification of an insulating MOF via chemical leaching.^{12,13} Stavila *et al.*¹⁴ have recently reviewed this topic.

Layered lead halides, in their own right, are an attractive set of materials with rich optoelectronic and semiconducting behavior. The extensive work of Mitzi and coworkers^{15,16} in the late 90's first highlighted how amines and diammines combine with main group halides to form unique layered perovskite structures, which motivated the chemistry of organic ammonium cation "spacers" to flourish. Following the work of Kojima et al.¹⁷ demonstrating the use of perovskite CH₃NH₃PbI₃ materials as solution deposited photovoltaics, there has been a huge flurry of research in this space to enhance the stability of the air unstable compound. This has included the use of ammonium "spacer" chemistry by Smith et al.¹⁸ which improved stability of a layered perovskite incorporating phenylethylammonium ions into the structure. Cao et al.¹⁹ also explored this idea with butylammonium instead of phenylethylammonium, and reported that the extent of ammonium spacer incorporation as well as growth direction of subsequent films had significant impact on functionality of their light absorbing devices. Maughan *et al.*²⁰ recently highlighted how the unusual aromatic cation tropylium⁺ can stabilize and electronically couple with tin and lead halide networks, helping to defining the position of either the conduction band maximum or valence band minimum.

Here we report a semiconducting hybrid compound with radical cation TTF+' stacks that separate condensed Pb₂I₅⁻ slabs in a manner that allows both the organic and the inorganic components of the hybrid to play an electronic role. (TTF)Pb₂I₅ crystals were grown under argon atmosphere by dissolving PbI₂ (110 mg, 0.239 mmol) and tetrabutylammonium iodide (168.2 mg, 0.455 mmol) in dimethylformamide (DMF, 1 cm³) at ambient temperature. Following complete dissolution, a separate solution of $(TTF)_2(BF_4)_2$ (20 mg, 0.0254 mmol) prepared following the literature procedure,²¹ in DMF (3 cm³) was added drop-wise. The resulting mixture is dark with red hue. This mixture was layered with acetonitrile (CH₂CN, 6 cm³) as non-solvent and left at room temperature for 2 days. Small dark bundles of crystals were formed beneath an unreacted PbI₂ layer. These bundles were isolated by washing with cold DMF, air dried, and finally dried in vacuum. The dark crystal bundles, when broken up, yield red rectangular single crystals. The material was observed stable in air for a month, but longer time study is needed.

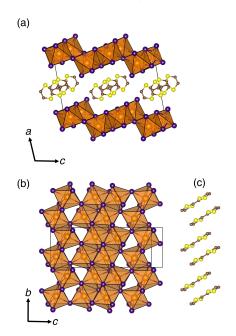


Figure 1. The crystal structure of the title compound (TTF)Pb₂I₅. (a) View down the *b* axis showing the corrugated Pb₂I₅⁻ layers. (b) A top view of the lead iodide layers, showing both edge and corner sharing of the octahedra. (c) TTF⁺ radical cations extend infinitely in stacks, with complete overlap as dimers, and partial overlap with the next nearest neighbor dimer pair. Space group $P_{2_1/c}$, a = 12.90(3) Å, b = 7.97(1) Å, c = 20.23(2) Å, $\beta = 100.25(7)^\circ$, Z = 4 $R_1 = 1.695$ %, $wR_r = -1.591$ %.

Single crystal X-ray diffraction carried out on multiple crystals confirmed the structure depicted in Figure 1(a). Figure 1(a) illustrates different views of the (TTF)Pb₂I₅ structure with some crystallographic details in the caption. The structure comprises sheets of corner and edge sharing lead iodide octahedra, separated by TTF+ cations that form stacks perpendicular to the anionic slabs. The TTF+ stacks contain eclipsed dimers (two TTF⁺) and maintain a consistent intra-dimer distance of 3.32 Å. The stacks extend through the crystal with an incomplete overlap with the next nearest TTF pairs, with an inter-dimer distance of 3.52 Å, as shown in Figure 1(c). It is interesting to note that this stacking structure of the TTF radical cations is commensurate with the corrugated sheet structure of the Pb-I sheets. In contrast to the previous work on EDT-TTF-I, iodoplumbates reported by Devic et al.,⁶ there is no disorder of the radical cation, which were reliably located over the temperature range of 373 K to 100 K. There is a slight contraction of the unit cell at 100 K, but no structural transition was observed in this temperature range.

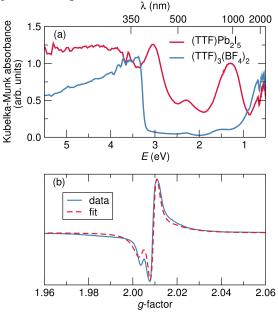


Figure 2. (a) Kubelka-Munk absorbance obtained from transformation of the diffuse reflectance UV-Vis spectra of $(TTF)Pb_2I_5$, as well as the precursor salt $(TTF)_3(BF_4)_2$. (b) Room-temperature ESR signal of polycrystalline $(TTF)Pb_2I_5$ sample, 9.2898 GHz, shown along with the simulation using *g* values of 2.010, 2.0083, and 2.053.

Figure 2(a) displays UV-Vis spectra obtained for $(TTF)Pb_2I_5$, as well as the starting material, $(TTF)_3(BF_4)_2$. The spectra were obtained using a Shimadzu UV3600 UV-NIR Spectrometer in diffuse reflectance mode using an integrating sphere and relative absorbance was obtained from Kubelka-Munk transformation of the reflectance spectra. Samples were diluted with $BaSO_4$ prior to measurement. Prior work from Sugano *et.* $al^{23,24}$ were the main sources used for interpreting the presented spectra. The insulating TTF salt, $(TTF)_3(BF_4)_2$, has

a structure comprising dimers and neutral TTF that are staggered with distances varying between 3.4 Å and 3.48 Å.²⁵ We observe noticeable absorptions for this salt near 0.6 eV which are attributed to intermolecular charge transfer from TTF^o to TTF⁺, as well as an absorption attributed to an *intra*molecular transition for a lone TTF⁺ within a dimer at 3.18 eV.^{22,24} The weak peaks at 1.5 eV and 2.3 eV are an intradimer transition and an intramolecular TTF⁺ transition, respectively.²⁴

Considering what is already known for monovalent and mixed valence TTF+' salts the absorption spectra of the title compound is not surprising. The broad band that peaks near 1.3 eV is from intradimer charge transfer, and the absorptions that peak at 2.3 eV and 3.0 eV are intramolecular transitions of the TTF+. What is unexpected however, is the low energy absorption at 0.6 eV which is attributed to *inter*molecular charge transfer from TTF^o to TTF^{+*}. Although no crystallographically observed TTF^o is present in our sample, we believe the cause of this absorption is back charge transfer from the Pb-I network to TTF^{+*}, in a manner similar to how TCNQ^{-*} back charge transfers to NMP+ in NMP-TCNQ.23 Furthermore, the sustained absorption over the 3.0 eV to 5.5 eV range is not usually seen for TTF^{+*} salts, which we attribute to absorption from the Pb-I network.

Figure 2(b) displays Electron spin resonance (ESR) spectra of powder samples of $(TTF)Pb_2I_5$, providing evidence of the radical cation state of TTF^+ at room temperature. The signal is centered close to the value for a free electron with an asymmetrical signal shape suggestive of some itinerant electron character, consistent with the observed free-carrier behavior.²⁶ The simulation of the EPR spectrum, which is depicted in Figure 2(b) as a dashed line, was carried out using Easy Spin²⁷ (details in SI).

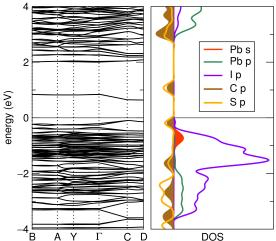


Figure 3. Band structure (PBE) and DOS (HSE06) for (TTF)Pb₂I₅. The DOS contributions of the C and S p states from TTF⁺ and the Pb and I contributions have been separated for added clarity.

Band structure calculations of (TTF)Pb₂I₅ based on density functional theory (DFT) were carried out using the Vienna ab initio Simulation Package (VASP)²⁸⁻³¹ which implements the Kohn-Sham formulation using a plane wave

basis and the projector-augmented wave formalism (PAW).^{32,33} The input structure was generated from the single crystal diffraction data by adding the aromatic H atoms to the TTF^{+*} radical cation and constraining them within the normal riding model of the HFIX function of the crystallographic refinement software SHELXL-2014.^{34,35} We decided to use this input model without any DFT structure optimization as the relaxed structure exhibited an increased volume (9%) and considerable deviations in polyhedral volumes for the complex Pb-I anions compared to the crystallographic results, potentially as a result of van der Waals interactions not being appropriately captured. The energy cut-off of the plane wave basis set was 500 eV and a Γ -centered k-mesh grid with 24 kpoints was employed, with the convergence criteria set at 0.01 meV. The Perdew-Burke-Ernzerhof (PBE)³⁶ version of the Generalized Gradient Approximation for employed for the exchange-correlation energy. The high-symmetry points were selected for the conventional Brillouin zone of the space group P_{2_1}/c using the Bilbao Crystallographic Server.³⁷ For more accurate band gap values and density of states (DOS), a screened hybrid functional (HSEo6)^{38,39} was employed. Spin-orbit coupling (SOC), which we found to be necessary to accurately describe the band structures of related lead⁴⁰ and bismuth⁴¹ halides was not employed in the results presented here: SOC affects only impacts the band dispersion of the Pb states, which do not contribute significantly to the states around the band gap in (TTF)Pb₂I₅. For comparison, the electronic structure of TTFI_{3}^{42} (P₂₁/n) was calculated using the same computational scheme, and is provided in the Supporting Information. The calculated band gap (HSE-level) of 1.02 eV is indirect between the valence band maximum (VBM, C) and the conduction band minimum (CBM, D) and corresponds to a transition between filled and empty sulfur p and carbon p states. Its magnitude is only marginally smaller than the smallest direct transition (1.03 eV) which is located at the special k point C.

We find the states of the TTF^{+*} radical cation to be energetically surrounded by filled states of the antibonding interaction between the lone-pair Pb s² states with the halogen p states in the valence band and by empty Pb p states in the conduction band. In other hybrid lead halides with simple aliphatic carbon hydrate counter cations, such as CH₃NH₃PbI₃, the filled counter-cation states are energetically far below the Fermi energy and the surrounding Pb-I states constitute the VBM and CBM. The band dispersion of the TTF^{+*} states around the band gap is relatively moderate, as would be expected of the 1D stacks.

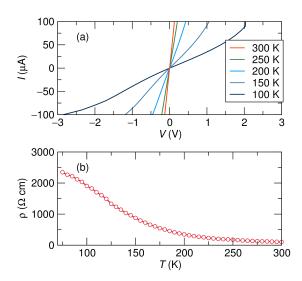


Figure 4. Four-probe measurements for (TTF)Pb₂I₅. (a) IV characteristics over the 100 K to 300 K range. (b) ρ vs. *T* between 75 K and 300 K. The room temperature resistivity of the pressed pellet is close to 100 Ω cm.

Four-probe electrical resistivity measurements were carried out on a pressed pellet of polycrystalline (TTF)Pb₂I₅ with evaporated gold contacts (schematic shown in the SI). Figure 4(a) shows *I*–*V* curves from 100 K to 300 K, suggesting nearly Ohmic behavior near the origin. As can be seen from Figure 4(b), as temperature is increased from 75 K to 300 K the resistivity of (TTF)Pb₂I₅ decreases by two orders of magnitude. The conductivity of the pressed pellet of the title compound is considerably higher than related monocation salts [(TTF)Br_{1.0} and (TTF)Cl_{1.0}] which have conductivities close to 10⁻⁶ S cm⁻¹ when measured similarly.⁴³ We believe the increased conductivity can be attributed to the synergistic effect of the inorganic network.

In summary, we have isolated a TTF^+ radical cation stabilized iodoplumbate that displays semiconducting behavior, with a room-temperature resistivity close to 100 Ω cm (conductivity greater than 0.01 S cm⁻¹). The electronic and optical properties of the compound are greatly influenced by the TTF⁺⁺ radical cation, which creates states in the electronic gap formed by the lead-iodide matrix. We believe these types of synergistic hybrid systems to be potentially promising materials as next generation photovoltaic materials and other optoelectronic applications, that encourages further exploration.

ASSOCIATED CONTENT

Supporting Information

TGA, PXRD, and crystallographic information including .cif of (TTF)Pb₂I₅. DOS structure and the methods used for related (TTF)I₃ compound. The Supporting Information is available free of charge on the ACS Publications website.

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ACKNOWLEDGMENT

This work was supported by the U.S. Department of Energy, Office of Science, Basic Energy Sciences under Award No. DE-SC-0012541. A.J.L. gratefully acknowledges support of Swiss National Science Foundation Fellowship No. PBSKP2-145825. O.B. gratefully acknowledges RISE Program for support. The research involved the use of shared experimental facilities of the Materials Research Laboratory and the Center for Scientific Computing at UCSB supported by the MRSEC Program of the National Science Foundation under Award Nos. DMR 1121053 and NSF CNS-0960316.

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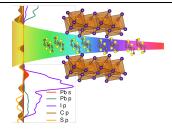
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Supporting Information for: (TTF)Pb₂I₅: a Radical Cation-Stabilized Hybrid Lead Iodide with Synergistic Optoelectronic Signatures

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TGA:

A TA Instruments Discovery TGA was utilized for the thermogravimetric analysis. A rate of 25 cm³/min dry nitrogen purge was employed with a temperature ramp rate of 10°C/min. The maximum temperature of the experiment was 800°C.

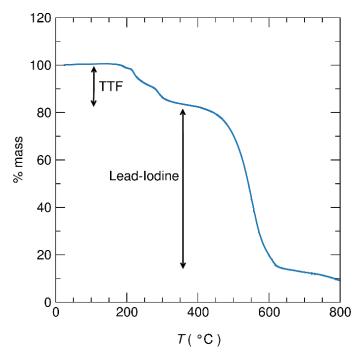
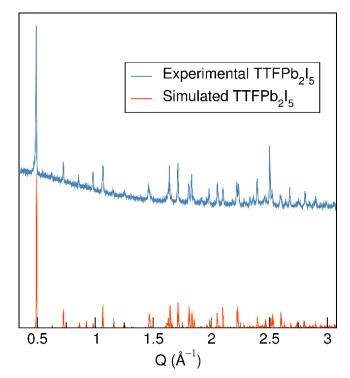


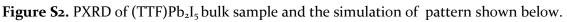
Figure S1. TGA data for the title compound (TTF)Pb₂I₅. The organic component (tetrathiafulvalene, TTF) is presumed to be the first part to degrade from the sample, at

temperatures beyond 200°C. This assertion agrees with the percent mass loss, matching the ~19% mass weight that TTF contributes to the (TTF)Pb₂I₅ formula weight.

Powder X-ray Diffraction (PXRD)

PXRD data was acquired on a Pananalytical Empyrean Powder XRD machine. PXRD data was utilized to establish bulk sample purity of isolated (TTF)Pb₂I₅ over the course of our experiments. We compared experimental PXRD with a simulated pattern generated from single crystal x-ray diffraction data, simulated with the program GSAS.^{1,2} The data shown in Figure S2 was carried out open to air.





Crystallographic information

Single Crystal X-ray diffraction data for (TTF)Pb₂I₅ was collected on a Bruker KAPPA APEX II diffractometer equipped with an APEX II CCD detector using a TRIUMPH monochromator with a Mo K α X-ray source (α = 0.71073 Å). The crystals were mounted on a cryoloop under Paratone-N oil and the data reported here was collected at room temperature. Absorption correction of the data was carried out using the multiscan method SADABS.³ Subsequent calculations were carried out using SHELXTL.⁴ Structure determinations were done using direct methods. All hydrogen atom positions were omitted. Structure solution, refinement, and creation of publication materials were performed using SHELXTL. The graphical depiction used in the main paper was done with the program VESTA.⁵

Empirical formula	$(C_6H_4S_4)Pb_2I_5$
Crystal habit, color	Block,red
Crystal size (mm)	0.05 × 0.05 × 0.025

Crystal system	Monoclinic
Space group	P 21/c
Vol (Å)	2048(6)
a	12.90(3)
b	7.97(1)
с	20.23(2)
α	90
β	100.25
γ	90
Ζ	4
fw (g/mol)	1249.18
Density (calcd) (g/cm ³)	4.051
Abs coeff (mm ⁻¹)	24.344
F _{ooo}	2116
Total no. reflections	13229
Unique reflections	4124
Final <i>R</i> indices $[I > 2\sigma(I)]$	$R_1 = 0.0643 \text{ w} R_2 = 0.0872$
Largest diff peak and hole $(e^- Å^{-3})$	1.685 and -1.591
GOF	0.945

Table S1. Crystallographic information for (TTF)Pb₂I₅ single crystal analysis performed at room temperature

<u>4-Probe Conductivity Measurement Setup</u>

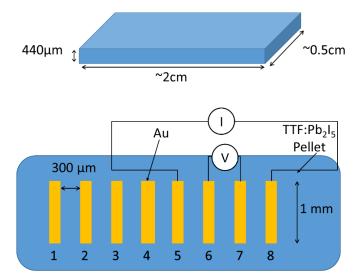


Figure S3. Cartoon depiction of the 4-probe setup employed for the IV measurements. Gold contacts were evaporated onto a pressed pellet of $(TTF)Pb_2I_5$, and contacts 5 – 8 proved most reliable for measurements. These were wired, and utilized in the reported measurements according to the above schematic.

TTFI₃ Band Structure

The computational methods that were employed for the simulation of (TTF)Pb₂I₅, which are described in the main body of the manuscript, were also used to acquire the band structure calculations for TTFI₃.

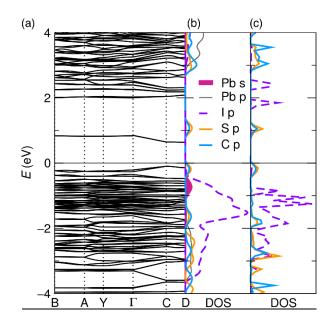


Figure s4: (a) Band structure (PBE) and (b) partial DOS (HSE) of (TTF)Pb2I5. The calculated band gap (HSE: 1.02 eV) is indirect between the special *k*-points C in the valence band and D in the conduction band. (c) Partial DOS of the triiodide (TTF)I₃ for comparison. The Fermi energy level is represented as a horizontal line.

Easy Spin Simulation⁶

The simulation was performed using the pepper function, which is a standard part of Easy Spin. The three *g* tensors reported were utilized as individual values with equal weighting. The crystal space group was included in the calculation as well as the possible angles of position. These angles helped to account for the random orientation of the powder solid-state ESR sample. The angles used were [0 o 0; 0 $\pi/4$ 0; 0 $\pi/8$ 0; 0 $\pi/16$ 0; 0 $\pi/32$ 0; 0 $\pi/64$ 0]. A Gaussian broadening was also applied at a value of 0.3 to all *g* tensors.

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